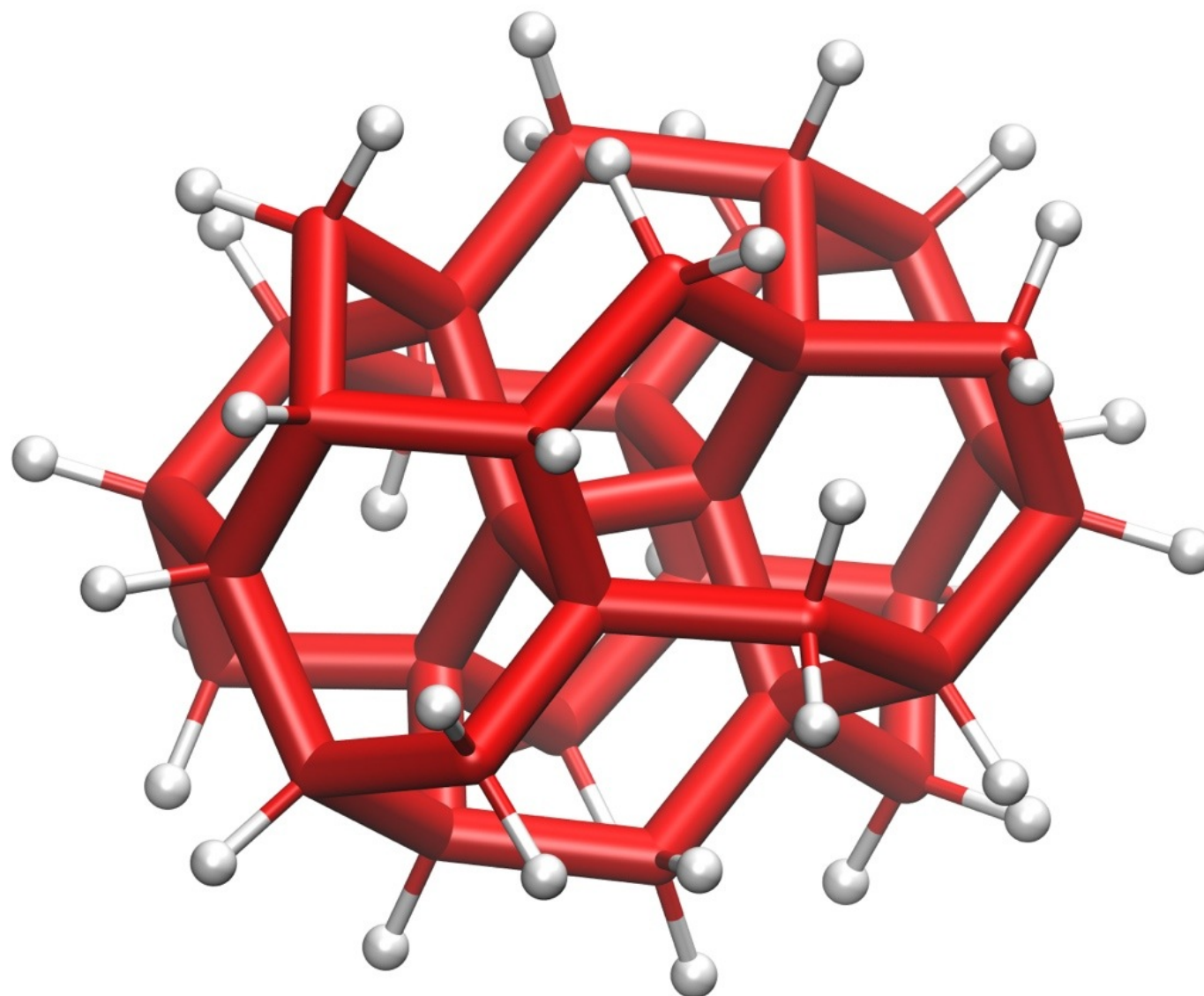




Multiple Exciton Generation in Si and Ge Nanoparticles with High Pressure Core Structures

S. Wippermann, M. Vörös, D. Rocca, A. Gali, G. Zimanyi, G. Galli



NanoMatFutur



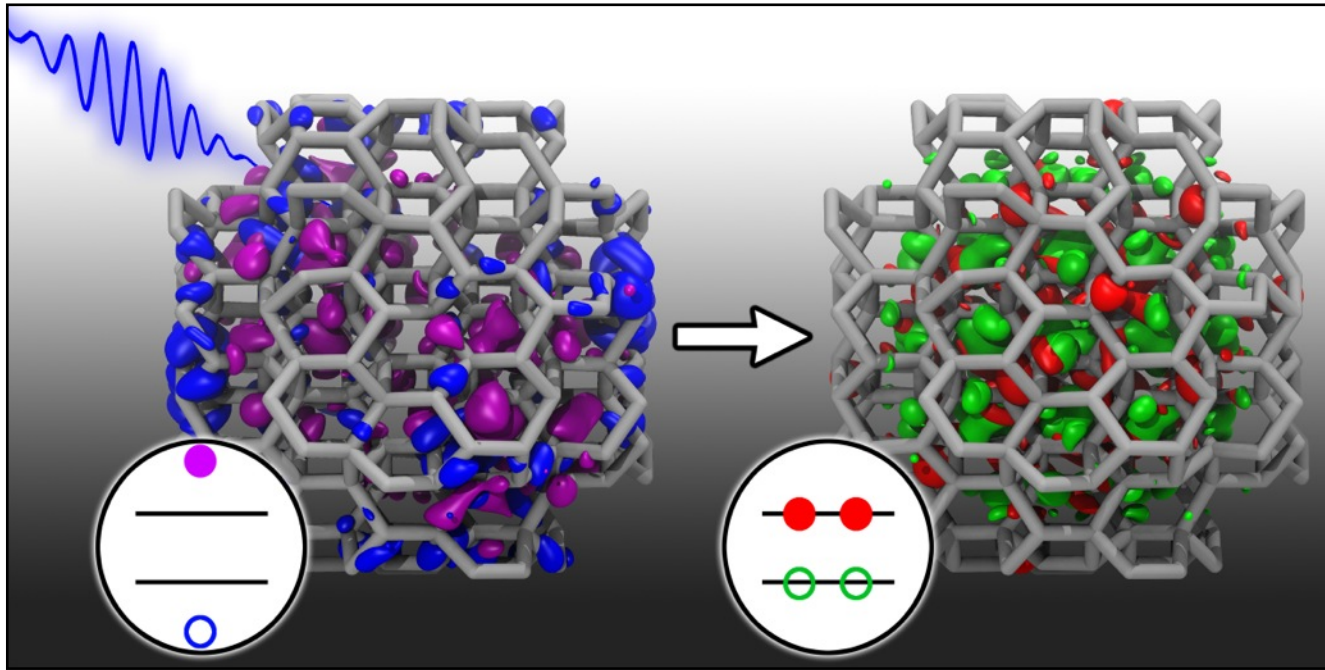
Bundesministerium
für Bildung
und Forschung



MAX-PLANCK-GESELLSCHAFT

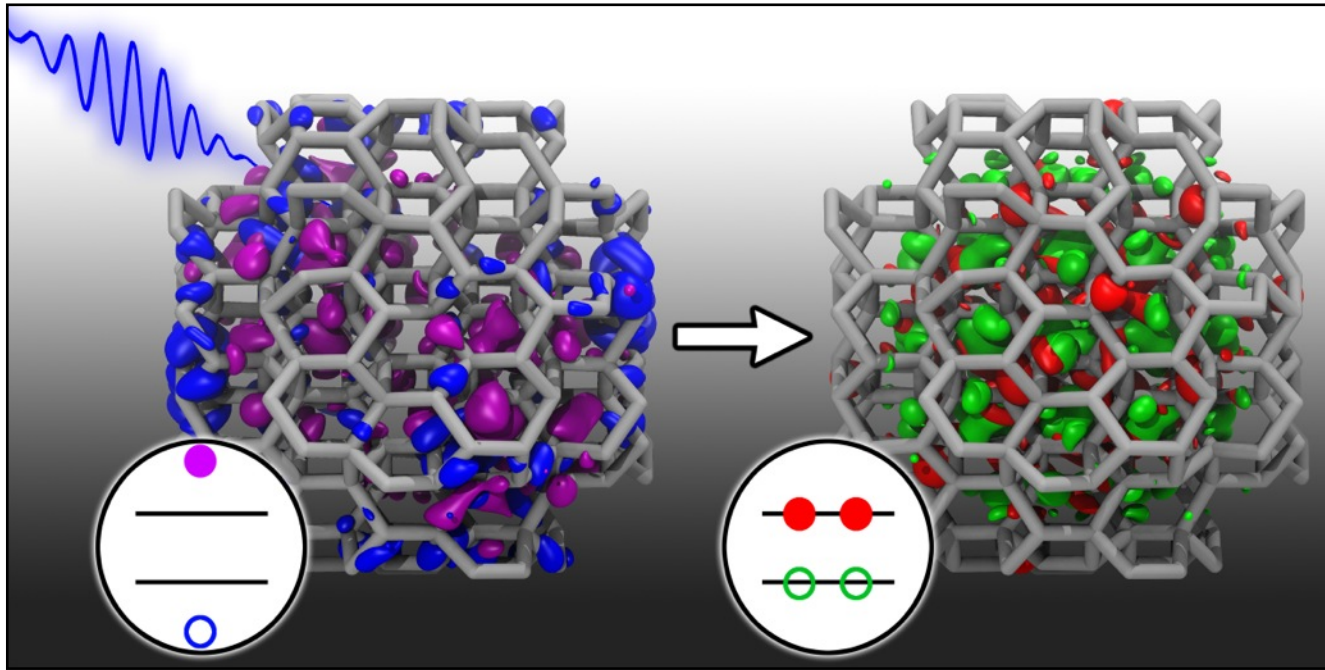
DPG-2014, 04/03/2014

Multiple Exciton Generation (MEG)



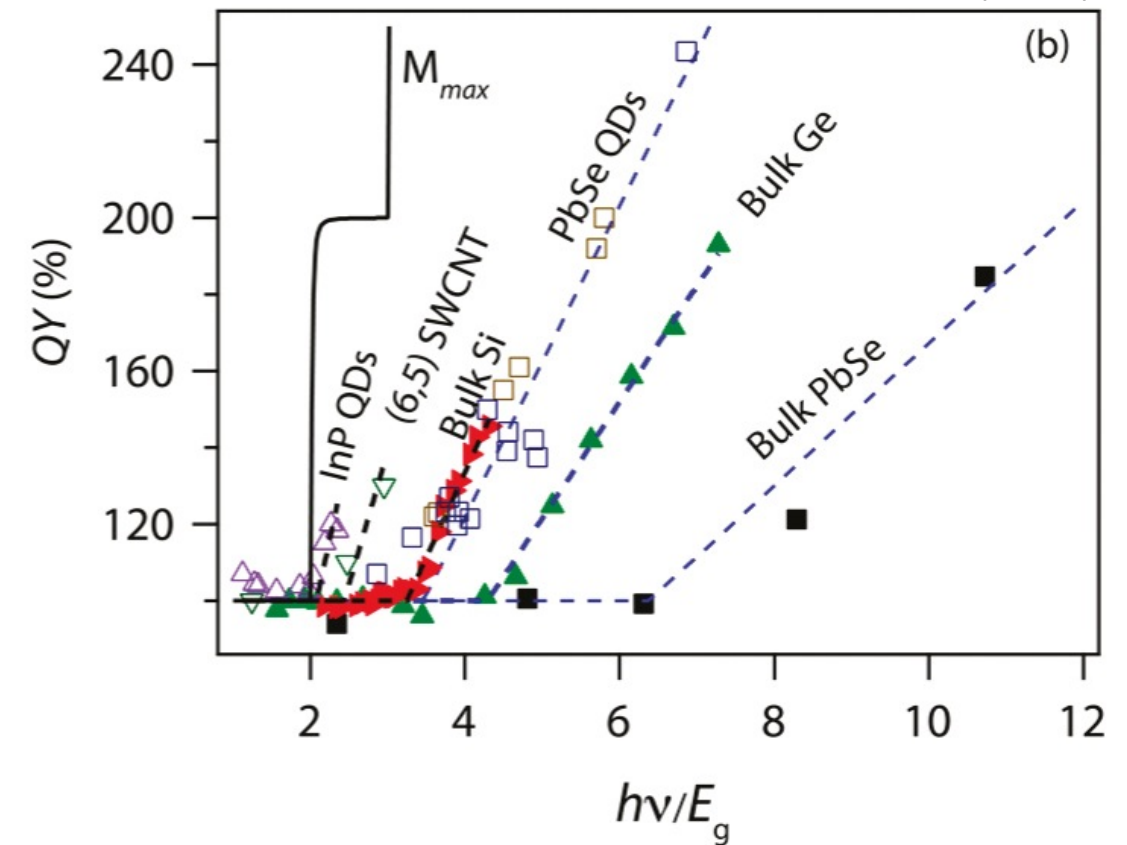
- **MEG:** hot exciton relaxes by exciting another exciton
- Promising pathway to exceed Shockley-Queisser limit in 3rd generation solar cells

