



The oxidation of soot particulate in the presence of NO₂

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Abstract

The microstructure of soot-like carbons is correlated with their reactivity toward combustion. The focus of this study is on Euro IV heavy duty diesel engine soot. Additionally, two soot samples providing lower and larger particle sizes as their most striking property are taken as references. The effect of NO₂ addition to the O₂-containing gas feed is investigated. It is found that NO₂ accelerates the oxidation of soot in the low-temperature region (250–400 °C) due to an increased surface functionalization with oxygen groups and a subsequent decomposition thereof. Accordingly, initially highly functionalized soot is less susceptible to this effect. The apparent activation energy of combustion is remarkably lowered in the presence of NO₂. It is found that the total reactivity correlates with microstructural features such as surface functionalization, size and curvature of basic structural units, stack height, and particle diameter.

Keywords: Carbon; Microstructure; Surface functionalization; Thermogravimetry; Kinetics; Combustion

1. Introduction

The global emissions of carbonaceous particulate matter from combustion of fossil fuels and burning of biomass are increasing. These particles, which are non-uniformly but widely distributed throughout the troposphere, have a high impact on the atmosphere due to their reactivity with gases such as NO_x, SO_x, or O₃. The physical and structural properties of environmentally relevant carbons have recently been investigated in detail [1], [2], [3] and [4]. The reactivity and adsorption properties of different substances toward carbon particles are important in order to monitor their behavior in the atmosphere. Studies resolving the relative role of soot and dust particles in the climate changes are also undertaken [5]. Their impact on the bio-organism or even on human health [6], [7] and [8] is assessed by investigation of carcinogenicity.

The majority of soot and carbon black related studies misses a detailed investigation of the microstructure or general morphology. Many studies ignore the fact that several types of carbon have been used in oxidation, reactivity, or atmospheric studies including flame soot, carbon black,

and amorphous carbon. This reflects the common opinion (or assumption) that the structure of the soot particles from various sources or synthesis conditions differs little. Most of the studies assume the soot density to be that of graphite. If, however, the nanostructure of soot and carbon black depends on the origin, the question has to be risen how the physical and chemical properties are affected [3], [9] and [10]. Reviews on soot reactivity studies point out that the origin of investigated samples indeed influences their properties [11] and [12]. The present study considers this important link of micromorphology and electronic structure [13] and [14] as well as surface functionalization to reactivity by investigating real diesel engine soot samples.

In order to introduce exhaust treatment systems it is necessary to give an overview of exhaust composition (Fig. 1). Harmful components are included in the exhaust gas to an amount of up to approx. 0.2 vol.%. The particulate matter (soot, ash, oil, and fuel) contributes to the exhaust with an amount of 20–200 mg m⁻³. This is 50 times as much as in the case of the Otto four stroke engine (1–10 mg m⁻³).

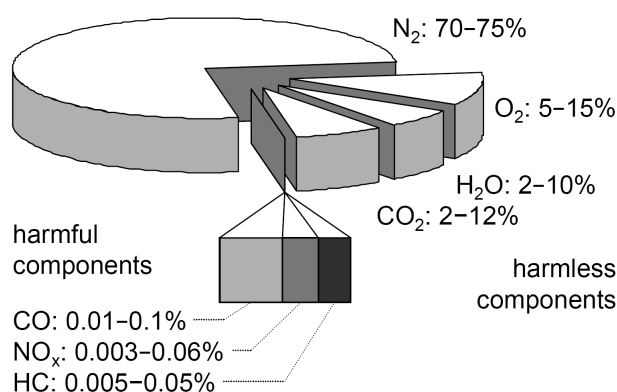
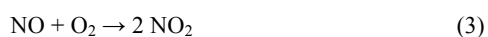


Fig. 1. Composition of heavy duty diesel engine exhaust (data compiled from literature [15]). The diagram shows the amount of basically harmless and harmful compounds in the exhaust gas of a diesel engine.

Practically any particulate matter generated in the diesel engine has to be removed from the exhaust according to the latest EU emission standards [16]. Various attempts have been made to reach the soot particle free diesel engine [12]. Efficient soot oxidation with O₂ can be supported by the addition of catalytically active metal oxide nanoparticles [17], [18] and [19] to the fuel. However, this approach involves the risk of potentially even more hazardous emissions. Another possibility is the oxidation of the diesel soot in filters with the catalytic interaction of NO₂ (Eqs. (1), (2) and (3)) [12].



