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A preliminary model for estimating the
first wall lifetime of a fusion reactor

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The estimation of the first wall lifetime is a necessary basis for predicting the overall lifetime of a fusion power plant. In order to do this, an analytical model was prepared and programmed for a computer which calculates the temperature and stress load of the first wall from the principal design parameters and gives the required relevant material properties. However, the analytical model requires information about the material properties, which is not complete so that the answers obtained from the model are very preliminary. This situation is underlined by the fact, that only calculations performed for the CTFR reactor module cell. The results obtained for vanadium and vanadium alloys show a strong dependence of the lifetime on the irradiation creep and the ductility of these materials. Completion of this model is envisaged as soon as the missing information becomes available.

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Abstract

The estimation of the first wall lifetime is a necessary basis for predicting the availability of a fusion power plant. In order to do this, an analytical model was prepared and programmed for the computer which calculates the temperature and stress load of the first wall from the principal design parameters and quotes them against the relevant material properties. Neither the analytical model nor the information about the material performance is yet complete so that the answers obtained from the program are very preliminary. This situation is underlined by the results of sample calculations performed for the CTRD blanket module cell. The results obtained for vanadium and vanadium alloys show a strong dependence of the lifetime on the irradiation creep and the ductility of these materials. Completion of this model is envisaged as soon as the missing information becomes available.

Contents:

	Page
1. Introduction	1
2. Principal design features of the module cell	1
3. Input quantities to the analytical model	2
4. Analysis of the temperature and stress load of the first wall	4
4.1 Determination of basic quantities	4
4.2 Heat transfer from the wall to the coolant	6
4.3 Temperatures inside the first wall	8
4.4 Stresses inside the first wall	8
5. Material properties and wall lifetime	10
5.1 Time rupture strength	10
5.2 Thermal creep	11
5.3 Irradiation creep	12
5.4 Open problems	12
6. Computer program	13
6.1 Description of the single programs	13
6.2 Description of the input	17
7. Sample calculations	19
7.1 Input specifications	19
7.2 Results for pure vanadium	20
7.3 Influence of titanium as alloying element	23
8. Conclusions	24
9. Acknowledgements	25
10. References	25

1. Introduction

The information about the first wall lifetime was recognized as important for predicting the frequency of shut-down periods for routine replacement. Together with the length of these periods which results from considerations of the feasibility and sequence of the replacement operations envisaged, this information should subsequently permit conclusions concerning the availability of a fusion power plant.

The purpose of this study was to provide the means for estimating the useful life to be expected for the first wall by means of an analytical model. This model should allow

- evaluation of the temperature, stress and radiation loads from the principal design parameters of a reactor
- quotation of these loads against the material performance and hence derivation of an estimated lifetime.

This model was developed in the course of the European Collaborative Tokamak Reactor Design study (CTRD). Its first part is therefore strongly related to this special design. Parametric studies by means of this model should already aid the design at an early stage of its development and provide a background against which the final decisions could be weighed. To meet these requirements, the analytical model as hitherto elaborated was programmed for the IBM 360/91 computer at IPP using basic FORTRAN-IV language.

2. Principal design features of the module cell

The present study is concerned with a modular cell (see Fig. 1) as proposed by J.T.D. Mitchell and J.A. Booth [1] and J.R. Stanbridge et. al. [2]. As a first guess it is assumed that the cross-section of this cell will be circular and that the material composition inside this cell is in close agreement with the blanket proposed by J. Darvas [3]. This implies the use of helium as a coolant.

To describe the geometrical environment of a single cell, the analytical model allows of two options: the arrangement in a square (see fig. 2a) or hexagonal (see fig. 2b) matrix.

In addition, the specification of a pitch ratio p is possible, this being the ratio of the distance d between the axes of two adjacent cells and their outer diameter D_C :

$$p = \frac{d}{D_C} \tag{1}$$

These options permit the definition of a package density β_p , which is the ratio of the cell cross-section A_C and that of a matrix element A_M . A simple calculation leads to the expression

$$\beta_p = \frac{\pi}{4} \cdot \frac{1}{\alpha p^2} \tag{2}$$

with $\alpha = 1.0$ for the square matrix and $\alpha = \frac{\sqrt{3}}{2} = 0.866$ for the hexagonal matrix.

3. Input quantities to the analytical model

Before beginning with the analytical treatment of this problem it first had to be decided which quantities shall be treated as independent variables. To gain the greatest possible flexibility it seemed realistic to choose those quantities about which information is most probably provided by other areas of reactor design:

- (1) - neutron wall loading P_{Wn} [W/cm²]

- bremsstrahlung power. Although there is a difference in the temperature dependence of the neutron and the bremsstrahlung wall loading it is assumed in this model that they are proportional to each other. Therefore, a constant factor f_{Wbr}

$$f_{Wbr} = \frac{P_{Wbr}}{P_{Wn}} \quad [-] \quad (3)$$

has been introduced. This factor describes the dependence exactly only if changes in the neutron wall loading are obtained by changing the ion density. In the case the ion temperature is changed it is only an approximation.

- The power multiplication factor M of the blanket, defined as the ratio of the total nuclear power of the blanket and the power carried by the 14 MeV neutrons from the plasma to the blanket.
- the structure material volume fraction ϵ_W permitted for the outer cell wall. This figure does not include further structure material inside the beryllium, lithium or graphite regions.
- a factor f_{HW} which relates the average power density \bar{q} in the first wall to the neutron wall loading. This factor can be derived from the results of a neutronics analysis. Knowing the power density profile $q(x)$ across the first wall with the thickness s_W for a given neutron wall loading P_{Wn} , f_{HW} becomes

$$f_{HW} = \frac{\bar{q}}{P_{Wn}} = \frac{\frac{1}{s_W} \int_0^{s_W} q(x) dx}{P_{Wn}}$$

- a factor f_{dpa} which relates the annual displacement rate in the first wall to the neutron wall loading.

- the outer cell diameter D_C [cm]
- the radial length of a cell L_C [cm]
- the helium pressure level P_C [bar]
- the helium inlet temperature to the blanket t_{Cin} [C]
- the helium outlet temperature from the blanket t_{Cout} [C]
- the width of the helium duct δ [cm]
- the actual helium temperature at the first wall t_{cx} [C]
 The choice of t_{cx} has to be made within the limits of t_{Cin} and t_{Cout} .
- a factor f_{HTC} [-]
 allowing for input of an enhanced heat transfer coefficient at the first wall. In the course of the calculation a heat transfer coefficient h is calculated for the annular duct in the straight section of the cell. Inside the spherical cup, however, the heat transfer may deviate from that. It depends on the amount of coolant which is diverted into the beryllium region along the path from the straight section to the bottom of the spherical cup as indicated in Fig. 1. The factor f_{HTC} has been introduced to take account of this fact. Actually it has to be determined by the thermal design of the cell.

4. Analysis of the temperature and stress load of the first wall

4.1 Determination of basic quantities

From the quantities defined in sections 2 and 3 the following general quantities can be derived:

The ratio of the cell wall thickness s_w and the outer cell diameter D_C can be calculated from the structure material volume fraction ξ_w and the package density ρ_p by

$$\frac{s_W}{D_C} = \frac{1}{2} \left(1 - \sqrt{1 - \frac{\epsilon_W}{\rho_p}} \right) \quad [-] \quad (4)$$

The actual wall thickness s_W is

$$s_W = D_C \cdot \frac{s_W}{D_C} \quad [\text{cm}] \quad (5)$$

The cell cross-section area A_C follows directly from

$$A_C = \frac{\pi}{4} D_C^2 \quad [\text{cm}^2] \quad (6)$$

and the cross-sectional area of a matrix element from

$$A_M = \frac{A_C}{\rho_p} \quad [\text{cm}^2] \quad (7)$$

For the investigation of heat transfer it is necessary to know the coolant duct cross-section area A_{CD}

$$A_{CD} = \pi \cdot \delta \cdot (d_C - \delta) \quad [\text{cm}^2] \quad (8)$$

with the inner diameter of the cell d_C

$$d_C = D_C - 2 s_W \quad [\text{cm}] \quad (9)$$

The hydraulic diameter of the duct is

$$d_h = 2 \cdot \delta \quad [\text{cm}] \quad (10)$$

To calculate the helium flow rate necessary for cooling the cell within the indicated temperature limits, first the total power produced inside a single cell, P_C , has to be evaluated. It is assumed that the entire energy incident on the area of one matrix element is converted to heat inside the corresponding cell. This means that

$$P_C = [P_{Wn} (M + f_{Wbr})] \cdot A_M \quad [\text{W}] \quad (11)$$

From this the coolant mass flow rate \dot{m}_c , volume flow rate \dot{v}_c and velocity w_c can be calculated by using the following equations:

$$\dot{m}_c = \frac{P_C}{c_p (t_{c\text{out}} - t_{c\text{in}})} \quad [\text{g/s}] \quad (12)$$

$$\dot{v}_c = \frac{\dot{m}_c}{\rho_c} \quad [\text{cm}^3/\text{s}] \quad (13)$$

$$w_c = \frac{\dot{v}_c}{A_{CD}} \quad [\text{cm/s}] \quad (14)$$

The specific heat at constant pressure, c_p , and the density ρ_c of the coolant are thereby taken at the mean coolant temperature t_c

$$t_c = \frac{t_{c\text{in}} + t_{c\text{out}}}{2} \quad [C] \quad (15)$$

4.2 Heat transfer from the wall to the coolant

The inner cell wall temperature t_{wi} is determined by the actual helium temperature t_{cx} and the temperature difference Δt_{cW} between the wall and coolant. The latter depends on the real heat flux density q_W at the wall and the heat transfer coefficient h of the coolant flow:

$$\Delta t_{cW} = \frac{q_W}{h} \quad [C] \quad (16)$$

The heat flux density q_W is calculated from the bremsstrahlung wall loading and the power density inside the wall by the following equation:

$$q_W = P_{Wn} \cdot (f_{Wbr} + s_W \cdot f_{HW}) \quad [W/\text{cm}^2] \quad (17)$$

The calculation of the heat transfer coefficient h is based on the Kraussold-Nusselt equation for a circular pipe flow presented by U. Grigull et al. [4]:

$$Nu = 0.032 \cdot Re^{0.8} \cdot Pr^{0.37} \cdot \left(\frac{L_c}{d_h} \right)^{-0.054} \quad (18)$$

with Nu , Re , and Pr being the Nusselt, Reynolds and Prandtl numbers. Introducing the coolant properties and the layout parameters into equ. (18) we arrive at the following expression for h :

$$h = \frac{k}{d_h} \cdot Nu =$$

$$h = 0.032 \cdot f_c(\mathcal{J}, p_c) \cdot w_c^{0.8} \cdot L_c^{-0.054} \cdot d_h^{-0.146} \quad [W/cm^2 \text{ grd}] \quad (19)$$

$f_c(\mathcal{J}, p_c)$ summarizes the coolant properties which, in general, are dependent on the coolant pressure p_c and the average film temperature \mathcal{J} :

$$f_c(\mathcal{J}, p_c) = \rho_c^{0.8} \cdot c_p^{0.37} \cdot k^{0.63} \cdot \eta^{-0.43} \quad (20)$$

Equation (20) yields the results in appropriate units for direct use in equ. (19), if the coolant properties are expressed in the following units:

density	ρ_c	[g/cm ³]
specific heat	c_p	[J/g grd]
thermal conductivity	k	[W/cm grd]
dynamic viscosity	η	[g/cm s]

To yield h in $W/cm^2 \text{ grd}$, the coolant velocity w_c , coolant duct length L_c , and hydraulic diameter d_h have to enter in the following units:

w_c	[cm/s]	L_c	[cm]	d_h	[cm]
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In the analytical model the heat transfer coefficient and the quantities depending on it are evaluated in a iterative way, so that the temperature difference between the wall and coolant is obtained with an accuracy of $\pm 1 \text{ C}$.

4.3 Temperatures inside the first wall

Having evaluated the inner wall temperature t_{W_i} by

$$t_{W_i} = t_{cx} + \Delta t_{cW} \quad [C] \quad (21)$$

the next step is to determine the outer and the mean wall temperatures t_{W_o} and t_{W_m} . To solve this problem, the differential equation describing the process of heat conduction was applied to the case of plane geometry, which should be a sufficiently good approximation. As boundary conditions the temperature gradient resulting from the heat transfer to the coolant was used at the inner side, while the bremsstrahlung wall loading treated as an external heat source was used at the outer side. This procedure yields the temperature difference across the wall:

$$\Delta t_W = \frac{P_{Wn} \cdot s_w}{k_W} \left(f_{Wbr} + \frac{1}{2} s_w \cdot f_{HW} \right) \quad [C] \quad (22)$$

The outer and mean wall temperatures t_{W_o} and t_{W_m} then become

$$t_{W_o} = t_{W_i} + \Delta t_W \quad [C] \quad (23)$$

$$t_{W_m} = t_{W_i} + \frac{1}{2} \Delta t_W \quad [C] \quad (24)$$

The thermal conductivity k_W of the wall material is thereby taken at the mean wall temperature. Therefore, in this case as well the analytical model provides an iterative procedure, yielding Δt_W with an accuracy of ± 0.5 C.

4.4 Stresses inside the first wall

Whereas the temperature load of the first wall has already been determined by the procedure described above, the problem of stress loading remains to be solved. At present, three sources of stress can be identified:

- tensile stresses due to the static coolant pressure
- thermal stresses due to the temperature gradients across the wall
- additional stresses due to differential swelling in the wall.

From these sources the contribution of stresses due to differential swelling has not yet been included in the analysis.

For calculating the tensile stresses the equation

$$\sigma_1 = \frac{P_c}{2 \left(\frac{s_w}{D_c} \right)} \quad [\text{kp/cm}^2] \quad (25)$$

is used, yielding the maximum tangential stress at any point along the straight section of the cell wall. This should also be the maximum stress in that part of the wall which is directly exposed to the neutron and bremsstrahlung radiation, at least at the point where the straight section is joined to the spherical cup.

