CO oxidation on metal-supported monolayer oxide films:
Do we know which phase (interface) is active?

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Abstract. Ultrathin ("monolayer") films of transition metal oxides grown on metal substrates have recently received considerable attention as promising catalytic materials, in particular for low-temperature CO oxidation. The reaction rate on such systems often increases when the film only partially covers the support, and the effect is commonly attributed to the formation of active sites at the metal/oxide boundary. By studying the structure and reactivity of FeO(111) films on Pt(111), here we show that, independently of the film coverage, CO oxidation takes place at the interface between reduced and oxidized phases in the oxide film formed under reaction conditions. The promotional role of a metal support is to ease formation of the reduced phase via reaction between CO adsorbed on metal and oxygen at the oxide island edge.

Keywords Ultrathin oxide films; iron oxide; CO oxidation; Metal/oxide interface; Inverse catalysts.

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Ultrathin films of transition metal oxides (TMO) supported on metal substrates show unique properties due to its reduced dimensionality and strong interaction with underlying support.[1] Among those, a "monolayer" FeO(111) film grown on Pt(111) is one of the most explored systems.[2] The interest in the films was reinforced after they showed considerable activity in CO oxidation at low temperatures.[3] Experimental studies and density functional theory (DFT) calculations[4] showed that a FeO(111) film transforms into an "O-rich" film with a close to FeO_2 stoichiometry and considered, for simplicity, as an O-Fe-O tri-layer. In fact, the O-rich films have a more complex structure, where FeO_2 tri-layer domains are "embedded" into the FeO film following the Moiré pattern present in FeO(111)/Pt(111)[5] (see Fig. 1). The reaction mechanism was rationalized in terms of direct CO reaction with a weakly bonded oxygen atom in the topmost layer thus forming CO_2 that desorbs and leaves oxygen vacancy behind which is then replenished by reaction with molecular O_2 in the gas phase.

The situation becomes more complex at sub-monolayer (sub-ML) film coverages as the surface additionally exposes oxide/metal interfacial cites which are commonly considered to be active. Bao and coworkers[6] addressed reactivity of FeO(111) islands on Pt(111) by scanning tunneling microscopy (STM) and ultraviolet photoelectron spectroscopy (UPS) under ultrahigh vacuum (UHV) conditions. Following the reaction by O_2 exposure to the CO pre-saturated surface and measuring CO consumption rate with UPS, the authors found a linear relationship between the activity and the island perimeter. This finding allowed the authors to conclude that the reaction takes place at the FeO(111)/Pt(111) boundary. The mechanism was studied by DFT, first for FeO(111)[6] and then for other TMO(111) islands on Pt(111).[7] It has been proposed that O_2 dissociates at coordinatively unsaturated Fe sites present at the island edges, thus resulting in O atoms bound both to Fe and Pt, which react with CO adsorbed on Pt to form CO_2. A recent STM study showed that the Fe-terminated edges may, indeed, host the catalytically active sites.[8]

Although this mechanism may be operative under UHV-based conditions, it hardly holds true at the realistic pressures, which force the formation of the FeO_{2-x} phase. Indeed, the transformation of a dense FeO film into FeO_{2-x} sets in at O_2 pressures as low as 10^{-2} mbar at 300 K[4] and even in 10^{-6} mbar O_2 at ~ 600 K for FeO islands.[9]

Pan et al.[10] studied CO oxidation on the FeO(111)/Pt(111) films at near atmospheric pressures as a function of the film coverage. The activity vs coverage plot showed a

maximum at ~ 0.4 ML, indicating that the oxide/metal boundary is more active than the film surface itself. In addition, temperature programmed reaction experiments only showed CO_2 production on "oxidized" FeO_{2-x} films, with a maximum reached at nearly the same coverage. All these results provided strong evidence that it is the FeO_{2-x}/Pt interface that catalyzes the reaction under realistic conditions. Accordingly, the enhanced reactivity was attributed to the reaction between CO adsorbed on Pt and oxygen species at the FeO_2 edges.

