

## Electroreduction

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Selective CO<sub>2</sub> Electroreduction to Ethylene and Multicarbon Alcohols via Electrolyte-Driven Nanostructuring

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**Abstract:** Production of multicarbon products (C<sub>2+</sub>) from CO<sub>2</sub> electroreduction reaction (CO<sub>2</sub>RR) is highly desirable for storing renewable energy and reducing carbon emission. The electrochemical synthesis of CO<sub>2</sub>RR catalysts that are highly selective for C<sub>2+</sub> products via electrolyte-driven nanostructuring is presented. Nanostructured Cu catalysts synthesized in the presence of specific anions selectively convert CO<sub>2</sub> into ethylene and multicarbon alcohols in aqueous 0.1M KHCO<sub>3</sub> solution, with the iodine-modified catalyst displaying the highest Faradaic efficiency of 80% and a partial geometric current density of ca. 31.2 mA cm<sup>-2</sup> for C<sub>2+</sub> products at -0.9 V vs. RHE. Operando X-ray absorption spectroscopy and quasi in situ X-ray photoelectron spectroscopy measurements revealed that the high C<sub>2+</sub> selectivity of these nanostructured Cu catalysts can be attributed to the highly roughened surface morphology induced by the synthesis, presence of subsurface oxygen and Cu<sup>+</sup> species, and the adsorbed halides.

## Introduction

The electrochemical production of fuels and chemical feedstocks from CO<sub>2</sub> and water using the electricity derived from renewable energy holds promise as a sustainable process that might help to mitigate some of our current energy and climate challenges. CO<sub>2</sub> electroreduction reaction (CO<sub>2</sub>RR)

to multicarbon hydrocarbons and oxygenates (C<sub>2+</sub>) with high energy density is highly desirable, but is severely limited by the slow kinetics of multiple proton and electron transfer steps during C–C coupling.<sup>[1–4]</sup> Cu, among the studied metals, is the only one producing hydrocarbons and alcohols in considerable amounts. However, polycrystalline Cu usually suffers from high overpotential and low C<sub>2+</sub> selectivity.<sup>[5]</sup> The formation of C<sub>2+</sub> products during CO<sub>2</sub>RR has been found to be extremely sensitive to the catalyst structure.<sup>[1,6–8]</sup> Therefore, nanostructured electrocatalysts capable of efficient generation of multicarbon products from CO<sub>2</sub>RR might be developed through rational design.

It is well-known that the activity and selectivity of CO<sub>2</sub>RR catalysts strongly depend on the precise control of their structure, such as the content of defects,<sup>[9]</sup> subsurface oxygen or Cu<sup>+</sup> species,<sup>[10–15]</sup> the specific shape of the nanocrystals,<sup>[16–18]</sup> or the surface composition and atomic ordering in bimetallic nanostructures.<sup>[19,20]</sup> Previous experimental and theoretical studies demonstrated that Cu(100) is the most favorable crystal orientation for the C–C coupling process.<sup>[21–23]</sup> However, the surface of Cu electrodes under electrochemical environments often undergoes reconstructions induced by applied potentials,<sup>[24–26]</sup> the intermediates formed during CO<sub>2</sub>RR,<sup>[27]</sup> as well as specifically adsorbed anions.<sup>[28–30]</sup> On the other hand, some anions either present in the electrolyte<sup>[12,31,32]</sup> or adsorbed on the electrode surface,<sup>[33]</sup> have been shown to play a vital role in the dynamic evolution of the catalyst structure under reaction conditions as well as the activity and selectivity of CO<sub>2</sub>RR. These important findings might be used in the design and development of new electrodes via electrochemical modifications.

The morphology and composition of electrochemically synthesized catalysts is strongly affected by the applied potential and electrolyte employed.<sup>[34]</sup> Herein we report an electrolyte-driven nanostructuring strategy for the facile synthesis of highly selective CO<sub>2</sub>RR catalysts. The nanostructured Cu catalysts synthesized in the presence of specific anions can selectively convert CO<sub>2</sub> to ethylene and multicarbon alcohols in aqueous 0.1M KHCO<sub>3</sub> solution, with the KI-pretreated catalyst displaying the highest FE of about 80% and partial current density of about 31.2 mA cm<sup>-2</sup> for C<sub>2+</sub> products at -0.9 V vs. RHE. The high C<sub>2+</sub> selectivity of these nanostructured Cu catalysts is attributed to their rough morphology, the presence of subsurface oxygen, Cu<sup>+</sup> species, and adsorbed halides on the surface.

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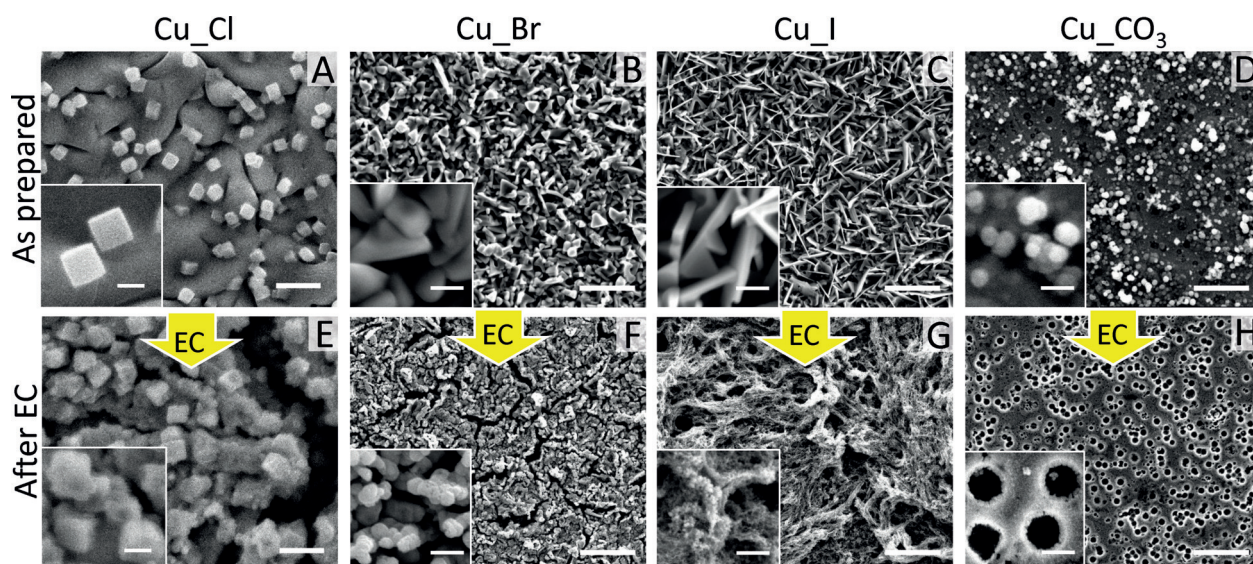
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**Figure 1.** SEM images of Cu<sub>Cl</sub>, Cu<sub>Br</sub>, Cu<sub>I</sub>, and Cu<sub>CO<sub>3</sub></sub> samples before and after 1 h of CO<sub>2</sub>RR at  $-1.0$  V vs. RHE in a CO<sub>2</sub>-saturated 0.1 M KHCO<sub>3</sub> solution. The scale bars in the main images and insets are 1  $\mu$ m and 200 nm for the Cu<sub>Cl</sub> sample (A,E), 5  $\mu$ m and 500 nm for Cu<sub>Br</sub> (B,F), Cu<sub>I</sub> (C,G), and Cu<sub>CO<sub>3</sub></sub> samples (D,H).