The thermal stresses are calculated using the formula given by J.R. Stanbridge et al. [2]:

$$\sigma_2 = \frac{1}{2} \cdot \frac{\Delta t_w \cdot \alpha \cdot E}{1 - \nu} \quad [\text{kp/cm}^2] \quad (26)$$

Here α is the thermal expansion coefficient, E the Young's modulus and ν the Poisson ratio of the wall material considered.

Since the thermal stress has a tensile characteristic at the colder side of the wall, the maximum total stress

$$\sigma = \sigma_1 + \sigma_2 \quad [\text{kp/cm}^2] \quad (27)$$

appears at the inner side of the cell wall.

5. Material properties and wall lifetime

Before any statement about the lifetime of the first wall can be made, the temperature and stress loads have to be quoted against the relevant material properties. As far as we are aware these are:

- the time rupture strength
- the creep strength
- the fatigue strength
- the ductility

In comparing the temperature and stress loads evaluated according to the procedure described in section 4 with those four properties, their variation with the radiation dose also has to be taken into account. As far as the fatigue strength is concerned, information about the thermal cycling should also be available. At the present stage only the time rupture behaviour, excluding the influence of irradiation, thermal creep and irradiation creep have been included in the analysis.

5.1 Time rupture strength

In the current literature (see, for example, H. Böhm, M. Schirra [5]) data for the time rupture strength σ_{tr} are presented as a function of the Larson-Miller parameter P. This parameter is defined as

$$P = T \cdot (C + \log t_1) \quad (28)$$

C being a material constant, T [K] the operating temperature and t_1 [h] the useful life.

The analytical model presented here makes use of information of this kind in the following way. If σ_{tr} is known as a function of P, the reverse procedure must yield a certain value for P when σ_{tr} is replaced by the total stress load σ evaluated in equ. (27). Introducing the working temperature T

(in the present case the inner wall temperature t_{Wi} as the point of maximum stress), equ. (28) can be solved for t_1 yielding

$$t_1 = 10 \left(\frac{P}{T_{Wi}} - C \right) \quad [h] \quad (29)$$

An interative procedure could be provided in the event of knowledge about the changes of σ_{tr} with the neutron dose becoming available.

5.2 Thermal creep

Literature on thermal creep, e.g. M. Schirra [6] or G. Schmidt [7], presents the dependenc of the creep rate $\dot{\epsilon}_t$ due to Norton's relationship:

$$\dot{\epsilon}_t = k \cdot \sigma^n \quad [h^{-1}] \quad (30)$$

Using σ in $[kp/mm^2]$ yields $\dot{\epsilon}_t$ in $[h^{-1}]$. M. Schirra [6] also presents an equation with which it is possible to calculate the lifetime t_2 due to thermal creep:

$$\log t_2 + m \log \dot{\epsilon}_t = \ell \quad (31)$$

Therefore, if the materials characteristics k , n , m , and ℓ are known, it is possible to arrive at a second figure t_2 for the expected lifetime due to

$$t_2 = 10^{(\ell - m \cdot \log \dot{\epsilon}_t)} \quad [h] \quad (32)$$

The analytical model makes use of this procedure and evaluates t_2 besides t_1 .

5.3 Irradiation creep

The importance of irradiation creep is outlined by J.R. Stanbridge et al. [2]. The equation given in this report can be transformed in such a way that a creep rate $\dot{\epsilon}_i$ can be obtained.

$$\dot{\epsilon}_i = C_i \cdot f_{dpa} \cdot P_{Wn} \cdot \zeta \quad [h^{-1}] \quad (33)$$

Information has, however, to be provided about the materials constant C_i , which is presented in this report only for stainless steel.

5.4 Open problems

As mentioned above, the analytical model for an estimation of the first wall lifetime as described so far is not yet complete. There is especially a lack of information about the irradiation effects. This becomes immediately obvious if the creep rates for both thermal and irradiation creep are quoted against any permissible limit. Equation (32) defines an expected lifetime only for the case of absence of irradiation and is therefore not necessarily true if $\dot{\epsilon}_t$ is replaced by the total creep rate $\dot{\epsilon}$:

$$\dot{\epsilon} = \dot{\epsilon}_t + \dot{\epsilon}_i \quad [h^{-1}] \quad (34)$$

The analytical model nevertheless makes use of this assumption to calculate a third lifetime t_3 .

$$t_3 = 10^{(l - m \cdot \log(\dot{\epsilon}_t + \dot{\epsilon}_i))} \quad [h] \quad (35)$$

To obtain a precise picture, obviously the total creep rate and the static elongations due to thermal and stress effects have to be quoted against the ductility of the

special material, which, in general, is strongly affected by irradiation embrittlement. This step in the analysis has not yet been done. Instead of this a fourth time t_4 describing the time in which a total of 1 % creep is reached is calculated:

$$t_4 = \frac{0.01}{\dot{\epsilon}_t + \dot{\epsilon}_1} \quad [h] \quad (36)$$

Summarizing the status of the development of the analytical model, it must be emphasized that further refinements and additions are necessary. This is especially true as far as the material properties are concerned. It is in this area that a lot of work will have to be done if this model is to be further developed to become a useful tool in reactor design.

6. Computer program

In order to have a convenient means of performing parameter studies, it was decided to program this model for the IPP computer. Figure 3 shows a flow diagram of this program, which shall be briefly explained in this section.

The program consists of a main program and 12 sub-programs, 8 of which exclusively provide material properties. These latter routines are written in such a way that they can easily be corrected and extended to any number of materials.

6.1 Description of the single programs

MAIN is a multi-purpose input/output program which can be applied to any calculation routine. It offers great flexibility for the following reasons:

- the number of input quantities is, in principle, unlimited.
- each input quantity can be varied in equal steps between two limits. In the present version, however, only the first two quantities with varying input are accepted to be changed.
- it provides either total output or selected output of special quantities.
- it provides additional plot output of selected quantities. In the present version, however, the plot routines have been removed to save storage location during the test period.

Communication with the first level subroutines (in this case WLOAD and LIFE) is ensured by COMMON statements. To adjust this program to any subroutine, only the COMMON and part of the DIMENSION statements have to be changed. In addition, the names of the input and output quantities have to be put to two DATA statements to provide a self-explanatory print output.

WLOAD performs all calculations described in section 4 of this paper. It makes use of two further subroutines for calculation: HTCOF and WALLTM.

HTCOF evaluates the heat transfer coefficient, the temperature difference between the wall and coolant and the inner wall temperature itself. The coolant properties needed for this calculation are supplied by the subroutine COOLNT.

WALLTM calculates the temperature difference within the wall using the thermal conductivity provided by COND.

LIFE finally takes over the figures for the temperature and stress load evaluated by WLOAD and quotes them against the material properties as described in sections 5 and 6.

COOLNT calculates the coolant properties for helium which are needed in WLOAD and HTCOF. Dependent on temperature and pressure, it yields the density, specific heat at constant pressure, thermal conductivity, and dynamic viscosity. The equations contained in this subroutine are taken from W. Zimmerer [8].

The remainder of the routines are concerned with properties of the wall materials. Each property needed is supplied by a special subprogram which may contain the data of various metals and alloys. Which of these data are used is decided by a material identification number.

COND supplies the thermal conductivity dependent on the temperature. At present this FUNCTION subprogram contains only data for vanadium and vanadium-titanium alloys. Data for pure vanadium and for the alloys V-3 Ti and V-20 Ti have been calculated according to H. Böhm et al. [9]. For any other titanium content these data are linearly interpolated. The actual titanium content which will be taken into account is supplied to this and any further routine by the material identification number.

EXPN calculates the thermal expansion coefficient dependent on the temperature. This FUNCTION subprogram at present contains only the data for pure vanadium which are also used for vanadium alloys. They are taken from F. Sperner [10].

EMOD supplies Young's modulus. Provision is made for taking into account a temperature dependence. At present, however, only the value for vanadium is supplied as a constant figure which is due to F. Sperner [10].

POIS supplies the Poisson ratio. For this the same is valid as for Young's modulus, the figure for vanadium being taken from Kieffer [11].

LARSON supplies the material constant C (see equ. (28)) and calculates the Larson-Miller parameter P for a given stress. As yet only data for two vanadium alloys, V-5 Ti and V-20 Ti, which were taken from H. Böhm, M. Schirra [5], have been evaluated. The curves given there were approximated by polynomials and interpolation is provided to obtain results for any reasonable titanium content.

CIC supplies the material constant C_i for irradiation creep (see equ. (33)). At present only the figure for stainless steel is known from the publication of J.R. Stanbridge et al. [2]. This figure is therefore also used for vanadium.

CREEP supplies the parameters for secondary creep, i.e. the creep constant k and the stress exponent n for Norton's equation (equ. (30)) and the material constants m and ℓ which are needed to solve equ. (32) and (35) respectively. This subroutine contains at present k and n for stainless steel according to G. Schmidt [7] dependent on temperature. Since m and ℓ are unknown, the same data as for vanadium are used here.

Data of k and n for vanadium and vanadium alloys were derived from the publication of H. Böhm and M. Schirra [5]. Because of the somewhat different definition of Norton's equation close agreement can only be expected for pure vanadium and the alloys V-2.8 Ti, V-5 Ti, and V-20 Ti. The constants m and ℓ were also taken from this publication.

Complete lists of all programs are included in the Appendix.

6.2 Description of the input

1st card (format: 8X, 4I4)

NA number of input quantities = 17 (fixed)
NR number of results (total) = 34 (fixed)
NPR number of results to be printed
 $NPR \leq NR$
 NPR = 0 : total output
NPL number of results to be plotted = 0 (fixed)

2nd card ff. (format: 8X, 3F12.5)

A1 lower limit of variable range
A2 upper limit of variable range
DA increment of variable variation

There have to be NA cards defining the input variables in the following sequence:

PWNU neutron wall loading P_{Wn} [W/cm²]
FWBR⁺ bremsstrahlung factor f_{Wbr} [-]
PMUL power multiplication factor M [-]
STRU structure material volume fraction ξ_w [-]
PRES coolant pressure P_c [bar]
TCI coolant inlet temperature t_{cin} [C]
TCO coolant outlet temperature t_{cout} [C]
TCX coolant temperature at first wall t_{cx} [C]
ARR option for cell arrangement
 = 1.0 cylindrical cells in square matrix
 > 1.0 cylindrical cells in hexagonal matrix
DCO outer cell diameter D_c [cm]
CHAN coolant duct width δ [cm]
XLEN cell or coolant duct length L_c [cm]
PITC cell pitch ratio p [-]

FHW⁺ power density factor f_{HW} [cm^{-1}]
FHTC factor for enhanced heat transfer f_{HTC} [-]
XMAT⁺⁺ material identification number [-]
FDPA⁺ displacement rate factor f_{dpa} [-]

+ The factors f_{Wbr} , f_{HW} , and f_{dpa} have to be normalized to $P_{Wn} = 100 \text{ W/cm}^2 = 1 \text{ MW/m}^2$.

++ The material identification number is of the following type:

XMAT = xx.yyzz

xx = charge number of the basic element
yy = percentage of the first alloying element
zz = percentage of the second alloying element

Examples:

Pure vanadium: XMAT = 23.0000
V-5 Ti : XMAT = 23.0500
V-10 Ti-10 Cr: XMAT = 23.1010

3rd card ff. (format: 8X, 9I4)

KPR identification numbers of the results to be printed.
There have to be NPR identification numbers. The numbers correspond to the order in the DATA Namer statement of MAIN. Their meaning can be looked up in the comment cards in WLOAD and LIFE.
This card can be omitted if NPR = 0

4th card ff. (format: 8X, 9I4)

KPL identification numbers of the results to be plotted.

This card has to be omitted since plot-output is not possible in the present version.

7. Sample calculations

7.1 Input specifications

To demonstrate the runnability of the program some sample calculations were performed. The following input parameters were chosen:

neutron wall loading	$P_{Wn} = 83 \text{ W/cm}^2$
bremsstrahlung factor	$f_{Wbr} = 2.7 \cdot 10^{-4}$
power multiplication factor	$M = 1.2$
structure material volume fraction	$\xi_W = 0.02$
coolant pressure	$P_c = 10 \text{ to } 50 \text{ bar}$
coolant inlet temperature	$t_{cin} = 350 \text{ C}$
coolant outlet temperature	$t_{cout} = 750 \text{ C}$
coolant temp. at first wall	$t_{cx} = 350 \text{ to } 750 \text{ C}$
arrangement option (see section 2)	$= 1.0$
cell outer diameter	$D_C = 30 \text{ cm}$
cell length	$L_C = 100 \text{ cm}$
pitch ratio	$p = 1.0$
heat source density factor	$f_{HW} = 0.04$
heat transfer factor	$f_{HTC} = 1.0$
material identific. number	see below
displacement rate factor	$f_{dpa} = 0.319$

Four runs were performed for four different vanadium alloys:

pure vanadium	XMAT = 23.0000
V-2.8 Ti	XMAT = 23.0300
V-5 Ti	XMAT = 23.0500
V-20 Ti	XMAT = 23.2000

The neutron wall loading, the bremsstrahlung factor and the power multiplication factor are in agreement with the CTRD outline specifications stated at the 1st General Meeting of the CTRD team in May 1974 at Garching. The characteristic data of the cell, i.e. D_C , L_C and ϵ_W correspond to the Culham design [1], while the arrangement of the cells and their pitch ratio were chosen arbitrarily. The heat source density factor and the displacement rate factor were derived from Steiner's publication [12]. For the thermodynamic cycle the temperature boundaries were taken from an equivalent fission reactor helium cycle [13].

The aim of the sample calculation was to see the influence of the coolant pressure level P_C and the actual helium temperature at the first wall t_{cx} . Therefore, P_C was varied between 10 and 50 bar, and t_{cx} within the limits of the blanket inlet and outlet temperatures, i.e. between 350 and 750 C.

7.2 Results for pure vanadium

From the four runs made for different materials here only the results for pure vanadium shall be reported in some detail. The influences of titanium additions to the basic material are summarized below.