Even though certain progress has recently been reached towards understanding the reactivity of the FeO(111)/Pt(111) systems (see also recent studies[11]), the reaction mechanism remains poorly understood. Basically, all proposed models considered solely the boundary between the oxide (either FeO or FeO_2) and the metal surface. In this work, we provide experimental results showing that the CO oxidation reaction takes place not at oxide/metal interface, but at sites provided by the boundary between reduced and oxidized FeO_x phases in the film, whose formation is promoted by the metal support at the initial stages of the reaction.

Figure 1a displays a typical STM image of a sub-ML "oxidized" FeO_{2-x} film, showing both individual islands and extended patches decorating terrace edges of the Pt(111) surface. A superstructure imaged as periodic protrusions of ~ 2.5 nm in size uniformly covers the oxide surface. The protruding spots reflect areas with a FeO_2 tri-layer structure within the Moiré pattern[5] (see the model in Fig. 1). Although we could not achieve high resolution to address the atomic structure of the island edges, analysis of numerous STM images (see also refs.[12]) suggest that the island edges are rarely represented by partial protrusions (see inset in Fig. 1a), indicating that an interface formed by a FeO_2 tri-layer domain to the Pt surface is a minority species. Recent STM studies[5b, 13] also found a rich variety of edge structures that may exist depending on the preparation and exposure conditions.
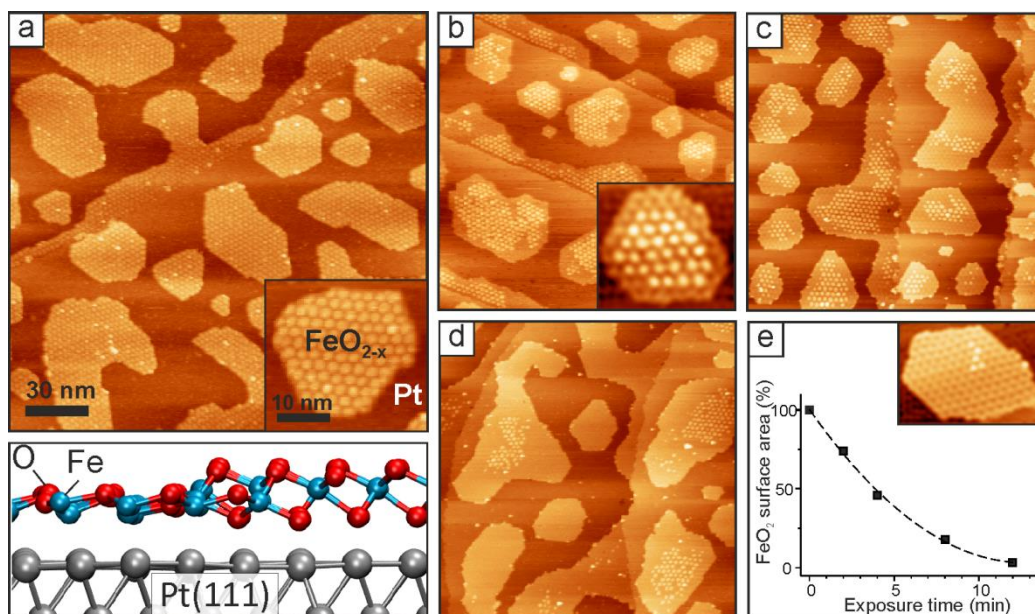
Figure 1. (a) Typical morphology of oxidized FeO_{2-x} films on Pt(111) at sub-monolayer coverages. The cross-view of a FeO_{2-x}/Pt(111) film is shown below. (b-d) STM images of 0.5 ML FeO_{2-x}/Pt(111) film exposed to 10^{-6} mbar CO at 450 K for 2 min (b), additional 2 min (i.e., 4 min in total) (c), and 8 min in total (d); Image size is 150 nm x 150 nm; tunneling bias and current are - 3 V and 0.1 nA, respectively. (e) The FeO_{2-x} surface area normalized to the area in the "as prepared" sample as a function of the accumulative exposure time.

