ORIGINAL PAPER



Machine learning for material characterization with an application for predicting mechanical properties

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Abstract

Currently, the growth of material data from experiments and simulations is expanding beyond processable amounts. This makes the development of new data-driven methods for the discovery of patterns among multiple lengthscales and time-scales and structure-property relationships essential. These data-driven approaches show enormous promise within materials science. The following review covers machine learning (ML) applications for metallic material characterization. Many parameters associated with the processing and the structure of materials affect the properties and the performance of manufactured components. Thus, this study is an attempt to investigate the usefulness of ML methods for material property prediction. Material characteristics such as strength, toughness, hardness, brittleness, or ductility are relevant to categorize a material or component according to their quality. In industry, material tests like tensile tests, compression tests, or creep tests are often time consuming and expensive to perform. Therefore, the application of ML approaches is considered helpful for an easier generation of material property information. This study also gives an application of ML methods on small punch test (SPT) data for the determination of the property ultimate tensile strength for various materials. A strong correlation between SPT data and tensile test data was found which ultimately allows to replace more costly tests by simple and fast tests in combination with ML.

KEYWORDS

machine learning, material characterization, small punch test, tensile properties, ultimate tensile strength

1 | INTRODUCTION

The field of materials science relies on experiments and simulation-based models as tools for material characterization [7]. Material properties, such as their structure and behavior, are critical to the potential application of the material of interest. More recently, the data generated by such experiments and simulations have created various opportunities for the application of data-driven methods. In addition to, for example, the experimental trial and error approach or a physical metallurgy approach, machine learning (ML) methods for property prediction and material design have attracted a lot of attention in recent years, see for example, [33, 109, 162].

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While experimental investigations (the so-called first paradigm of materials science) have been carried out since the stone and copper age, scientists of the 16th century started to describe physical relations by equations (second paradigm). Thus, analytical equations became a central instrument of theoretical physics which were able to complement the empirical and experimental sciences. The 1950s marked the beginning of computational materials science and simulations, the third paradigm. Within this framework, computer experiments and simulations became possible, with the corresponding results being analyzed and interpreted like measured ones. It had to be recognized that many properties of materials cannot be described by a closed mathematical form as they are determined by several multilevel, intricate theoretical concepts. With the help of large amounts of data, hidden correlations, reflected in terms of structure and patterns in the data can be discovered that are not normally visible in small data sets. Thus, the fourth paradigm, data-driven science, of materials research was born [7, 34].

However, it is not only an advantage to have a large data volume but it can also be a challenge to cope with tremendous amounts of data. Today, data are indeed more and more easily acquired and stored, due to huge progresses in sensors and ways to collect data on one side, and in storage devices on the other side. Nowadays, there is no hesitation in many domains in acquiring very large amounts of data without knowing in advance if they will be analyzed and how. The spectacular increase in the amount of data is not only found in the number of samples collected for example over time, but also in the number of attributes, or characteristics, that are simultaneously measured on a process. Data are gathered into vectors whose dimension corresponds to the number of simultaneous measurements on the process. Growing dimensions result in high dimensional data, as each sample can be represented as a point or vector in a high-dimensional space. Working with high-dimensional data means working with data that are embedded in high-dimensional spaces [159]. The curse of dimensionality is the expression of all phenomena that appear with high-dimensional data, and that have most often unfortunate consequences on the behavior and performances of learning algorithms.

Contrary to the curse of dimensionality, databases in materials science are often limited in size due to expensive and time consuming data acquisition via experiments or simulations [136]. Then the insufficient data size for the training of a ML model compromises the learning success and suitable new approaches for small datasets have to be found.

This work contains a literature survey which covers an overview of ML for materials science and specifically for metallic material characterization. As the measurement of such parameters is often expensive and time consuming obtained via experiments, alternative basic tests, such as the small punch test (SPT) can be an option if it can be shown that the same material property information can be extracted.

There is a wide range of ML approaches based on SPT data which will be presented. Furthermore, in Section 3 an example is described which uses ML for the prediction of tensile properties of an insert-material-type based on SPT data. The objective of this study is to investigate whether it is possible to find a ML model which predicts/determines the tensile properties of a material from SPT data [11]. Section sec4 concludes this paper by giving an outlook on further research perspectives.

2 | STATE OF THE ART

2.1 | Overview—ML for materials science

With ML, given enough data and a data-driven algorithm for rule discovery, a computer is able to determine physical laws which lead to the given data without human input [19, 68]. Traditional computational approaches use the computer for the employment of a hard-coded algorithm provided by a human expert. By contrast, ML approaches learn the rules that underlie a dataset by assessing a portion of that data and building a model to make predictions [19]. However, the human still needs to choose suitable ML models which supposedly represent the data well and do manual (sub-)tasks in preprocessing and feature generation.

The existence of large amounts of data makes the use of ML models possible and enables data-driven knowledge to be obtained and patterns to be discovered. On the other hand, big data and their high dimensionality lead to difficult computational and statistical challenges, such as scalability and memory shortage, noise accumulation, interference correlation, incidental endogeneity, and measurement errors [40].