Figure 4 shows the results obtained for t_1 , which is the useful life due to the time rupture strength. As was to be expected, the life decreases as temperature and pressure increase.

The same is true of t_2 , which is the useful life due to thermal creep, as can be seen from the solid lines in fig. 5. Two differences, however, can be observed:

- the temperature dependence shows a steeper slope in the case of thermal creep as compared to the time rupture behaviour.
- the pressure dependence shows an increasing slope in the time rupture behaviour but a decreasing slope in the creep behaviour.

As far as the absolute figures are concerned, it can be clearly stated that the limits set by the time rupture strength are more stringent than those set by thermal creep. This is more obvious from table I, in which the temperature limits are summarized as dependent on the pressure level for three different lifetimes, namely 1, 5, and 20 years. From this table it can be concluded that the first wall could last the whole life of the reactor, if the helium pressure does not exceed about 40 bar and the wall temperature remains in the range of 400 C, as was anticipated in the first guess of the blanket thermal design.

Figure 5 also shows the influence of irradiation creep, which is represented by the dotted lines. At this point it should be repeated, however, that two assumptions are involved in this consideration:

- the irradiation creep constant C_i is the same for vanadium as for stainless steel.
- the lifetime due to the sum of thermal and irradiation creep obeys the same law as the lifetime due to thermal creep alone.

This figure shows that irradiation creep limits the lifetime at lower temperatures whereas at higher temperatures thermal creep is the limiting effect. As is already expressed by equ. (33), there is only a dependence on pressure, not on temperature. The absolute values of the lifetime where irradiation creep becomes important, however, are far from being realistic. Therefore strong doubts should be cast on this line of reasoning.

A more realistic picture of the material performance can most probably be achieved if the actual creep rates, i.e. the sum of thermal and irradiation creep, are considered. In Fig. 6 the total creep rates are plotted versus the temperature t_{cx} with the pressure level as the parameter. The zero slope at low temperatures is again due to the irradiation creep, whereas the strong increase at higher temperatures is due to thermal creep. The straight lines in this diagram designate those creep rates which can be allowed to reach a certain amount of total creep in a certain time. This picture clearly shows that irradiation creep may become very important from this point of view.

At this stage again the question arises what amount of total creep or, to be more precise, of total elongation including static elongation and creep can be permitted from the ductility point of view. No relevant information is available at present. Therefore, the amount of total creep has to be treated as a variable in the following considerations.

Figures 7 to 9 now show the limits for the first wall lifetime of 1, 5, and 20 years respectively, within the parameter fields of both pressure and temperature. Common to all the three figures is that the limit of thermal creep is irrelevant in all cases. Whether or not the time rupture strength will be a limiting property ultimately depends on the change of this limit due to irradiation and on the total permissible creep.

If a useful life of 1 year (see Fig. 7) is envisaged, then the time rupture strength should become a limit only if helium pressures above 40 bar are chosen. At lower pressure levels the ductility should always present the significant limitation. This is the more true the longer the lifetimes expected are (see Figs. 8 and 9). The reason for this is

the shift of the curves for equal total creep to lower levels of temperature and pressure with increasing time, and the most probable shift of the total permissible creep to lower values with increasing time and neutron fluence.

The most interesting information from these three pictures, however, is the cut-off of the smooth curves characterizing the constant amounts of total creep at certain pressures. This is a consequence of the effect of irradiation creep. As can be seen from Figs. 7 to 9, these pressure limits are dependent on the lifetime chosen. Assuming a helium temperature at the first wall close to the inlet coolant temperature (to be precise, $t_{cx} \approx 420$ C for $t_{cin} = 350$ C), the choice of the helium pressure level is exclusively determined by the lifetime desired and the total permissible creep. This relationship is shown in Fig. 10. This picture can be interpreted in the following way:

If the ductility loss during 5 years of operation should allow for 5 % creep during this period, then the helium pressure should not exceed 34 bar. If, however, the helium pressure chosen were to be only 20 bar, then a lifetime of about 10 years could be expected.

7.3 Influence of titanium as alloying element

In this section it is not intended to repeat the entire procedure once more for the three vanadium-titanium alloys. The results of the equivalent calculations shall only be summarized with regard to their tendency.

As compared to pure vanadium the V-2.8 Ti alloy offers a higher lifetime due to its time rupture strength. This, however, can only be considered as an advantage, as was stated in section 7.2, if the total creep rates are smaller and/or if the irradiation effects on the ductility are of less importance than in the case of pure vanadium. Indeed, the total creep rate has to be expected to be higher at

low temperatures but very much smaller at higher temperatures. The reason for this is that the thermal creep behaviour is somewhat more balanced, yielding higher lifetimes at high temperatures and lower, but still sufficiently high lifetimes at low temperatures. This, in turn, means that the irradiation creep becomes more important at low temperatures than in the case of pure vanadium. Since the model does assume identical irradiation creep constants for all the alloys considered there should be no significant consequence for the choice of the helium pressure if a low first wall temperature is envisaged. Under these circumstances the choice of this alloy would only be profitable if higher temperatures should occur.

Similar conclusions can be drawn about the two remaining alloys V-5 Ti and V-20 Ti. These, however, again show very disadvantageous behaviour of the thermal creep rate at higher temperatures. At present no necessity can therefore be seen for using alloys with a high titanium content unless significant differences in the ductility and irradiation creep behaviour should be detected.

8. Conclusions

The work done up to now shows that the analytical model and the computer program based on it is able to derive statements concerning the material performance and especially the first wall lifetime. The reliability of these statements mainly depends, however, on the accuracy and reliability of the material data. In this respect the model is far from being complete.

At present only some data on vanadium and a few vanadium alloys are included. Already the sample calculations show that the information about these materials is insufficient to arrive at reasonably credible estimates of the lifetime. As far as these alloys are concerned, neither the time rupture nor the

thermal creep behaviour seem to be properties governing the first wall life. It is far more probable that especially at the low temperatures envisaged the irradiation creep and the ductility under neutron irradiation will be the properties that will impair operation of the wall. It is in these very fields, however, that information is lacking.

It is hoped by the author that the work performed hitherto can be completed to some extent by incorporating the information arising in the field of material investigation. It will then surely be possible to arrive at somewhat more credible statements about the first wall lifetime to be expected for this special design.

It should also be mentioned that neither the method applied nor the computer program is restricted to the blanket design of the CTRD. In the same way as was done here, it could be applied to any other design now existing or arising in the future. In this case modifications of the program will be necessary especially in the subroutine WLOAD. The amount of work entailed, however, should not be prohibitive.

Besides this, the program could be sustained in an operable mode if the stress, temperature and irradiation loads were used as the only input quantities. With this modification performed the program could be made a module of a more comprehensive systems analysis program system.

9. Acknowledgements

The author gratefully acknowledges his nomination to the CTRD study group by K.H. Schmitter in that it enabled him to deal with this interesting problem. He is also indebted to Dr. A. Knobloch, member of der Coordinating Committee, for providing him with useful information. He also thanks Dr. S. Förster (KFA Jülich), Dr. G. McCracken and Mr. J.T.D. Mitchell (Culham Laboratory) for many helpful discussions and some valuable hints during the progress of this work. He finally thanks A. Nicol for reading the manuscript.

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TABLE I: Temperature limits $\dot{\gamma}$ for the first wall coolant to reach a lifetime t for a certain coolant pressure P_C for pure vanadium.

$\dot{\gamma}$ [Cl; P_C [bar]

$\dot{\gamma}_{TR}$: limit due to time rupture strength

$\dot{\gamma}_{CR}$: limit due to thermal (+ irradiation) creep

P_C	$t = 1a$		$t = 5a$		$t = 20a$	
	$\dot{\gamma}_{TR}$	$\dot{\gamma}_{CR}$	$\dot{\gamma}_{TR}$	$\dot{\gamma}_{CR}$	$\dot{\gamma}_{TR}$	$\dot{\gamma}_{CR}$
10	650	780	615	760	585	750
20	640	690	600	670	575	655
30	590	640	560	625	540	610
40	520	610	495	590	460	580
50	410	585	380	570	360	560