A wide range of particle trapping and exhaust treatment systems have been introduced and are still under development, for example, ceramic diesel particulate filters, particle traps, and oxidation catalysts with open deposition structures. The general aim is to separate particles in a structure that in principle remains open, and thus does not need to be exchanged. However, care must be taken to ensure that the deposited particles cannot shear off as a result of the aerodynamic forces associated with sudden surges of the gas flow. Removal of particulate matter is done, for instance, by oxidizing the carbon with NO₂. For this purpose an oxidation catalyst, in which NO₂ is formed, could be placed in front of the filter. At temperatures exceeding 200 °C this NO₂ should continuously remove the deposited soot particles. The present study aims at the general feasibility of this strategy.

Heavy-duty (HD) Euro IV soot, soot produced under black smoke (BS) conditions, and aerosol soot produced by spark discharge are investigated. These materials are chosen in order to demonstrate the variety of pollutant diesel exhaust

particles. Moreover, the investigation of bulk and surface properties [1], [2], [3] and [4] are important factors playing a key role in the understanding of the overall behavior of environmentally relevant carbons. The reactivity toward O₂ and NO₂ is investigated by means of thermogravimetric analysis (TGA).

2. Experimental

2.1. Selection of soot samples

Euro IV soot is collected from a HD test diesel engine (6.9 l displacement, 228 kW) equipped with a double-step-controlled supercharging and external-controlled cooled exhaust gas recirculation [20]. The maximal exhaust gas flow at rated speed and full load is 1200 N m³ h⁻¹. This engine is optimized in order to fulfill the Euro IV conditions for HD trucks [16]. The BS soot sample originates from another test diesel engine operating at 30% load and artificially adjusted for high soot emission by air throttling and reducing rail pressure (blackening number 5). Spark discharge soot (GfG soot) is produced with an aerosol generator (GfG 1000, Palas GmbH, Karlsruhe) [21] operated with two graphite electrodes (CRG München, 200 ppm ash), 150 Hz discharge frequency, and 4 L min⁻¹ argon carrier gas flow.

2.2. Thermogravimetric analysis of soot oxidation

For TGA, the procedures are as follows: the TG/DSC data is acquired using a Netzsch-STA 449 instrument with Al₂O₃ crucibles. The samples are evacuated and the sample chamber is re-filled with 10% O₂, 1.5% H₂O, and balance N₂, which is maintained at a total flow rate of 100 mL min⁻¹. Steam is added to the feed gas to approach realistic exhaust conditions. The NO₂ content is varied between 0 and 1000 ppm and a heating rate of β = 5 K min⁻¹ is used. The gas phase products are transferred through a heated quartz capillary to a Balzers Thermostar quadrupole mass spectrometer (QMS) operated in SIM mode. The only products observed are CO₂ (m/e 44) and H₂O (m/e 18). The ion currents for these masses are divided by the ion current for m/e 14 (N₂) to compensate for changes in sensitivity of the QMS. A background is subtracted and the resulting intensity is normalized with regard to the mass of soot. The sample charge used for TGA measurements is about 1 mg.

2.3. Kinetic analysis

Apparent activation energies E_a and frequency factors A₀ are calculated from the TGA profiles using the temperature of maximum combustion rate T_{max}. As shown by Brukh and Mitra [22] (Eq. (4)), the plot of ln(T_{max}²/β) as a

function of $1/(RT_{\max})$ results in a straight line with the slope E_a and intercept $\ln(E_a/A_0)$, from which E_a and A_0 can be determined.

$$\ln\left(\frac{T_{\max}^2}{\beta}\right) = \frac{E_a}{RT_{\max}} + \ln\left(\frac{E_a}{A_0}\right) \quad (4)$$

It should be stressed that due to the non-isothermicity of the catalyst bed (hot spots) and especially due to the anisotropic structure of (partly) sp^2 hybridized carbonaceous materials, which strongly affects the kinetics of combustion [23] on the micro- and macro-kinetic scale (Fig. 3d), this method yields only rough estimates of apparent kinetic constants.