Materials science is an interesting field of application for big data methods and ML approaches which is beginning to show enormous promise. Four primary elements are critical in materials science and engineering: processing, structure, properties, and performance [110, 115]. There is no general agreement, however, on how these elements are interconnected. ML methods can be applied to the so-called process-structure-property-performance chain for



learning more about the intrinsic interrelations of these components. One main goal is the enabling, acceleration, and simplification of the discovery and development of novel materials based on the convergence of high-performance computing, automation, and ML [27]. Another aim of using such approaches in the field of materials science is to achieve high-throughput identification and quantification of essential material properties [15].

Besides experimentally obtained datasets, numerous studies draw required information from simulation-based data mining. Altogether, it is shown that experiment- and simulation-based data mining in combination with ML tools provide exceptional opportunities to enable highly reliant identification of fundamental interrelations within materials for characterization and optimization in a scale-bridging manner [15].

For more detailed information on recent ML applications in materials science we refer to the general reviews of Mueller et al [109], Wagner et al [162], Dimiduk et al [33], or Wei et al [168]. Examples for successful applications of ML techniques in materials science are, for example, to represent inorganic materials [61, 134, 135], predict fundamental properties [29, 79, 100], create atomic potentials [80], identify functional candidates [81, 89, 94], analyze complex reaction networks [157], or guide experimental design [71, 122], high-throughput phase diagrams and crystal structure determination [49].

2.1.1 | Open problem—interpretability

However, one of the major criticisms of ML algorithms in science is the lack of novel understanding and knowledge arising from their use. This is mostly because more complex ML algorithms are often treated as black boxes. Those machine-built models are hard to understand for humans [133].

For a better acceptance of ML models, data scientists aim to establish clear causal relations between materials structure defined broadly across length scales and properties. Especially scientific models have further constraints such as a minimal number of parameters and adherence to physical laws. It is the obligation of the data scientist to translate the results of their work into knowledge other scientists can use in aiding for example materials discovery or deployment [162]. Useful techniques for finding simple, reduced and interpretable models are for example principal component analysis (PCA) [172], cross-validation and regularization, and a thoughtful choice of model.

PCA is a powerful technique for data dimensionality reduction. Large datasets are increasingly widespread. In order to interpret such datasets, PCA can be applied to drastically reduce their dimensionality in an interpretable way, such that most of the information in the data is preserved [67]. PCA extracts the orthogonal directions with the greatest variance from a dataset, the resulting principal components being linear combinations of the original variables. However, principal components are not necessarily simple to interpret physically but as the extracted features are linear combinations of the original variables they can still be intuitively explained. Moreover, it allows a very straightforward data visualization through data projection onto the main extracted components [158]. However, PCA might be the wrong choice if features are not covariant.

Another way to achieve interpretable ML models is intelligent feature selection for dimension reduction and thus easier interpretability. Regularization of a model entails adding a tunable penalty on model parameter size to the cost function being minimized leading to a reduced feature space [162].

Furthermore, the choice of ML model has an immediate impact on its explainability. Regressions lead to coefficients whose size gives information about the relative size effect of modifying an input on the output. Decision trees are set up like flow charts and therefore easy to read. More complex models such as artificial neural networks (ANNs) are missing a clear explanation of the machine's "thinking" due to complex node interactions. But methods such as feature visualization [37, 142] or attribution [44, 150, 176] exist, which allow a better understanding and interpretability of black box models. However, sometimes it might be reasonable to trade model accuracy for better explainability.

2.1.2 | Open problem—small data

Contrary to the curse of dimensionality, there is often the problem of small data in material characterization, because the experimental or simulative generation of data is complex and expensive. Also the availability of the testing units limits data generation. Special test environments and material properties that are difficult to implement in the laboratory are often required, such as the creation of a corroded workpiece.



Often models are fitted to extremely small training sets which does not play to the strength of ML and will not allow the replication of the success ML methods had in other fields. It is of course possible to use ML methods as a simple fitting procedure for small low-dimensional datasets [133].

A few approaches are known to tackle this problem. For example, a ML model can be constructed by restricting the configurational space of materials, such as predicting the band gaps of selected families of semiconductors with fixed composition or crystalline structure instead of modeling compounds spanning a wide chemical space [32, 119, 179]. Another approach by Zhang et al [179] proposes to incorporate the crude estimation of the property in the feature space to establish ML models using small sized materials data, which increases the accuracy of prediction without the cost of higher degree of freedom.

Another approach for insufficient training data is the additional integration of prior knowledge into the training process, which leads to the notion of informed ML [161], or more specifically physics-informed ML [124, 177, 180]. Domain knowledge is often given as a set of additional constraints [149]. The integration of additional domain knowledge generates a new hybrid formulation of the ML problem which then ideally leads to physically meaningful and significantly more accurate interpretations of the data [38]. Besides adding constraints, expert knowledge can be incorporated in different ways. Up to recently, it was mostly limited to labeling data for supervised learning and setting prior probabilities in Bayesian networks [90]. However, in semi-supervised clustering applications, user guidance can be given by partial labeling information, which can be incorporated using hard constraints [24]. The domain knowledge can also be integrated into the process of building a ML model by the application of data visualization which often improves the accuracy of the resulting model [90]. Another approach is the monotonization of ML functions based on known physical relationships [31].

In general, ML systems are rarely viewed in the context of small data, where an insufficient data size for the training model compromises the learning success. The bottleneck of the database size especially limits applications, in which the construction of a database via experiments is time consuming and costly [136]. Thus, the recent development of materials databases might be helpful in tackling the small data problem.