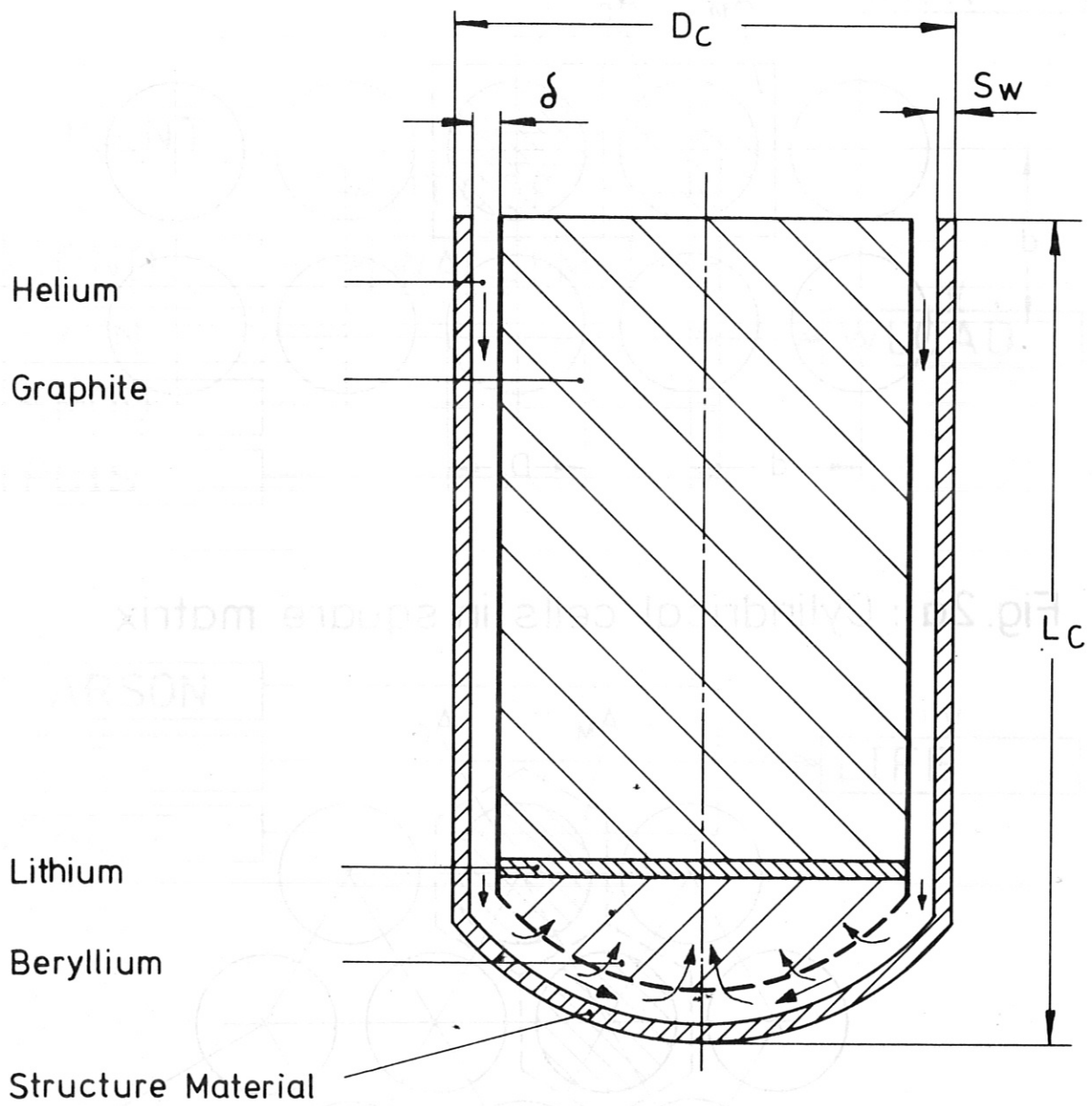


Fig.1.: Scheme of a blanket module cell

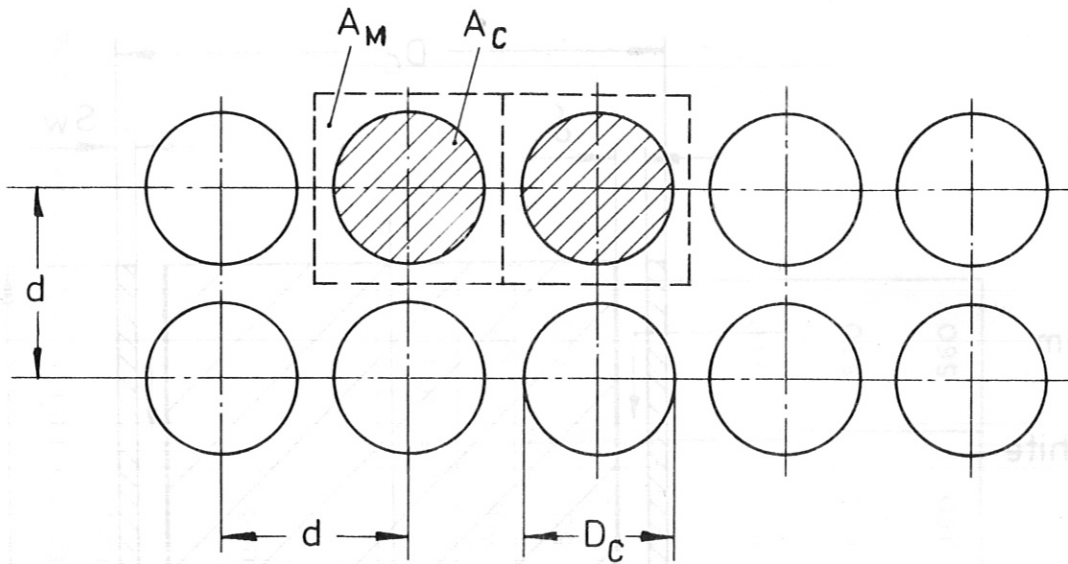


Fig.2a : Cylindrical cells in square matrix

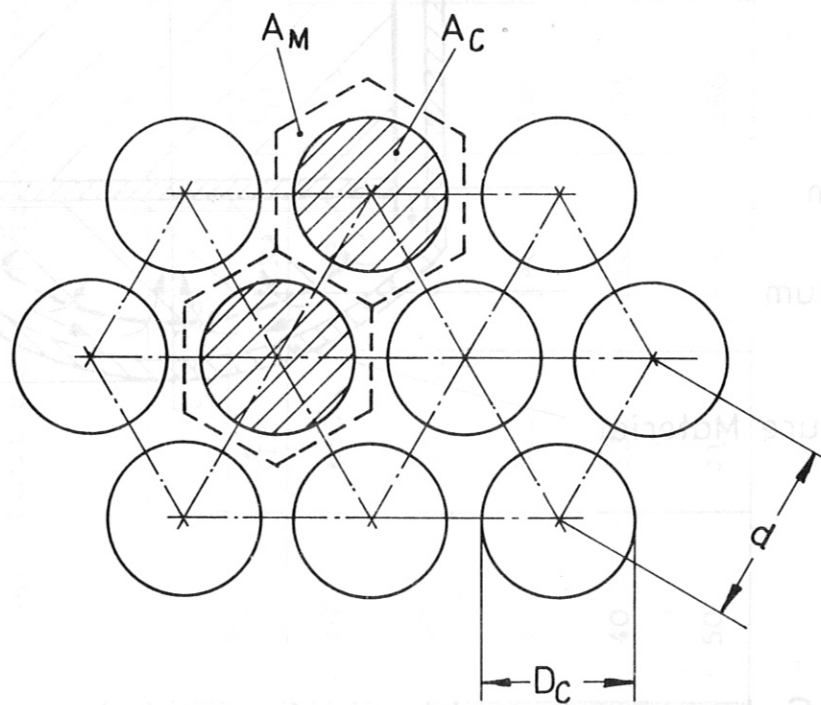


Fig.2b : Cylindrical cells in hexagonal matrix

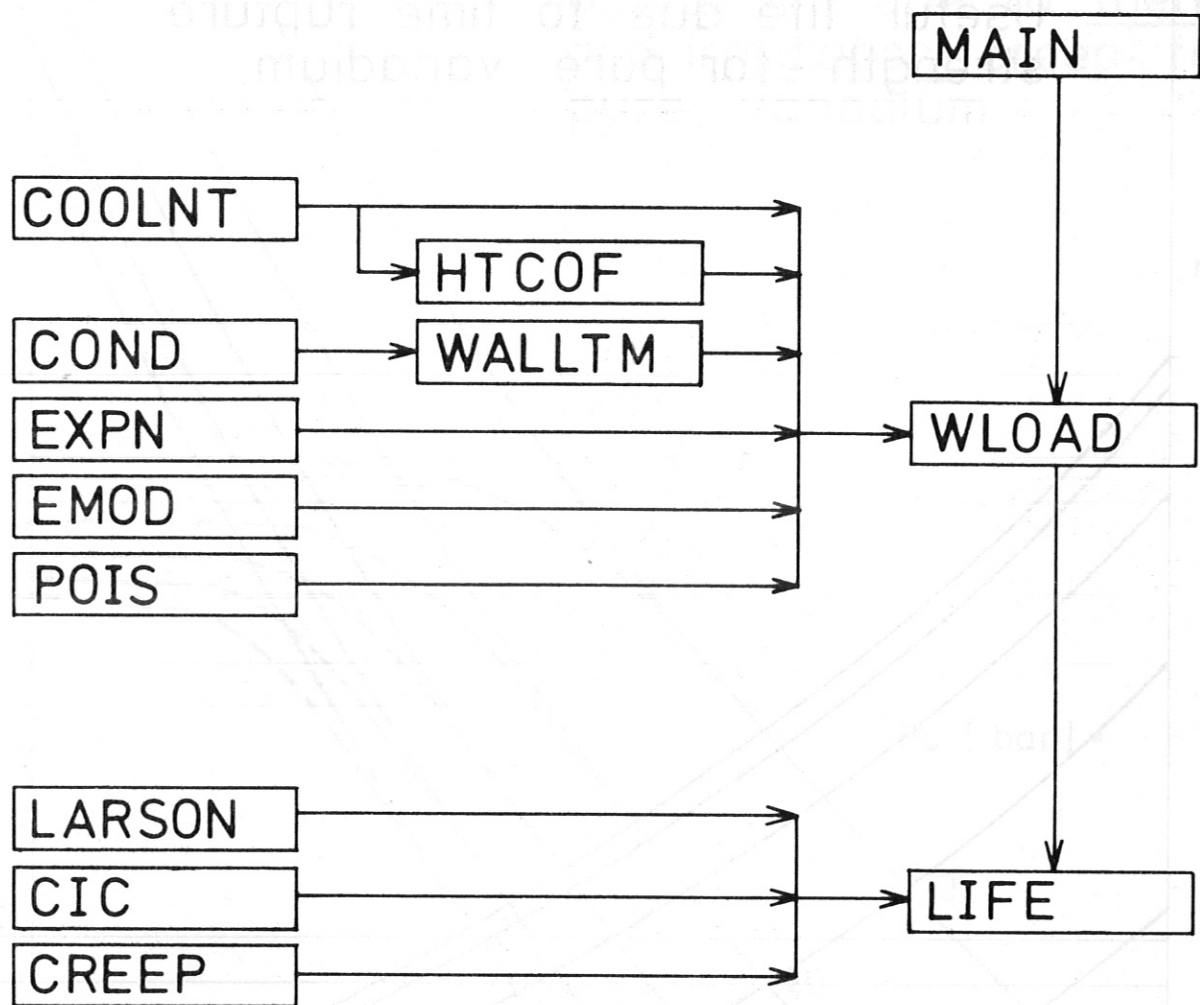


Fig. 3. : Flow Diagram of "WALLIFE "

Fig. 4 :
Useful life due to time rupture
strength for pure vanadium

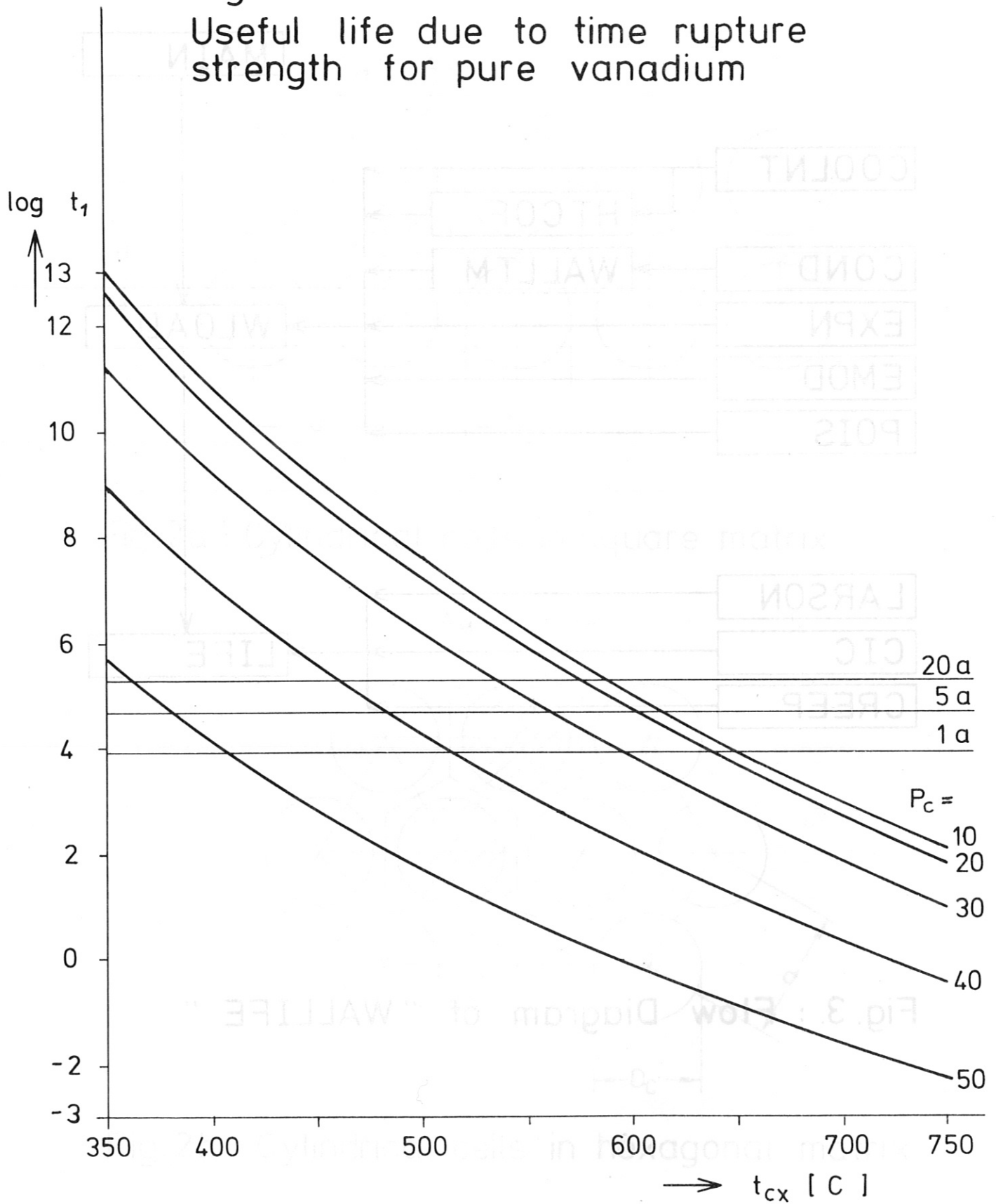
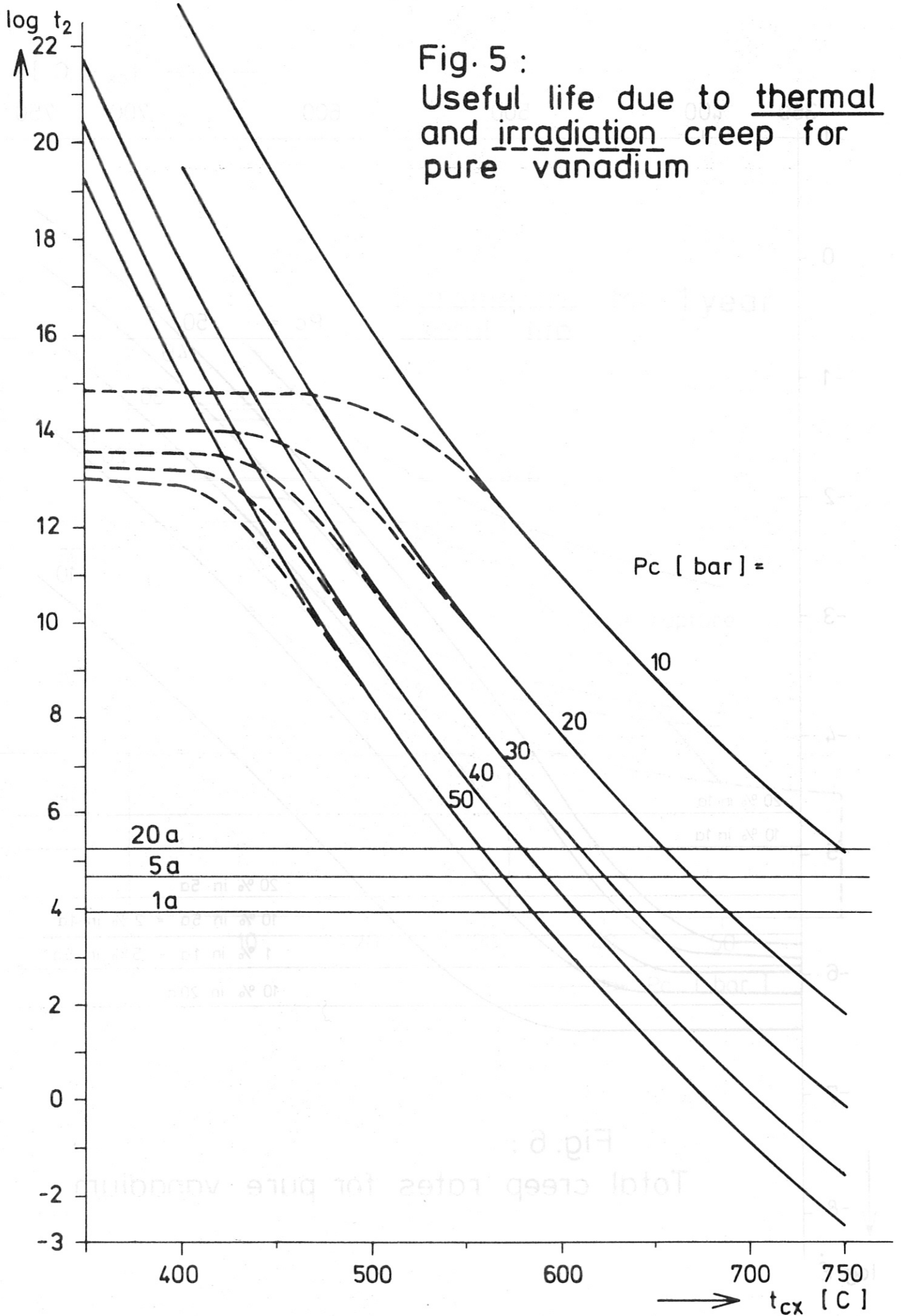