3. Results and discussion

3.1. Structural and electronic characterization

Representative HRTEM micrographs of the investigated soot samples are shown in Fig. 2. The morphological differences are listed in Table 1. Data from statistical analysis of HRTEM micrographs and quantitative evaluation of EELS spectra [4] indicate that the structure of Euro IV soot is somehow in between BS and GfG soot. BS soot consists of relatively large spheres with rather flat basic structural units (BSUs) in agreement with a high sp^2/sp^3 hybridization ratio. Contrarily, GfG soot comprises very small and curved fullerene-like subunits with a high degree of sp^3 hybridization.

IR adsorption bands according to $C-O$, $C=O$, and $O-H$ vibrations are identified on each soot sample, however, in highest intensity on the GfG soot, whereas the BS soot provides a comparably low degree of surface functionalization with oxygen groups. The oxygen content in the order $GfG > Euro\ IV > BS$ is confirmed by XPS analysis. A detailed analysis of investigated soot samples is published elsewhere [4].

3.2. Oxidation of soot

Fig. 2a shows the oxidation behavior of Euro IV soot. The influence of the oxidizing agent NO₂ is revealed. In the range of 200–500 °C the mass loss strongly increases in the presence of NO₂. The NO₂ concentration was varied between 0 and 1000 ppm and a steady increase of reactivity is observed [4]. Beyond 500 °C the oxidation is not accelerated anymore and the total burn-off temperature of the soot remains unaffected. The remaining weight in the TG experiments (Fig. 2 and Fig. 3 show normalized data) is ash from residual engine lubricant oil. Its amount varies between 0 and 10 wt.%. However, a catalytic impact on soot oxidation can be excluded as repeated TG experiments under identical conditions resulted in good reproducibility of (normalized) weight loss profiles. Fig. 3a reveals differences in

Table 1. Structural and electronic differences of investigated soot samples (data compiled from literature [4]).

	Euro IV	BS	GfG
particle diameter (nm)	18 ± 6	24 ± 7	< 3
BSU size (nm)	1.4 ± 0.8	1.6 ± 0.9	1.3 ± 0.5
BSU curvature	0.83 ± 0.06	0.87 ± 0.04	0.79 ± 0.08
sp^2/sp^3 ratio	66:34	76:24	54:46
surface oxygen (at.%)	12.0	7.4	16.9

CO₂ evolution during oxidation of Euro IV diesel engine soot. In the absence of NO₂, the development of CO₂ begins at 300 °C, which is in agreement with the initial mass loss as observed in Fig. 2a. Note that the initial mass loss below 100 °C is due to desorption of H₂O. The CO₂ signal passes a maximum at 580 °C, where the rate of mass loss also reaches its maximum. The profile sharply drops to zero and the soot is completely oxidized at 630 °C. The addition of NO₂ notably changes the oxidative behavior of Euro IV soot. A pronounced evolution of CO₂ at lower temperatures is observed. The inset in Fig. 3a magnifies the curve shape in the lower temperature range, which is characterized by an onset at 200 °C followed by a plateau of rather constant rate of CO₂ formation rate up to 400 °C. Then, similar to the experiment without NO₂, the maximum of CO₂ release is at 580 °C in agreement with the maximum rate of mass loss at this temperature. Finally the soot is burnt at around 610 °C and the CO₂ signal returns to zero.

BS soot apparently contains less volatile compounds than Euro IV soot as indicated by the negligible weight loss in the low temperature region (Fig. 2b). Here, the addition of NO₂ does not influence the mass loss profile up to 200 °C. In the temperature range between 250 and 600 °C, an accelerating impact of NO₂ on the oxidation rate is revealed. As for Euro IV soot the mass loss over temperature increases with higher NO₂ concentration, however, at the heating rate of 2 K min⁻¹ the total burn-off temperatures are essentially unaffected at 630 K. Fig. 3b shows the m/e 44 traces of the TG experiments with and without the addition of 500 ppm NO₂. Differences in CO₂ evolution during oxidation of the BS soot are revealed. In the case of the experiment without NO₂, the evolution of CO₂ begins at 250 °C as shown in the inset in Fig. 3b. The signal passes a maximum at 645 °C and reaches zero at 670 °C indicating that the soot is completely oxidized. In the case of NO₂ addition, the MS signal rises at lower temperatures. The formation of CO₂ already starts at 150 °C and at lower temperatures (150–600 °C) the generation of CO₂ is higher as compared to the experiment without NO₂. However, the signal shows a maximum at a slightly higher temperature of