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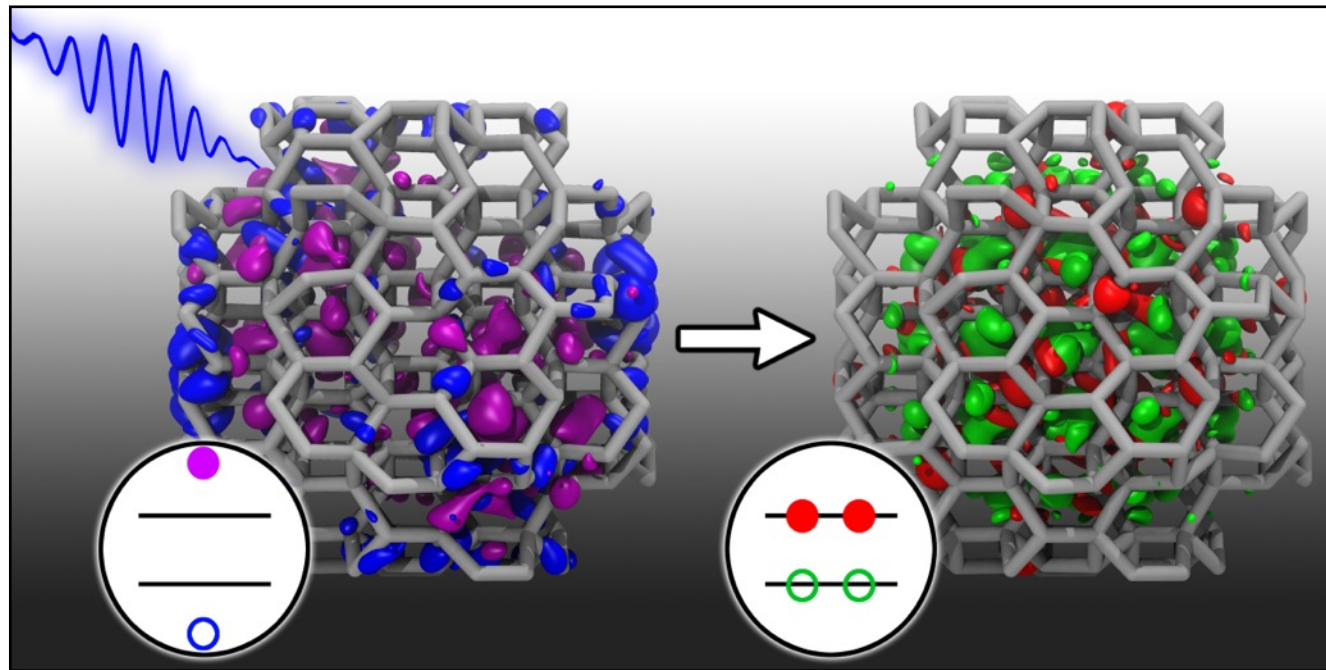
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M. Beard, JPCL 2, 1282 (2011)

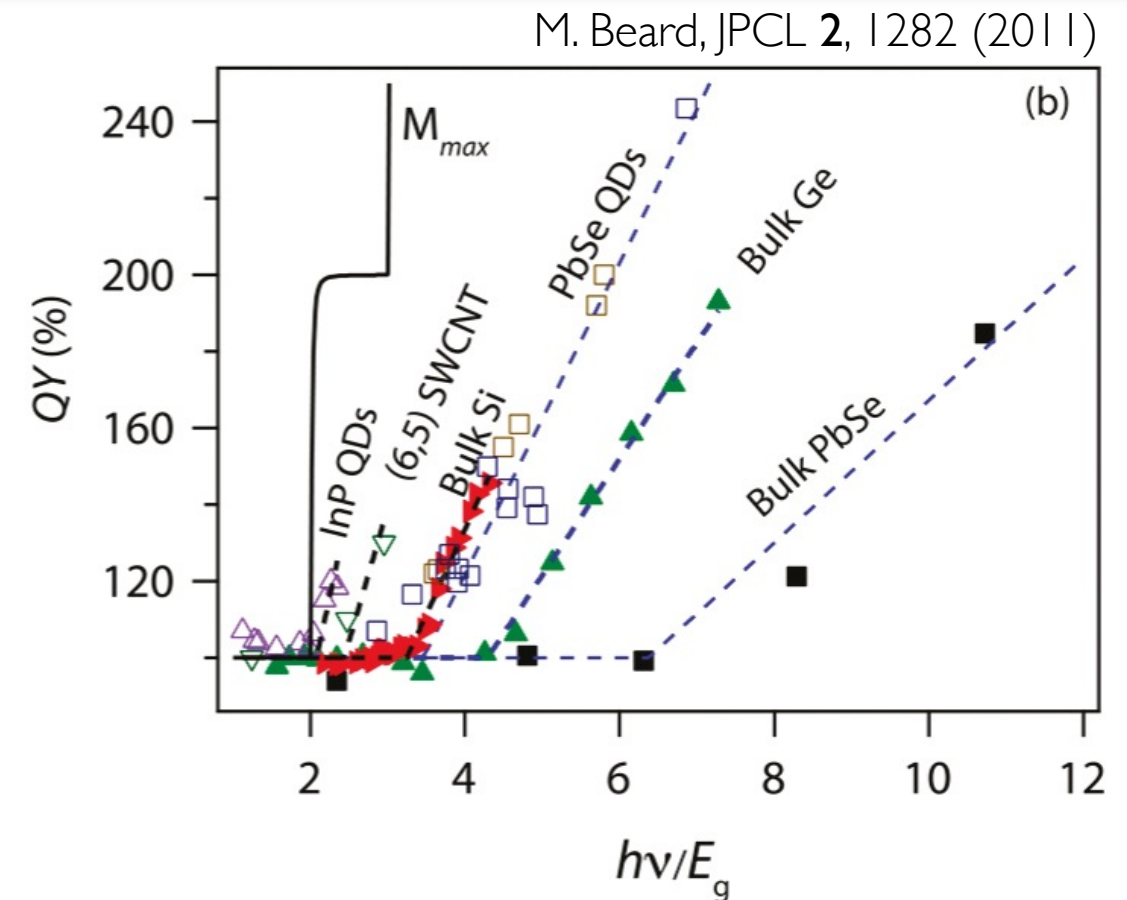


- enhanced MEG in quantum-confined nanostructures, e. g. nanocrystals

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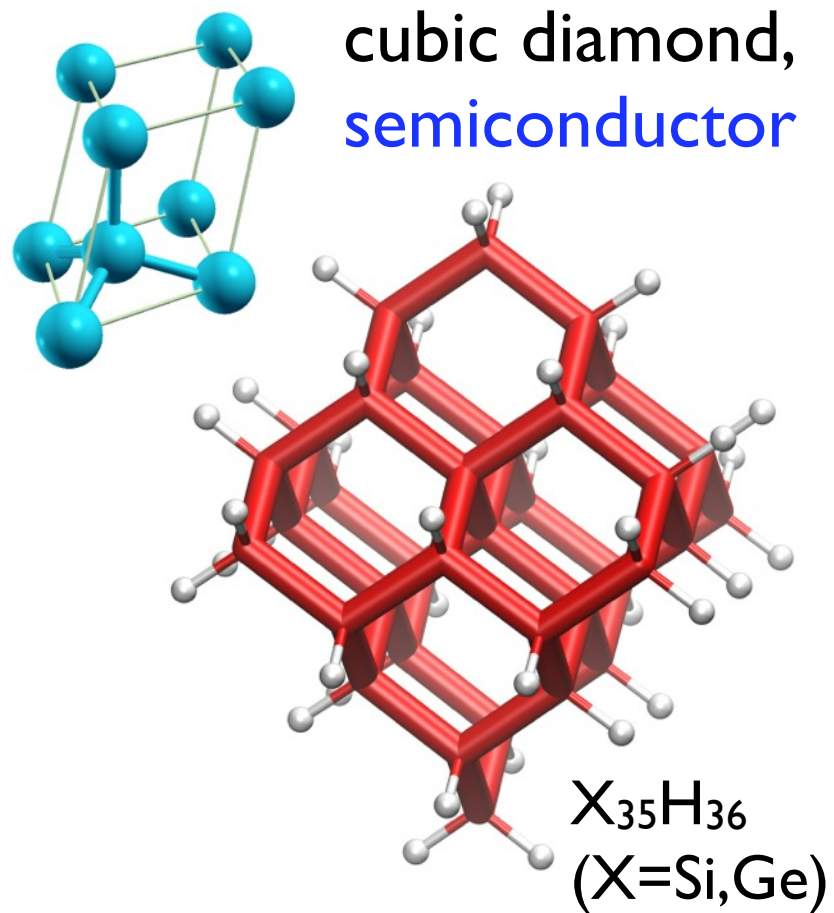


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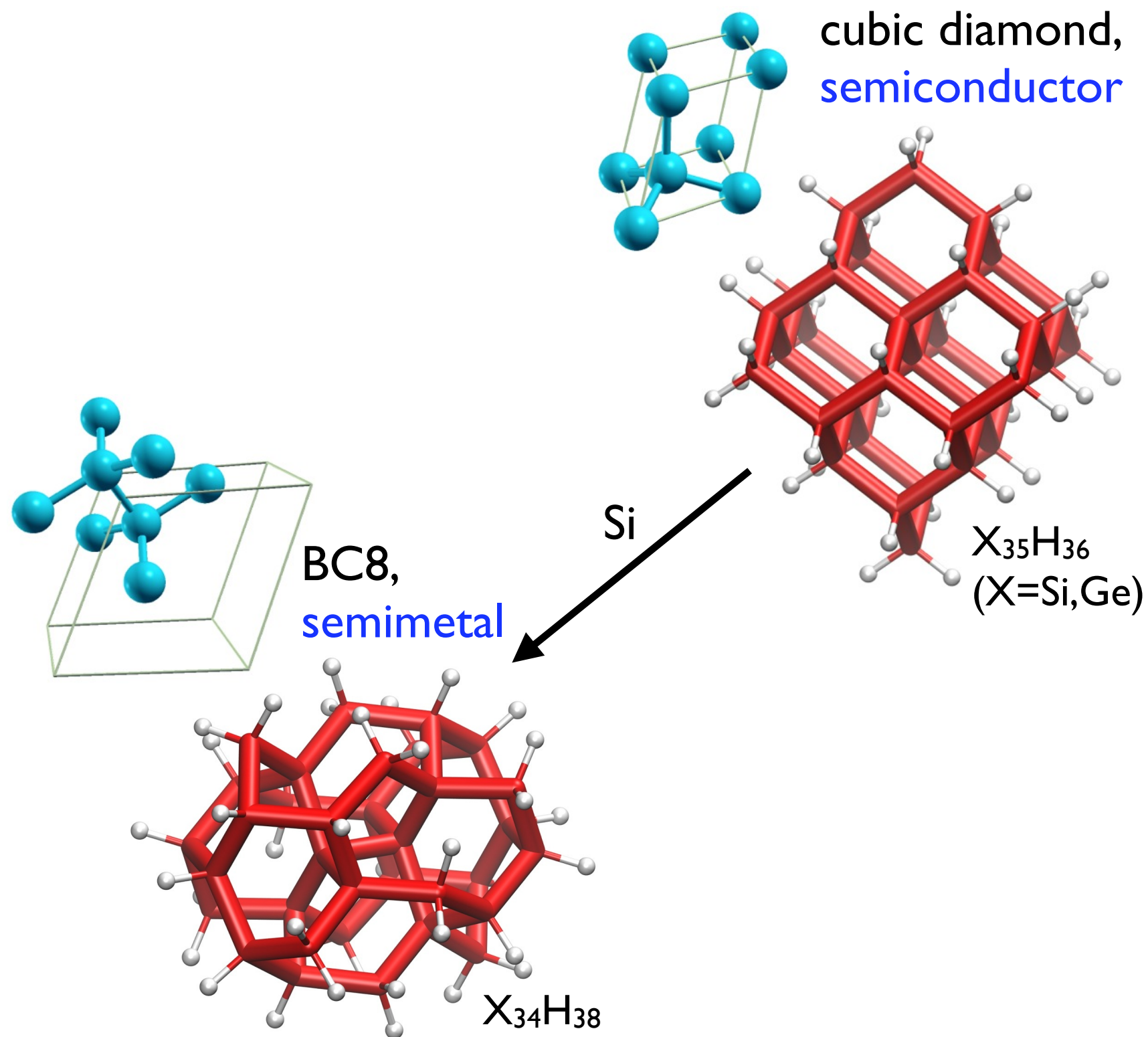
Key problem

- Quantum confinement required for efficient MEG, but pushes electronic gap beyond solar spectrum
- MEG observed in Si and Ge: gap/wavefunction engineering possible?
=> investigate nanocrystals with high pressure core structures