Fig. 5 :
Useful life due to thermal
and irradiation creep for
pure vanadium



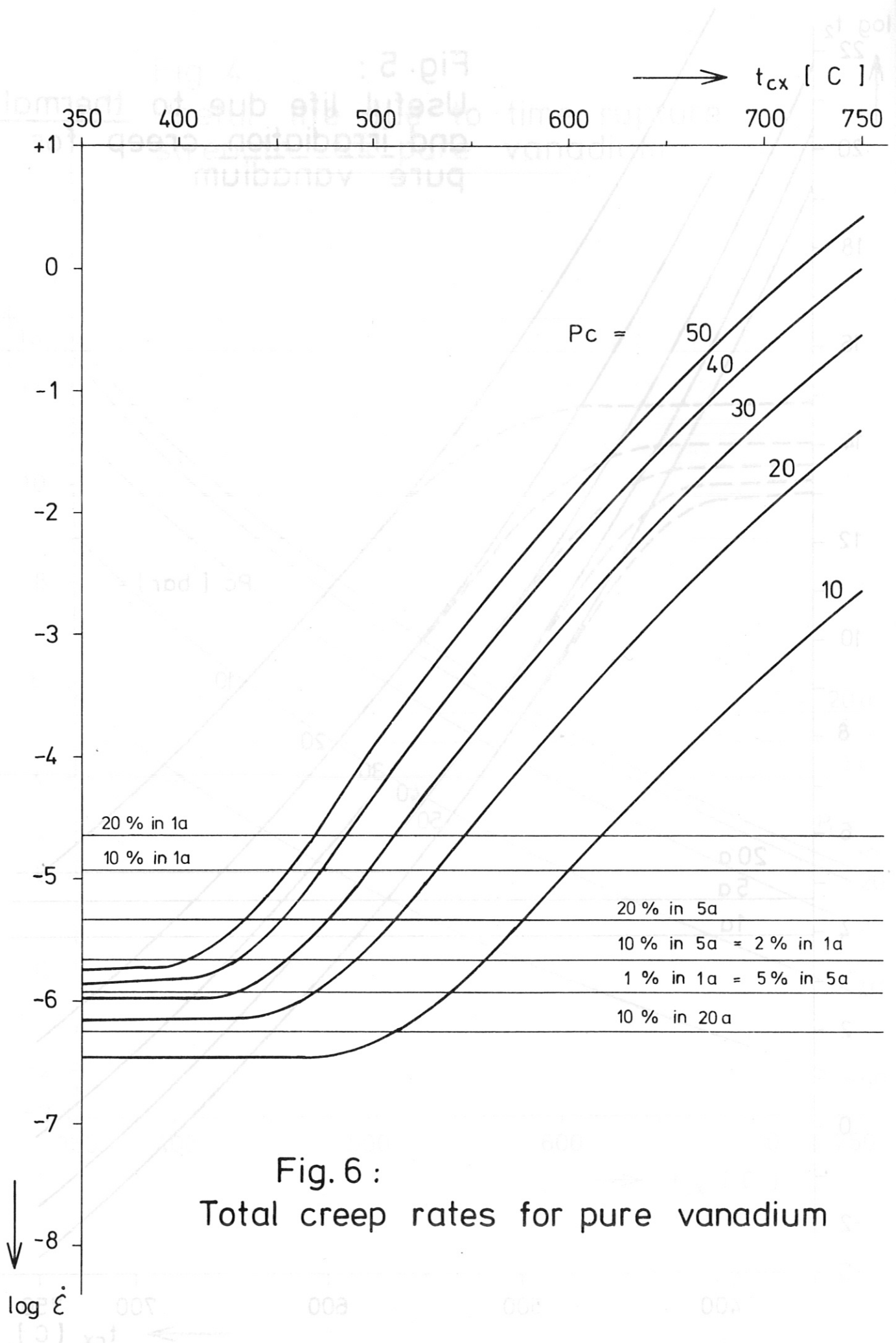


Fig. 6 :
 Total creep rates for pure vanadium

Fig. 7 :
Parameters for 1 year
useful life

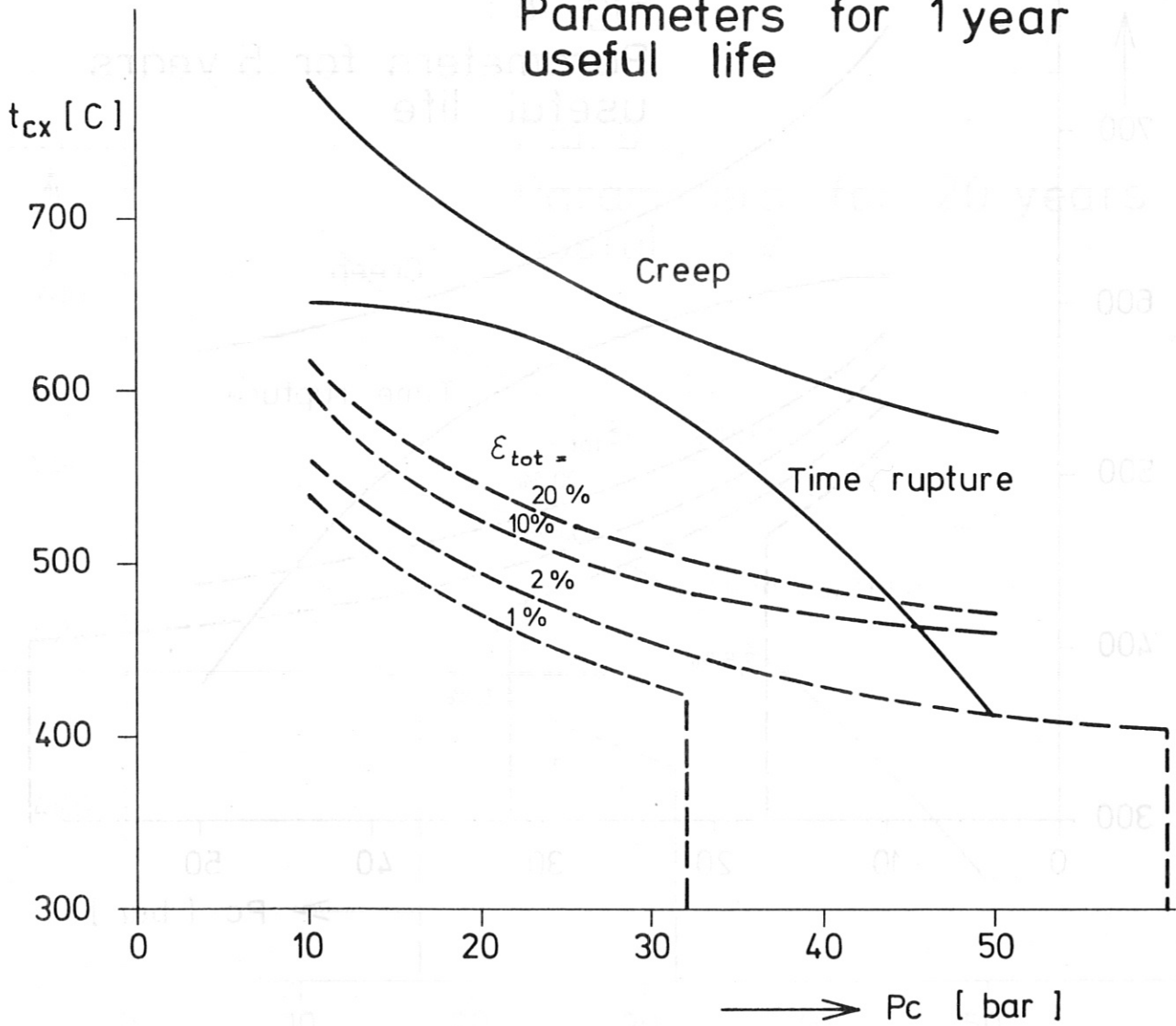


Fig. 8 :
Parameters for 5 years
useful life

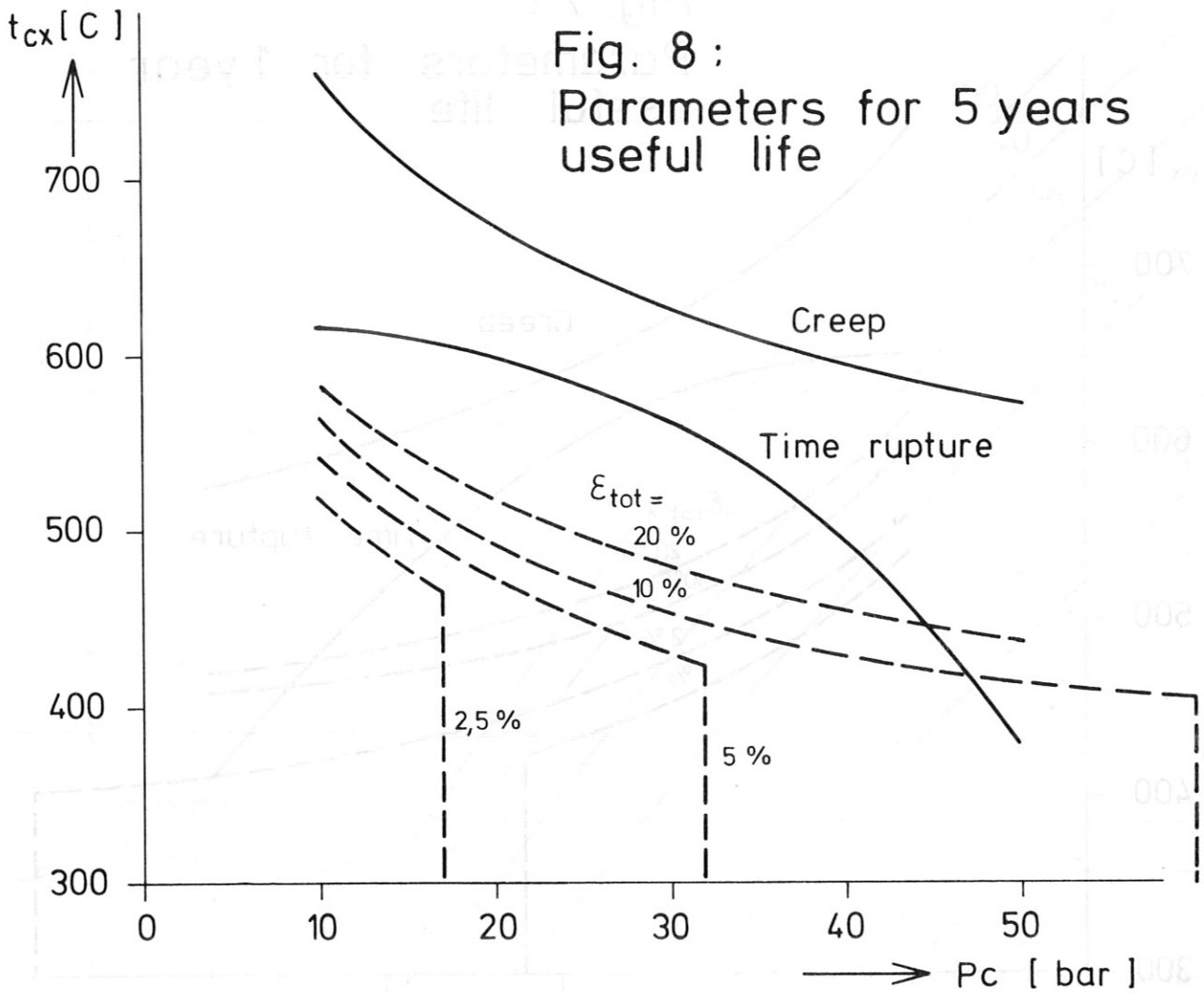
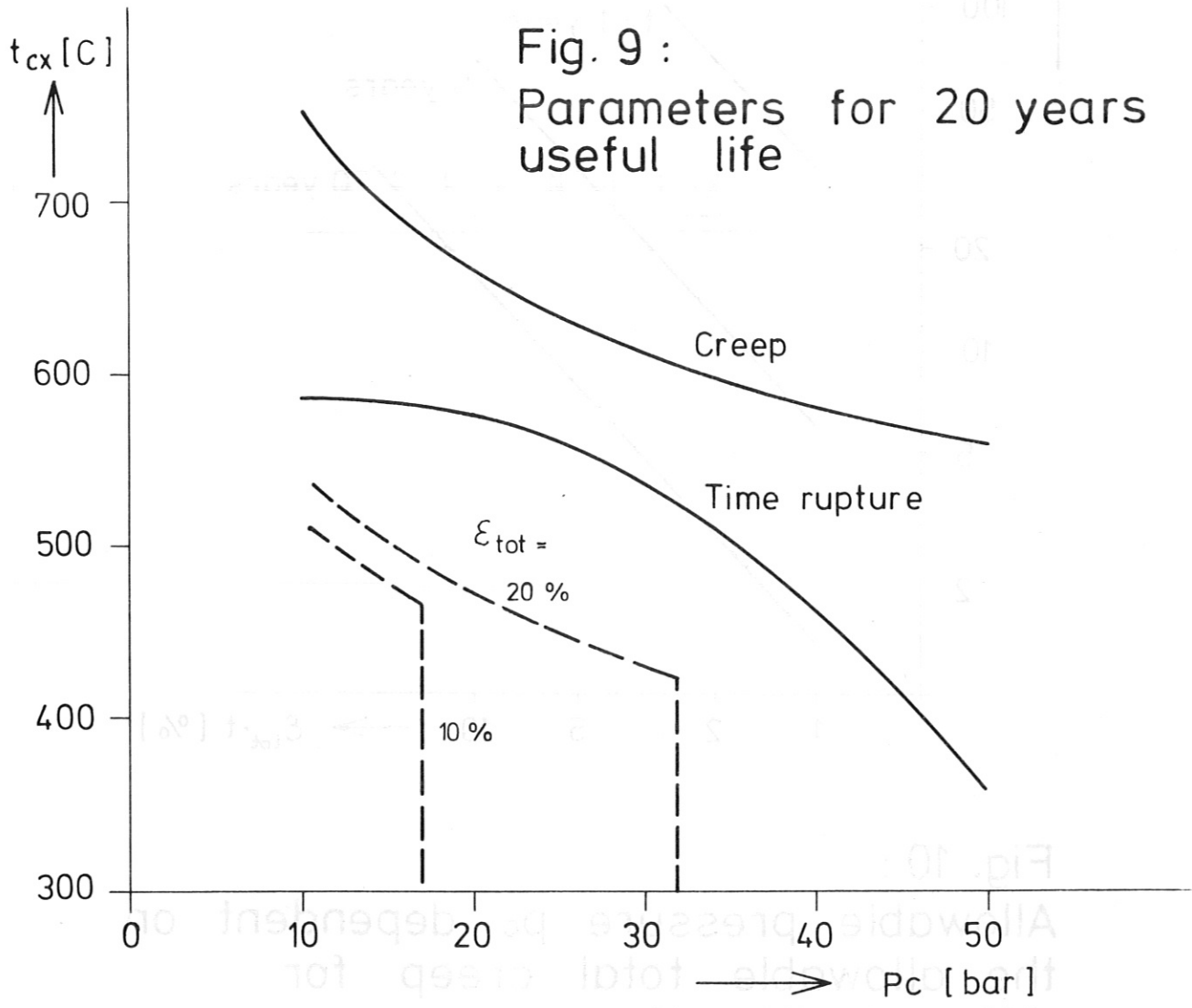