## Results and Discussion

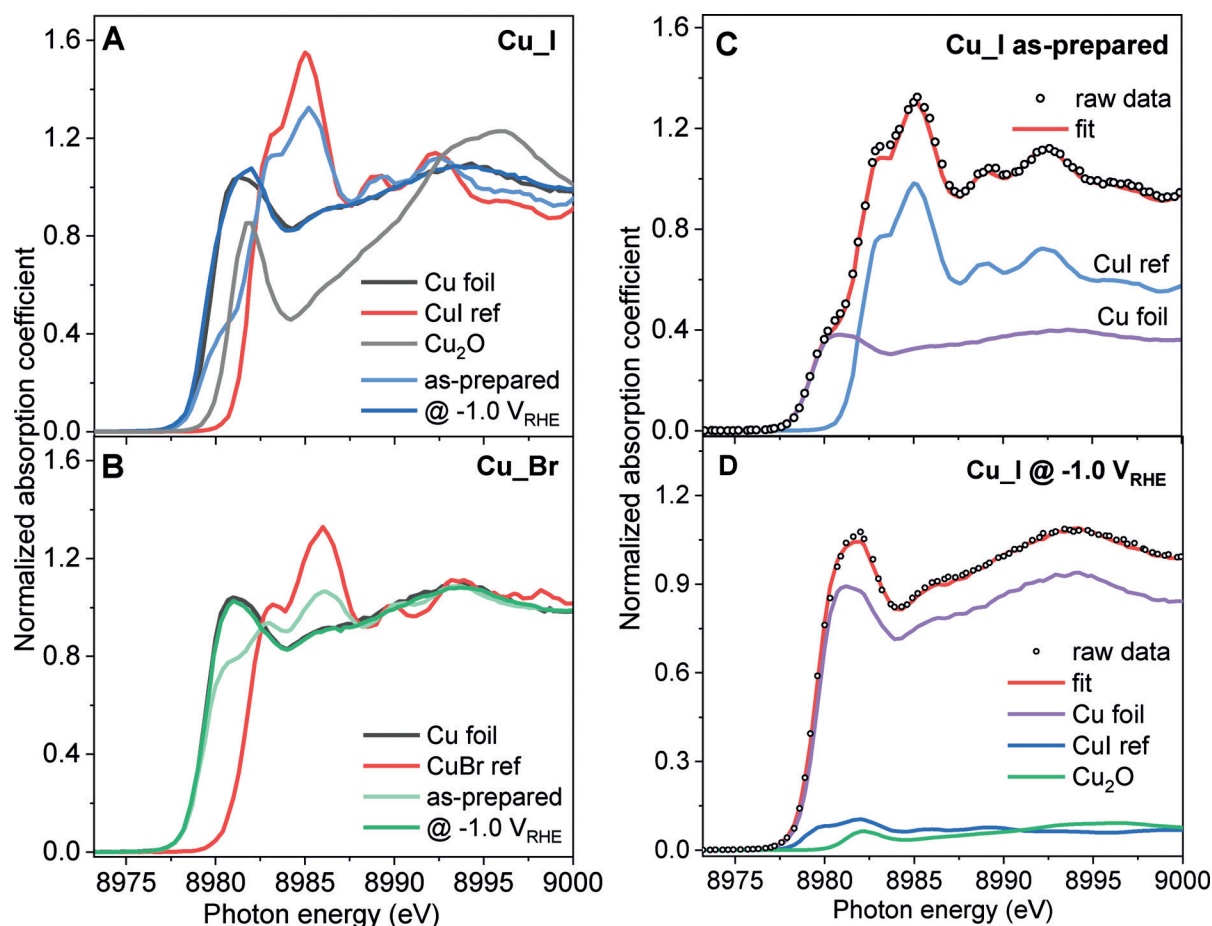
Nanostructured Cu catalysts were synthesized by cycling electropolished Cu foils in different 0.1M potassium salt solutions (between 0.3 and 2.2 V vs. RHE) and were denoted by Cu<sub>X</sub> (X = Cl, Br, I) and Cu<sub>CO<sub>3</sub></sub>, respectively. Additional details on the synthesis parameters are shown in the Supporting Information, Table S1. Figure 1 and the Supporting Information, Figure S1 show scanning electron microscopy (SEM) images of these samples as-prepared, after immersion in 0.1M KHCO<sub>3</sub> solution, as well as after 1 h of CO<sub>2</sub>RR at  $-1.0$  V vs. RHE. Cu nanocubes with an edge size of 250–300 nm are formed on the surface of the Cu<sub>Cl</sub> (Figure 1A) sample as we discussed previously.<sup>[18]</sup> In clear contrast, the Cu<sub>Br</sub> and Cu<sub>I</sub> samples show a faceted crystal morphology characterized by flatter larger structures for Cu<sub>Br</sub> (Figure 1B) and needle-like shapes for Cu<sub>I</sub> (Figure 1C). A composition of CuBr and CuI was confirmed by energy-dispersive X-ray spectroscopy (EDX; Supporting Information, Table S2). The Cu<sub>CO<sub>3</sub></sub> sample shows particles dispersed on the underlying Cu surface (Figure 1D). After the former electrolyte-driven surface nanostructuring, when the different samples are subsequently immersed in the same 0.1M KHCO<sub>3</sub> solution for 30 min before applying any potential, the edges of CuBr and CuI crystals as well as nanocubes in the Cu<sub>Cl</sub> sample become slightly roughed (Supporting Information, Figure S1), and an increased oxygen content and decreased halide content are observed due to the slow decomposition of the Cu halide (CuX) in the aqueous solution (Supporting Information, Table S2). The highest halide content of the three Cu<sub>X</sub> samples is consistent with the highest stability of CuI among the possible CuX compounds that could be formed.<sup>[33]</sup> The crystalline structure of the copper halide layer in the as prepared state was further confirmed by grazing incidence X-ray diffraction (GI-XRD) measurements, as shown in the Supporting Information, Figure S2. All three