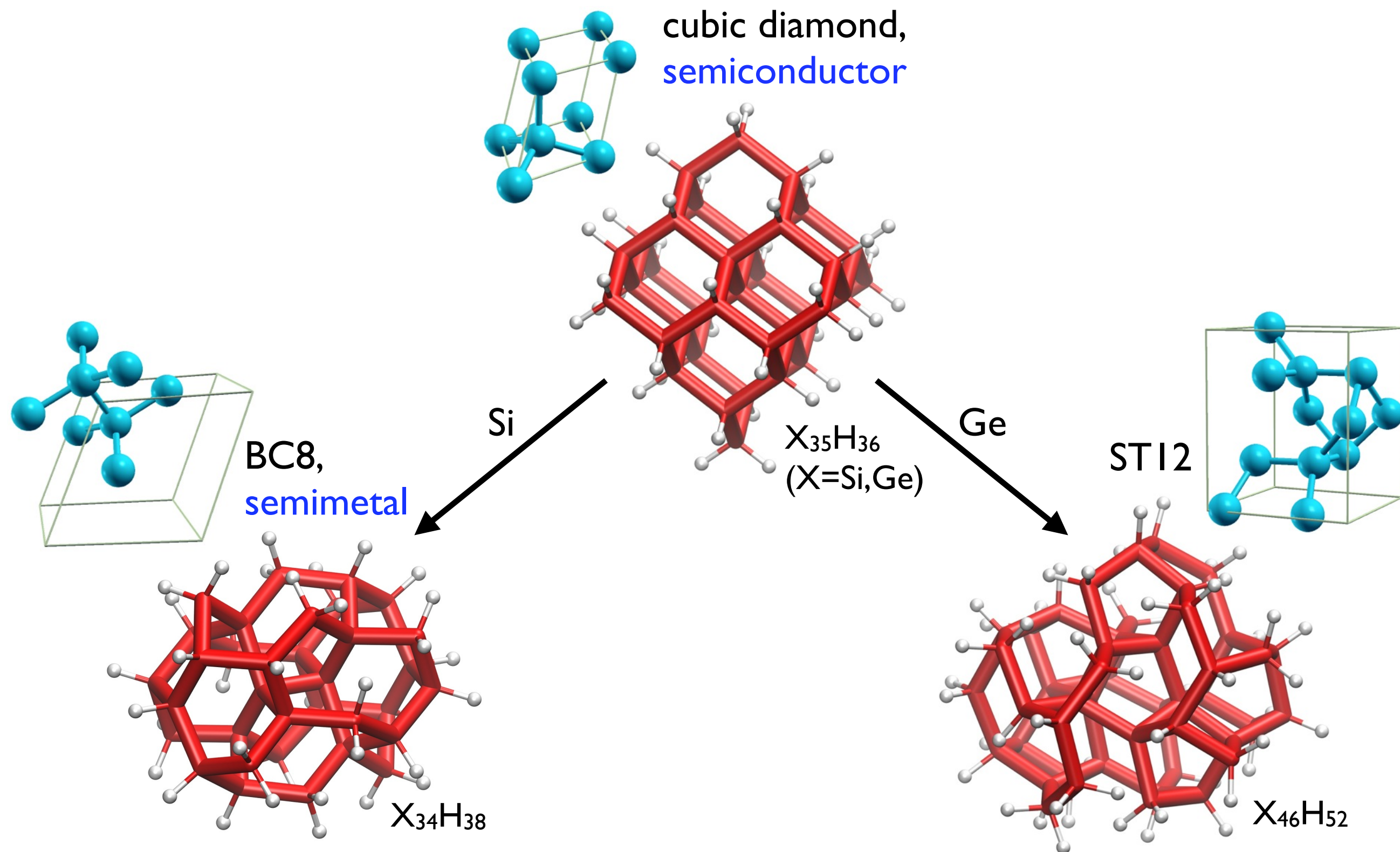
Allotropes of Si and Ge



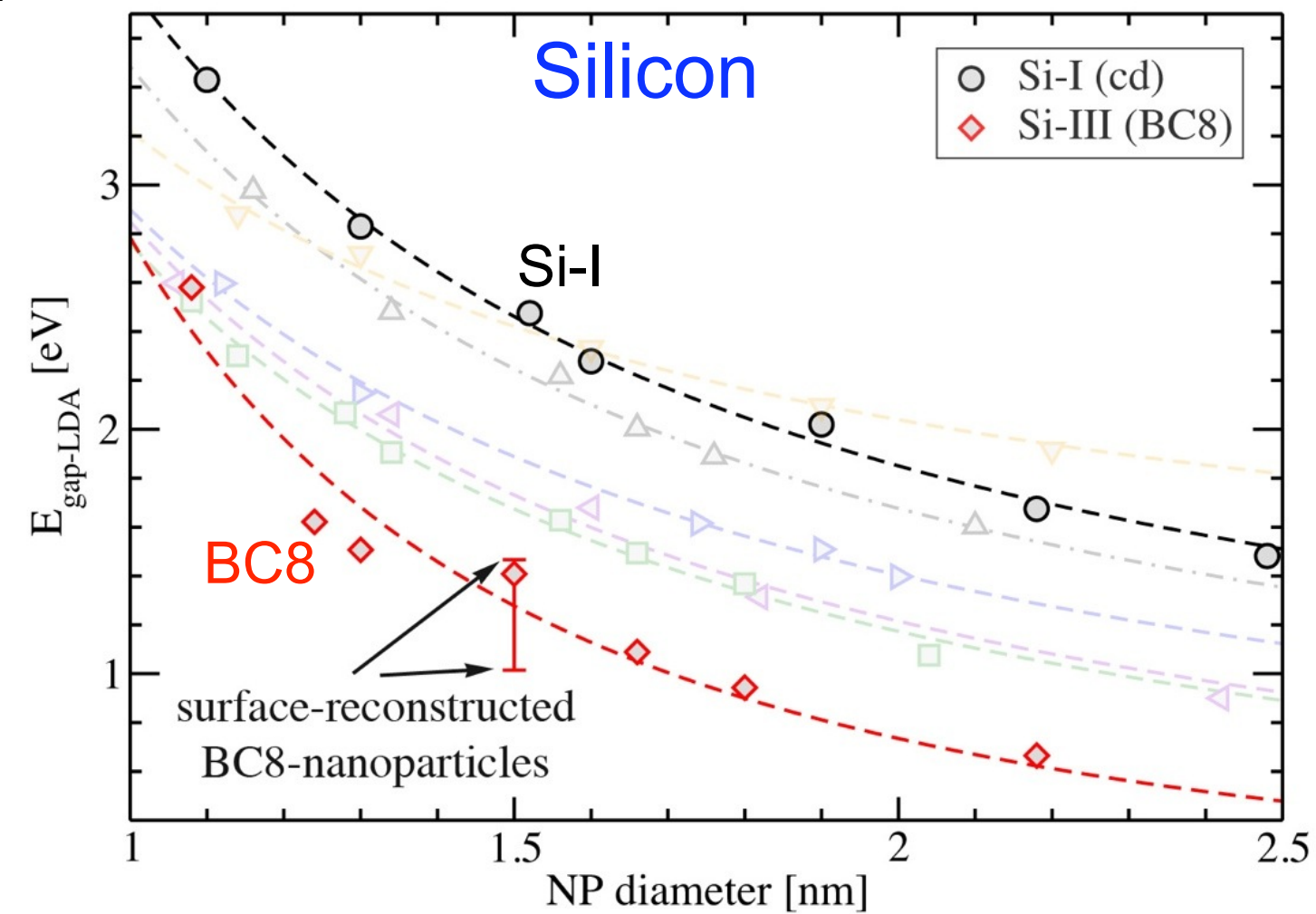
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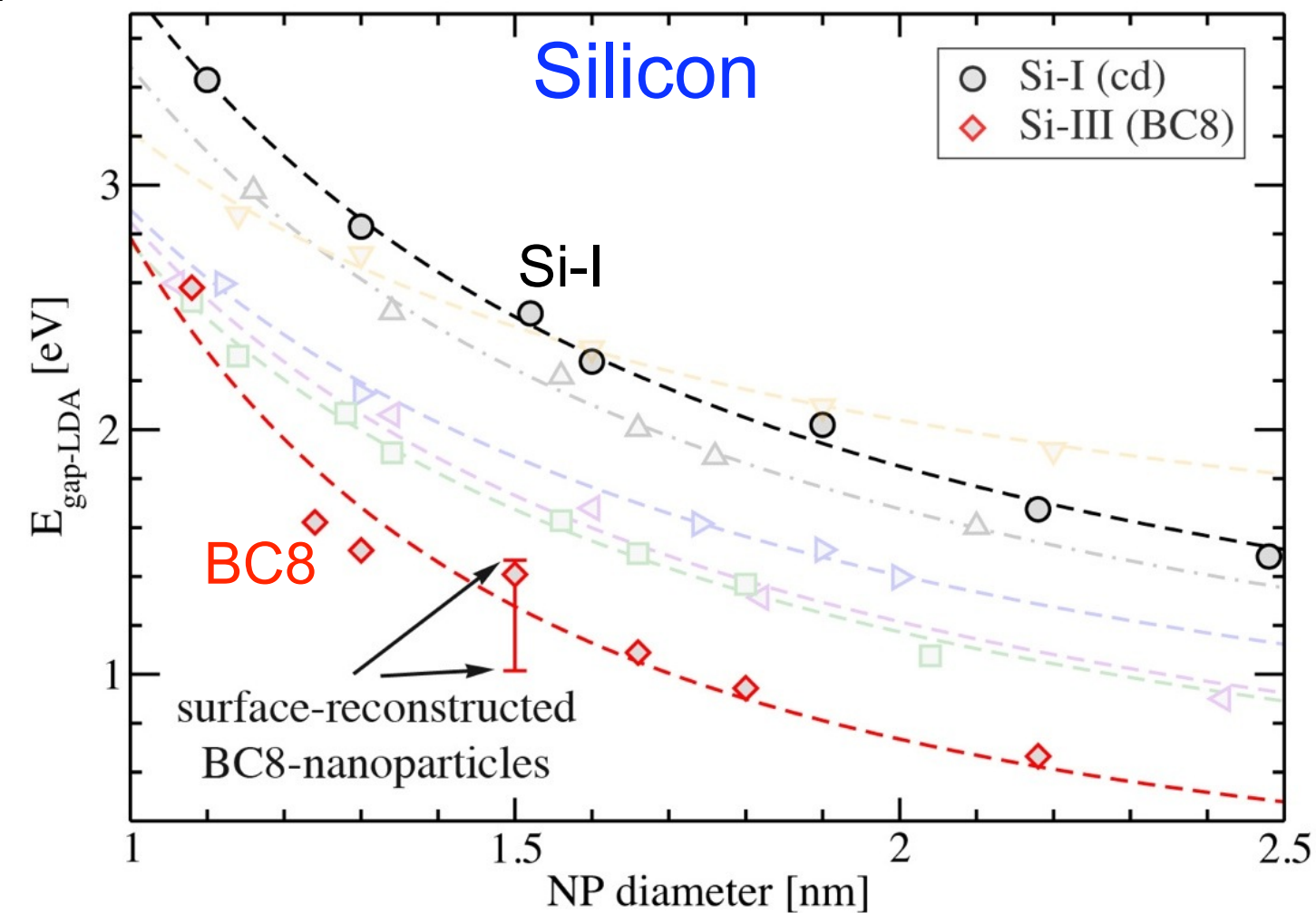
Allotropes of Si and Ge



Reduced gaps for BC8 and ST12 nanocrystals

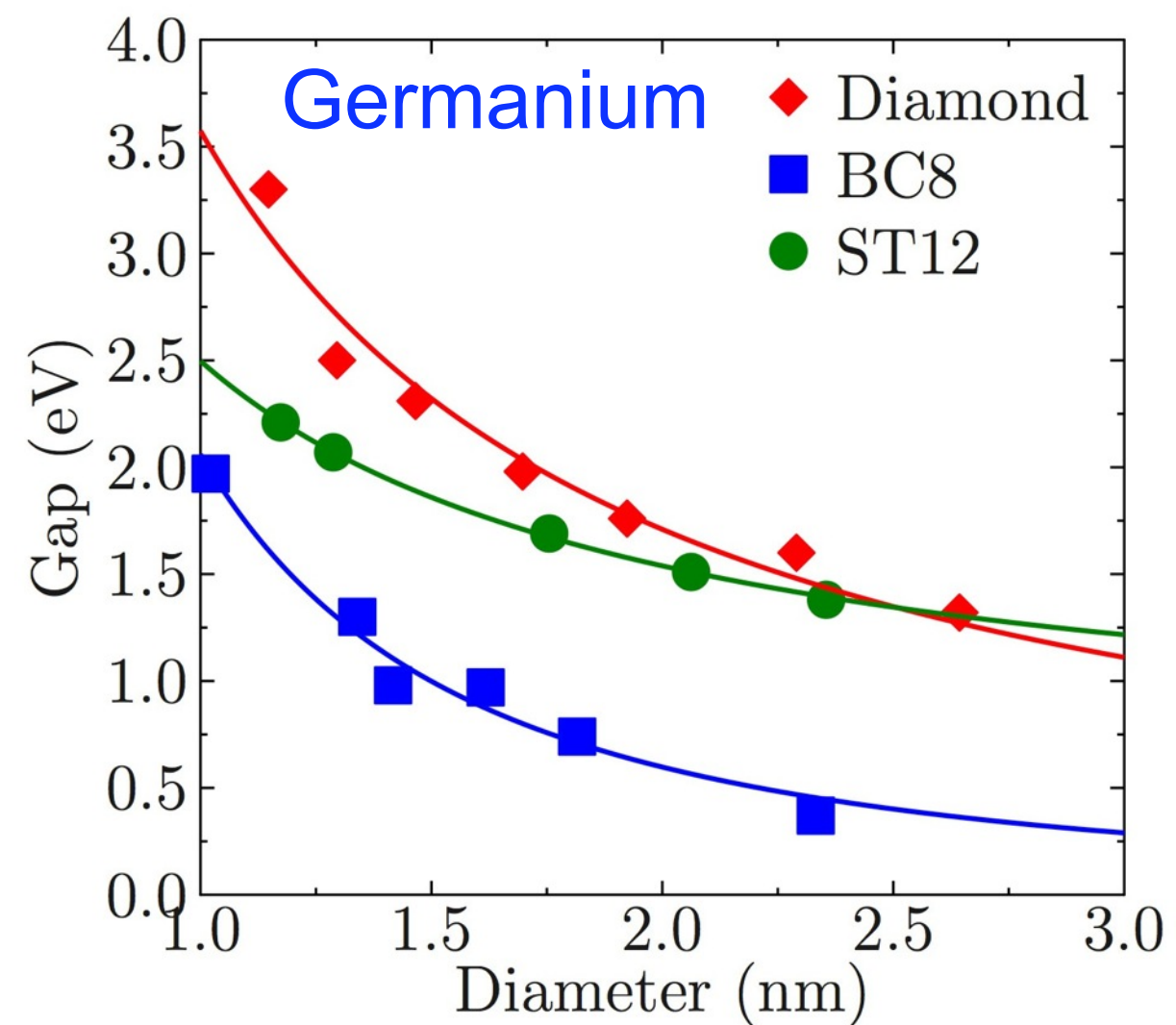
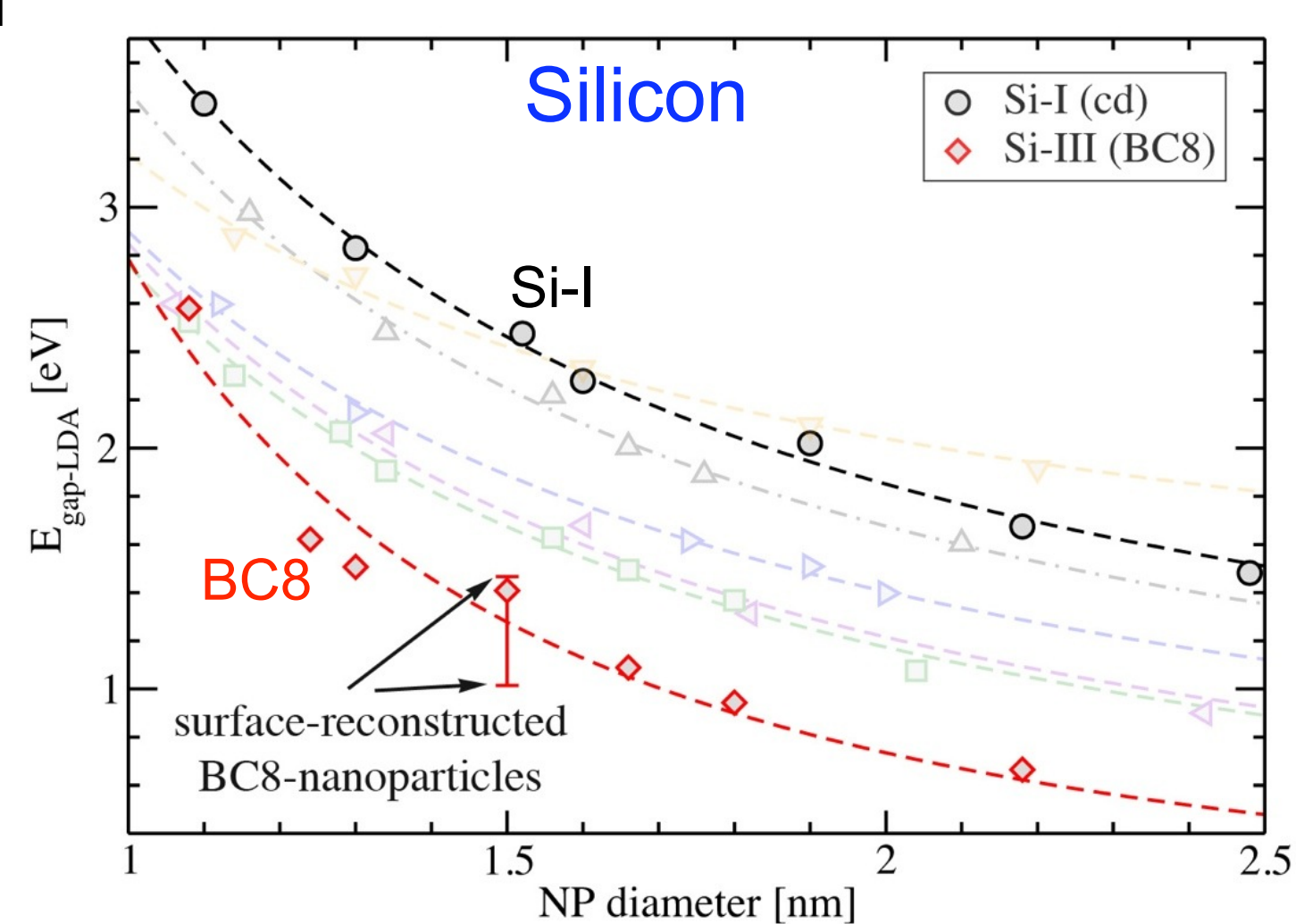