We first studied the interaction of FeO_{2-x} islands with pure CO by exposing it to 10^{-6} mbar at 450 K for 2 min several times. Room temperature STM images, after each treatment (Fig. 1(b-d)), clearly show that CO induced reduction starts at the island edge and propagates into the interior region, leaving a FeO(111)-like structure behind (insets in Fig. 1b,e). Apparently, the reaction front velocity depends on edge orientation and size. Figure 1e depicts global kinetics of the FeO_{2-x} -> FeO reduction by measuring the FeO_{2-x} surface area normalized to the total islands area. The best fit revealed areas decaying as $1 - 0.15t + 0.006t^2$ (the R factor 0.995). Such a decay is typical for etching of two-dimensional islands involving bond-breaking at the edge sites following a first-order kinetics. [14] In our case, the rate-limited step is most likely the CO reaction with an O atom to form CO_2, which desorbs into the gas phase. Indeed, the reaction considerably slows down by lowering the temperature (see Figure S1 in Supporting Information).

In addition, the STM images (Fig. 2) show that the protrusions disappear starting from the side that is closer to the island edge. The interior region remains intact, indicating that no reaction occurs on the oxidized FeO_{2-x} surface. The progressive reduction only takes place at the interface between the compact FeO_{2-x} domain and the reduced surface formed by the reaction with CO.
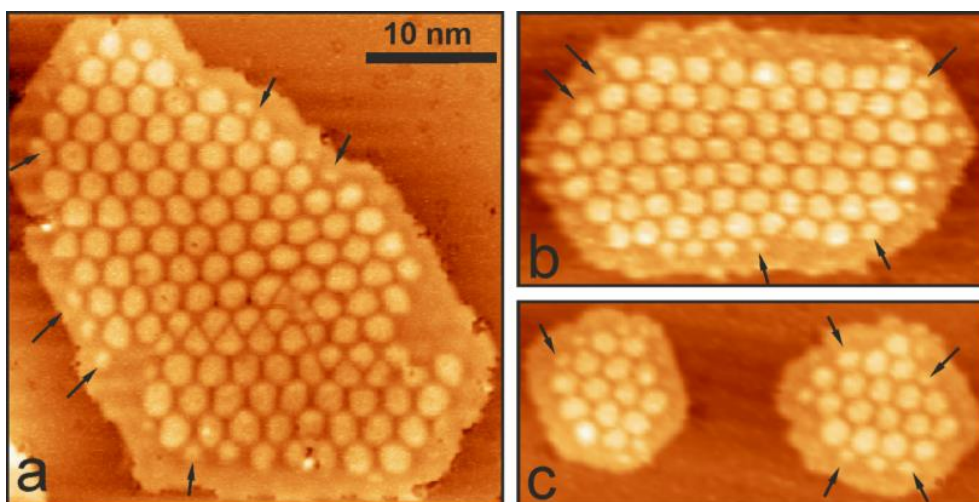
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Figure 2. STM images of the FeO$_{2-x}$ islands after exposure to 10^{-6} mbar CO at 400 K for 4 min. Tunneling conditions: bias -3 V, current 0.05 nA. The arrows highlight non-uniform disappearance of the Moiré spots close to the island edge.