halide-modified Cu samples consist of their copper halide phases, with the exception of the Cu<sub>Cl</sub> sample, which also includes minor Cu<sub>2</sub>O peaks owing to the instability of CuCl.

More interestingly, these samples show very different morphologies after 1 h of CO<sub>2</sub>RR, although most of the halides and carbon atoms have been removed during the reaction (Supporting Information, Table S2). Numerous particles with average size of  $220 \pm 60$  nm are formed on the surface of the Cu<sub>Br</sub> sample (Figure 1F), while the Cu<sub>I</sub> sample shows a rough surface (Figure 1G), with a very high roughness factor determined by measuring the double-layer capacitance (Supporting Information, Table S3). The particles on the surface of Cu<sub>CO<sub>3</sub></sub> are removed during CO<sub>2</sub>RR, leading to a porous Cu surface with average pore size of  $430 \pm 130$  nm (Figure 1H).

To monitor the evolution of the chemical state and local environment of Cu during CO<sub>2</sub>RR we conducted operando X-ray absorption spectroscopy (XAS) measurements. While extended X-ray absorption fine-structure spectroscopy (EXAFS) data measured in total fluorescence yield mode did not show any significant difference from the metallic structure apart from a highly defective structure (Supporting Information, Figures S3–S7, Table S4), high-energy resolution fluorescence detected X-ray near edge structure (HERFD-XANES) spectra are more sensitive to the chemical state and coordination environment of Cu under reaction conditions.<sup>[35,36]</sup> Figure 2 shows the Cu K-edge HERFD-XANES spectra of the as prepared Cu<sub>X</sub> samples as well as data from the same samples obtained during CO<sub>2</sub>RR at  $-1.0$  V vs. RHE along with the reference spectra of bulk Cu, Cu<sub>2</sub>O, CuI, and CuBr. The spectra suffer from significant self-absorption effects, but a normalization was used here to have equal self-absorption in the reference spectra and halide- and carbonate-modified samples. The spectra of the as-prepared Cu<sub>I</sub> and Cu<sub>Br</sub> samples show distinctive features of CuI (Figure 2A; Supporting Information, Figure S8) and CuBr (Fig-





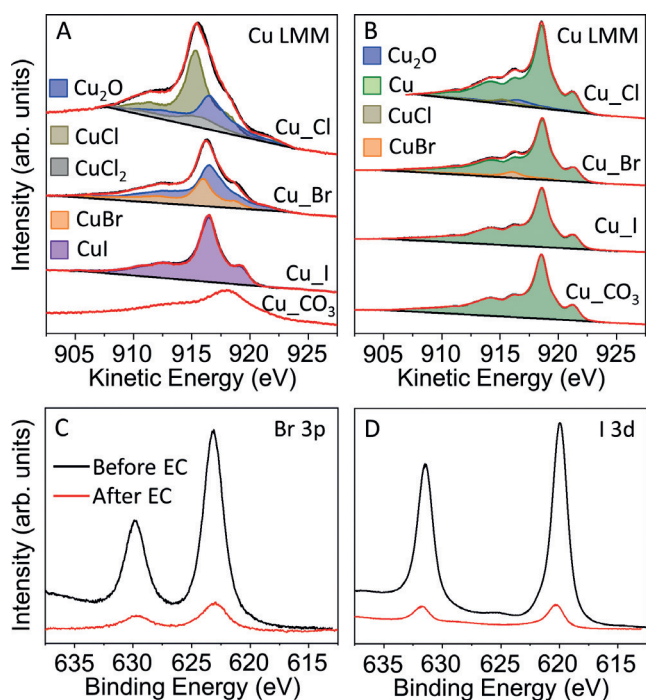
**Figure 2.** HERFD-XANES spectra of the Cu\_I (A) and Cu\_Br (B) samples in the as-prepared state and measured under operando CO<sub>2</sub>RR conditions in 0.1 M KHCO<sub>3</sub> after 1 h of CO<sub>2</sub>RR at  $-1.0$  V vs. RHE. Reference spectra of bulk Cu, CuI, CuBr, and Cu<sub>2</sub>O are also plotted. Linear combination analysis (LCA) of operando HERFD-XANES spectra of the Cu\_I sample measured in the as-prepared state (C) and during CO<sub>2</sub>RR at  $-1.0$  V vs. RHE (D) are shown. The corresponding subspectral components needed to fit the data (metallic Cu, CuI, and Cu<sub>2</sub>O) are scaled according to their weighting parameters.

ure 2B). A broad pre-edge feature between 8979 and 8981 eV, most likely corresponding to metallic Cu, is also detected for the Cu\_I sample. The more intense pre-edge feature at 8981 eV in the Cu-Br sample points out a higher amount of metallic Cu. The as-prepared Cu\_Cl and Cu\_CO<sub>3</sub> samples on the other hand show the presence of a dominant metallic Cu component (pre-edge feature at 8981 eV; Supporting Information, Figures S9, S10, respectively). A slightly more intense feature at 8986 eV and lower intensity of the above mentioned metallic feature in the as-prepared Cu\_Cl indicate the presence of CuCl species. Finally, there is no clear evidence of cationic Cu in the as-prepared Cu\_CO<sub>3</sub> sample.