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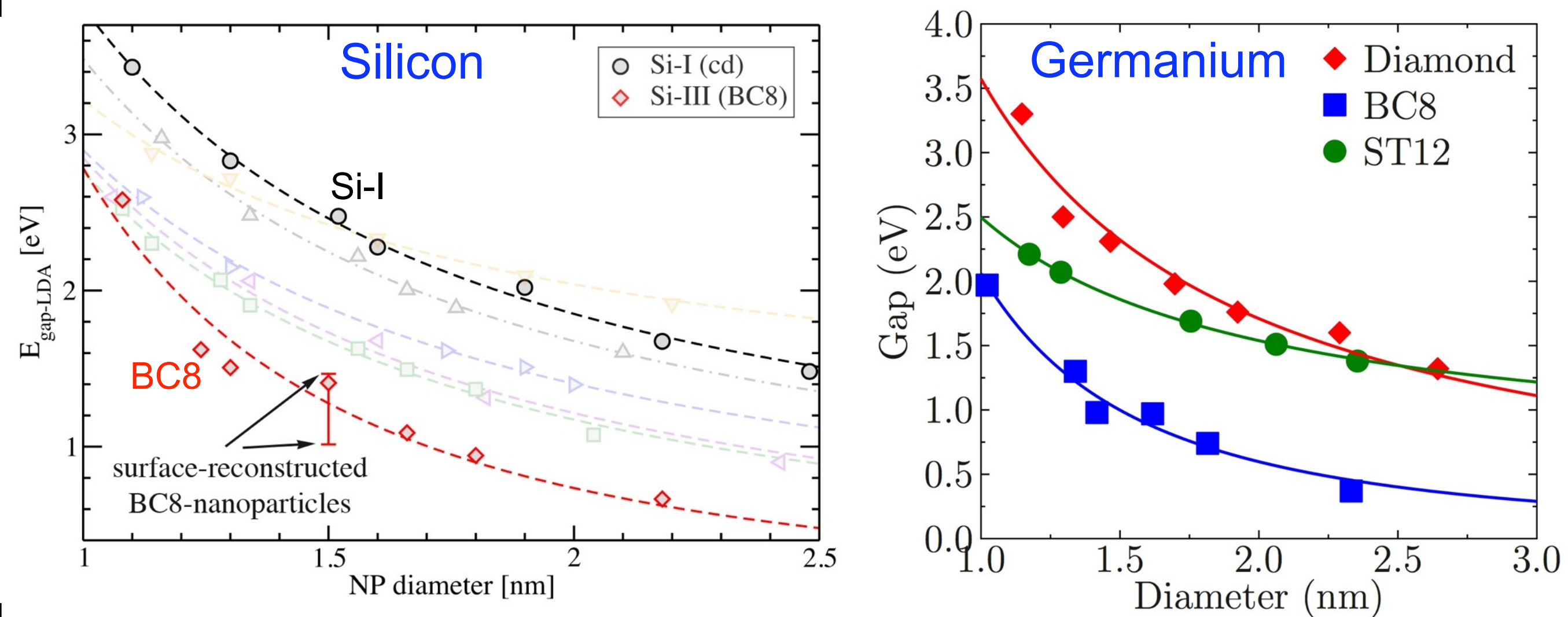
- BC8 is semimetal in the bulk, quantum confinement opens small gap in BC8 nanocrystals => significantly reduced gaps compared to Si-I

Reduced gaps for BC8 and ST12 nanocrystals



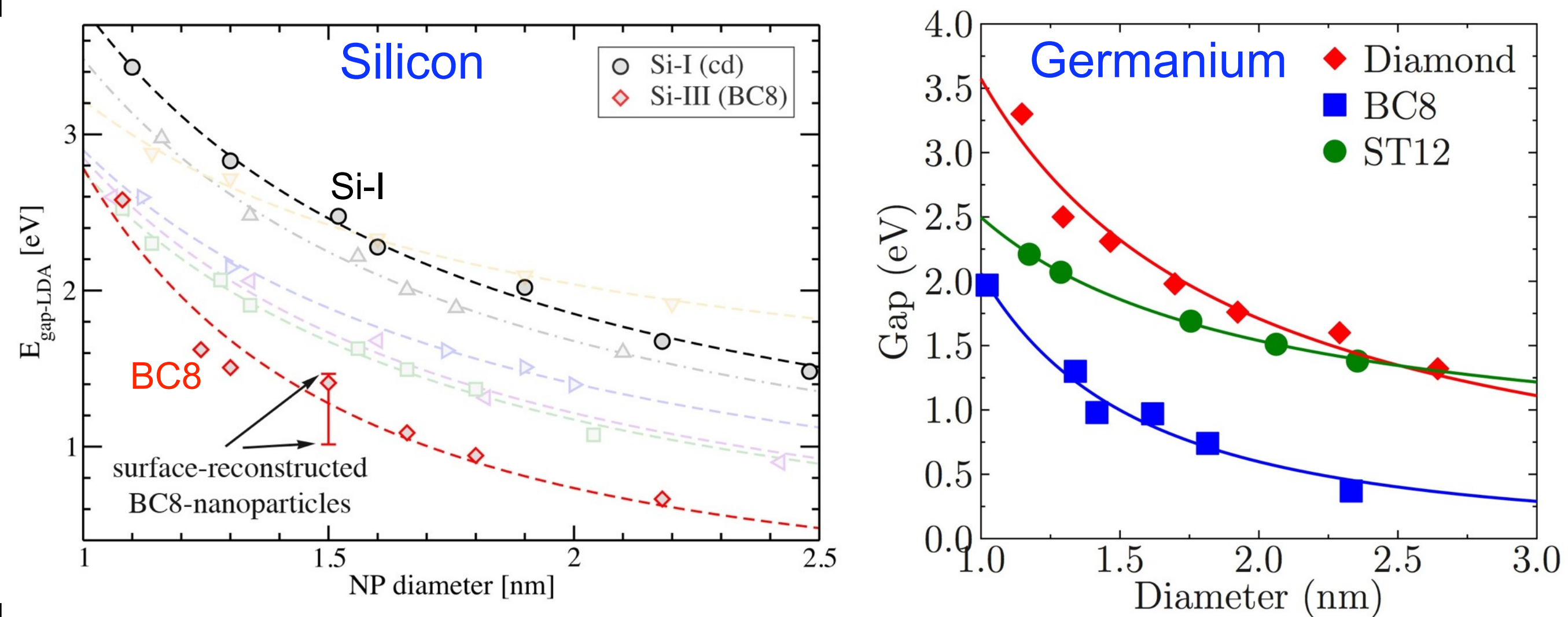
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- Ge ST12 features reduced gaps for $d < 2.5\text{nm}$, significantly increased electronic density of states at band edges
- Get realistic estimate of electronic gaps from quasiparticle calculations in GW approximation**

GW for large systems

- Calculation, storage & inversion of dielectric matrix ϵ is major computational bottleneck => **spectral representation of ϵ (RPA)**

$$\tilde{\epsilon} = \sum_{i=1}^N \tilde{\mathbf{v}}_i \lambda_i \tilde{\mathbf{v}}_i^H = \sum_{i=1}^N \tilde{\mathbf{v}}_i (\lambda_i - 1) \tilde{\mathbf{v}}_i^H + I$$

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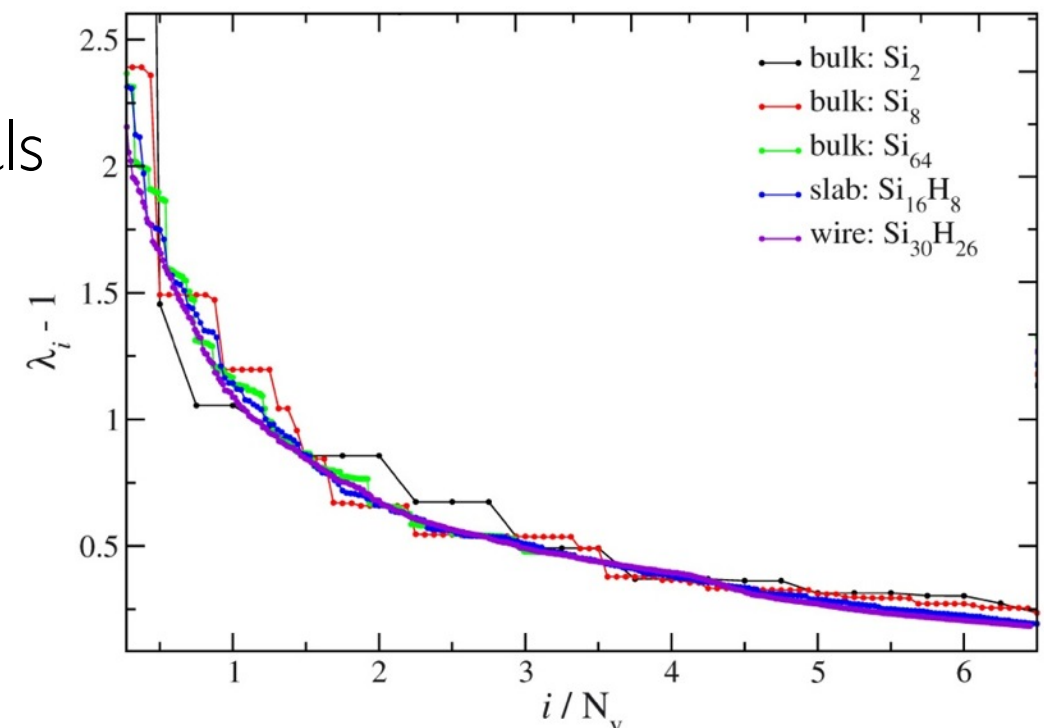
- Calculating eigenvectors/-values does NOT require explicit knowledge of the matrix itself; knowing the **action of ϵ on an arbitrary vector** is sufficient
- In linear response: $(\epsilon - I)\Delta V_{SCF} = -v_c \Delta n$
- **Charge density response Δn to perturbation of self-consistent field ΔV_{SCF} can be evaluated from density functional perturbation theory**

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- In RPA fast monotonous decay of dielectric eigenvalue spectrum
- Single parameter N_{eig} to control numerical accuracy
- No summation over empty states, no inversion**