To rationalize this behavior, we invoke DFT calculations[10] for periodic structures which found the lowest O vacancy formation energy (E_{vac} = 1.31 eV) for the oxygen atoms at the "intrinsic" FeO$_2$/FeO interface within the FeO$_{2-x}$ structure. The respective energies for oxygen atoms at the FeO$_2$/Pt edge are 1.58 and 1.53 eV, i.e. considerably lower than for surface oxygen in the FeO$_2$ tri-layer (1.67 eV), and all these are much more weakly bound than the O atoms at the FeO/Pt edge (2.15 eV). The proposed scenario is schematically shown in Fig. 3. For simplicity, we considered the exposed FeO$_2$/metal interface. First, CO adsorbed on Pt reacts with the O atom at the island edge. The Fe atoms around the O vacancy relax and bind to Pt locally forming a FeO structure. However, further CO reaction with this site is hardly possible (under the assumptions made) as it requires about 2.15 eV to extract oxygen from FeO/Pt. Therefore, the reaction pathway involving CO on Pt(111) is terminated. However, a new interface is created which is similar to the "intrinsic" FeO/FeO$_2$ one. Extraction of oxygen at this interface only costs 1.31 eV,[10] thus rendering the reaction with CO feasible.
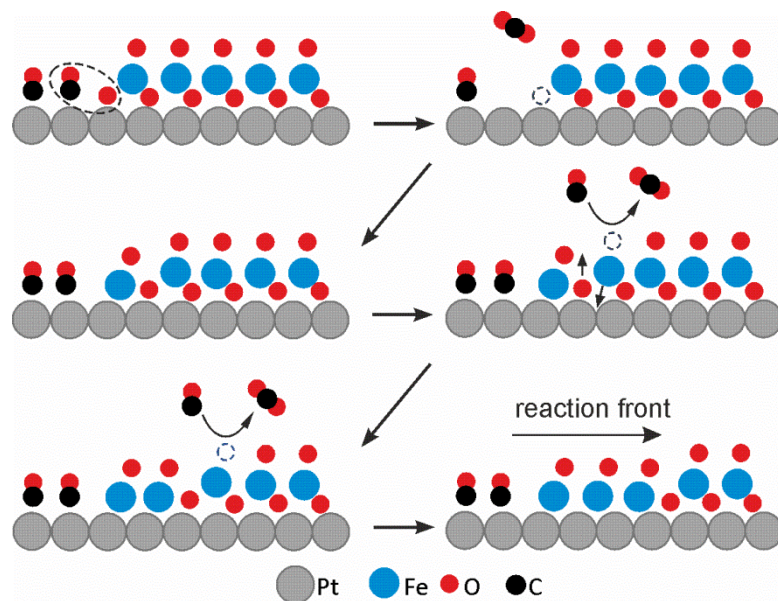
Figure 3. Schematic scenario for CO reaction with FeO_{2-x}/Pt(111) islands . CO adsorbed on Pt reacts with the O atom at the island edge to form CO_2. The atoms surrounding the O vacancy relax and locally form a FeO bilayer structure. Further reaction occurs at the interface between the original oxidized FeO_2 and the reduced FeO-like phases, providing the most weakly bonded oxygen (see text).

CO may either react directly with the O atom via an Eley-Rideal type mechanism or first adsorb on a neighboring low-coordinated Fe atom and then react via a Langmuir-Hinshelwood type mechanism. The process further repeats itself, and the interface spatially shifts towards the interior region. In principle, such a scenario can be adopted for other possible FeO_x/Pt structures at the island edge in the oxidized films. In addition, once triggered the reaction may also propagate *along* the island edge via the same mechanism. Note that, even though the surface left behind the reaction front showed STM fingerprints of FeO(111), its atomic structure and hence interface to the FeO_{2-x} domains may not be exactly the same as discussed in the schematic model. However, we took the scheme as a guideline.