2.1.3 | Existing databases

With the launch of the materials genome initiative [155] in 2011 and the coming of the big data era, a large effort has been made in the materials science community to collect extensive datasets of materials properties and to provide materials engineers with ready access to the properties of known materials. Existing databases are the materials project [63, 98], the inorganic crystal structure database [39, 126], the Materials genome initiative [155, 171], the NOMAD archive [34, 113], the Topological Materials Database [156], Supercon [97], or the National Institute of Materials Science 2011 [111] with many databases of material properties of metal alloys, or the National Institute of Standards and Technology [112] with databases of properties for material classes such as structural ceramics, oxide glasses, superconductors [50, 73, 121, 127, 167, 175]. A more comprehensive list of material databases can be found in Correa-Baena et al [27].

Traditionally, negative results are often discarded and left unpublished. However, negative data are often just as important for ML algorithms as positive results in order to prevent bias in the data. In some disciplines with a longer tradition of data-based research (like chemistry), such databases already exist. In a similar vein, data that emerge as a side product but are not essential for a publication are often left unpublished producing so-called publication bias [108]. This eventually results in a waste of resources because other researchers then have to repeat the work in order to produce a balanced dataset for ML applications [86].

However, few of these available databases are ready to use with informatics techniques because they lack the uniform data formats or application programming interfaces required for informatics software [167].

2.1.4 | Materials informatics

The databases mentioned above contain information on numerous properties of known materials and are essential for the success of materials informatics [86]. For more general information, the reader is referred to [145], which gives an introducing review on materials informatics, that aptly describes the concept of big data in materials science. Another publication on materials informatics is, for example, [169] which introduces four main research areas in materials informatics: standardization of representation and exchange of material data; organization, management, retrieval, filtration



and correlation of material data; material graphics; and data mining and knowledge discovery of material data. Another review on materials informatics mostly focuses on atomic-scale modeling [167]. But it also promotes the idea to expand materials databases to make more data easily accessible to informatics. The widespread use of such data requires the digitalization and structuring of materials data. The data must also be easily sharable and accessible. Services that provide software interfaces to allow for automated data querying, processing, and access are evolving, for example, the Materials Data Facility [14] and Citrination [47, 116]. Wagner et al [162] propose a workflow for a materials informatics problem focusing on (a) the assembly of primary features, (b) the construction of an exploratory model, (c) refinement of the model to satisfactory accuracy, and (d) final training and deployment. By proceeding in an iterative fashion upwards in complexity, the final model will be as simple as possible improving its explainability and interpretability. The shorter, but more philosophical report of Rajan et al [125] focuses on the role of materials informatics that allows one to survey complex, multiscale information in a high-throughput, statistically robust, and yet physically meaningful manner.

2.2 | ML for metallic material characterization

Mechanical material properties are characteristics to be precisely predicted and controlled as they are strongly linked to and highly affected by process parameters and resulting microstructures [15]. The basic idea of using ML methods for material property prediction is to analyze and map the relationships (nonlinear in most cases) between the properties of a material and their related characteristics by extracting knowledge from existing experimental or simulated data [86]. Mechanical behavior in simulations is often described by means of constitutive equations [15].

Research on the macroscopic performance of materials mainly focuses on the structure-activity relationship between the macroscopic (eg, mechanical and physical) properties of a material and its microstructure [86]. Many material parameters can be estimated to within an order of magnitude using elementary physical ideas. Whenever these parameters cannot be reliably estimated as such, ML approaches can be helpful which then require experimental or simulative data [88].

Experimental testing methods which can be used on metals help understand materials and their properties better. Typical destructive tests are bend test, impact test, hardness test, tensile test, fatigue test, corrosion resistance test, or wear test, see for example [104]. In the following efforts of the materials community to enhance such tests and their results with ML methods will be discussed.

2.2.1 | Corrosion

Corrosion detection and monitoring are essential diagnostic and prognostic means for preserving material "health" and reducing life-cycle cost of industrial infrastructures, ships, aircrafts, ground vehicles, pipelines, and so on [6]. More recently, ML approaches have shown great potential to improve corrosion detection [66]. They will aid a human inspector and significantly cut down on the time and cost associated with inspecting for example civil infrastructure and eliminate the need for dependence on prior knowledge and human effort in designing features [12]. Popular approaches are the application of ANNs for image processing-based corrosion detection.

Convolutional neural networks (CNNs) were used for corrosion detection by investigation of images and identification of rusty parts in the image [12]. The proposed CNN outperforms state of the art vision-based corrosion detection approaches that are developed based on texture and color analysis using a simple multilayered perceptron network. Model input is an image of the material and region of interest, output is the classification information: corroded/not corroded for a sliding window over the image. Overfitting due to small datasets can be avoided by using pretrained networks. Another example for the application of CNNs for corrosion detection is proposed by Bastian et al [13]. Similar to Atha et al [12] images are used as input and classified into any of the four classes: no corrosion, low-level corrosion, medium-level corrosion, and high-level corrosion. More investigations of similar CNN-based publications for corrosion detection are listed in [13].

Fang et al [41] proposed a novel hybrid methodology combining genetic algorithms and support vector regression, which is capable of forecasting the atmospheric corrosion depth of metallic materials such as zinc and steel. This hybrid approach is capable of solving nonlinear regression estimation problems in materials science. The genetic algorithms are adopted to automatically determine the optimal hyper-parameters for support vector regression. The inputs for the support vector regression are temperature, time of wetness, exposure time, sulfur dioxide concentration, and chloride concentration, respectively. The outputs are the predicted corrosion depth for zinc or steel.