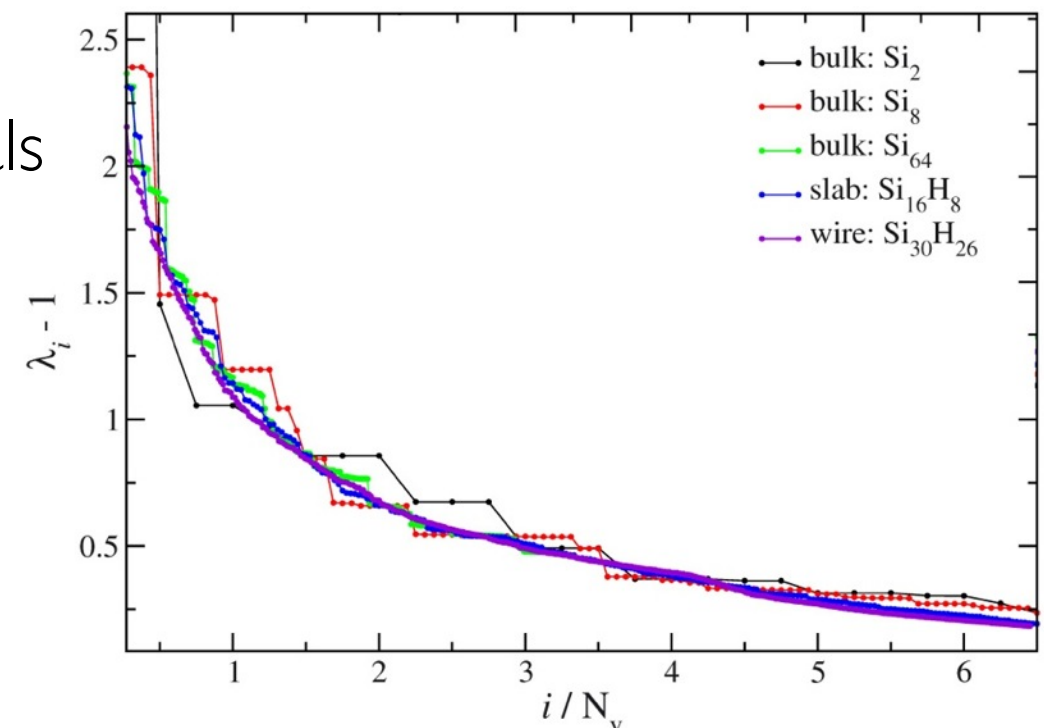
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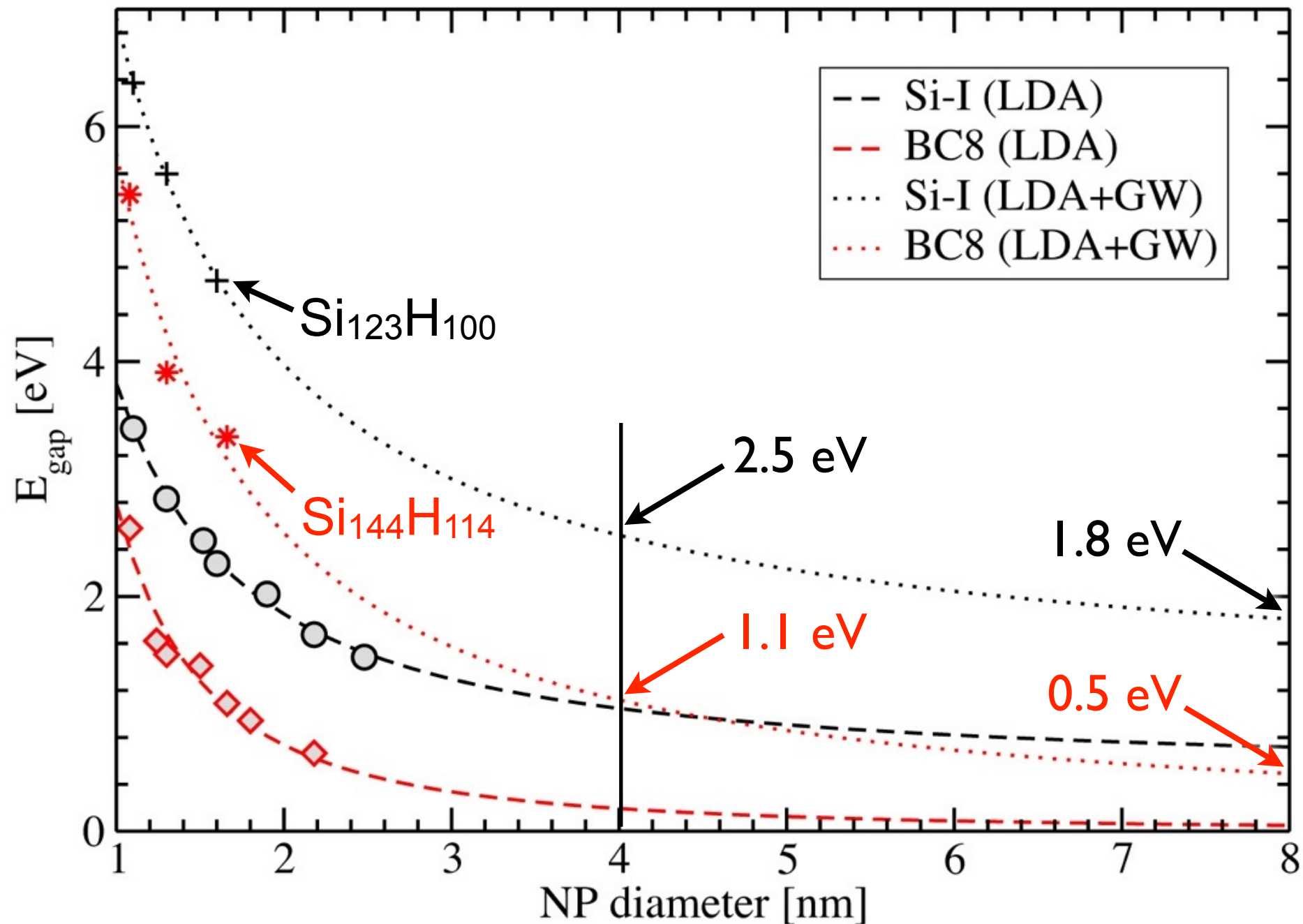
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D. Rocca, H.-V. Nguyen,
T.A. Pham (UCD)

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Optimum gap for MEG in 4-8nm BC8 NPs



- GW calculations up to $d=1.6\text{nm}$ ($\text{Si}_{144}\text{H}_{114}$) confirm trends observed in LDA
- Optimum gap** for PV cells employing MEG ($E_g = 0.5\text{-}1.0\text{ eV}$) [5] found for **BC8 NPs** within typical experimental **size range of $d = 4\text{-}8\text{ nm}$** (extrapolation of GW gaps)

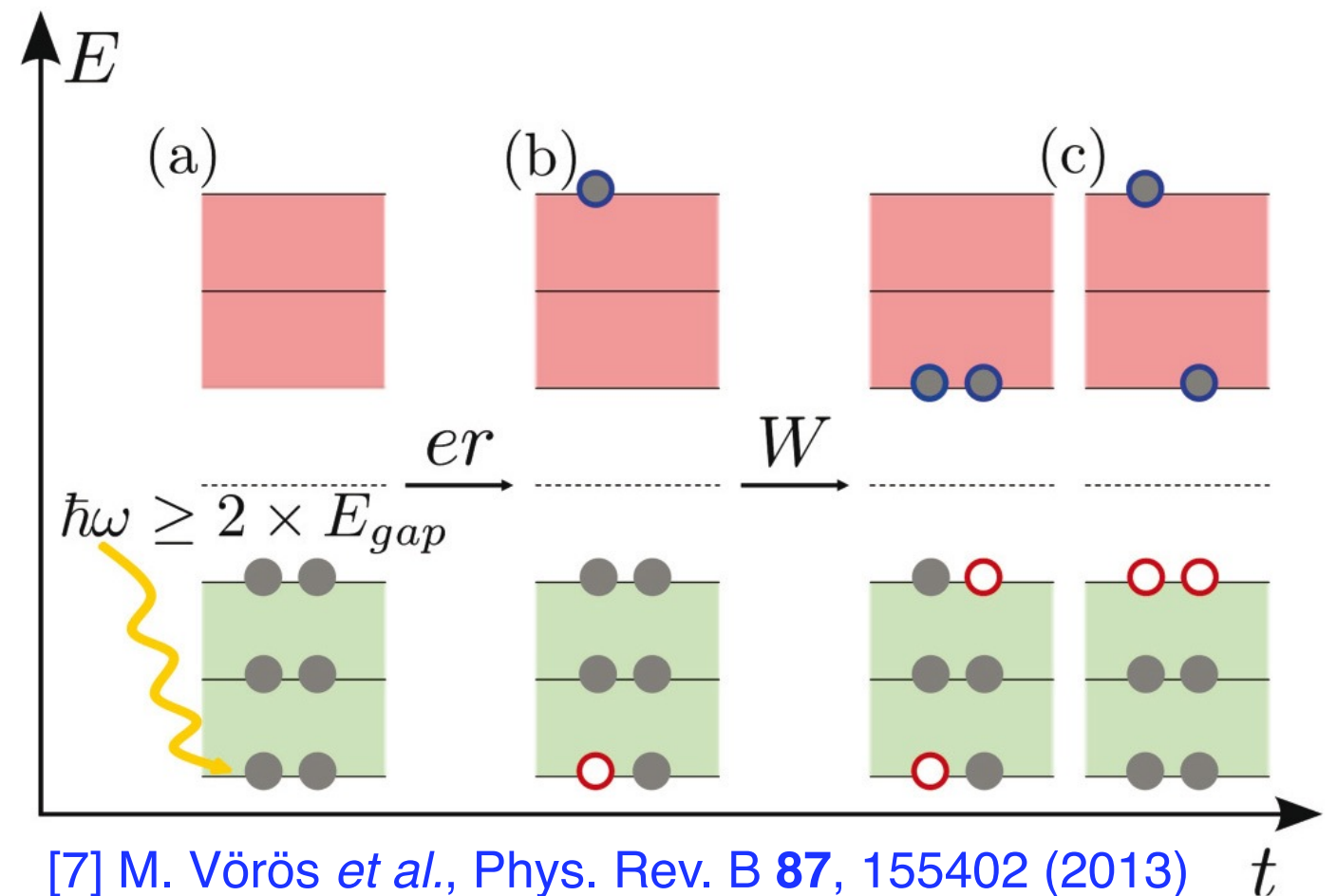
Calculating Multiple Exciton Generation (MEG) Rates

- Impact Ionization (II) is dominating contribution to MEG [6]
=> **approximate MEG rates with II rates [7]**

- Calculate II rates from Fermi's Golden Rule:

$$\Gamma_i^{II} = \frac{2\pi}{\hbar} \sum_f |\langle X_i | W | XX_f \rangle|^2 \delta(E_i - E_f)$$

- Approximate initial exciton (X_i) and final biexciton states (XX_f) as singly and doubly excited Slater determinants, built up from DFT orbitals
- Screened Coulomb interaction W calculated using same technique as during GW calculations



[6] A. Piryatinski *et al.*, J. Chem. Phys. **133**, 084508 (2010)
K. Velizhanin *et al.*, Phys. Rev. Lett. **106**, 207401 (2011)