In the next step, we studied morphological changes induced by mixtures of CO and O_2. After exposure to a stoichiometric $CO/O_2 = 2/1$ mixture at 450 K, STM images showed the same morphology as in pure CO (*cf.* Fig. 4b and 4c). Subsequent 6 min exposure to the O-rich mixture ($CO/O_2=1/5$) caused no further changes in the film (*cf.* Fig. 4c and 4d), whereas the film would be fully reduced in pure CO (Fig. 1). Obviously, oxygen in excess suppresses oxide reduction. On the other hand, the "front line" does not return back toward

the island edges, either. Moreover, STM images of the samples directly exposed to the O-rich mixtures (CO/O_2 = 1/5 and 1/9, see Fig. 4e,f) show that even excess of oxygen in the mixture cannot prevent edges from reduction. Although the reaction does not propagate far from the island edge, it is clear that the "oxidized" FeO_{2-x}/Pt interface does not survive under reaction conditions.
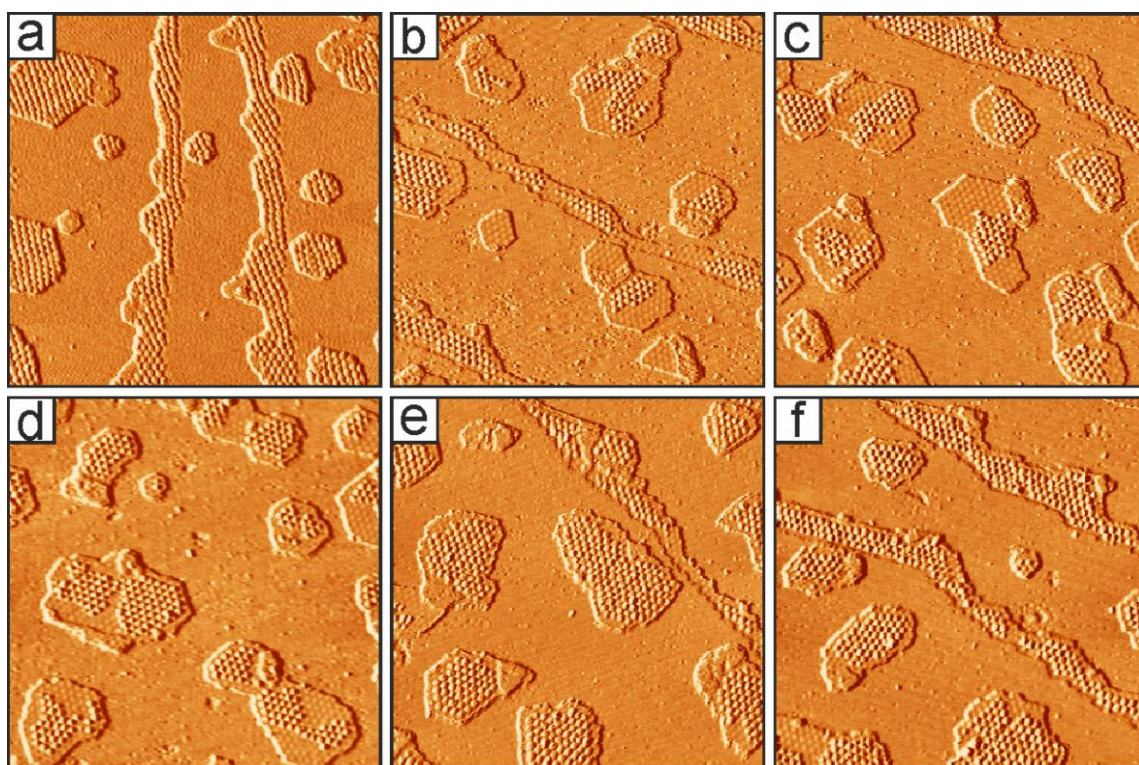

Figure 4. STM images (presented in differentiated contrast) of the oxidized FeO_{2-x} sample before (a) and after exposure to 10^{-6} mbar CO (b). The same sample was re-oxidized and exposed to the mixture of 5×10^{-7} mabr O_2 and 10^{-6} mbar CO (c). Shown on (d) is the same sample as on (c) after additional exposure to the mixture of 5×10^{-6} mbar O_2 and 10^{-6} mbar CO. (e) Oxidized FeO_2 sample after exposure to the mixture of 5×10^{-6} mbar O_2 and 10^{-6} mbar CO. (f) Oxidized FeO_2 sample after exposure to the mixture of 9×10^{-6} mbar O_2 and 10^{-6} mbar CO. All gas exposures were performed at 450 K for 6 min. (Image size is 150 nm × 150 nm, the tunneling conditions are (a) -2.3 V, 1.9 nA; (b-d) -2.1 V, 1.9 nA; (e,f) -3.0 V, 1.9 nA).