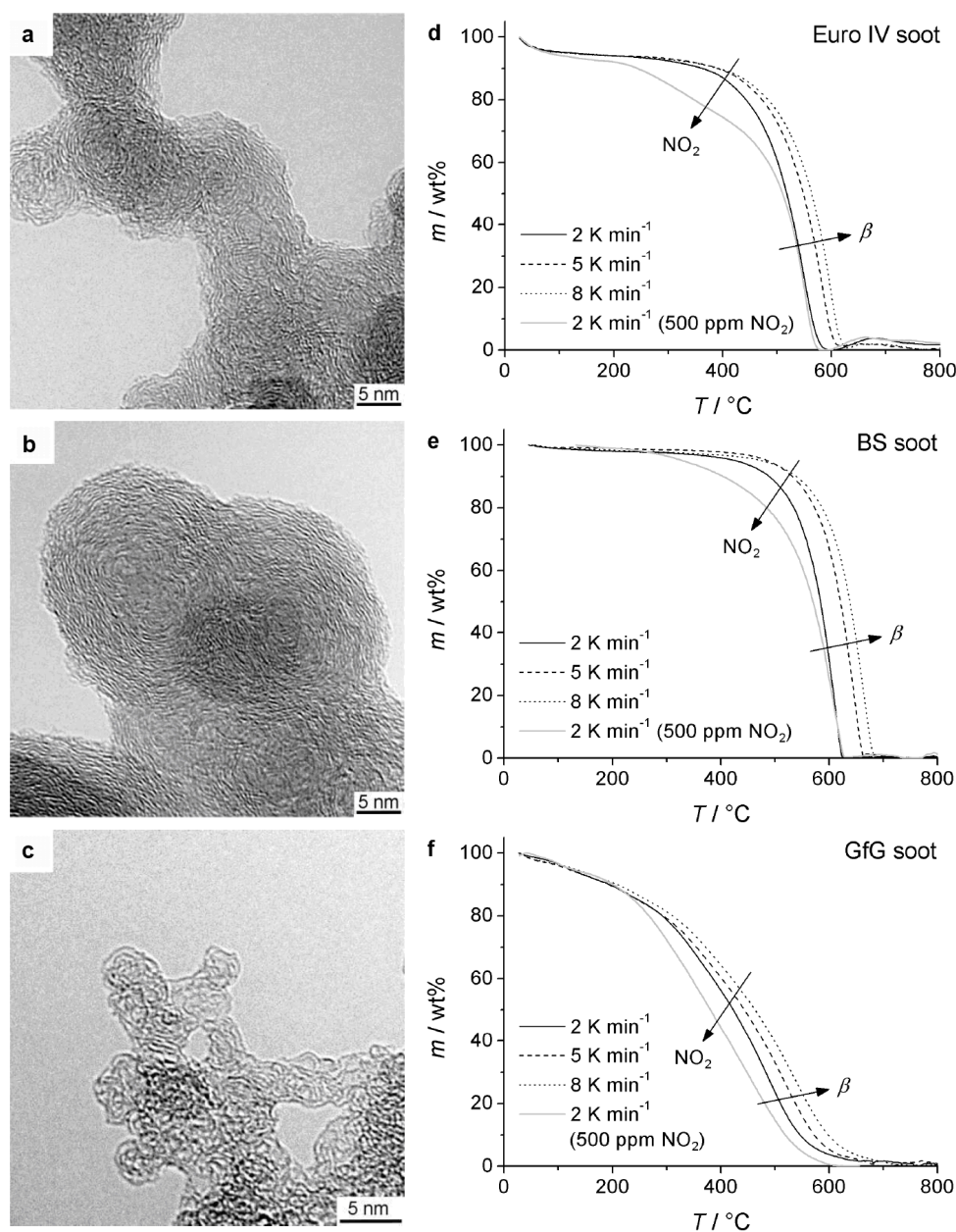


Fig. 2. HRTEM micrographs of (a) Euro IV, (b) BS, and (c) GfG soot. Reprinted from literature [1]. Reproduced by permission of the PCCP Owner Societies. (d–f) TGA measurements of soot samples in 10% O₂/1.5% H₂O/bal. N₂ and in 500 ppm NO₂/10% O₂/1.5% H₂O/bal. N₂; arrows indicate the effects of NO₂ addition and increasing heating rate β .