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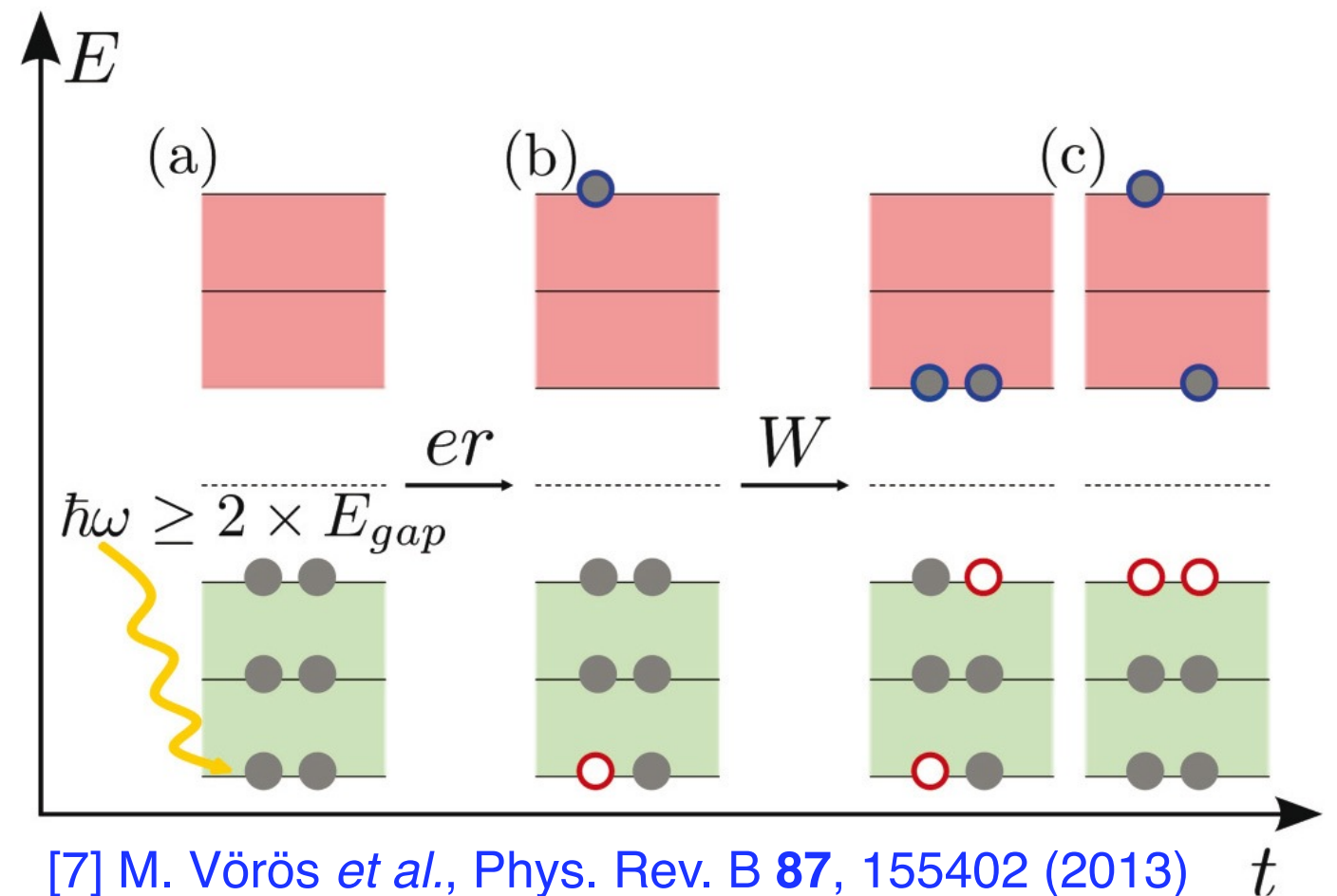
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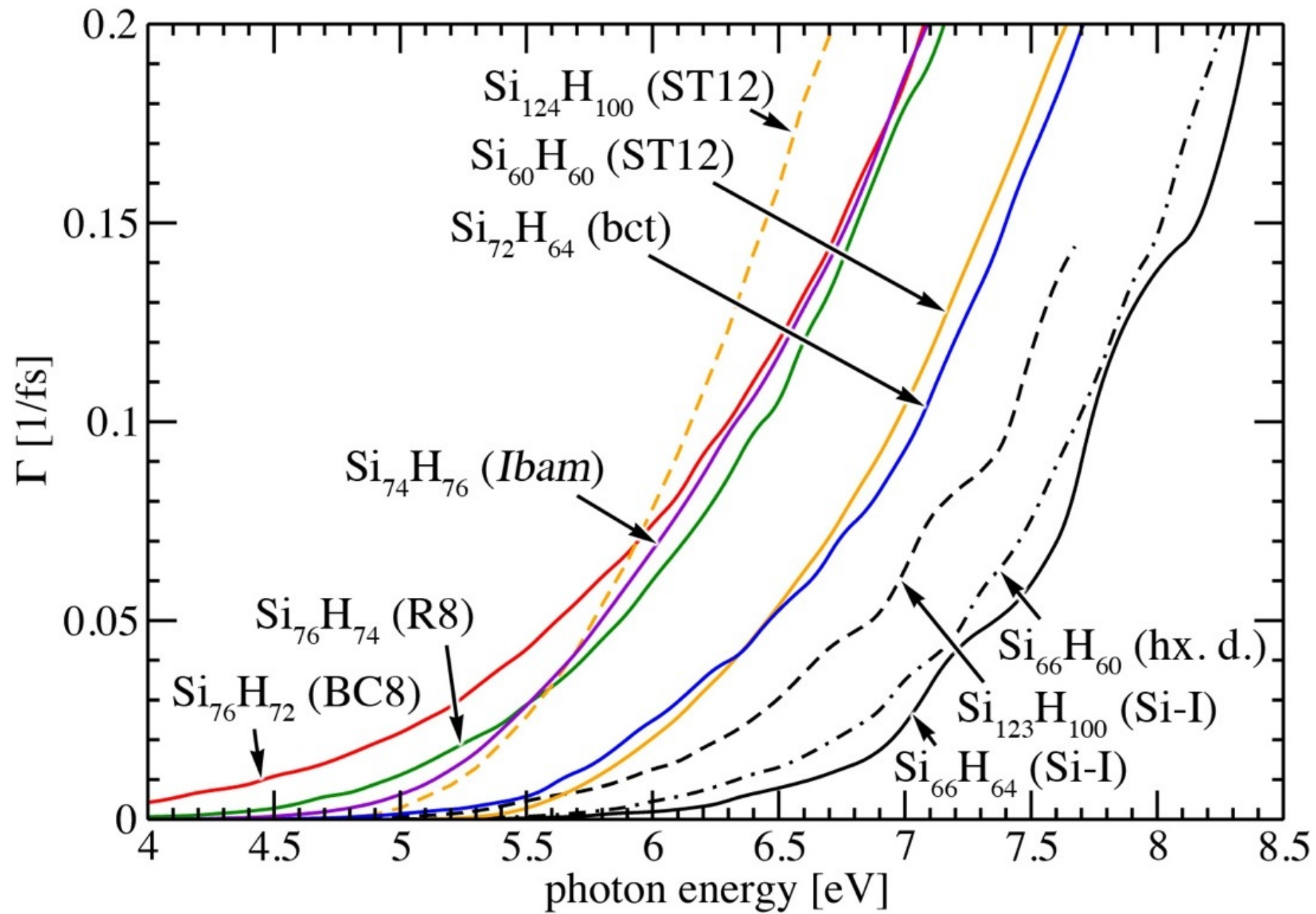
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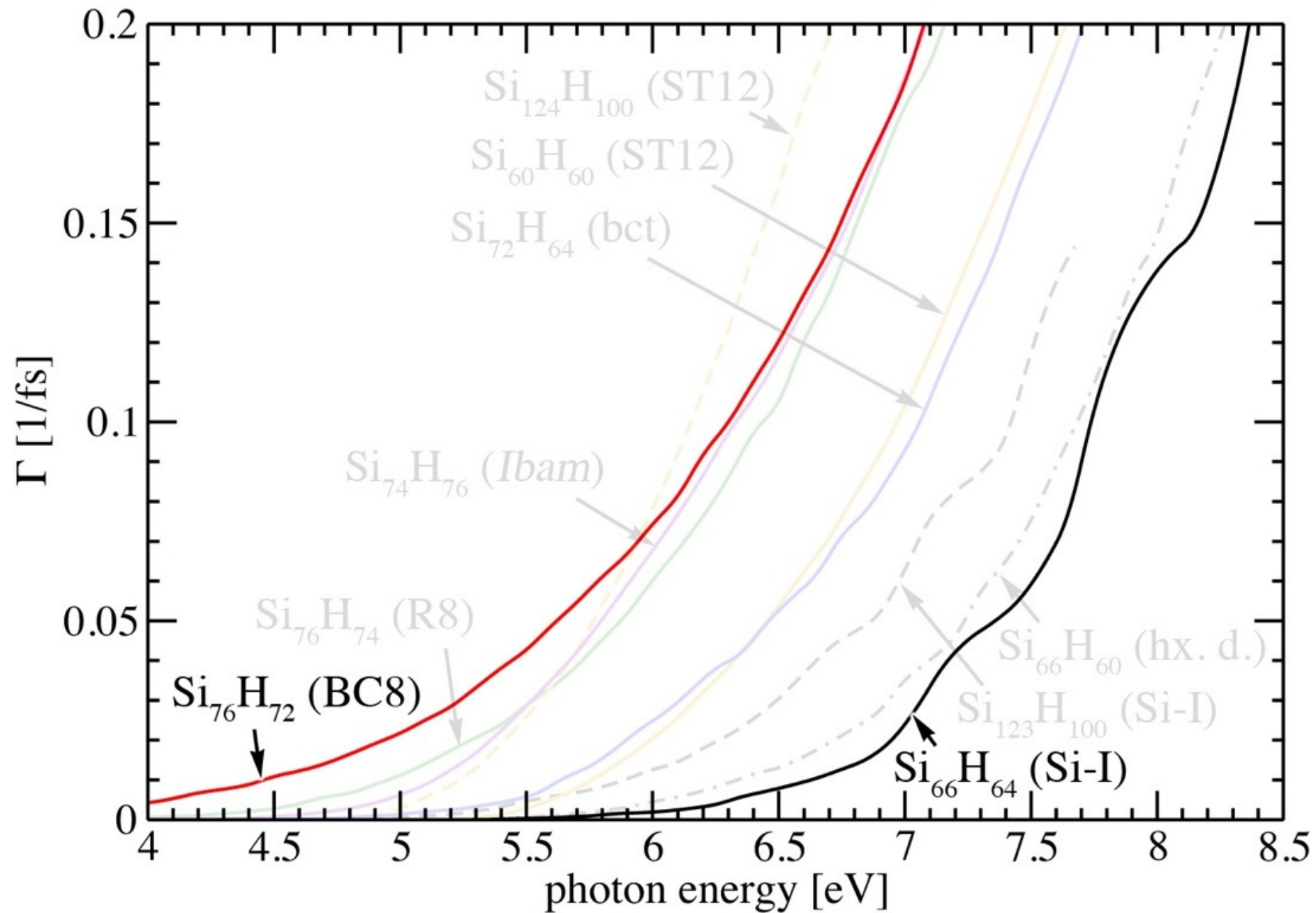
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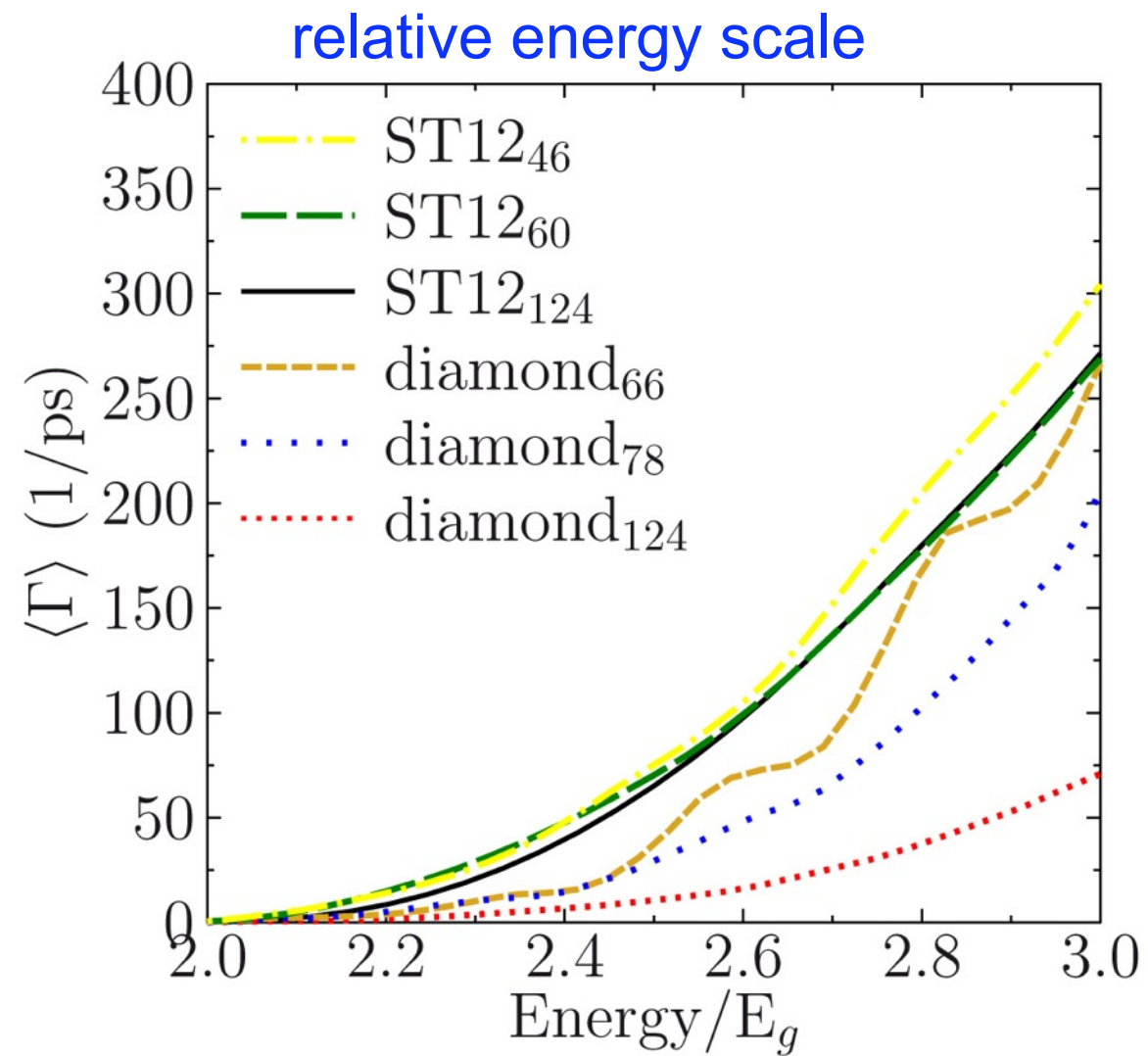


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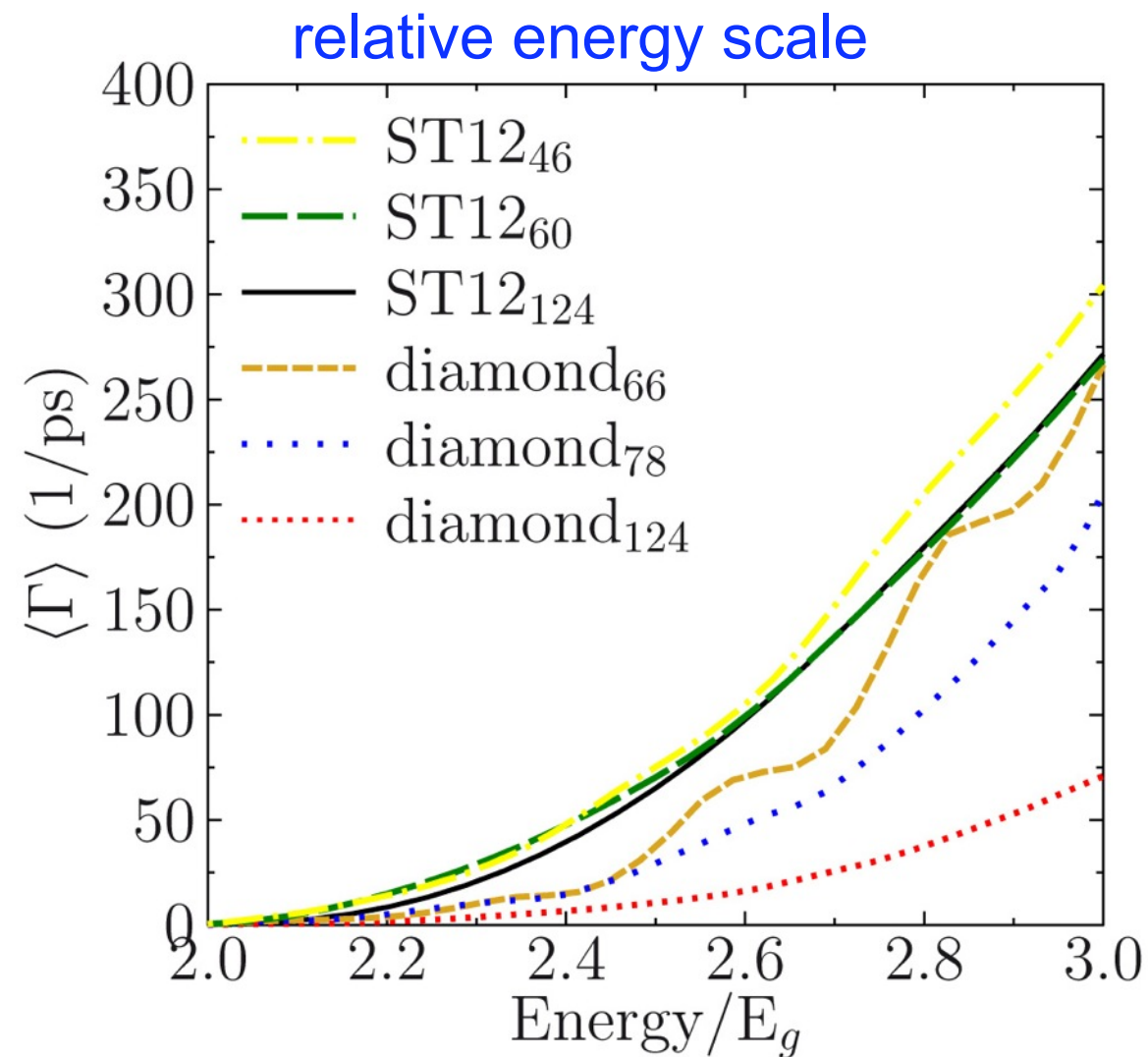


BC8 NPs feature lower activation threshold on absolute energy scale & order of magnitude higher impact ionization rate at same energies and same NP size!