To link the observed structure and reactivity, we carried out reaction tests in another UHV chamber using a quadrupole mass-spectrometer (QMS) placed close (~ 1 mm) to the film surface. Figure 5 (red curve) depicts the CO_2 signal recorded on the 0.5 ML FeO_{2-x}/Pt(111) sample in an O-rich mixture consisting of 1×10^{-6} mbar of CO and 5×10^{-6} mbar of O_2. After the crystal was heated up to the reaction temperature (450 K), sustained CO_2 production is observed, suggesting that the prepared surface is catalytically active. The reaction rate stays fairly constant, at least within the first 10 min. In the next experiment,

we stopped the oxygen flow after 2 min of the reaction at 450 K (Fig. 5, black curve). As expected, the CO_2 production drops to zero and recovers after re-introducing O_2. However, the CO_2 signal does not reach the original level. This is more clearly seen by switching oxygen off and on. The steady-state activity almost linearly decreases with increasing the number of "pulses". Bearing in mind that FeO_{2-x} islands in pure CO ambient (i.e. when O_2 is off) exhibit progressive reduction propagating from the edge (see Fig. 2), which cannot be recovered even in oxygen-rich ambient (see Fig. 4c,d), the observed rate attenuation can readily be explained by decreasing the total perimeter length of the boundary between the oxidized, FeO_{2-x} and reduced, FeO-like phases. Therefore, the results provide compelling evidence that this interface provides the most active sites.
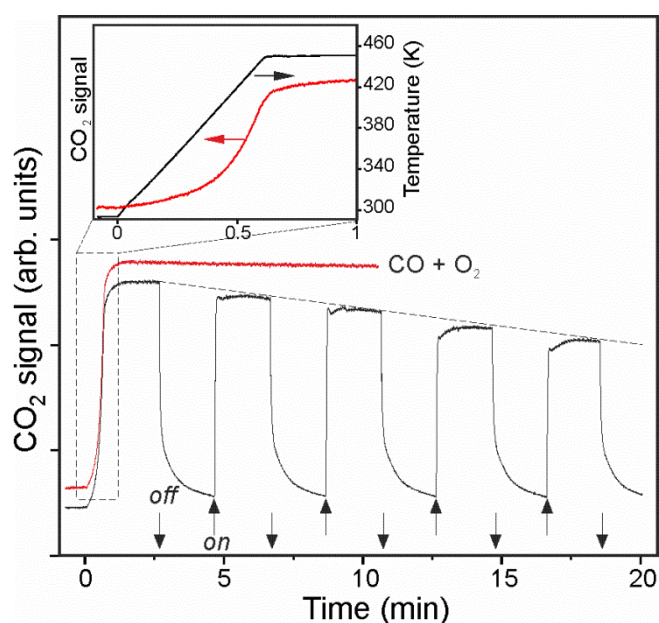

Figure 5. CO_2 (44 amu) signal measured by QMS in front of the sub-monolayer FeO_{2-x} film. The reaction (CO/O_2=1/5) mixture consisting of 10^{-6} mbar of CO and 5×10^{-6} mbar of O_2. At time zero, the sample was heated up to the reaction temperature 450 K with a rate 2 K/s. The red curve shows CO_2 production under steady state conditions. Black curve shows CO_2 kinetics upon switching O_2 flow in the mixture off and on, as indicated by the arrows, while keeping the CO pressure constant.