HERFD-XANES spectra of all four samples measured under CO<sub>2</sub>RR conditions after 1 h of activation indicate nearly complete reduction of Cu, showing a close resemblance to the reference spectrum of the Cu foil. An exception however is the Cu\_I sample measured at  $-1.0$  V vs. RHE, which has the position of the pre-edge feature shifted to the energy typical for Cu<sub>2</sub>O. To support the described qualitative observations, linear combination analysis (LCA) of the XANES spectra was carried out using various combinations of reference spectra as basis sets (Figure 2C,D; Supporting

Information, Figures S9–S11). According to the LCA analysis, the as-prepared Cu\_X samples contain Cu<sup>+</sup> species (CuI, CuBr, CuCl, and/or Cu<sub>2</sub>O), the relative content of which follows the sequence Cu\_I > Cu\_Br > Cu\_Cl (63, 32, and 24 at% correspondingly; Supporting Information, Table S5). Notably, no CuO<sub>x</sub> could be detected in the as-prepared Cu\_I and Cu\_Br samples, while the Cu\_Cl and Cu\_CO<sub>3</sub> samples contained correspondingly Cu<sub>2</sub>O (14 at%) and CuO (3 at%). As mentioned above, the Cu\_CO<sub>3</sub> sample is fully reduced after 1 h under CO<sub>2</sub>RR, while Cu\_I, Cu\_Br, and Cu\_Cl show the presence of Cu<sub>2</sub>O in the amounts of 8, 3 and 1 at%, respectively, albeit the latter two numbers are within the experimental error. The reduced Cu\_I sample still contained ca. 7 at% of residual CuI. These results also reveal the higher sensitivity of HERFD-XANES measurements acquired in grazing configuration to Cu<sub>2</sub>O and CuI species as compared to GI-XRD which only shows metallic Cu peaks after CO<sub>2</sub>RR (Supporting Information, Figure S2).<sup>[36,37]</sup>

Quasi in situ X-ray photoelectron spectroscopy (XPS) measurements were performed to probe the chemical state and composition of the surface of the Cu catalysts during CO<sub>2</sub>RR. The electrochemical cell was directly attached to the

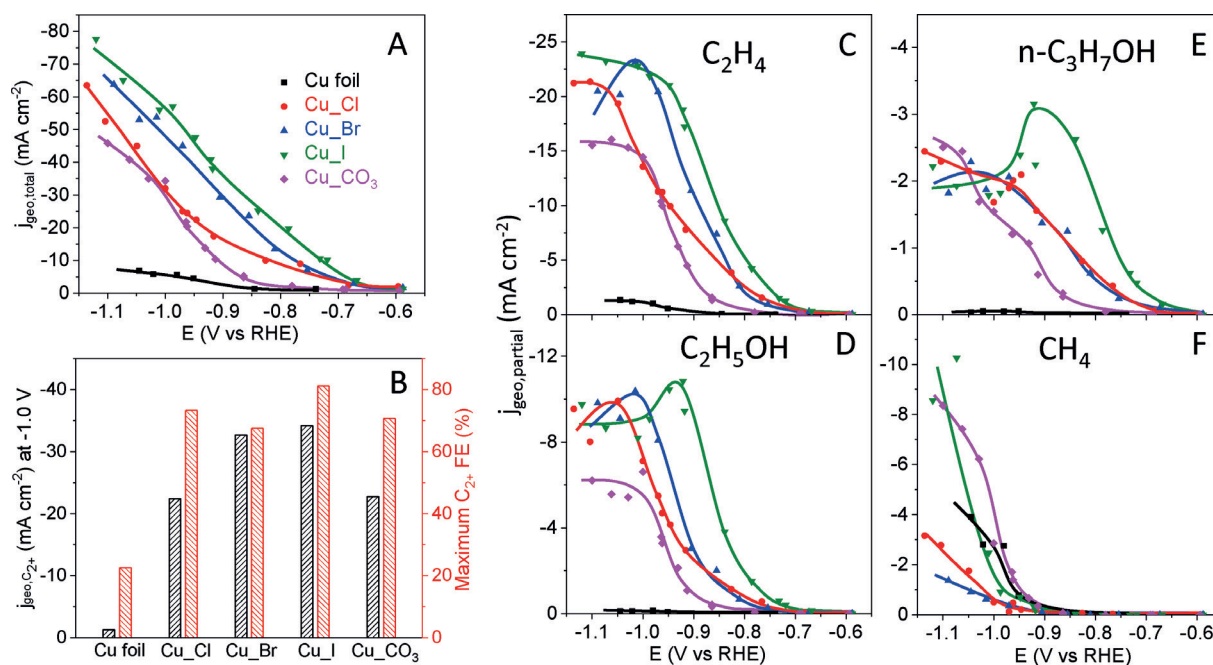


**Figure 3.** Quasi in situ Cu Auger LMM XPS spectra of Cu<sub>X</sub>CO<sub>3</sub>, Cu<sub>Br</sub>, and Cu<sub>I</sub> before (A) and after (B) 1 h of CO<sub>2</sub>RR at  $-1.0$  V vs. RHE in a CO<sub>2</sub>-saturated 0.1 M KHCO<sub>3</sub> solution. Br 3p and I 3d XPS spectra of the Cu<sub>Br</sub> (C) and Cu<sub>I</sub> (D) measured before and after CO<sub>2</sub>RR are also shown.