Another support vector machine (SVM) approach was applied by Hoang et al [54] for image processing-based detection of pipe corrosion. The image texture including statistical measurements of image colors, gray-level co-occurrence matrix, and gray-level run length is employed to extract features of the pipe surface. SVM optimized by differential flower pollination is then used to construct a decision boundary that can recognize corroded and intact pipe surfaces by blockwise classification of the original image. Also with the application of a support vector regression and based on a much smaller database (trained on only 46 samples), the corrosion rate of 3C steel in different environments was predicted based on five different seawater environment factors, including temperature, dissolved oxygen, salinity, pH-value, and oxidation-reduction potential. The prediction error was very small [170].

Jimenez et al [66] compare various ML approaches (ANN, SVM, classification tree, and *k*-nearest neighbor) for automatic pitting corrosion detection in 316L stainless steel. Model input are environmental variables such as chloride concentration, pH, and temperature while the output is an information about the material being corroded or not. The models based on ANNs and SVM with linear kernel were demonstrated to be a valuable tool to be applied for this purpose. The classification performance for ANN and SVM is much better compared to *k*-nearest neighbor and classification tree models for this application. The principal advantage compared with the traditional techniques is that it is not necessary to apply a surface analysis technique to study corrosion behavior of a material.

2.2.2 | Fatigue

Fatigue as the weakening of a material caused by cyclic loading that results in progressive and localized structural damage and the growth of cracks can also be predicted with the incorporation of ML methods. The prediction of fatigue in welded structures for a wide range of structural materials by multiscale FEM and ML was proposed by Shiraiwa et al [140]. Two ML algorithms are applied: one is deterministic ML based on the traditional methods, and the other is model-based ML. Deterministic ML such as multivariate linear regression and ANNs use chemical composition, processing parameters (reduction ratio, heat treatment), inclusion sizes, and fatigue strength as input features to accurately predict fatigue strength. For model-based ML, microstructures and stress-strain curves in 40 low carbon steels with different chemical compositions and heat treatment conditions were prepared to create the learning dataset which was used to train an ANN. This approach allows for incorporation of prior knowledge of structure and property, and it can account for uncertainty such as scattering of fatigue life.

Another attempt to identify novel connections between fatigue properties and a variety of material parameters with the help of ML algorithms has been made by Agrawal et al [8] for the prediction of fatigue strength of steel from composition and processing parameters such as chemical composition, upstream processing details, heat treatment conditions and mechanical properties. Various ML methods such as basic regression, decision trees, SVM, ANN were used. Most success showed ensemble methods and individualized methods for different types of materials.

Machine learning techniques have also been utilized to predict material fatigue life for P91 steel base metal based on the hold time in fatigue tests [178]. A combined approach of genetic algorithms and SVM is used to predict the fatigue life with high accuracy.

Abdalla et al [2] use an ANN radial basis function model, taking the maximum tensile strain and pressure ratio as input, and put forward the model of fatigue life of steel reinforcing bars.

2.2.3 | Creep

Creep is a type of metal deformation that occurs at stresses below the yield strength of a metal, generally at elevated temperatures. Creep rupture is becoming increasingly one of the most important problems affecting behavior and performance of power production systems operating in high temperature environments and potentially under irradiation as is the case of nuclear reactors. Creep rupture forecasting and estimation of the useful life are required to avoid unanticipated component failure and cost ineffective operation [25]. The material behavior is influenced by the multidimensional interdependencies between the individual elements of the chemical composition, the heat treatment parameters, product form, tensile properties and microstructure, which are difficult to describe using simple analytical methods. Modeling with ML techniques therefore seems to be an interesting alternative. Moreover, the application of ML takes away the requirement for long and expensive experiments [46].

For the design of materials, creep is considered an important material property. However, quite often such designs only focus on one objective (eg, creep) without considering the comprehensive design of multi-property [164]. For the



investigation of creep rupture life and rupture strength of austenitic stainless steels [146] once again ANNs are popular models. For the prediction of creep rupture life and the creep rupture stress for a given stress, the training database for the input parameters contains test conditions (stress and temperature), chemical composition, solution treatment temperature, and time (the latter being available in a very limited number of cases), nature of the quench following, grain size, and logarithm of ruptured life for a broad variety of stainless steels.

Chatzidakis et al [25] employ and compare general regression neural networks, ANNs and Gaussian processes to capture the underlying trends and provide creep rupture forecasting. Input parameters are experimental creep rupture data. However, the overall performance of the developed models was insufficient.

In the study of Shin et al [139], the five different ML models random forest (RF), linear regression, *k*-nearest neighbor, kernel ridge, Bayesian ridge are applied for the prediction of Larson-Miller parameters which represent the creep behavior. From a wide range of available features (466), relevant ones are selected with optimization approaches and different set-ups of features and models evaluated. Highest accuracy was obtained by RF for a varying number of top ranking features between 5 and 21.

For the prediction of rupture and creep rupture stress of 9%Cr steels, a multilayer perceptron neural network was applied with the input parameters chemical composition, heat treatment information, geometrical form of the investigated components [46].