655 °C and under these conditions the soot is totally burnt at 670 °C.

In case of the GfG soot, the NO₂ concentration does not significantly influence the mass loss behavior in the temperature range between 50 and 200 °C (Fig. 2c). Similar to the other soot samples the temperature range between 200 and 590 °C shows an acceleration of soot oxidation in the presence of NO₂. The combustion is completed at 600 °C. The CO₂ profiles recorded during the TG experiments reveal only slight differences for the GfG soot (Fig. 3c). The evolution of CO₂ begins at 100 °C and two maxima are observed at 380 and 500 °C, respectively. The sig-

nal slowly drops to zero at 750 °C. In the case of NO₂ addition, the CO₂ signal rises at a lower temperature of around 70 °C. Here, the signal rises faster, showing an increase in the oxidation rate at lower temperatures. The first maximum in CO₂ generation is similar in position and intensity with the experiment without NO₂, however, the second peak is somewhat less intense and weakly shifted to the higher temperature of 510 °C. Complete oxidation is achieved at 700 °C.

The mechanistic complexity of soot oxidation is reflected in the non-homogeneous curve shapes of TG experiments, especially as observed for the GfG soot sample. The pro

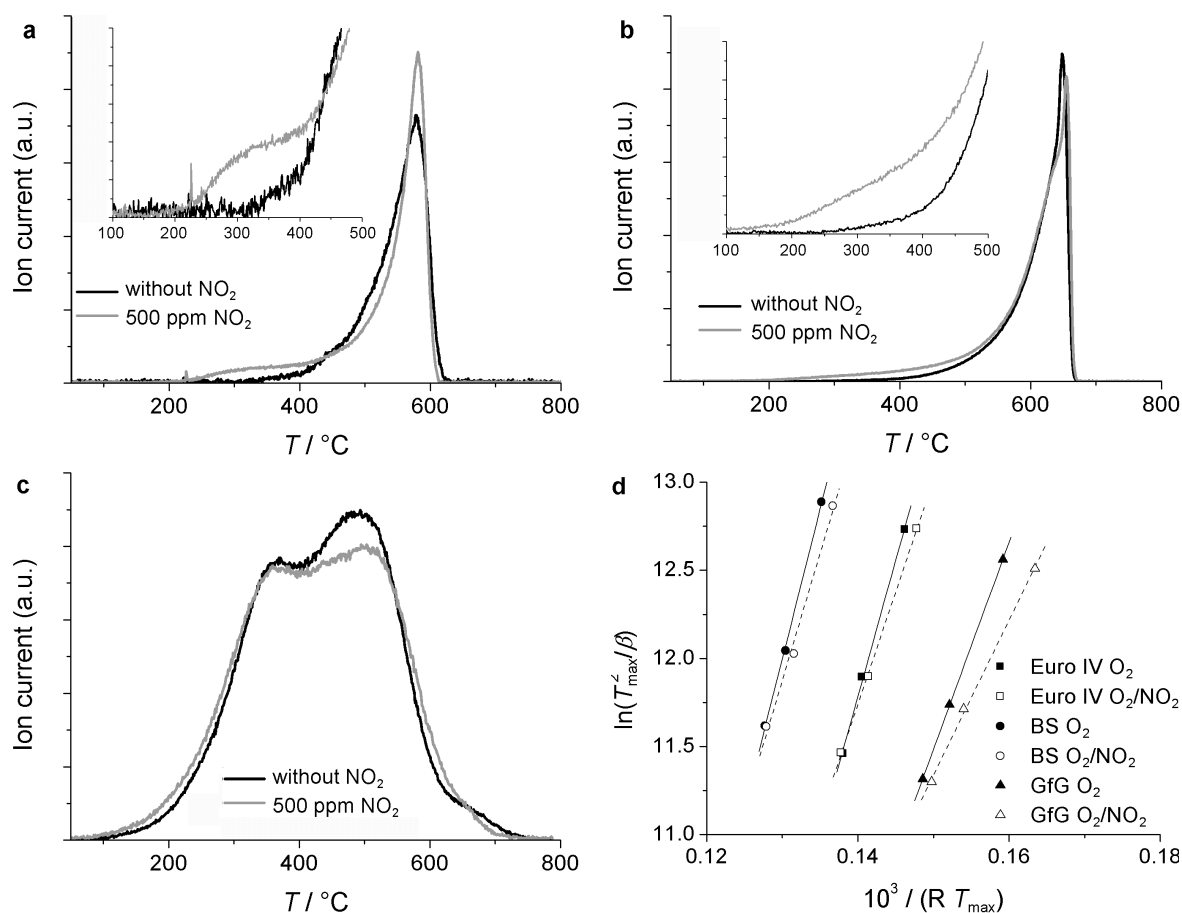


Fig. 3. (a-c) Comparison of CO₂ profiles recorded during TGA (5% O₂/1.5% H₂O/bal. N₂) in the absence and presence of 500 ppm NO₂ (inset: magnification of low-temperature CO₂ formation); (d) determination of apparent kinetic parameters.

cess is illustrated in Fig. 4 and begins with initial surface oxidation and evaporation of volatile compounds such as adsorbed hydrocarbons and water. The latter process is nicely resolved for Euro IV soot (Fig. 2a). The spherical soot particles are then subjected to pervasive formation of small pores, which increase in size due to enhanced access to gas phase O₂ and finally the spheres reduce in size until total combustion. All these processes can contribute to the complex curve shapes as observed in Fig. 2 and Fig. 3.