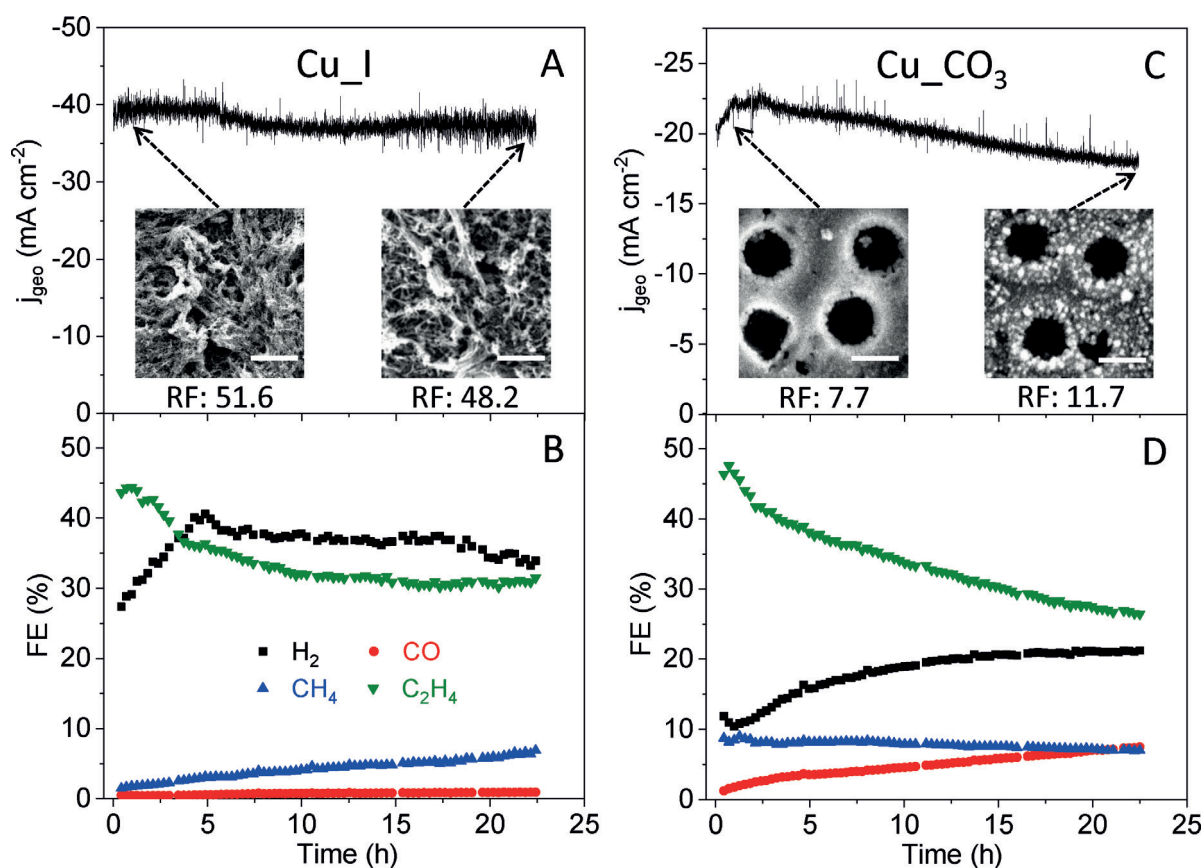
XPS analysis system and the sample transfer was conducted in UHV. The Cu<sub>X</sub> and Cu<sub>CO<sub>3</sub></sub> samples were measured with XPS in the as-prepared state and after 1 h of CO<sub>2</sub>RR at  $-1.0$  V vs. RHE as shown in Figure 3 and the Supporting

Information, Figures S12 and S13. Among the Cu<sub>X</sub> samples, the Cu<sub>I</sub> sample shows an almost pure CuI surface in its as prepared state, while the surface of the Cu<sub>Br</sub> is composed of Cu<sub>2</sub>O (65 at%) and CuBr (35 at%), and that of Cu<sub>Cl</sub> is composed of Cu<sub>2</sub>O (31 at%), CuCl (56 at%) and CuCl<sub>2</sub> (12 at%). The composition difference is also consistent with the relative stability of the three Cu halides.<sup>[33]</sup> However, the Cu<sub>CO<sub>3</sub></sub> sample has a starting Cu oxidation state of Cu<sup>2+</sup> in the form of CuO and CuCO<sub>3</sub>. After CO<sub>2</sub>RR, most of the Cu<sup>+</sup> and Cu<sup>2+</sup> species in the as prepared samples were reduced to metallic Cu. However, a small amount of CuBr still survived (Figure 3B). Moreover, considerable amounts of halides (Figure 3C,D) were observed on the Cu surface before but also after the reaction and subsequent in situ rinsing in water. The metallic Cu surface of the Cu<sub>I</sub> sample seen by XPS, together with the Cu<sub>2</sub>O and CuI species revealed by the more bulk-sensitive XANES measurements (Figure 2), indicate the presence of subsurface oxygen and Cu<sup>+</sup> species in the halide-derived CO<sub>2</sub>RR Cu catalysts.

The catalytic activity and selectivity of the nanostructured Cu catalysts were obtained by performing chronoamperometry measurements in a CO<sub>2</sub>-saturated 0.1 M KHCO<sub>3</sub> solution. All the nanostructured Cu catalysts show significantly higher geometric current density than an electropolished Cu foil (EP\_Cu), and Cu<sub>I</sub> shows the highest current density in the measured potential range (Figure 4A). However, when the current densities were normalized by the electrochemically active surface area (ECSA), the three halide-modified Cu catalysts show similar specific activity (Supporting Information, Figure S17A). Therefore, the high surface area (thus high density of surface reactive sites) of the roughened Cu catalysts plays a very important role in the significantly improved apparent activity compared to a flat Cu surface.



**Figure 4.** Total geometric current density (A), total Faradaic efficiency and geometric partial current density of (B) C<sub>2+</sub> products, and partial geometric current densities of (C) C<sub>2</sub>H<sub>4</sub>, (D) C<sub>2</sub>H<sub>5</sub>OH, (E) n-C<sub>3</sub>H<sub>7</sub>OH, and (F) CH<sub>4</sub> as a function of the applied potential after 1 h of CO<sub>2</sub>RR in a CO<sub>2</sub>-saturated 0.1 M KHCO<sub>3</sub> solution.



**Figure 5.** Time-dependent geometric current densities and Faradaic efficiencies of gas products for the Cu\_I sample at  $-0.9$  V vs. RHE (A, B) and the Cu\_CO<sub>3</sub> sample at  $-0.95$  V vs. RHE (C, D) in a CO<sub>2</sub>-saturated 0.1 M KHCO<sub>3</sub> solution. The insets in (A) and (C) are SEM images of the Cu\_I and Cu\_CO<sub>3</sub> samples after 1 h and after 22 h of CO<sub>2</sub>RR. Scale bars: 5  $\mu\text{m}$  (A) and 500 nm (C). Roughness factors (RF) after 1 h and after 22 h of CO<sub>2</sub>RR are also indicated.

Nevertheless, apart from the roughness, the increased activity over the nanostructured Cu catalysts versus EP\_Cu could also be ascribed to defects, their porous structures, and the presence of adsorbed halide species, Cu<sup>+</sup> or subsurface oxygen species, as we observed by XPS (Figure 3) and XANES (Figure 2).