2.2.4 | Flow behavior and work hardening

In sheet metal forming operations, mechanical properties of the sheet material such as flow stress or stress-strain curves greatly influence metal flow and product quality [52]. The flow stress can be determined by tensile tests which provide uniaxial information about stress-strain behavior. Hardness tests measure the resistance of material to an indenter, and hardness correlates well with flow stress. Compression testing reveals flow stress but is complicated by friction and buckling. Bend testing and torsion testing provide a good measure of flow stress and fracture resistance but are complicated by radial stress and strain gradients [173].

One of the main advantages of applying ML approaches is that it is not necessary to postulate a mathematical model at first, which is quite difficult because of the nonlinearities in the response of the deformation behaviors of the materials under elevated temperatures and strain rates and the factors affecting the flow stress. Lin et al [82] propose a feed forward back propagation ANN model to predict the constitutive flow behaviors of 42CrMo steel during hot deformation, and investigate the general nature of the influence of strain, strain rate and temperature on the compressive deformation characteristics of 42CrMo steel. The capability of the developed ANN model to predict the flow stress level, the strain hardening, and flow softening stages is also investigated. The inputs of the ANN are deformation temperature, log strain rate, and strain whereas flow stress level, the strain hardening, and flow softening stages are the output. Low absolute relative errors for training and test data promise a good generalizing model.

Gupta et al [51] predict flow stress based on a feed forward ANN trained with the back propagation algorithm. Input features were strain, strain rate, and temperature. The implemented ANN produced more accurate results than conventional mathematical models such as Johnson Cook, modified Zerilli-Armstrong, and modified Arrhenius. support vector regression based flow stress prediction for austenitic stainless steel 304 with strain, strain rate and, temperature as inputs and the flow stress as output was proposed by Desu et al [30] and provides more accurate results than the conventional mathematical models.

The identification of work hardening properties of steel and an aluminum alloy from indentation tests is described by Meng et al [101]. The authors propose a material parameter identification protocol based only on the imprint shape of the indentation test using orthogonal decomposition and manifold learning for the prediction of the strain hardening exponent and yield stress.

ANN modeling for anisotropic mechanical properties and work hardening behavior of Inconel 718 alloy at elevated temperatures was proposed by Mahalle et al [91]. Strain and temperature are used to successfully predict material properties such as ultimate strength, yield strength, elongation, and strain hardening coefficient.

2.2.5 | Tensile properties

Tensile properties indicate how the material will react to forces being applied in tension. Determining the tensile properties is crucial because it provides information about the modulus of elasticity, elastic limit, elongation, proportional 8 of 21 Mitteilu

limit, reduction in area, tensile strength, yield point, yield strength, and other tensile properties [123] which then define the state of the material, its longevity, or its ability to perform in an application. Thus, the accurate prediction of tensile properties has great importance for the service life assessment of structural materials [163].

The ultimate tensile strength (UTS) of iron castings which gives information about the capacity of a metal to resist deformation when subject to a certain load was predicted based on a variety of input features such as composition, size of casting, cooling speed or thermal treatment (25 variables) to gain information about the mechanical properties of a foundry and thus predict foundry defects [131]. Estimating the value of UTS is one of the hardest issues in foundry production, due to many different circumstances and variables that are involved in the casting process. Bayesian networks, k-nearest neighbor, and ANNs were used for classification of the UTS. All of the investigated approaches perform well, but ANNs outperformed the other classifiers. Similar to the aforementioned study, Sterjovski et al [148] also used ANNs (of back propagation type ANN) to predict mechanical properties of steel, such as the impact toughness of quenched and tempered pressure vessel steel exposed to multiple postweld heat treatment cycles; the hardness of the simulated heat affected zone in pipeline and tap fitting steels after in-service welding; and the hot ductility and UTS of various microalloyed steels over the temperature range for strand or slab straightening in the continuous casting process. Input parameters were composition, cooling rate, temperature, and thickness. It was shown that ANNs could successfully predict all mechanical properties investigated. Another example for the application of ANNs is proposed by Sankar et al [130] who predict elongation, self-tempering temperature, and yield strength for reinforcement steel bars subjected to thermomechanical treatment based on two input parameters (bar diameter and quenching duration). The numerical results of the ANN are compared with experimental results and are found to be in good agreement.

Pruning and predator prey algorithms were applied by Datta et al [28] which were able to extract more knowledge from the input data, than typically possible with conventional ANN analysis. Alloy composition and the thermo-mechanical controlled processing parameters, deformation in different temperature zones, finish rolling temperature, and cooling rate of high strength steels have been taken as input parameters, whereas UTS, yield strength, and percentage elongation were predicted. It was shown that in this type of steel the yield strength depends mostly on the solid solution hardening and the microstructural constituents while UTS is more influenced by the precipitation hardening, but all these strengthening mechanisms have a negative effect on the ductility of the steel.

Pattanayak et al [117] investigated the role of the composition and processing parameters on the mechanical properties of API grade microalloyed pipeline steel, in respect to its strength, impact toughness, and ductility. ANN models, capable of prediction and diagnosis in nonlinear and complex systems, are used to obtain the relationship of composition and processing parameters with said mechanical properties. Then the models are used as objective functions for the multiobjective genetic algorithms for evolving the tradeoffs between the conflicting objectives of achieving improved strength, ductility and impact toughness. The Pareto optimal solutions were analyzed successfully to study the role of various parameters for designing pipeline steel with such improved performance.