Apparent activation energies were determined from the change of the temperatures of maximum weight loss when varying the heating rate of TGA (arrows in Fig. 2). As shown in Fig. 3d good linearity is obtained according to Eq. (4). The data are listed in Table 2. It is seen that the apparent activation energies obtained for Euro IV, BS, and GfG soots are lower with 500 ppm NO₂. The values drop from 155 to 127 kJ mol⁻¹, from 170 to 144 kJ mol⁻¹ and from 117 to 88 kJ mol⁻¹, respectively.

4. Discussion

4.1. Influence of microstructure on reactivity

Various models have been developed to describe the structure of soot [25] and [26]. The BSU of soot consists of carbon atoms located at basal plane and edge site positions. Small graphene segments, i.e., BSUs, are frequently aggregated in short stacks with a turbostratic structure. The dimensions of the crystallite stacks considerably vary between different soots and carbon blacks. The relative number and accessibility of potential reactive carbon layer edge sites (aryl C—H bonds) depends on the soot nanostructure. These sites are potentially accessible for reaction with O₂ or NO₂. Theoretical models for carbon gas phase reactions have been developed [27]. Larger graphene layer planes, larger (and hence fewer) crystallites, and concentrically oriented crystallites result in fewer potential reaction sites [28], or sites of lower reactivity. As revealed by the combination of TEM and TG experiments, the fullerenoidic small graphenes are more easily oxidized than the

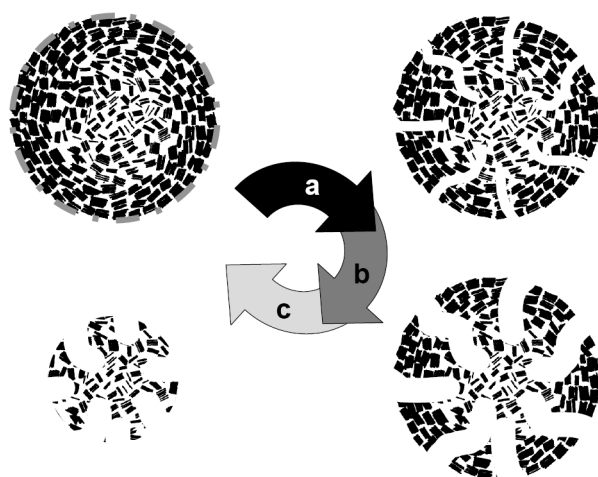


Fig. 4. Illustration of the oxidation of soot spheres: initial surface oxidation and removal of volatile parts is followed by (a) formation of small pores by anisotropic oxidation of graphene layers, (b) pore widening, and (c) size reduction. Carbon sphere reprinted from literature [25], © 1997, with permission from Elsevier.