The total FEs of C<sub>2+</sub> products are shown in Figure 4B and the Supporting Information, Figure S14, and FEs for each individual product are shown in the Supporting Information, Figure S15. The highest C<sub>2+</sub> FE of about 80% was achieved at  $-0.9$  V vs. RHE over Cu\_I, while the C<sub>2+</sub> FEs of about 66–73% at  $-1.0$  V vs. RHE were obtained over Cu\_Cl, Cu\_Br, and Cu\_CO<sub>3</sub>. The C<sub>2+</sub> FEs of all nanostructured Cu catalysts were 3-fold higher than those of the EP\_Cu sample (ca. 20%) at  $-0.9$  V vs. RHE. At more negative potentials ( $<-1.0$  V vs. RHE), all the catalysts show decreased C<sub>2+</sub> FE and increased H<sub>2</sub> FE, which is due to CO<sub>2</sub> mass-transport limitations. The potential-dependent partial current densities of the various CO<sub>2</sub>RR products are shown in Figure 4C–F and the Supporting Information, Figures S16, S17. Cu\_I showed the highest partial current densities for ethylene, ethanol, and n-propanol. Compared to the previously reported systems, where EP\_Cu was studied in an electrolyte mixture containing KX (X = Cl, Br, I) and KHCO<sub>3</sub>,<sup>[17]</sup> our new nanostructured catalytic systems (synthesized by cycling the Cu foil in KX

followed by washing in water) avoided the complexity of the co-existence of structural and chemical electrolyte effects since all CO<sub>2</sub>RR measurements were done in KHCO<sub>3</sub>. Furthermore, higher activity, lower onset potential and higher C<sub>2+</sub> selectivity were obtained for the present electrolyte pre-nanostructured catalysts as compared to the case where iodine was added to the electrolyte during the reaction (Supporting Information, Figure S18). Although the Cu\_CO<sub>3</sub> sample showed high partial current density for C<sub>2+</sub>, its methane partial current density was also higher, comparable to that of EP\_Cu. The simultaneously favorable production of ethylene and methane over the porous Cu\_CO<sub>3</sub> is slightly different from that reported for Cu foam catalysts, which simultaneously favored the production of ethylene and ethane, which is probably due to the smaller pore size (430 nm) in our Cu\_CO<sub>3</sub>. Overall, the iodine modified Cu catalyst (Cu\_I) synthesized via electrolyte-driven nanostructuring showed a high C<sub>2+</sub> geometric current density of about 31.2 mA cm<sup>-2</sup> at  $-0.9$  V vs. RHE (Supporting Information, Figure S19), superior to previously reported Cu catalysts (Supporting Information, Table S6).

Stability tests were carried out for the Cu\_I and Cu\_CO<sub>3</sub> samples at potentials for which the highest C<sub>2+</sub> FEs were detected (Figure 5). The current density of Cu\_I remained almost stable within our 22 h test, while a 20% decrease was



observed for the Cu\_CO<sub>3</sub> sample. Furthermore, a decrease in the C<sub>2</sub>H<sub>4</sub> FE was observed for both samples, although in the case of the Cu\_I this took place only during the first 5 h, becoming stable subsequently. The decrease in the C<sub>2</sub>H<sub>4</sub> FE was accompanied by an increase in methane FE for the Cu\_I sample (a fast decrease in C<sub>2</sub>H<sub>4</sub>/CH<sub>4</sub> FE ratio; Supporting Information, Figure S20) and of CO and H<sub>2</sub> FE for Cu\_CO<sub>3</sub>. However, the morphology (inserts in Figure 5A) and roughness (Supporting Information, Table S3) of Cu\_I after the stability test were similar to those after 1 h of CO<sub>2</sub>RR, suggesting that the increased production of CH<sub>4</sub> (or CO) at the expense of C<sub>2</sub>H<sub>4</sub> was not caused by morphological changes, but most likely by the gradual depletion of subsurface oxygen species<sup>[38–40]</sup> and Cu<sup>+</sup> species as well as the loss of adsorbed iodine ions in the Cu\_I sample. In clear contrast, a large number of NPs were formed during CO<sub>2</sub>RR on the surface of the Cu\_CO<sub>3</sub> sample (inserts in Figure 5C) and the roughness increased accordingly (Supporting Information, Table S3). The presence of these new low-coordinated sites in the form of NPs is expected to favor the formation of CO and H<sub>2</sub><sup>[41]</sup> and could also lead to the deactivation of the Cu\_CO<sub>3</sub> sample.

Although the activity, selectivity, and stability of CO<sub>2</sub>RR catalysts are determined by multiple complex factors such as roughness, defects, shape, and oxidation state, the present data feature that the C–C coupling process over the Cu\_X samples, especially Cu\_I, is strongly related to the presence and stabilization of Cu<sup>+</sup> species as well as of the adsorbed halides, as confirmed by operando HERFD-XANES (bulk-sensitive) and quasi in situ XPS measurements (surface sensitive). We found a positive correlation between the production of C<sub>2+</sub> and the amount of Cu<sup>+</sup> species in the halide-modified Cu catalysts in the following order: Cu\_I > Cu\_Br > Cu\_Cl. Previous theoretical studies predicted that subsurface oxygen as well as the presence of a Cu<sup>+</sup>/Cu<sup>0</sup> interface plays a crucial role in CO<sub>2</sub> activation and CO dimerization, ultimately resulting in higher C<sub>2+</sub> selectivity.<sup>[11,42]</sup> Interestingly, the adsorbed halides are known to bind more strongly to the oxidized Cu surface<sup>[12,31]</sup> and to facilitate the formation and stabilization of the intermediates during CO<sub>2</sub>RR required to obtain C<sub>2+</sub> products. On the other hand, the Cu\_CO<sub>3</sub> sample with only metallic Cu species under reaction conditions, showed higher CH<sub>4</sub> selectivity than all Cu\_X samples. The latter is probably attributed to its nanoporous structure.<sup>[43]</sup> At the end, we should also highlight the role of the high ECSA of the present nanostructured Cu catalysts.<sup>[44]</sup> Apart from having a higher surface area, a drastic increase in the ECSA during nanostructuring a flat Cu surface is usually coupled with the creation of highly reactive surface sites such as defects and low-coordinated sites. These surface sites might be more favorable for C–C coupling during CO<sub>2</sub>RR,<sup>[1,45]</sup> not only improving the apparent activity but also helping to tune the selectivity towards multicarbon products. In this work we were able to modify the surface morphology and its composition and chemical state (Cu<sup>+</sup>) via an electrolyte-driven nanostructuring pre-treatment strategy, which was found to lead to enhanced C<sub>2+</sub> selectivity.