For the ANN-based prediction of yield strength, UTS, ductility (elongation, and reduction of area) of ferritic steel weld metals appropriate for the welding of high strength low alloy steels, Metzbower et al [103] use the chemical composition of as deposited weld beads, and the cooling rate. The various established ANN models are found to work well once combined to an ensemble of best models. The ensemble produced a more reliable prediction than an individual model and reproduced known metallurgical trends well. Poudel et al [120] also compare different ML models and ensembles for the prediction of UTS, yield strength, elongation of steel bars instead of performing tensile tests. Input parameters were process parameters from the rebar manufacturing process such as for example material composition, temperatures, rod diameter, rod speed, and cooling rate. Investigated classification algorithms were among others multivariate linear regression, principal components regression, partial least squares regression, ANNs, and locally weighted regression. The main conclusion from this research is that any single model cannot efficiently describe the complicated relationship of the input-output space of the rebar manufacturing process. Ensemble methods, however, as combination of ML approaches used together in conjunction with some model selection or model weighting techniques lead to robust prediction systems.

Applications of traditional ML algorithms for the prediction of tensile properties were proposed by [137, 138, 151, 163, 164]. Shigemori et al [137, 138] reported about the successful application of locally weighted regression in predicting the tensile strength for a certain type of steel product which is produced by hot rolling. As input, 18 items were selected from chemical composition, heating, rolling, and cooling temperature. These variables have a clear physical causal relationship with the output variable. Furthermore, least squares SVMs are suitable approaches for the prediction of the elastic modulus and yield stress of materials. In [151] FEM-simulated load-indentation curves of Al6061 and Al7075 are investigated for the determination of these material parameters based on a training set of large strain-large deformation FEM for the



simulation of indentation tests. Characteristic features are extracted from the load-indentation curves and used in the ML model. The proposed least squares SVM model is capable of predicting reasonably accurately the elastic modulus and yield stress of materials based on the load-indentation curves of dual conical indenters with different half-angles. For material design, a RF model in combination with an optimization algorithm was found to relate yield strength, impact toughness and total elongation with material composition information and treatment parameters for the production of RAFM steels [163, 164]. For yield strength, highly correlated features were tempering temperature and C content; and tempering time and Cr content for elongation. The accuracy and generalization ability of the RF were acceptable ($R^2 > 85\%$).

An interesting study different from most approaches using material composition information as input features for material parameter prediction was proposed by Fragassa et al [45] for the prediction of the tensile behavior of cast alloys such as yield strength, ultimate strength, ultimate strain, and Young's modulus, by a pattern recognition analysis on experimental data and the application of RF, ANN, and *k*-nearest neighbor. All information is directly taken from micrographs. For the prediction of UTS and yield strength ANNs show the best results.

Another application of ANNs is the interpretation of acoustic emission data for failure prediction. Christopher et al [26] propose the prediction of the ultimate strength of aluminum/silicon carbide (Al/SiC) composites by using acoustic emission parameters through ANN analysis. This approach was earlier pursued for the prediction of the ultimate strength of unidirectional T-300/914 tensile specimens using acoustic emission response and an ANN back propagation algorithm [132].

To prevent surface cracks on cast steel, its hot ductility must be monitored. Experimental investigations are difficult to execute. Thus, ML models are proposed for the prediction of hot ductility. For example, a multivariate linear regression [85] predicted hot ductility and grouped 12 chemical elements that had similar experimental effects on ductility. The cooling condition used here is different from actual continuous-casting conditions, so the model is difficult to apply in commercial setups. A back-propagation ANN method has been used [148] to predict hot ductility for various microalloyed steels over the temperature range for strand or slab straightening in the continuous casting process. However, the recorded data are limited and do not lead to a generalizing model. Additionally, the ANN used only one hidden layer, so it has a limited ability to describe the complex relationship between input and output. Also an ANN model was used to predict high-temperature ductility of various steel grades from their composition and thermal history (described by five experimental variables) [78]. The developed model can predict ductility for a wider composition range and thermal history than previous studies have achieved. Therefore, it can be used effectively in commercial production.

2.3 | ML for the SPT

As mentioned above, the estimation of tensile properties of materials such as elastic modulus, yield strength, or strain hardening exponent is considered to be of fundamental importance. Conventional tests are destructive in nature and require reasonable specimen cross-section and volume. There are situations where limited volume of material is available for property assessment, like in material development, failure analysis and remaining life assessment of in-service components, materials for pressure vessels, turbines, thermal power plants, or chemical processing industries [9, 60]. In these situations, test techniques using small volume specimens become more attractive. Small specimen test techniques have been established as a reliable alternative to the traditional tensile test, as the results of these test techniques are in good agreement with the tensile test and are reproducible when tested under controlled conditions [11].

The extraction of mechanical properties of in-service materials through such small specimen test techniques has become more popular in recent years. Of the available small specimen test techniques, SPT methods have proven to be promising [11, 92, 93, 165]. In the SPT, a thin disk like specimen of small diameter is deformed in a miniaturized deep drawing experiment. The measurable output is the load displacement curve of the punch, which contains information about the elasto-plastic deformation behavior and about the strength properties of the material [3]. Such test methods are basically nondestructive in nature and are proficient enough to extract the flow properties of the materials only using small volumes of material. After more than 40 years' development, there are some standards in SPTs, such as ASTM-F2183 [147], GB/T-29459 [143], and CEN CWA-15627 [22].