Table 2. Apparent activation energies (kJ mol⁻¹) of soot samples determined from Fig. 3d in the absence and presence of NO₂.

	Euro IV	BS	GfG
without NO ₂	155 ± 6	170 ± 6	117 ± 1
500 ppm NO ₂	127 ± 3	144 ± 14	88 ± 3

stacked flat graphenes. Thus, the number of such potential sites is a direct reflection of the detailed soot particle nanostructure. Graphite oxidation proceeds anisotropically, i.e., the reactivity of basal plane carbon atoms is far lower than that of edge site carbon atoms [29]. Thus the oxidation rates of graphitized carbon blacks exhibit higher threshold temperatures and much slower rates than the non-graphitized GfG and Euro IV HD diesel engine soots. Geometrically, carbon atoms in edge sites can form bonds with chemisorbed oxygen due to the availability of unpaired sp² electrons. Carbon atoms in basal planes can only share one of four electrons for chemical bonds to heteroatoms. Consequently, the observed reactivity will be an average of basal vs. edge site carbon atom reactivities for graphitic layer planes of finite dimensions [30]. As the layer plane size decreases, the number of edge site carbon atoms will necessarily increase in proportion to the number of basal plane carbon atoms, allowing one to expect the overall reactivity to increase. In addition to the size of the graphene segments, their relative curvature will also influence their oxidation rate. Curvature arises from non-6-membered carbon rings within the aromatic framework. This curvature imposes bond strain as the orbitals overlap,

and the electronic resonance stabilization is lessened [31]. Therein the C—C bonds are weakened and individual atoms are more exposed, i.e., they are more susceptible to oxidative attack [32]. This leads to the observed structure reactivity correlation. It is for these same reasons that fullerenes and carbon nanotubes are less resistant toward oxidation than planar graphite [23]. The GfG soot is the most reactive of all three investigated materials. The reason is the highly defective structure, which is observed in the TEM micrographs. The material consists of fine subunits not larger than 3 nm building up agglomerates. Strongly bent graphenes lead to localized double bonds resulting in an olefinic structure [33]. Theoretical investigations of nanocarbons predict the influence of geometric changes on chemical properties (increasing reactivity upon increasing curvature) [34]. TEM and TPO results are in perfect agreement, as evidenced by the low apparent activation energy of 117 kJ mol⁻¹ that is needed to oxidize GfG soot. The Euro IV HD diesel engine soot is less reactive than the GfG soot but still more than the BS soot. The analysis of the apparent activation energy that is necessary to oxidize the Euro IV soot results in an intermediate value of 155 kJ mol⁻¹. One reason is seen in the size distribution of primary particles – the majority of the particles have a size of 10–15 nm. Comparing the TEM images the Euro IV soot appears more compact and thus is less accessible to oxygen than the GfG soot. However, the crucial reason is seen in the multi-shell fullerene-like structure with a defective surface. The defective non-six-membered carbon rings may produce highly localized olefinic electronic structures prone to the addition of molecular oxidants [35], as already described for the GfG soot. Not only the morphology but also the surface chemistry of the Euro IV soot differs from that of the GfG soot.

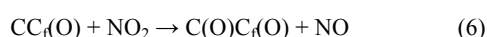
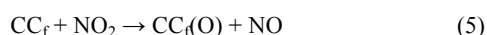
The BS soot is less prone to oxidation due to the well developed graphitic properties. The BSUs are flat, indicating a less defective structure. One observes domains where graphenes are stacked forming graphitic nanocrystallites. The ratio of edge to in-plane carbon atoms is low. These structures decrease the reactivity [11] and [36]. Additionally the local density is higher. The higher apparent activation energy of 170 kJ mol⁻¹ as obtained from the TG measurements is a clear indication for these influences [37]. A lower amount of defects should result in fewer functional groups and this is reflected in the DRIFT spectra [4]. The high rate of oxidation of the GfG and the Euro IV soot is due to the defective graphenes that are more reactive than these of the BS soot.