## Conclusion

We have presented an electrolyte-driven nanostructuring strategy for the facile synthesis of CO<sub>2</sub>RR electrocatalysts highly selective to C<sub>2+</sub> products. The proposed synthesis not only leads to strong morphological modifications of the sample surface, but also to the presence of residual halides and cationic Cu species. These Cu electrocatalysts can selectively convert CO<sub>2</sub> into ethylene and multicarbon alcohols in a KHCO<sub>3</sub> solution, with the iodine-modified catalysts showing the highest FE for C<sub>2+</sub> of about 80% and partial geometrical current density of about 31.2 mA cm<sup>-2</sup> at –0.9 V vs. RHE. The superior C<sub>2+</sub> selectivity of the halide-modified Cu catalysts was attributed to their rough surface morphology combined with electronic and chemical effects arising from the stabilization of subsurface oxygen as well as Cu<sup>+</sup> species and adsorbed halides on the surface. Cu\_CO<sub>3</sub> shows both high C<sub>2+</sub> and methane selectivity, which is attributed to its particular nanoporous structure. Stability tests suggested that the gradual depletion of subsurface oxygen/Cu<sup>+</sup> species and the increased number of low-coordinated sites formed under reaction conditions are behind the distinct catalytic performance of the halide- and carbonate-modified Cu catalysts, respectively. This work provides new insights required for the design of highly active C<sub>2+</sub>-selective CO<sub>2</sub>RR catalysts.

## Experimental Section

Cu\_Cl, Cu\_Br, Cu\_I, and Cu\_CO<sub>3</sub> catalysts were prepared by electrochemically cycling an electropolished Cu foil in 0.1 M KCl, KBr, KI, and K<sub>2</sub>CO<sub>3</sub> solutions with triangular potential scans at a rate of 500 mV s<sup>-1</sup>, respectively. During each cycle, the potential was held at the negative (E<sub>1</sub>) and positive (E<sub>2</sub>) limits for 5 and 10 s, respectively. The cycled Cu catalysts were prepared with the indicated potential ranges and number of cycles as shown in the Supporting Information, Table S1. For other experimental details, including operando and ex situ characterizations and electrochemical measurements, see the Supporting Information.

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## Conflict of interest

The authors declare no conflict of interest.

**Keywords:** adsorbed halides · CO<sub>2</sub> electroreduction · copper(I) · electrolyte-driven nanostructuring · multicarbon products