Fig. 9 :
Parameters for 20 years
useful life



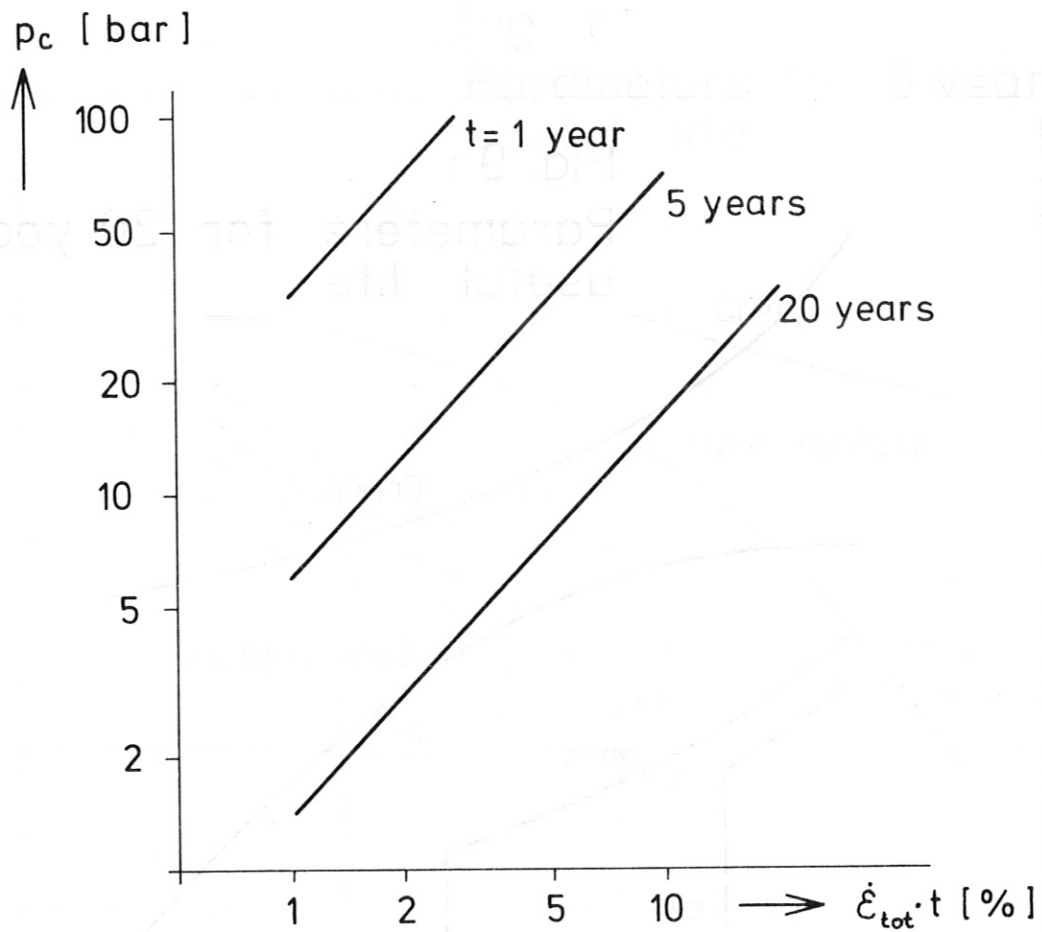


Fig. 10 :

Allowable pressure p_c dependent on the allowable total creep for different life times

A P P E N D I X

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C*****	M A I N *****	10
C		20
C*****	THIS IS A MULTI-PURPOSE INPUT/OUTPUT PROGRAM WHICH CAN BE	30
C	ADJUSTED TO A NUMBER OF SUBROUTINES BY CHANGING ONLY A FEW	40
C	STATEMENTS.	50
C		60
C*****	IN THIS VERSION IT IS APPLIED TO THE PROBLEM OF EVALUATING	70
C	THE LIFETIME EXPECTED FOR THE FIRST WALL OF A CTRD BLANKET	80
C	MODULE CELL.	90
C		100
C*****	AUTHOR: DR.W.M.DAENNER, IPP GARCHING	110
C		120
C*****	INPUT DESCRIPTION:	130
C	CARD 1 NA NUMBER OF ARGUMENTS TO BE READ AND TRANSFERRED	140
C	TO THE CALCULATION SUBROUTINE(S)	150
C	NR NUMBER OF RESULTS TO BE TRANSFERRED FROM THE	160
C	SUBROUTINE(S) AND SHIFTED TO OUTPUT	170
C	NPR NUMBER OF RESULTS TO BE PRINTED (NPR.LE.NR)	180
C	NPR = 0 TOTAL OUTPUT	190
C	NPR > 0 SELECTIVE OUTPUT	200
C	NPL NUMBER OF RESULTS TO BE PLOTTED (MAXIMUM = 10)	210
C	IN THIS VERSION NO PLOT-OUTPUT IS POSSIBLE,	220
C	THEREFORE NPR = 0 HAS TO BE USED.	230
C	CARD 2 A1 LOWER LIMIT OF ARGUMENT RANGE	240
C	A2 UPPER LIMIT OF ARGUMENT RANGE	250
C	DA INCREMENT TO BE USED FOR THE VARIATION OF THIS	260
C	ARGUMENT BETWEEN THE LIMITS A1 AND A2	270
C	THIS CARD HAS TO BE REPEATED FOR EACH OF THE	280
C	NA ARGUMENTS.	290
C	CARD 3 KPR IDENTIFICATION NUMBER(S) OF THE RESULTS TO BE	300
C	PRINTED (MAXIMUM NPR)	310
C	THIS CARD CAN BE OMITTED IF NPR = 0.	320
C	CARD 4 KPL IDENTIFICATION NUMBER(S) OF THE RESULTS TO BE	330
C	PLOTTED (MAXIMUM 10, BUT EQUAL NPL)	340
C	THIS CARD CAN BE OMITTED IF NPL = 0.	350
C		360
C	DIMENSION APR1(10),APR2(10),KPL(10),RPL(10,10,10),ZZ(6)	370
C		380
C*****	THE FOLLOWING CARDS HAVE TO BE ADJUSTED TO THE SPECIAL PROBLEM.	390
C		400
C*****	THE LENGTH OF THE ARRAYS IN THIS DIMENSION STATEMENT HAS TO BE	410
C	EQUAL TO 'NA':	420
C		430
C	DIMENSION A1(17),A2(17),DA(17),NAMEA(17)	440
C		450
C*****	THE LENGTH OF THE ONEDIMENSIONAL ARRAYS AND THAT OF THE LAST	460
C	DIMENSION IN THE THREEDIMENSIONAL ARRAY HAS TO BE EQUAL TO 'NR'.	470
C		480
C	DIMENSION NAMED(34),KPR(34),RPR(10,10,34)	490
C		500
C*****	THE LENGTH OF THE FOLLOWING COMMON ARRAYS HAS TO BE	510
C	'NA' AND 'NR' RESPECTIVELY.	520
C		530
C	COMMON ARG(17)	540
C	COMMON RES(34)	550
C		560
C*****	THE FOLLOWING DATA STATEMENTS SHOULD CONTAIN THE 'NA' VARIABLE	570

C	NAMES OF THE INPUT QUANTITIES AND THE 'NR' VARIABLE NAMES OF THE	580
C	OUTPUT QUANTITIES. THEY ARE ONLY USED TO IDENTIFY THE DIFFERENT	590
C	COLUMNS IN THE PRINT OUTPUT LIST.	600
C		610
	DATA NAMEA/'PWNU','FWBR','PMUL','STRU','PRES','TCI ','TCD ','	620
1	'TCX ','ARR ','DCO ','CHAN','XLEN','PITC','FHW '	630
2	'FHTC','XMAT','FDPA' /	640
	DATA NAMER/'PCK ','SOD ','SW ','DCI ','DHY ','TCM ','ACH ','	650
1	'AC ','AM ','PWBR','PC ','DTC ','XMC ','XVC ','	660
2	'WC ','QW ','HFW ','HTC ','DTCW','TWI ','TWO '	670
3	'TWM ','DTW ','STTH','STTN','STTO','PLM ','ULTR'	680
4	'CRIC','CRSC','ULSC','CRTO','ULC ','ULIC' /	690
C		700
C	***** IF THIS ADJUSTMENT IS DONE IN THE RIGHT WAY THE PROGRAM IS NOW	710
C	READY FOR USE. THE OUTPUT IS SELFEXPLAINING.	720
C		730
	CALL DATE (ZZ)	740
	WRITE(6,206) ZZ	750
	WRITE(6,200)	760
	READ(5,100) NA,NR,NPR,NPL	770
	DO 1 N=1,NA	780
	READ(5,101) A1(N),A2(N),DA(N)	790
1	WRITE(6,201) NAMEA(N),A1(N),A2(N),DA(N)	800
	IF(NPR) 2,3,2	810
2	READ(5,100) (KPR(K),K=1,NPR)	820
3	IF(NPL) 4,5,4	830
4	READ(5,100) (KPL(K),K=1,NPL)	840
5	I=0	850
	DO 6 N=1,NA	860
	IF(DA(N)) 7,6,7	870
7	I=I+1	880
	IF(I-2) 8,9,6	890
8	I1=N	900
	GO TO 6	910
9	I2=N	920
6	CONTINUE	930
	DO 10 N=1,NA	940
10	ARG(N)=A1(N)	950
	LOOP1=1+IFIX((A2(I1)-A1(I1))/DA(I1))	960
	LOOP2=1+IFIX((A2(I2)-A1(I2))/DA(I2))	970
	DO 11 L1=1,LOOP1	980
	ARG(I1)=A1(I1)+FLOAT(L1-1)*DA(I1)	990
	APR1(L1)=ARG(I1)	1000
	DO 12 L2=1,LOOP2	1010
	ARG(I2)=A1(I2)+FLOAT(L2-1)*DA(I2)	1020
	APR2(L2)=ARG(I2)	1030
C		1040
C	***** AT THIS PLACE INSERT YOUR CALCULATION SUBROUTINE(S). *****	1050
C		1060
	CALL WLOAD	1070
	CALL LIFE	1080
C		1090
	IF(NPR) 13,14,13	1100
13	DO 15 L3=1,NPR	1110
	LPR=KPR(L3)	1120
15	RPR(L1,L2,L3)=RES(LPR)	1130
	GO TO 17	1140

```

14 DO 16 L3=1,NR 1150
16 RPR(L1,L2,L3)=RES(L3) 1160
17 IF(NPL) 18,12,18 1170
18 DO 20 L4=1,NPL 1180
    LPL=KPL(L4) 1190
20 RPL(L1,L2,L3)=RES(LPL) 1200
12 CONTINUE 1210
11 CONTINUE 1220
    IF(NPR) 19,21,19 1230
19 NTAB=1+(NPR-1)/8 1240
    GO TO 22 1250
21 NTAB=1+(NR-1)/8 1260
22 DO 23 L1=1,LOOP1 1270
    WRITE(6,202) NAMEA(I1),APR1(L1) 1280
    DO 24 NT=1,NTAB 1290
        WRITE(6,203) 1300
        IF(NPR) 26,25,26 1310
25 NT2=8*NT 1320
    NT1=NT2-7 1330
    IF(NT-NTAB) 28,27,27 1340
27 NT2=NR 1350
28 WRITE(6,204) NAMEA(I2),(NAMER(NN),NN=NT1,NT2) 1360
    WRITE(6,207) 1370
    DO 29 L2=1,LOOP2 1380
29 WRITE(6,205) APR2(L2),(RPR(L1,L2,L3),L3=NT1,NT2) 1390
    GO TO 24 1400
26 NT2=8*NT 1410
    NT1=NT2-7 1420
    IF(NT-NTAB) 30,32,32 1430
32 NT2=NPR 1440
30 WRITE(6,204) NAMEA(I2),(NAMER(KPR(NN)),NN=NT1,NT2) 1450
    WRITE(6,207) 1460
    DO 31 L2=1,LOOP2 1470
31 WRITE(6,205) APR2(L2),(RPR(L1,L2,L3),L3=NT1,NT2) 1480
24 CONTINUE 1490
23 CONTINUE 1500
    STOP 1510
100 FORMAT(8X, 9I4) 1520
101 FORMAT(8X,3F12.5) 1530
200 FORMAT(/' INPUT PARAMETERS'//) 1540
201 FORMAT(' VARIATION OF ',A4,' FROM ',F12.5,' TO ',F12.5,' BY STEPS 1550
    1OF ',F12.5) 1560
202 FORMAT(//' R E S U L T S F O R ',A4,' = ',F12.5) 1570
203 FORMAT(/'5X,' ARGUMENT',7X,' R E S U L T S F O R'//) 1580
204 FORMAT(5X,A4,11X,8(A4,8X)) 1590
205 FORMAT(' ',F12.5,5X,8(1PE12.4)) 1600
206 FORMAT('1DATE = ',2A4/' TIME = ',2A4/' JOB = ',2A4'//) 1610
207 FORMAT(' ') 1620
    END 1630