A prerequisite for using this test is to establish correlations between SPT and conventional tests [10, 65], such as relations between the tensile test and the Erichsen cupping test for the determination of the critical damage value curves, the initiation time and location of fracture [72]. Other examples are the correlation between the SPT and tensile tests for the estimation of the tensile strength [74] or the construction of stress-strain curves only from indentation tests [174].

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Here the link between mean pressure and the indent diameter, obtained from indentation tests, to the stress and strain determined from a tensile test was obtained by the empirical relation of Tabor [152].

Exploiting force-displacement curves (FDCs) from spherical indentation simplifies the identification of mechanical properties. Parameters such as hardness, deformation mode, yield stress, Young's modulus of the indented material can also be extracted out of the imprint shape, which is a valid alternative for using the indentation curve [102]. For example, Milivcka et al found linear correlations between the SPT maximum force and the tensile strength of a 9% Chromium steel for creep resistant high temperature applications [105].

Detailed analyses of stress and strain in the SPT disc have been performed by means of analytical elastic-plastic modeling, see for example [20, 23] and by FEM to underpin the empirical correlations [4, 21, 141]. FEM simulations are mainly used to generate FDCs based on an assumed material constitutive law which are then compared with experimental FDCs. This approach is called inverse analysis [83, 87]. For more information, see for example [118, 168]. However, the solution to the inverse identification problem is nonunique. Furthermore, there are other problems such as insufficient accuracy, indentation frame/machine compliance, noisy input data, difficulty in determining the exact starting point of the load (force)-displacement curve, bending of the specimen in thin-sheet indentation and FE mesh dependence [102].

Various mechanical properties that can be extracted through SPT are for example fracture mode [69, 99, 106], yield stress [21, 70, 84], UTS [59, 96, 105, 165] (based on hydraulic bulge tests which are similar to SPT but high-pressure hydraulic oil is used instead of punch to cause specimen deformation); Young's modulus [42, 57, 114, 153], fracture toughness [4,18,76,83,95,107,166], creep properties [36, 48], or elastic plastic properties [43, 128].

These examples show that the SPT is a widely used approach for the determination of various material parameters from a small amount of material. However, the approach also has a few disadvantages, which have to be kept in mind. To begin with, the selected sample size may not represent the bulk material. The sample size effect has an important influence on the mechanism of the fracture of SPT, because often the thickness of the SPT sample is only 5 to 8 times of the mean grain size. The changes in grain-boundary distribution and the grain orientation, that resulted from the sample size effect, may affect the mechanical behaviors of the SPT samples. These parameters may affect the mechanism of fracture on small scale material samples [144]. In addition, the SPT response is sensitive to various test parameters such as specimen shape, specimen thickness, test speed, ball diameter, clamping force, and material. For more information, the reader is referred to [10], where an extensive collection of SPT configurations is listed.

In recent years, using ML approaches on SPT data has become popular. Most approaches incorporate ANNs for the identification of material properties from SPT data. Abendroth et al [5] identify ductile damage and fracture parameters from the SPT using ANNs. FEM is used to compute the load displacement curves. Via a systematic variation of the material parameters a data base is built up, which is used to train the ANNs. This neural network can be used to predict the load displacement curve of the SPT for a given material parameter set. The identified material parameters are validated by independent tests on notched tensile specimens. A similar approach was followed by Abendroth et al [3, 4] earlier for the estimation of the hardening and damage parameters of high strength steels. The combination of FEM, SPT data, and ANNs was used to identify the parameters of the Gurson-Tvergaard-Needleman model for ductile damage and fracture parameter prediction.

Linse et al [83] use synthetic load displacement curves generated via FEM for a variety of material parameters. This database is then used to train ANNs, which approximate the load displacement curves of the SPT as a function of the material parameters. The identification procedure itself consists of an optimization algorithm, minimizing the difference between the measured load displacement curves and its approximation by the neural networks until the true set of material parameters is found. Prerequisite is an accurately working FEM-Model. The approach was applied for the identification of hardening parameters and Weibull-parameters in the brittle and brittle-ductile transition region of two nonirradiated reactor vessel steels.

Another employment of FEM simulations of tensile, bulge, Erichsen tests is proposed by Abbassi et al [1] for the calculation of damage parameters. The Gurson-Tvergaard-Needleman model was employed. An identification procedure based on ANNs is used to determine the material parameters of the Gurson-Tvergaard-Needleman model. The ANN was trained by using the FEM results of the notched tensile test with varying the damage parameters. A good efficiency for the identification of damage parameters was proven. The Poisson ratio can also be predicted using a similar approach (combination of SPT and FEM and ANN) [56]. Using FE simulations, the relation between the material parameters and the quantities characterizing the depth-load response is calculated. An approximate inverse function represented by an ANN is derived on the basis of these data.

In contrast to all the ANN-based approaches described so far, the study of Meng et al [102] suggests the use of manifold learning for indentation-based material characterization. However, FEM was also used there. Input parameters



were features describing the imprint shape of SPTs. The manifold learning approach was able to iteratively reduce the distance between the FE-simulated and the experimental imprint shapes in order to identify the material hardening parameters. The approach was successfully shown for three different materials: AISI1095 steel and two aluminum alloys EN AW-2017F and EN AW-5754F.