Multiple Exciton Generation in Ge allotrope nanocrystals



Multiple Exciton Generation in Ge allotrope nanocrystals

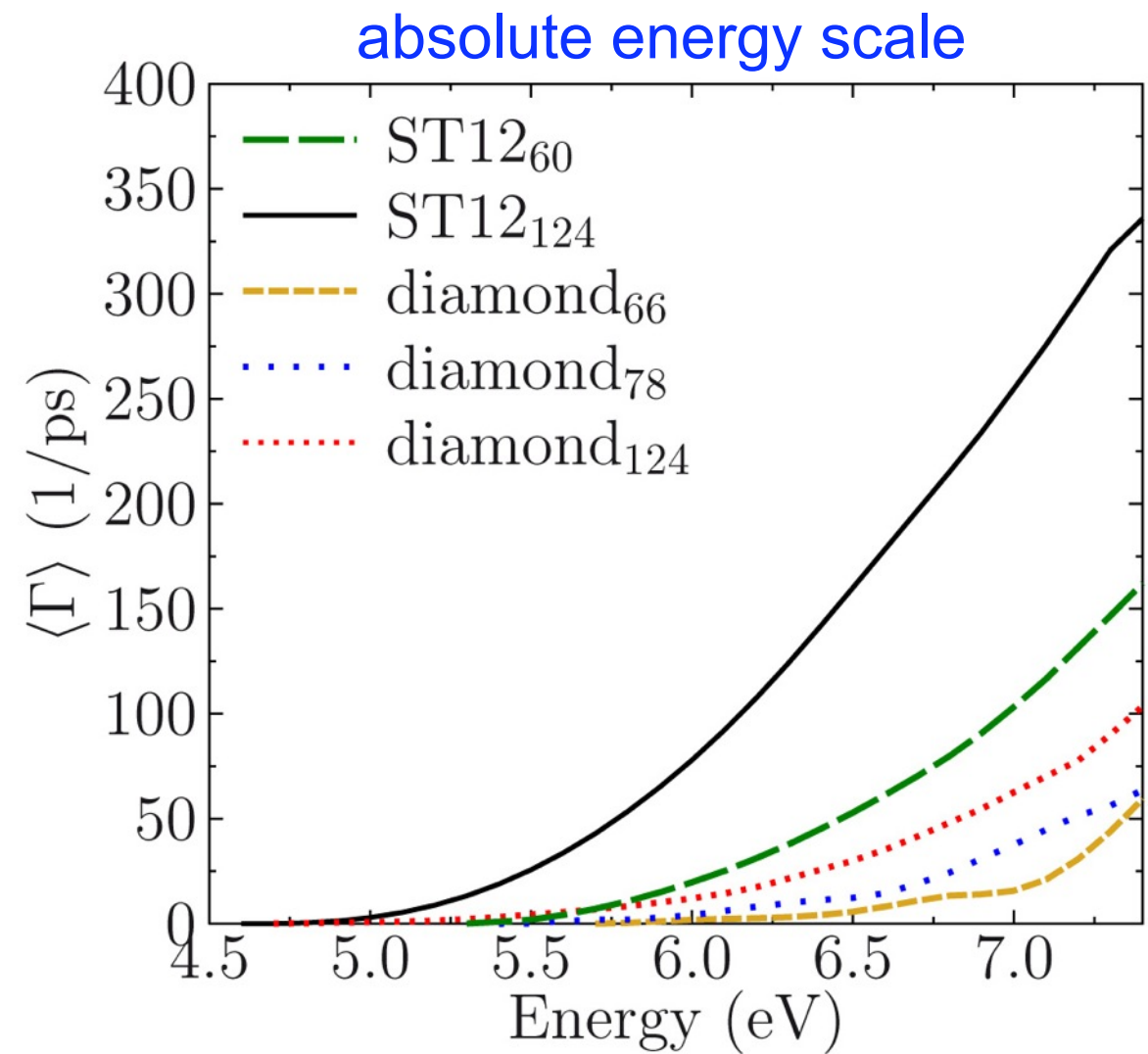
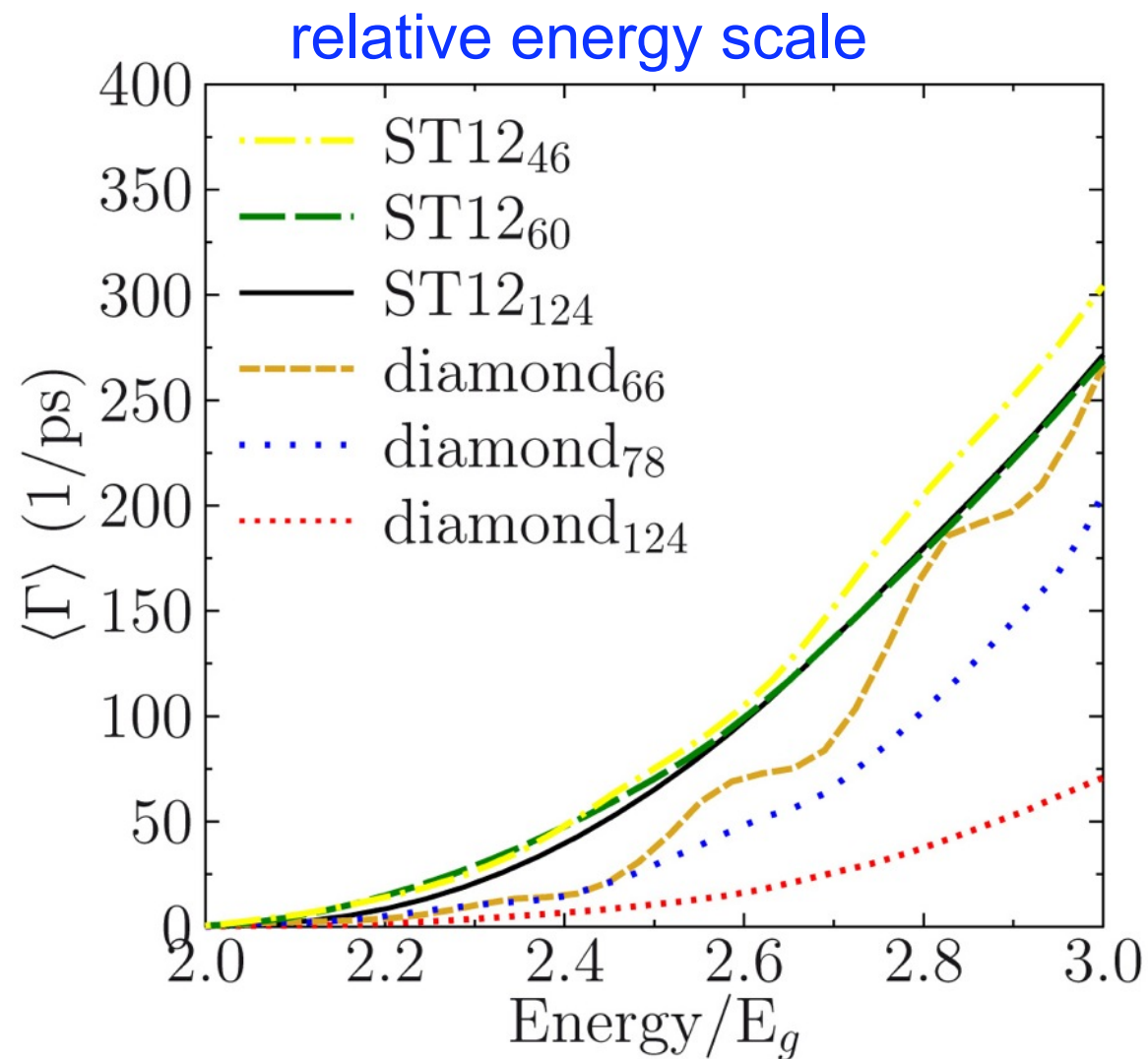


ST12 II rates size-independent



Increasing EDOS at band edges counterbalances loss of confinement

Multiple Exciton Generation in Ge allotrope nanocrystals

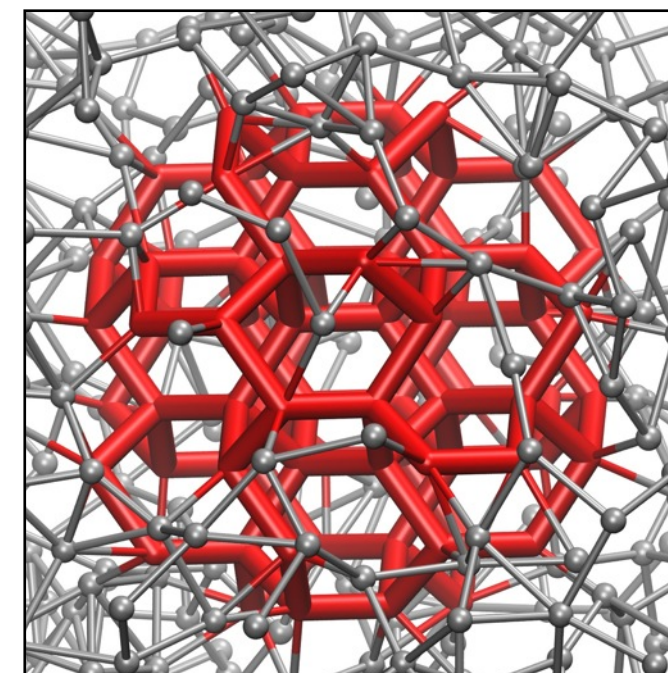


ST12 II rates size-independent

- Increasing EDOS at band edges counterbalances loss of confinement
- Simultaneously lower electronic gaps and higher relative II efficiency translate to significantly improved MEG on absolute energy scale

Summary

- (Metastable) high pressure phases of elemental semiconductors allow for gap engineering of nanoparticles, while retaining efficient MEG (Si-BC8 and Ge-ST12)
- Nanoparticles with high pressure core structures can attain optimum gap range for MEG-based solar energy conversion
- High pressure nanoparticles can be formed via the high pressure route or directly at ambient pressure in solution by chemical bottom-up synthesis from a precursor



Talk on Si NPs embedded in charge transport matrix today at 12:30pm, Tre Ma

PRL **110**, 046804 (2013)

PHYSICAL REVIEW LETTERS

week ending
25 JANUARY 2013

High-Pressure Core Structures of Si Nanoparticles for Solar Energy Conversion

S. Wippermann,^{1,2} M. Vörös,³ D. Rocca,¹ A. Gali,^{3,4} G. Zimanyi,² and G. Galli^{1,2}

¹Department of Chemistry, University of California, Davis, California 95616, USA

²Department of Physics, University of California, Davis, California 95616, USA

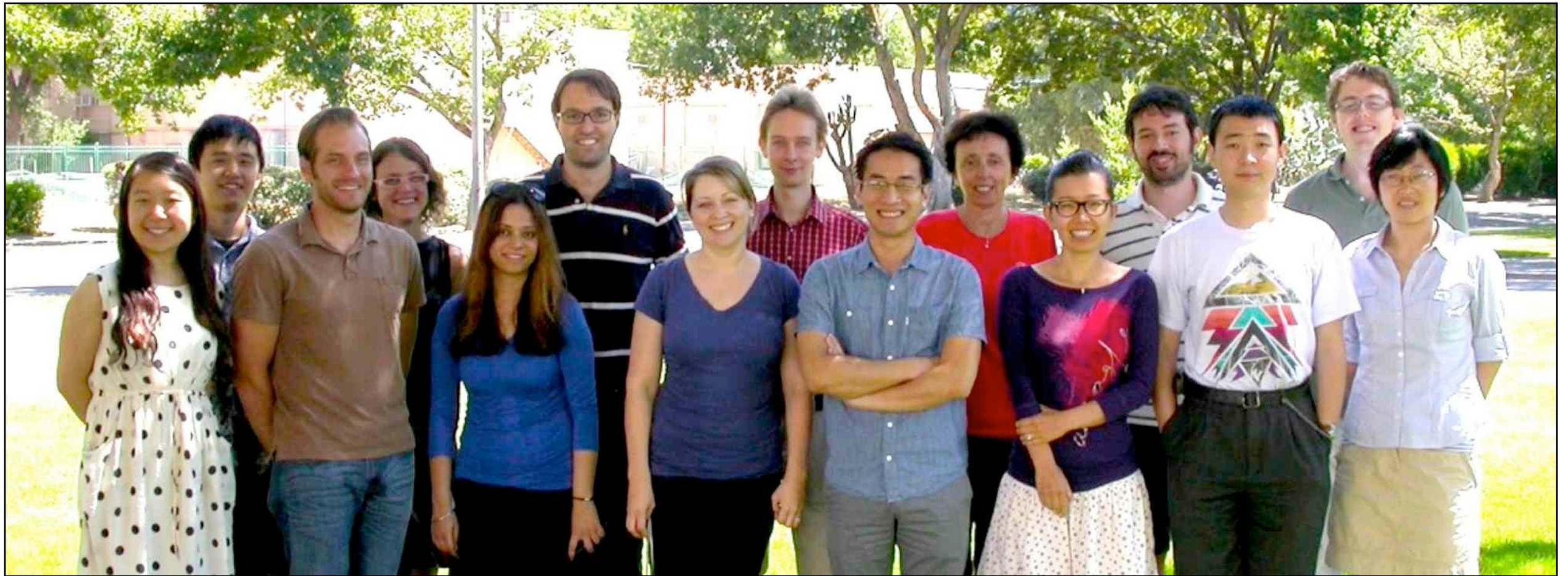
³Department of Atomic Physics, Budapest University of Technology and Economics, Budafoki út 8, H-1111 Budapest, Hungary

⁴Institute for Solid State Physics and Optics, Wigner Research Center for Physics, Hungarian Academy of Sciences,
P.O. Box 49, H-1525 Budapest, Hungary

(Received 1 August 2012; published 24 January 2013)

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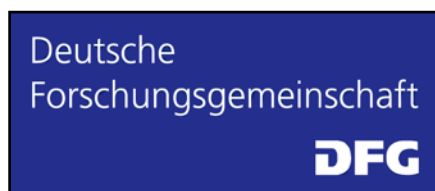
Giulia Galli and her group at UChicago, especially Marton Vörös and Tuan Anh Pham



Francois Gygi (UC Davis), Gergely Zimanyi (UC Davis),
Adam Gali (Budapest Univ.), Dario Rocca (Univ. Lorraine)