4.2. Influence of gas phase on reactivity

The influence of NO₂ and H₂O on diesel engine soot oxidation has been investigated in various studies prior to this work. The oxidation of the carbon black Printex-U, graphite and activated carbons in presence of NO₂, O₂, and NO on soot aerosols as well as on diesel engine soot have

been investigated [38], [39] and [40]. Phenomenological models for soot oxidation were presented [37], [41] and [42]. Other groups investigated the influence of O₂ and NO₂ on the generation of surface oxygen groups [43] and [44]. The influence of an optimized design of particulate traps on the soot oxidation was tested [45], as well as the influence of engine settings and exhaust gas temperatures [46].

It is generally accepted, that NO₂ enhances the oxidation of soot and carbon black. The effect of induced reactivity with higher amount of NO₂ is an effect of enhanced graphene functionalization with oxygen groups via the following reactions (Eqs. (5) and (6)) [43],



where CC_f indicates a free edge carbon atom, CC_f(O) and C(O)C_f(O) both represent surface oxidized carbon atoms, which can decompose yielding CO or CO₂. This effect takes place due to the easier dissociation of NO₂ on graphenes in comparison with the O₂. It has been shown in a recent ab initio modeling study of the oxidation of the basal plane of graphene that the initial step is chemisorption of an oxygen atom across a C—C bond of the graphene sheets. The calculated binding energy of the chemisorbed oxygen atom is 250–300 kJ mol⁻¹, which must be compared with the bond dissociation enthalpies E_d(O—O) = 498 and E_d(ON—O) = 306 kJ mol⁻¹, respectively. The reaction is exothermic, but enhanced with NO₂ due to the lower dissociation energy. Transferred to the results of soot and carbon black oxidation, this means that at lower temperatures the first step to the oxidation of the carbons, being the functionalization, is enhanced. The total oxidation of the carbon to CO₂ occurs at lower temperatures. When comparing the oxidation in various amounts of NO₂, it is apparent that highly functionalized carbons are less prone to an enhancement of oxidation with NO₂. The GfG soot does not display such significant differences in the oxidation curves as the Euro IV or the BS soot. From the TG experiments it is apparent that the influence of the oxidation of the soot through NO₂ is dominant at temperatures in the range of 200–450 °C. The first step of creating an oxygen functional group is easier due to the strong oxidative properties of the NO₂. The addition of NO₂ increases the oxidation of the graphitic parts of the soot. Apparently the oxidation of

graphitic materials only in the presence of O₂ does not take place at temperatures below 400 °C. NO₂ lowers the apparent activation energy at temperatures relevant for diesel soot oxidation. For temperatures between 50 and 350 °C, activation energies in the range of 50–80 kJ mol⁻¹ are reported. This correlates at least qualitatively with the data obtained in this study (Table 2). Nevertheless, the general soot nanostructure plays an important role in the overall reactivity. Different reaction rates and activation energies are reported in the literature. In many cases this is attributed to different experimental setups. In this work it is obvious, that the differences in activation energies and rates stem from the differences in microstructure of the soot. So even in the case of gasification of soot with NO₂, the structure reactivity correlation previously discussed holds. It is important to mention that in the case of the GfG soot the overall reaction is enhanced but not as significantly as in the case of the Euro IV or the BS soot. It seems that the more functionalized the graphenes are, the less the soot oxidation can be enhanced by NO₂. Probably, NO₂ generates thermodynamically favored surface functional groups that play the key role in carbon oxidation at temperatures below 400 °C.

5. Conclusion

The widely accepted structure–reactivity–correlation for the oxidation of nanostructured carbonaceous materials could be confirmed for three structurally differing soot samples. The NO₂ rich atmosphere enhances the soot oxidation at lower temperatures by generation of active surface functional groups. The enhanced reactivity of soot emitted from modern diesel engines plays an important role in the exhaust aftertreatment. With both of these issues the diesel engine exhaust aftertreatment is feasible at temperatures typical of the exhaust train.

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