```

C***** W L O A D *****

C

C**** SUBPROGRAM FOR EVALUATING THE TEMPERATURE AND STRESS LOAD OF THE
C FIRST WALL OF THE CTRD MODULE CELL

C

C***** AUTHOR: DR.W.M.DAENNER, IPP GARCHING

C

C***** INPUT

PWNU	(W/CM2)	NEUTRON WALL LOADING	10
FWBR	(-)	BREMSSTRAHLUNG FACTOR	20
PMUL	(-)	POWER MULTIPLICATION FACTOR OF THE BLANKET	30
STRU	(-)	STRUCTURE MATERIAL VOLUME FRACTION FOR THE CELL WALL	40
			50
PRES	(BAR)	COOLANT PRESSURE	50
TCI	(C)	COOLANT INLET TEMPERATURE	70
TCO	(C)	COOLANT OUTLET TEMPERATURE	80
TCX	(C)	COOLANT TEMPERATURE AT THE FIRST WALL	90
ARR	(-)	ARRANGEMENT OPTION	100
		= 1.0 : CYL. CELLS IN SQUARE MATRIX	110
		> 1.0 : CYL. CELLS IN HEXAGONAL MATRIX	120
DCD	(CM)	OUTER CELL DIAMETER	130
CHAN	(CM)	COOLANT DUCT WIDTH	140
XLEN	(CM)	CELL OR COOLANT DUCT LENGTH	150
PITC	(-)	CELL PITCH RATIO	160
FHW	(1/CM)	HEAT SOURCE DENSITY FACTOR	170
FHTC	(-)	FACTOR FOR ENHANCED HEAT TRANSFER	180
XMAT	(-)	WALL MATERIAL IDENTIFICATION NUMBER	190
FDPA	(DPA*CM2/W*A)	DISPLACEMENT RATE FACTOR	200

C

C***** OUTPUT

PCK	(-)	PACKAGE DENSITY	210
SOD	(-)	RATIO OF WALL THICKNESS AND OUTER CELL DIAMETER	220
			230
SW	(CM)	FIRST WALL THICKNESS	240
DCI	(CM)	INNER CELL DIAMETER	250
DHY	(CM)	COOLANT DUCT HYDRAULIC DIAMETER	260
TCM	(C)	MEAN COOLANT TEMPERATURE	270
ACH	(CM2)	COOLANT DUCT CROSS-SECTION AREA	280
AC	(CM2)	CELL CROSS-SECTION AREA	290
AM	(CM2)	CROSS-SECTION AREA OF MATRIX ELEMENT	300
PWBR	(W/CM2)	BREMSSTRAHLUNG WALL LOADING	310
PC	(W)	TOTAL POWER PER CELL	320
DTC	(C)	COOLANT TEMPERATURE RISE	330
XMC	(G/SEC)	COOLANT MASS FLOW RATE	340
XVC	(CM3/SEC)	COOLANT VOLUME FLOW RATE	350
WC	(CM/SEC)	COOLANT VELOCITY	360
QW	(W/CM3)	HEAT SOURCE DENSITY INSIDE THE FIRST WALL	370
HFW	(W/CM2)	HEAT FLOW FROM WALL TO COOLANT	380
HTC	(W/CM2*C)	HEAT TRANSFER COEFFICIENT	390
DTCW	(C)	TEMPERATURE DIFFERENCE WALL -> COOLANT	400
TWI	(C)	INNER WALL TEMPERATURE	410
TWO	(C)	OUTER WALL TEMPERATURE	420
TWM	(C)	MEAN WALL TEMPERATURE	430
DTW	(C)	TEMPERATURE DIFFERENCE INSIDE THE WALL	440
STTH	(KP/CM2)	MAXIMUM THERMAL STRESS INSIDE THE WALL	450
STTN	(KP/CM2)	TENSILE STRESS INSIDE THE WALL	460
STTO	(KP/CM2)	MAXIMUM TOTAL STRESS INSIDE THE WALL	470

C

C***** SUBPROGRAMS NEEDED: COOLNT

C	HTCOF	580
C	WALL TM	590
C	EXPN	600
C	EMOD	610
C	POIS	620
C		630
C	SUBROUTINE WLOAD	640
C	COMMON PWNU,FWBR,PMUL,STRU,PRES,TCI,TCO,TCX,ARR,DCO,CHAN,XLEN,	650
1	PITC,FHW,FHTC,XMAT,FDPA	660
C	COMMON PCK,SOD,SW,DCI,DHY,TCM,ACH,AC,AM,PWBR,PC,DTC,XMC,XVC,WC,	670
1	QW,HFW,HTC,DTCW,TWI,TWO,TWM,DTW,STTH,STTN,STTO	680
C	CHARACTERISTIC MATRIX AREA FIGURE 'ALPHA'	690
C	IF(ARR-1.0) 1,1,2	700
C	'ALPHA' FOR CYLINDER IN SQUARE MATRIX	710
1	ALPHA=1.0	720
C	GO TO 3	730
C	'ALPHA' FOR CYLINDER IN HEXAGONAL MATRIX	740
2	ALPHA=0.866	750
3	PCK=0.7854/(ALPHA*PITC*PI TC)	760
C	SOD=0.5*(1.0-SQRT(1.0-STRU/PCK))	770
C	SW=SOD*DCO	780
C	DCI=DCO-2.0*SW	790
C	DHY=2.0*CHAN	800
C	TCM=0.5*(TCI+TCO)	810
C	ACH=3.14159*CHAN*(DCI-CHAN)	820
C	AC=0.7854*DCO*DCO	830
C	AM=AC/PCK	840
C	PWBR=PWNU*FWBR	850
C	PC=(PWNU*PMUL+PWBR)*AM	860
C	DTC=TCO-TCI	870
C	CALL COOLNT (TCM,PRES,DENM,CAPM,CONM,VISM)	880
C	XMC=PC/(CAPM*DTC)	890
C	XVC=XMC/DENM	900
C	WC=XVC/ACH	910
C	QW=FHW*PWNU	920
C	HFW=PWBR+QW*SW	930
C	CALL HTCOF (HFW,WC,XLEN,DHY,TCX,PRES,FHTC,HTC,DTCW,TWI)	940
C	CALL WALL TM (TWI,QW,PWBR,DCO,DCI,XMAT,TWO,TWM,DTW)	950
C	STTH=0.5*DTW*EXPN(XMAT,TWM)*EMOD(XMAT,TWM)/(1.0-POIS(XMAT,TWM))	960
C	STTN=0.50986*PRES/SCD	970
C	STTO=STTH+STTN	980
C	RETURN	990
C	END	1000

C*****	L I F E	*****	10
C			20
C	SUBPROGRAM FOR EVALUATING THE USEFUL LIFE OF THE FIRST WALL		30
C	OF THE CTRD MODULE CELL		40
C			50
C*****	AUTHOR: DR.W.M.DAENNER, IPP GARCHING		60
C			70
C*****	INPUT	PWNU (W/CM2) NEUTRON WALL LOADING	80
C		FWBR (-) BREMSSTRAHLUNG FACTOR	90
C		PMUL (-) POWER MULTIPLICATION FACTOR OF THE BLANKET	100
C		STRU (-) STRUCTURE MATERIAL VOLUME FRACTION FOR	110
C		THE CELL WALL	120
C		PRES (BAR) COOLANT PRESSURE	130
C		TCI (C) COOLANT INLET TEMPERATURE	140
C		TCO (C) COOLANT OUTLET TEMPERATURE	150
C		TCX (C) COOLANT TEMPERATURE AT THE FIRST WALL	160
C		ARR (-) ARRANGEMENT OPTION	170
C		= 1.0 : CYL. CELLS IN SQUARE MATRIX	180
C		> 1.0 : CYL. CELLS IN HEXAGONAL MATRIX	190
C		DCO (CM) OUTER CELL DIAMETER	200
C		CHAN (CM) COOLANT DUCT WIDTH	210
C		XLEN (CM) CELL OR COOLANT DUCT LENGTH	220
C		PITC (-) CELL PITCH RATIO	230
C		FHW (1/CM) HEAT SOURCE DENSITY FACTOR	240
C		FHTC (-) FACTOR FOR ENHANCED HEAT TRANSFER	250
C		XMAT (-) WALL MATERIAL IDENTIFICATION NUMBER	260
C		FDPA (DPA*CM2/W*A) DISPLACEMENT RATE FACTOR	270
C			280
C*****	OUTPUT	PLM (-) LARSON-MILLER-PARAMETER	290
C		ULTR (H) USEFUL LIFE DUE TO TIME-RUPTURE-STRENGTH	300
C		CRIC (H-1) IRRADIATION CREEP RATE	310
C		CRSC (H-1) THERMAL CREEP RATE	320
C		ULSC (H) USEFUL LIFE DUE TO THERMAL CREEP	330
C		CRTO (H-1) TOTAL CREEP RATE	340
C		ULC (H) USEFUL LIFE DUE TO TOTAL CREEP	350
C		UL1C (H) TIME TO REACH 1% TOTAL CREEP	360
C			370
C*****	SUBPROGRAMS NEEDED: LARSON		380
C		CIC	390
C		CREEP	400
C			410
C			420
	SUBROUTINE LIFE		430
	COMMON	PWNU,FWBR,PMUL,STRU,PRES,TCI,TCO,TCX,ARR,DCO,CHAN,XLEN,	440
1		PITC,FHW,FHTC,XMAT,FDPA	450
	COMMON	PCK,SOD,SW,DCI,DHY,TCM,ACH,AC,AM,PWBR,PC,DTC,XMC,XVC,WC,	460
1		QW,HFW,HTC,DTCW,TWI,TWO,TWM,DTW,STTH,STTN,STTO,PLM,ULTR,	470
2		CRIC,CRSC,ULSC,CRTO,ULC,UL1C	480
	CALL	LARSON(XMAT,STTO,PLM,C)	490
		ULTR=10.0**{(PLM/(TWI+273.15))-C}	500
		CRIC=CIC(XMAT)*FDPA*PWNU *STTO	510
	CALL	CREEP(XMAT,TWM,CCO,CEX,CL1,CL2)	520
		CRSC=CCO*(0.01*STTO)**CEX	530
		ULSC=10.0**{(CL1 -CL2*ALOG(CRSC))	540
		CRTO=CRIC+CRSC	550
		ULC=10.0**{(CL1 -CL2*ALOG(CRTO))	560
		UL1C=0.01/CRTO	570
		RETURN	580
		END	

C***** H T C O F *****	10
C	20
C**** SUBPROGRAM FOR EVALUATING THE HEAT TRANSFER COEFFICIENT FOR A	30
C HELIUM GAS FLOW	40
C	50
C***** AUTHOR: DR.W.M.DAENNER, IPP GARCHING	60
C	70
C***** INPUT HFW (W/CM2) HEAT FLUX DENSITY AT THE WALL	80
C WC (CM/SEC) COOLANT VELOCITY	90
C XLEN (CM) COOLANT DUCT LENGTH	100
C DHY (CM) HYDRAULIC DIAMETER OF THE COOLANT DUCT	110
C TX (C) COOLANT TEMPERATURE	120
C PR (BAR) COOLANT PRESSURE	130
C FHTC (-) FACTOR FOR ENHANCED HEAT TRANSFER	140
C***** OUTPUT HTC (W/CM2*C) HEAT TRANSFER COEFFICIENT	150
C DTCW (C) TEMPERATURE DIFFERENCE BETWEEN WALL AND	160
C COOLANT	170
C TW (C) WALL TEMPERATURE	180
C	190
C***** SUBPROGRAM NEEDED: COOLNT	200
C	210
SUBROUTINE HTCOF (HFW,WC,XLEN,DHY,DX,PR,FHTC,HTC,DTCW,TW)	220
FWLD=WC**0.8 / (XLEN**0.054 * DHY**0.146)	230
IT=1	240
TF=TX	250
3 CALL COOLNT (TF,PR,DEN,CAP,CON,VIS)	260
FTP=DEN**0.8 * CAP**0.37 * CON**0.63 / VIS**0.43	270
HTC=0.032*FTP*FWLD*FHTC	280
IF(IT-1) 1,1,2	290
1 DT1=HFW/HTC	300
TF=TX+0.5*DT1	310
IT=IT+1	320
GO TO 3	330
2 DT2=HFW/HTC	340
DDT=ABS(DT2-DT1)	350
IF(DDT-1.0) 4,4,5	360
5 DT1=DT2	370
IT=IT+1	380
TF=TX+0.5*DT1	390
GO TO 3	400
4 DTCW=DT2	410
TW=TX+DTCW	420
RETURN	430
END	440

C***** W A L L T M *****	10
C	20
C**** SUBPROGRAM FOR EVALUATING THE TEMPERATURE DIFFERENCE WITHIN THE	30
C FIRST WALL OF THE CTRD MODULE CELL	40
C	50
C***** AUTHOR: DR.W.M.DAENNER, IPP GARCHING	60
C	70
C***** INPUT TW1 (C) INNER WALL TEMPERATURE	80
C QW (W/CM3) HEAT SOURCE DENSITY INSIDE THE WALL	90
C PWBR (W/CM2) HEAT FLUX DENSITY FROM OUTSIDE THE WALL	100
C DCO (CM) OUTER WALL DIAMETER	110
C DCI (CM) INNER WALL DIAMETER	120
C XMAT (-) WALL MATERIAL IDENTIFICATION NUMBER	130
C***** OUTPUT TW2 (C) OUTER WALL TEMPERATURE	140
C TWM (C) MEAN WALL TEMPERATURE	150
C DTW (C) TEMPERATURE DIFFERENCE ACROSS THE WALL	160
C	170
C***** SUBPROGRAM NEEDED: COND	180
C	190
SUBROUTINE WALLTM (TW1,QW,PWBR,DCO,DCI,XMAT,TW2,TWM,DTW)	200
SW=0.5*(DCO-DCI)	210
FTW=SW*(0.5*SW*QW+PWBR)	220
IT=1	230
TW=TW1	240
3 CON=COND(XMAT,TW)	250
IF(IT-1) 1,1,2	250
1 DT1=FTW/CON	270
TW=TW1+0.5*DT1	280
IT=IT+1	290
GO TO 3	300
2 DT2=FTW/CON	310
DDT=ABS(DT2-DT1)	320
IF(DDT-0.5) 4,4,5	330
5 DT1=DT2	340
TW=TW1+0.5*DT1	350
IT=IT+1	360
GO TO 3	370
4 TW2=TW1+DT2	380
TWM=0.5*(TW1+TW2)	390
DTW=TW2-TW1	400
RETURN	410
END	420