NanoMatFutur I3N12972



Wi3879/I-I



NISE-project 35687



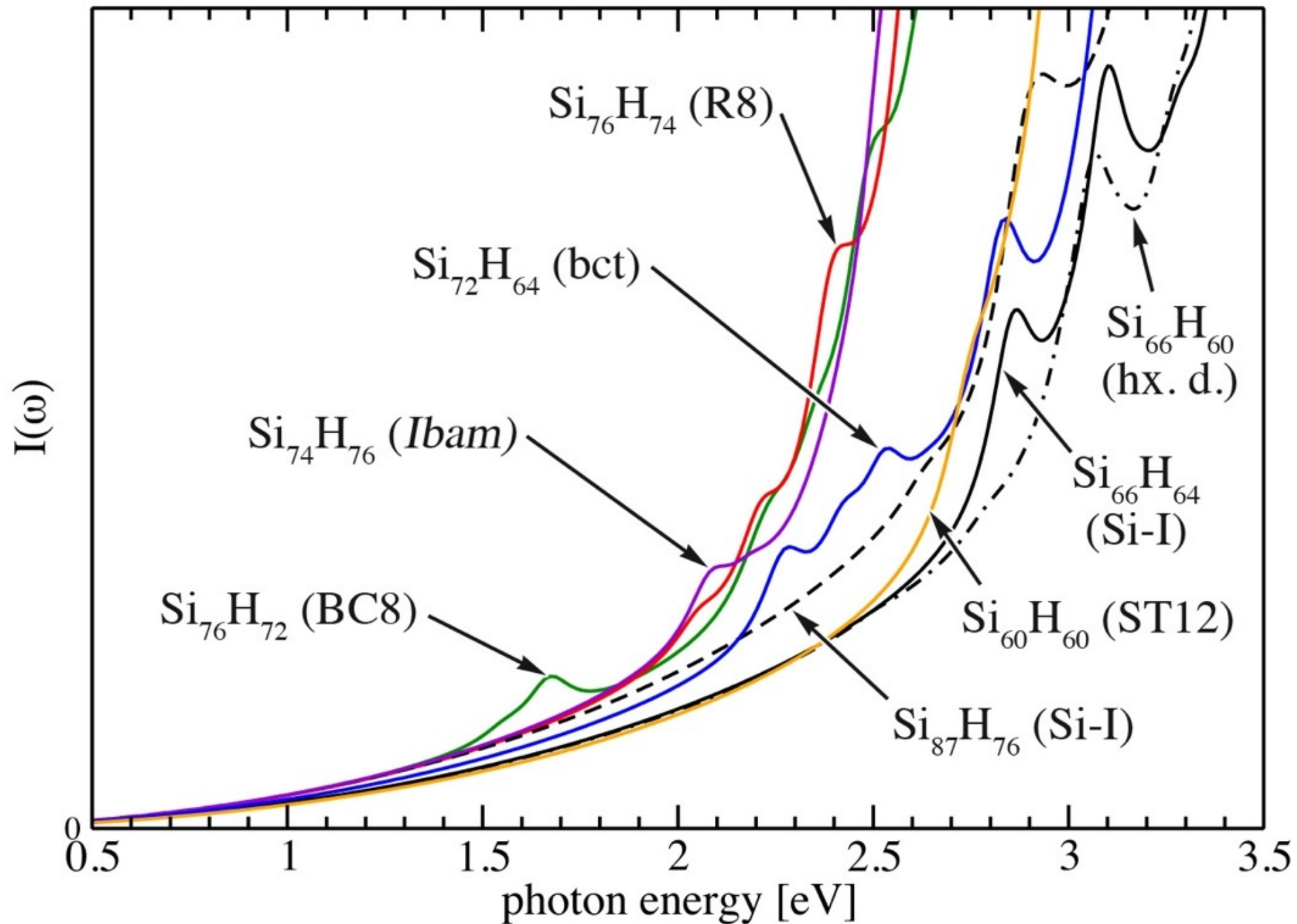
NSF/Solar DMR-1035468



Max-Planck-Institute for Iron
Research, Düsseldorf

Additional Slides

Optical properties (TD-DFT RPA)



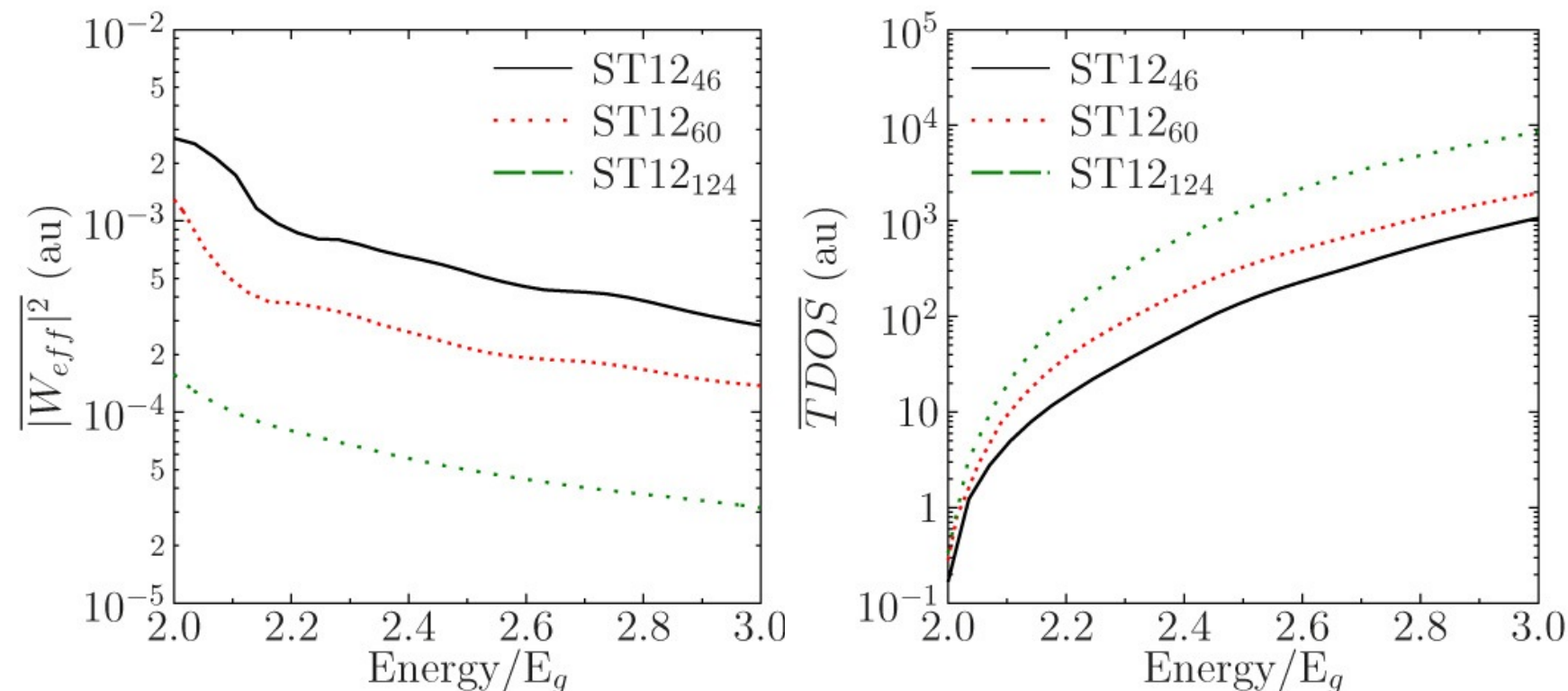
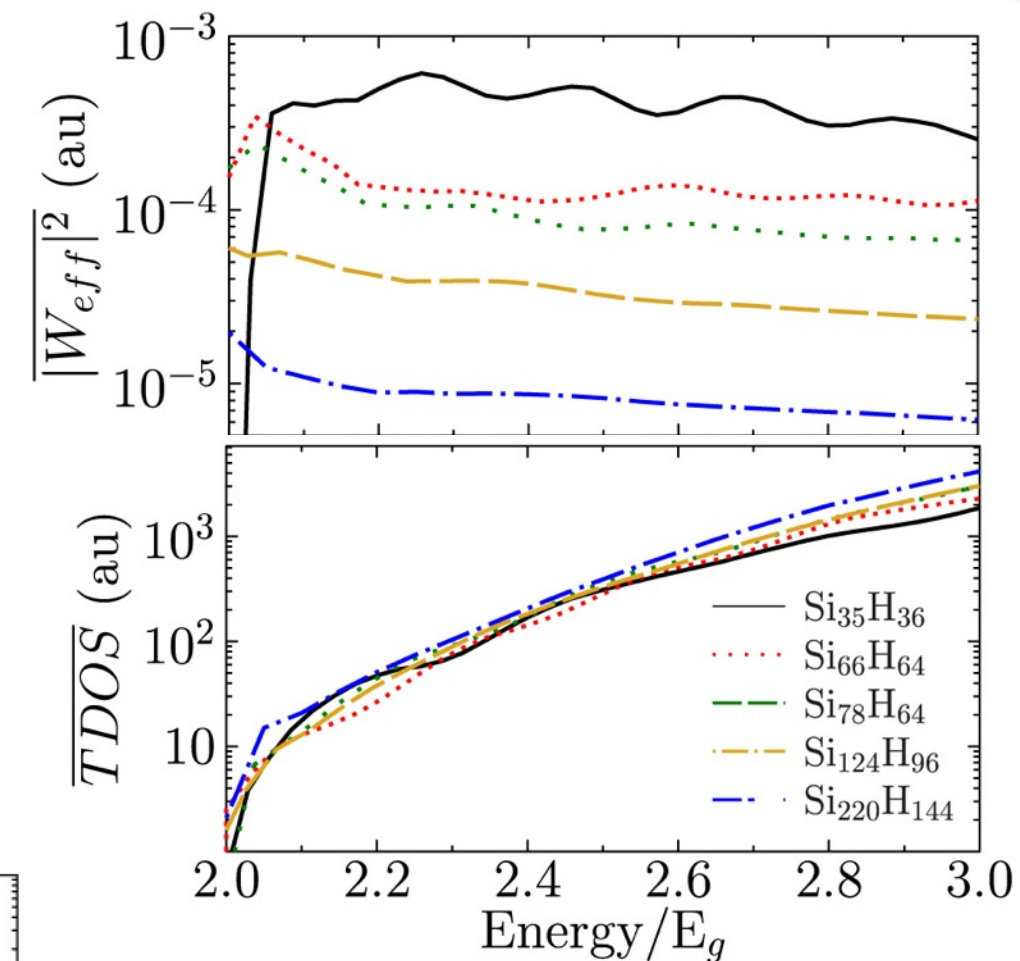
- **red-shifted optical absorption** onset for high density phases (BC8, R8, *Ibam*)
- less pronounced for ST12 and low density phases (bct, hd)


cubic diamond vs. ST12

$$\begin{aligned}\Gamma_i^{II} &= \frac{2\pi}{\hbar} \sum_f |\langle X_i | W | X X_f \rangle|^2 \delta(E_i - E_f) \\ &= \frac{2\pi}{\hbar} |W_{eff}^i|^2 \cdot TDOS_i\end{aligned}$$

 **cubic diamond:** NP size increase reduces Coulomb interaction W_{eff} , trion DOS almost constant

=> impact ionization rate drops



 **ST12:** W_{eff} reduced as for cubic diamond, **but TDOS increases**

=> impact ionization rate remains almost constant with increasing size



Density functional theory (DFT):

 Hohenberg-Kohn theorem:

$$E_{XC}[n] \approx E_{XC}^{LDA}[n] = \int n(\mathbf{r}) \epsilon_{XC}^{hom}(n(\mathbf{r})) d\mathbf{r}$$

$$E_e[n] = T_0[n] + \frac{1}{2} \int \frac{n(\mathbf{r})n(\mathbf{r}')}{|\mathbf{r} - \mathbf{r}'|} d\mathbf{r} d\mathbf{r}' + \int n(\mathbf{r}) V(\mathbf{r}) d\mathbf{r} + E_{XC}[n]$$

[Walter Kohn, Nobel Prize for chemistry in 1998]

  **atomic positions**

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atomic positions

Kohn-Sham **self-consistent**
electron structure

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Starting point: initial geometry



external potential

interatomic forces

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 spectroscopic and
transport properties

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- ⦿ DFT disregards **screened e^-e^- interaction** and **e^-h interaction for excited states** \Rightarrow band gap underestimation, wrong distribution of spectral weights
- ⦿ Perturbative approaches for including screening (GW) and e^-h interaction (Bethe-Salpeter), starting from Quantum Liouville equation

$$i \frac{d\hat{\rho}(t)}{dt} = [\hat{\mathcal{H}}(t), \hat{\rho}(t)] \quad \rho(\mathbf{r}, \mathbf{r}', t) = \sum_v \phi_v^*(\mathbf{r}, t) \phi_v(\mathbf{r}', t)$$

single particle occ. orbitals

$$\int \hat{\mathcal{H}}(\mathbf{r}, \mathbf{r}', t) \phi(\mathbf{r}', t) d\mathbf{r}' = \left(-\frac{1}{2} \nabla^2 + v_H(\mathbf{r}, t) + v_{ext}(\mathbf{r}, t) \right) \phi(\mathbf{r}, t)$$
$$+ \int \Sigma(\mathbf{r}, \mathbf{r}', t) \phi(\mathbf{r}', t) d\mathbf{r}'$$

time-dep. perturbation,
i. e. electromagn. field

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time-dep. perturbation, i. e. electromagn. field

self-energy

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statically screened
Bethe-Salpeter
equation (BSE)

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statically screened
Bethe-Salpeter
equation (BSE)

screened Coulomb
interaction

Excited states (GW/BSE)

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- Solution: **spectral representation** of RPA dielectric matrix; obtain matrix from **directly calculating eigenvectors and eigenvalues**

$$\tilde{\epsilon} = \sum_{i=1}^N \tilde{\mathbf{v}}_i \lambda_i \tilde{\mathbf{v}}_i^H = \sum_{i=1}^N \tilde{\mathbf{v}}_i (\lambda_i - 1) \tilde{\mathbf{v}}_i^H + I$$

=> no summation over empty states, no inversion, storage of eigenvector/-value pairs only!

How to calculate the screening

Obtaining the eigenvectors/-values does NOT require explicit knowledge of the matrix; knowledge of the action of the matrix on an arbitrary vector is sufficient!

in linear response: $(\epsilon - I)\Delta V_{SCF} = -v_c \Delta n$

charge density response Δn
to perturbation of self-consist.
field ΔV_{SCF} can be evaluated
from density functional
perturbation theory

orthogonal iteration procedure to
obtain eigenvectors/-values, using
 ΔV_{SCF} as trial potentials

in RPA fast monotonous decay of
dielectric eigenvalue spectrum

single parameter N_{eig} to control
numerical accuracy

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