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- [1] D. Gao, R. M. Arán-Ais, H. S. Jeon, B. Roldan Cuenya, *Nat. Catal.* **2019**, *2*, 198–210.
- [2] Z. W. Seh, J. Kibsgaard, C. F. Dickens, I. Chorkendorff, J. K. Nørskov, T. F. Jaramillo, *Science* **2017**, *355*, eaad4998.
- [3] A. D. Handoko, F. Wei, Jennady, B. S. Yeo, Z. W. Seh, *Nat. Catal.* **2018**, *1*, 922–934.
- [4] Z. Weng, Y. Wu, M. Wang, J. Jiang, K. Yang, S. Huo, X. Wang, Q. Ma, G. W. Brudvig, V. S. Batista, et al., *Nat. Commun.* **2018**, *9*, 415.
- [5] M. Gattrell, N. Gupta, A. Co, *J. Electroanal. Chem.* **2006**, *594*, 1–19.
- [6] Y. Zheng, A. Vasileff, X. Zhou, Y. Jiao, M. Jaroniec, S.-Z. Qiao, *J. Am. Chem. Soc.* **2019**, *141*, 7646–7659.
- [7] S. Hanselman, M. T. M. Koper, F. Calle-Vallejo, *ACS Energy Lett.* **2018**, *3*, 1062–1067.
- [8] A. Bagger, L. Arnarson, M. H. Hansen, E. Spohr, J. Rossmeisl, *J. Am. Chem. Soc.* **2019**, *141*, 1506–1514.
- [9] A. Verdaguier-Casadevall, C. W. Li, T. P. Johansson, S. B. Scott, J. T. McKeown, M. Kumar, I. E. L. Stephens, M. W. Kanan, I. Chorkendorff, *J. Am. Chem. Soc.* **2015**, *137*, 9808–9811.
- [10] H. Mistry, A. S. Varela, C. S. Bonifacio, I. Zegkinoglou, I. Sinev, Y. W. Choi, K. Kisslinger, E. A. Stach, J. C. Yang, P. Strasser, et al., *Nat. Commun.* **2016**, *7*, 12123.
- [11] M. Favaro, H. Xiao, T. Cheng, W. A. Goddard, J. Yano, E. J. Crumlin, *Proc. Natl. Acad. Sci. USA* **2017**, *114*, 6706–6711.
- [12] D. Gao, I. T. McCrum, S. Deo, Y. W. Choi, F. Scholten, W. Wan, J. G. Chen, M. J. Janik, B. Roldan Cuenya, *ACS Catal.* **2018**, *8*, 10012–10020.
- [13] P. De Luna, R. Quintero-Bermudez, C.-T. Dinh, M. B. Ross, O. S. Bushuyev, P. Todorović, T. Regier, S. O. Kelley, P. Yang, E. H. Sargent, *Nat. Catal.* **2018**, *1*, 103–110.
- [14] S. Lee, D. Kim, J. Lee, *Angew. Chem. Int. Ed.* **2015**, *54*, 14701–14705; *Angew. Chem.* **2015**, *127*, 14914–14918.
- [15] Y. Zhou, F. Che, M. Liu, C. Zou, Z. Liang, P. De Luna, H. Yuan, J. Li, Z. Wang, H. Xie, et al., *Nat. Chem.* **2018**, *10*, 974–980.
- [16] K. Jiang, R. B. Sandberg, A. J. Akey, X. Liu, D. C. Bell, J. K. Nørskov, K. Chan, H. Wang, *Nat. Catal.* **2018**, *1*, 111–119.
- [17] F. S. Roberts, K. P. Kuhl, A. Nilsson, *Angew. Chem. Int. Ed.* **2015**, *54*, 5179–5182; *Angew. Chem.* **2015**, *127*, 5268–5271.
- [18] D. Gao, I. Zegkinoglou, N. J. Divins, F. Scholten, I. Sinev, P. Grosse, B. Roldan Cuenya, *ACS Nano* **2017**, *11*, 4825–4831.
- [19] S. Ma, M. Sadakiyo, M. Heim, R. Luo, R. T. Haasch, J. I. Gold, M. Yamauchi, P. J. A. Kenis, *J. Am. Chem. Soc.* **2017**, *139*, 47–50.
- [20] Y. W. Choi, F. Scholten, I. Sinev, B. R. Cuenya, *J. Am. Chem. Soc.* **2019**, *141*, 5261–5266.
- [21] Y. Hori, I. Takahashi, O. Koga, N. Hoshi, *J. Phys. Chem. B* **2002**, *106*, 15–17.
- [22] J. H. Montoya, C. Shi, K. Chan, J. K. Nørskov, *J. Phys. Chem. Lett.* **2015**, *6*, 2032–2037.
- [23] E. Pérez-Gallent, M. C. Figueiredo, F. Calle-Vallejo, M. T. M. Koper, *Angew. Chem. Int. Ed.* **2017**, *56*, 3621–3624; *Angew. Chem.* **2017**, *129*, 3675–3678.
- [24] P. Grosse, D. Gao, F. Scholten, I. Sinev, H. Mistry, B. Roldan Cuenya, *Angew. Chem. Int. Ed.* **2018**, *57*, 6192–6197; *Angew. Chem.* **2018**, *130*, 6300–6305.
- [25] J. Huang, N. Hörmann, E. Oveisi, A. Loiodice, G. L. De Gregorio, O. Andreussi, N. Marzari, R. Buonsanti, *Nat. Commun.* **2018**, *9*, 3117.
- [26] D. Kim, C. S. Kley, Y. Li, P. Yang, *Proc. Natl. Acad. Sci. USA* **2017**, *114*, 10560–10565.
- [27] C. M. Gunathunge, X. Li, J. Li, R. P. Hicks, V. J. Ovalle, M. M. Waegle, *J. Phys. Chem. C* **2017**, *121*, 12337–12344.
- [28] B. Rahn, R. Wen, L. Deuchler, J. Stremme, A. Franke, E. Pehlke, O. M. Magnussen, *Angew. Chem. Int. Ed.* **2018**, *57*, 6065–6068; *Angew. Chem.* **2018**, *130*, 6173–6176.
- [29] Y. Deng, Y. Huang, D. Ren, A. D. Handoko, Z. W. Seh, P. Hirunsit, B. S. Yeo, *ACS Appl. Mater. Interfaces* **2018**, *10*, 28572–28581.
- [30] M. D. Regulacio, Y. Wang, Z. W. Seh, M. Han, *ACS Appl. Nano Mater.* **2018**, *1*, 3042–3062.
- [31] D. Gao, F. Scholten, B. Roldan Cuenya, *ACS Catal.* **2017**, *7*, 5112–5120.
- [32] Y. Huang, C. W. Ong, B. S. Yeo, *ChemSusChem* **2018**, *11*, 3299–3306.
- [33] I. T. McCrum, S. A. Akhade, M. J. Janik, *Electrochim. Acta* **2015**, *173*, 302–309.
- [34] S. Y. Lee, H. Jung, N. K. Kim, H. S. Oh, B. K. Min, Y. J. Hwang, *J. Am. Chem. Soc.* **2018**, *140*, 8681–8689.
- [35] O. Hirsch, K. O. Kvashnina, L. Luo, M. J. Süess, P. Glatzel, D. Koziej, *Proc. Natl. Acad. Sci. USA* **2015**, *112*, 15803–15808.
- [36] T. Gu, H. W. P. Carvalho, D. E. Doronkin, T. Sheppard, P. Glatzel, A. J. Atkins, J. Rudolph, C. R. Jacob, M. Casapu, J. Grunwaldt, *Chem. Commun.* **2015**, *51*, 9227–9230.
- [37] M. Scherzer, F. Girgsdies, E. Stotz, M.-G. Willinger, E. Frei, R. Schlögl, U. Pietsch, T. Lunkenbein, *J. Phys. Chem. C* **2019**, *123*, 13253–13262.
- [38] C. Liu, M. P. Lourenço, S. Hedström, F. Cavalca, O. Diaz-Morales, H. A. Duarte, A. Nilsson, L. G. M. Pettersson, *J. Phys. Chem. C* **2017**, *121*, 25003–25009.
- [39] F. Cavalca, R. Ferragut, S. Aghion, A. Eilert, O. Diaz-Morales, C. Liu, A. L. Koh, T. W. Hansen, L. G. M. Pettersson, A. Nilsson, *J. Phys. Chem. C* **2017**, *121*, 25010–25017.
- [40] A. Eilert, F. Cavalca, F. S. Roberts, J. Osterwalder, C. Liu, M. Favaro, E. J. Crumlin, H. Ogasawara, D. Friebel, L. G. M. Pettersson, et al., *J. Phys. Chem. Lett.* **2017**, *8*, 285–290.
- [41] R. Reske, H. Mistry, F. Behafarid, B. Roldan Cuenya, P. Strasser, *J. Am. Chem. Soc.* **2014**, *136*, 6978–6986.
- [42] H. Xiao, W. A. Goddard, T. Cheng, Y. Liu, *Proc. Natl. Acad. Sci. USA* **2017**, *114*, 6685–6688.
- [43] A. Dutta, M. Rahaman, N. C. Luedi, M. Mohos, P. Broekmann, *ACS Catal.* **2016**, *6*, 3804–3814.
- [44] L. Wang, S. A. Nitopi, A. B. Wong, J. L. Snider, A. C. Nielander, C. G. Morales-guio, M. Orazov, D. C. Higgins, C. Hahn, T. F. Jaramillo, *Nat. Catal.* **2019**, *2*, 702–708.
- [45] R. M. Arán-Ais, D. Gao, B. Roldan Cuenya, *Acc. Chem. Res.* **2018**, *51*, 2906–2917.

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