It is important to note that a similar picture has previously been observed by STM on continuous FeO_{2-x} films on Pt(111) treated in the mbar pressure range, where compact domains of FeO_{2-x} were surrounded by areas formed upon reduction with CO.[15] It has been proposed that the reduction process starts on some defects. In the case of partially covered films, the transformation is definitely triggered at the island edges via reaction with CO adsorbed on Pt. Certain similarities between the results on dense FeO films at high

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pressures[15] and those obtained in this study for FeO islands at low pressures, suggest that the reaction in both cases occurs at the interface between the reduced (red) and the oxidized (ox) phases of the film formed under reaction conditions, i.e. on the "oxide$_{red}$/oxide$_{ox}$" interface, irrespectively of the film coverage. The observed coverage effect[10] may be linked to the easy formation of the reduced phase via reaction between CO adsorbed on Pt and oxygen at the oxide island edge in partially covered films. Therefore, Pt may be considered as a promoter for the formation of the active interface rather than directly participating in the catalytic reaction.

In summary, by studying structure and reactivity of sub-monolayer FeO(111)/Pt(111) films we provide strong evidence that the CO oxidation reaction primarily takes place at the "oxide/oxide" interface in the film, the formation of which is promoted by the Pt support, and not directly at the oxide/Pt interface as considered previously. In addition, the results show that certain precautions must be taken into account while applying the concept of "inverted", i.e. oxide-on-metal model catalysts to mimic reactions occurring on conventional metal catalysts supported on transition metal oxides. It becomes evident that the two-dimensional character of the iron oxide film is important for the structure-reactivity relationships observed. Indeed, iron oxide nanoparticles formed on Pt(111) as a result of FeO film dewetting (ref.[3]) were found to be inactive, although expose a metal/oxide interface. This may hold true for other ultrathin film systems as well. Finally, our results may shed light on the origin of reactivity of unsupported iron oxide catalysts, whose activity is usually associated with low-coordinated sites which may readily be formed at the above-discussed "oxide/oxide" interfaces.

Experimental section

The experiments were performed in an UHV chamber equipped with LEED, AES, and STM. FeO(111) films were prepared using the recipe reported in ref. [9] by evaporating Fe onto the Pt surfaces in 1.3×10^{-7} mbar O$_2$ at 300 K followed by annealing in 1×10^{-6} mbar O$_2$ at 700 K for 5 min. "As prepared" FeO(111) samples were oxidized in 5×10^{-6} mbar O$_2$ at 570 K to transform into the FeO$_{2-x}$ phase. STM images were obtained using electrochemically etched W-tips. Reactivity measurements were performed in another UHV chamber

equipped with QMS. The films were prepared in the same manner as described above, and the coverage was determined by CO titration of Pt(111) using TPD spectra of 2 L CO adsorbed at 300 K.[10]

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Ultrathin Films in Action

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CO oxidation on metal-supported monolayer oxide films: Do we know which phase (interface) is active?
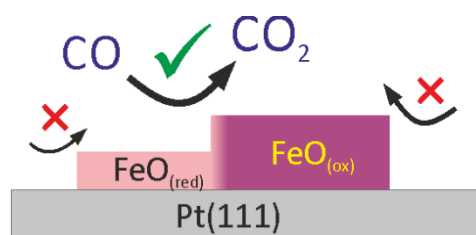
Enhanced activity of "monolayer" transition metal oxides on metal surfaces in low-temperature CO oxidation is often described in terms of active sites at the metal/oxide boundary. By studying the structure and reactivity of FeO(111) films on Pt(111), we show that, independently of the film coverage, CO oxidation takes place at the interface between reduced and oxidized phases of the oxide film formed under reaction conditions. The promotional role of a metal support is to ease formation of the reduced phase via reaction between CO adsorbed on metal and oxygen at the oxide island edge.