C*****	H E L I U M	*****	10
C			20
C****	A SUBROUTINE PROGRAM FOR EVALUATING THE THERMAL PROPERTIES OF		30
C	HELIUM AS A COOLANT		40
C			50
C*****	AUTHOR: DR.W.M.DAENNER, IPP GARCHING		55
C			56
C****	INPUT	T (C) TEMPERATURE	60
C		P (BAR) PRESSURE	70
C****	OUTPUT	DEN (G/CM3) DENSITY	80
C		CAP (W*S/G*K) SPECIFIC HEAT AT CONSTANT PRESSURE	90
C		CON (W/CM*K) THERMAL CONDUCTIVITY	100
C		VIS (G/CM*S) DYNAMIC VISCOSITY	110
C			120
C*****	THE INDICATED PROPERTIES ARE CALCULATED USING FORMULAS		121
C	PRESENTED BY W.ZIMMERER: MAPLIB-FUNKTIONEN ZUR BERECHNUNG DER		122
C	ZUSTANDSGROESSEN VON HELIUM, LUFT, KOHLENDIOXID UND WASSER.		123
C	KFK 1403 (1971)		124
C			125
C	SUBROUTINE COOLNT (T,P,DEN,CAP,CON,VIS)		130
C	TA=T+273.15		140
C	PA=1.0E5*P		150
C	DENSITY 'DEN'		160
C	D=2076.2*(TA/PA+1.13E-5/TA**0.3333-2.37E-2/(TA*TA))		170
C	DEN=1.0E-3/D		180
C	SPECIFIC HEAT AT CONSTANT PRESSURE 'CAP'		190
C	C=5196.0-1043.0*(1.0/TA**1.3333-28300.0/TA**3.0)*1.0E-5*PA		200
C	CAP=1.0E-3*C		210
C	THERMAL CONDUCTIVITY 'CON'		220
C	CON=1.44E-3*(TA/273.15)**0.7*(1.0+2.0E-9*PA)		230
C	DYNAMIC VISCOSITY 'VIS'		240
C	V=1.855E-5*(TA/273.16)**0.68		250
C	VIS=10.0*V		260
C	RETURN		270
C	END		280

C*****	C O N D	*****	10
C			20
C	FUNCTION SUBPROGRAM FOR PROVIDING THE THERMAL CONDUCTIVITY OF		30
C	STRUCTURAL METALS AND ALLOYS		40
C			50
C*****	AUTHOR: DR.W.M.DAENNER, IPP GARCHING		60
C			70
C*****	INPUT XMAT (-) MATERIAL IDENTIFICATION NUMBER		80
C	T (C) TEMPERATURE		90
C			100
C*****	OUTPUT COND (W/CM*C) THERMAL CONDUCTIVITY		110
C			120
C*****	CURRENT CONTENTS:		130
C	=====		140
C	XMAT = 23.XX00 V-XXTI ALLOYS		150
C	FOR PURE VANADIUM AND THE ALLOYS V-3TI AND		160
C	V-20TI POLYNOMIAL APPROXIMATION OF THE DATA		170
C	FROM H.BOEHM E.A., 6. PLANSEE-SEMINAR.		180
C	LINEAR INTERPOLATION FOR OTHER TI-CONTENTS.		190
C			200
	FUNCTION COND (XMAT,T)		210
	IF(IFIX(XMAT)-23) 1,1,1		220
1	TIC=XMAT-FLOAT(IFIX(XMAT))		230
	IF(TIC) 2,2,3		240
3	IF(TIC-0.03) 2,4,5		250
2	C1=0.2899+1.0312E-4*T-2.1875E-8*T*T		260
	IF(TIC) 6,6,4		270
6	COND=C1		280
	RETURN		290
4	C2=0.255+1.8E-4*T-1.0E-7*T*T		300
	IF(TIC-0.03) 8,7,9		310
7	COND=C2		320
	RETURN		330
8	COND=C1-TIC*(C1-C2)/0.03		340
	RETURN		350
5	IF(TIC-0.2) 4,9,9		360
9	C3=0.14375+2.175E-4*T-5.625E-8*T*T		370
	IF(TIC-0.2) 11,10,10		380
10	COND=C3		390
	RETURN		400
11	COND=C2-(TIC-0.03)*(C2-C3)/0.17		410
	RETURN		420
	END		430

C*****	E X P N	*****	10
C			20
C	FUNCTION SUBPROGRAM FOR EVALUATING THE THERMAL EXPANSION COEFFI-		30
C	CIENT OF STRUCTURAL MATERIALS		40
C			50
C*****	AUTHOR: DR.W.M.DAENNER, IPP GARCHING		60
C			70
C*****	INPUT XMAT (-) MATERIAL IDENTIFICATION NUMBER		80
C	TC (C) TEMPERATURE		90
C*****	OUTPUT EXPN (CM-1) THERMAL EXPANSION COEFFICIENT		100
C			110
C*****	CURRENT CONTENTS:		120
C	=====		130
C	XMAT = 23.0000 PURE VANADIUM		140
C	IN 3 TEMPERATURE RANGES LINEAR INTERPOLATION OF		150
C	DATA FROM F.SPERNER, METALL 15,10(1961),988-994		160
C			170
C	FUNCTION EXPN(XMAT,TC)		180
C	IF(IFIX(XMAT)-23) 1,1,1		190
1	IF(TC-500.0) 2,2,3		200
2	EXPN=7.975E-6+3.25E-9*TC		210
	RETURN		220
3	IF(TC-900.0) 4,4,5		230
4	EXPN=9.6E-6+2.0E-9*(TC-500.0)		240
	RETURN		250
5	EXPN=10.4E-6+2.5E-9*(TC-900.0)		260
	RETURN		270
	END		280

C***** E M O D *****	10
C	20
C FUNCTION SUBPROGRAM FOR PROVIDING THE YOUNG'S MODULUS OF	30
C STRUCTURAL MATERIALS	40
C	50
C***** AUTHOR: DR.W.M.DAENNER, IPP GARCHING	60
C	70
C***** INPUT XMAT (-) MATERIAL IDENTIFICATION NUMBER	80
C TC (C) TEMPERATURE	90
C***** OUTPUT EMOD (KP/CM2) YOUNG'S MODULUS	100
C	110
C***** CURRENT CONTENTS:	120
C =====	130
C XMAT = 23.0000 PURE VANADIUM	140
C CONSTANT VALUE DUE TO F.SPERNER, METALL 15,10	150
C (1961),988-994	160
C	170
C FUNCTION EMOD(XMAT,TC)	180
C IF(IFIX(XMAT)-23) 1,1,1	190
C 1 EMOD=1.5E6	200
C RETURN	210
C END	220

C***** P O I S *****	10
C	20
C FUNCTION SUBPROGRAM FOR PROVIDING THE POISSON RATIO OF	30
C STRUCTURAL MATERIALS	40
C	50
C***** AUTHOR: DR.W.M.DAENNER, IPP GARCHING	60
C	70
C***** INPUT XMAT (-) MATERIAL IDENTIFICATION NUMBER	80
C TC (C) TEMPERATURE	90
C***** OUTPUT POIS (-) POISSON RATIO	100
C	110
C***** CURRENT CONTENTS:	120
C =====	130
C XMAT = 23.0000 PURE VANADIUM	140
C CONSTANT VALUE DUE TO KIEFFER, JANG, ETTMAYER:	150
C SONDERMETALLE, SPRINGER-VERLAG WIEN/NEW YORK	160
C 1971	170
C	180
C FUNCTION POIS(XMAT,TC)	190
C IF(IFIX(XMAT)-23) 1,1,1	200
C 1 POIS=0.35	210
C RETURN	220
C END	230

C*****	L A R S O N	*****	100
C			190
C	SUBPROGRAM FOR PROVIDING THE ALLOY CONSTANT C AND THE		200
C	LARSON-MILLER PARAMETER PLM NECESSARY FOR A GIVEN SRESS		300
C			400
C*****	AUTHOR: DR.W.M.DAENNER,IPP GARCHING		401
C			402
C*****	INPUT	XMAT (-) MATERIAL IDENTIFICATION NUMBER	403
C		STR (KP/CM2) STRESS LOAD	404
C*****	OUTPUT	PLM (-) LARSON-MILLER-PARAMETER REQUIRED	405
C		C (-) MATERIAL CONSTANT	406
C			407
C*****	CURRENT CONTENTS:		408
C	=====		409
C	XMAT = 23.XX00	V - XX TI ALLOYS	410
C		CONSTANT VALUE FOR ALLOY CONSTANT C.	411
C		POLYNOMIAL APPROXIMATION FOR V-5TI AND V-20TI	412
C		CORRESPONDING TO DATA OF H.BOEHM, M.SCHIRRA:	413
C		J.LESS-COMM.MET.,12(1967),280-293. LINEAR	414
C		INTERPOLATION FOR OTHER TITANIUM CONTENTS.	415
C			416
	SUBROUTINE LARSON(XMAT,STR,PLM,C)		500
	IF(IFIX(XMAT)-23) 1,1,1		600
1	C=15		700
	TIC=XMAT-FLOAT(IFIX(XMAT))		800
	P5=18143.0+5.206E-2*STR-2.837E-4*STR*STR		900
	P20=19933.0-2.402*STR+4.436E-4*STR*STR-2.658E-8*STR*STR*STR		1000
	PLM=P5+(P20-P5)*(TIC-0.05)/0.15		1100
	RETURN		1200
	END		1300

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C***** C I C ***** 10
C 20
C FUNCTION SUBPROGRAM FOR SUPPLYING THE IRRADIATION CREEP CONSTANT 30
C OF A STRUCTURAL METAL OR ALLOY 40
C 50
C***** AUTHOR: DR.W.M.DAENNER, IPP GARCHING 60
C 70
C***** INPUT XMAT (-) MATERIAL IDENTIFICATION NUMBER 80
C***** OUTPUT CIC (-) IRRADIATION CREEP CONSTANT 90
C 100
C***** CURRENT CONTENTS: 110
C ===== 120
C XMAT = 26.0000 STAINLESS STEEL 316SS 130
C CONSTANT VALUE DUE TO J.R.STANBRIDGE E.A.: 140
C CLM-R 127 (1974) 150
C 160
C FUNCTION CIC(XMAT) 170
C IF(IFIX(XMAT)-26) 1,1,1 180
1 CIC=1.679E-11 190
C RETURN 200
C END 210

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C*****	C R E E P	*****	10
C			20
C	SUBROUTINE FOR EVALUATING THE SECONDARY CREEP PARAMETERS FOR		30
C	STRUCTURAL METALS AND ALLOYS		40
C			50
C*****	AUTHOR: DR.W.M.DAENNER, IPP GARCHING		60
C			70
C*****	INPUT	XMAT (-) MATERIAL IDENTIFICATION NUMBER	80
C		TC (C) TEMPERATURE	90
C*****	OUTPUT	C (-) CREEP CONSTANT FOR NORTON'S EQUATION	100
C		EX (-) STRESS EXPONENT FOR NORTON'S EQUATION	110
C		C1 (-) MATERIAL CONSTANT	120
C		C2 (-) MATERIAL CONSTANT	130
C		C1 AND C2 ARE FOR LIFE CALCULATION	140
C		DUE TO M.SCHIRRA: KFK 1925 (1974).	150
C			160
C*****	CURRENT CONTENTS:		170
C	=====		180
C	XMAT = 26.0000	STAINLESS STEEL	190
C		C1 AND C2 ARE ARBITRARILY TAKEN TO BE THE SAME	200
C		AS FOR VANADIUM (SEE BELOW). C AND EX TEMPER-	210
C		ATURE DEPENDENT DUE TO G.SCHMIDT: KFK 808 (1968)	220
C	XMAT = 23.XX00	V - XX TI ALLOYS	230
C		CONSTANT VALUES FOR C1 AND C2. EVALUATION OF	240
C		C AND EX FOR PURE VANADIUM AND THE ALLOYS	250
C		V-2.8TI, V-5TI, AND V-20TI FROM DATA PRESENTED	260
C		BY H.BOEHM, M.SCHIRRA: J.LESS-COMM.METALS 12	270
C		(1967), 280-293. NO INTERPOLATION FOR OTHER	280
C		TITANIUM CONTENTS.	290
C			300
C	SUBROUTINE CREEP(XMAT,TC,C,EX,C1,C2)		310
C	TK=273.15+TC		320
C	IF(IFIX(XMAT)-26) 2,1,2		330
1	C1=-1.55		340
C	C2=1.1		350
C	C=1.0E-12** (923.15/TK) * 10.0** (15.0*(1.0-923.15/TK))		360
C	EX=5538.0/TK		370
C	RETURN		380
2	C1=-1.55		390
C	C2=1.1		400
C	TIC=XMAT-FLOAT(IFIX(XMAT))		410
C	IF(TIC) 3,3,4		420
3	EX=4.4		430
C	C=9.3E6*EXP(-6.5E4/(1.987*TK))		440
C	RETURN		450
4	IF(TIC-0.03) 5,5,6		460
5	EX=9.3		470
C	C=3.65E-10*EXP(-3.1E4/(1.987*TK))		480
C	RETURN		490
6	IF(TIC-0.05) 7,7,8		500
7	EX=8.4		510
C	C=4.5E-4*EXP(-5.1E4/(1.987*TK))		520
C	RETURN		530
8	EX=4.2		540
C	C=7.0E22*EXP(-13.6E4/(1.987*TK))		550
C	RETURN		560
C	END		570