Another innovative approach suggests the application of digital video processing of the forming process created by the Erichsen cup test [129]. The OpenCV library [16] was used to develop a sheet-metal-forming image analysis program which identifies the indent shape and position to calculate the anisotropic coefficient which showed good agreement with tensile test results. However, results were biased by the environment around the testing area.

A code of practice on SPT was established in 2010 in order to harmonize the various test set-ups and to achieve a better comparability of the results of different labs [58].

3 | AN APPLICATION OF ML IN MATERIAL PROPERTY PREDICTION

Similar to the studies mentioned above, we want to apply ML models to extract material properties from SPT data since the SPT has long been recognized as a supportive means for the development and monitoring of structural materials.

Conventionally, the following empirical correlation has been used for the estimation of the UTS σ_{UTS} from FDCs [17, 21, 35, 77, 96]

$$\sigma_{UTS} = \beta_{UTS} \cdot F_m / (h_0 \cdot \nu_m) \tag{1}$$

with F_m being the maximum force, v_m the corresponding punch displacement, h_0 the initial specimen thickness, and β_{UTS} an empirical coefficient. However, various researchers [9, 77] propose the determination of σ_{UTS} based on the force F_i instead of F_m for the correlation with the UTS in order to avoid a strong dependence of the correlation factor β_{UTS} on the tensile properties of the material. $F_i = F(v_i)$ is the force at the intersection point v_i of FDCs for varying uniform elongations and constant UTS σ_{UTS} and constant initial flow stress σ_{y0} .

$$\sigma_{UTS} = \beta_{UTS} \cdot F(v_i) / (h_0^2)$$
⁽²⁾

with β_{UTS} being geometry and material dependent. More information about this empirical approach can be found in [9]. In addition to the empirical approach (1), (2), we propose two ML models to determine the UTS σ_{UTS} from SPT data, shown in Figure 1B.

3.1 | Experiments

SPT data of three ferritic-martensitic (f/m) Cr-steels and two bainitic reactor pressure vessel steels [9] were used to predict the UTS σ_{UTS} based on ML approaches. The UTS σ_{UTS} was obtained by tensile tests. A materials overview can be found in Table 1, which was adopted from Altstadt et al [9]. The P92 was available in the, as received condition, and in four different heat treatments. The specimens were of 0.5 mm thickness. SPT was conducted with a punch diameter of 2.5 mm, a receiving hole diameter of 4 mm, edge size 0.2 mm, and the edge being beveled. All tests were performed at a displacement rate of 0.5 mm/min. The displacement *v* was measured by an inductive sensor with an accuracy of $\pm 1 \mu$ m. The FDCs *F*(*v*) of SPTs at room temperature are shown in Figure 2A (mean of all available curves per material at room temperature). The *F*(*v*) curves of the material P91 for different temperatures are shown in Figure 2B. The curve for *T* = -177° C represents a brittle failure, the other test curves represent ductile failure. By using nine different materials and heats, and by testing selected materials at different temperatures for each type of material. Figure 1 shows the UTS σ_{UTS} for all investigated materials and temperatures. For more detailed information on the experimental set-up the reader is referred to Altstadt et al [9].

3.2 | Preprocessing

First, unified sampling points for the displacement were defined with a distance of $44 \,\mu m$ based on a sampling rate of roughly $1 \,\mu m$ in the raw data. Then each of the displacement values is defined as a feature, leading to 151 features for

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Material	Туре	Product form	Heat treatment	Referenc
P91	f/m steel	Hot rolled pipe ø $360 \times 50 \text{ mm}$	Normalization 1040-1100°C/30 min	[75]
			tempering 730-780°C/60 min	
P92	f/m steel	Hot rolled pipe ø 219 × 22 mm	AR: standard normalization + tempering	[55]
			HT1: AR + 800°C/2 h	
			HT2: AR + 760°C/2 h	
			HT3: AR + 750°C/2 h	
			HT4: AR + 740°C/2 h	
Eurofer97	f/m steel	Hot rolled plate 14.5 mm	Normalization 980°C/27 min/air cooling	[53, 154]
			tempering 760°C/90 min/air cooling	
22NiMoCr 37	Bainitic steel	Reactor pressure vessel Biblis C	890°C/4 h/water quenching	[181]
			650°C/7 h/air cooling	
15Kh2MFA	Bainitic steel	Reactor pressure vessel Greifswald	Original RPV manufacturing technology	[160]
		unit 8		
[atomia]	Number of tests	Test tomportures	TABLE 2 Overview of SPTs	
	Number of tests			
	23	$-1// \dots + 331$		
2-AK	5	Room temperature		
2-H11	5	Room temperature		
2-H12	4	Room temperature		
2-H13	5	Room temperature		
2-H14	5	Room temperature		
Iroler97	X	24 1 250		
	21	-24 + 250		
NiMoCr37	31	$-24 \dots + 250$ $-151 \dots + 332$		
2NiMoCr37 5Kh2MFA	31 33	$-24 \dots + 250 \\ -151 \dots + 332 \\ -150 \dots + 332$		
Kh2MFA	31 33	$\begin{array}{c} -24 \dots + 250 \\ -151 \dots + 332 \\ -150 \dots + 332 \end{array}$	(B)	n SPTc
NiMoCr37 Kh2MFA (A)	31 33	$-24 \dots + 250$ $-151 \dots + 332$ $-150 \dots + 332$ ((B) force-displacement curves from	n SPTs
NiMoCr37 Kh2MFA (A)	31 33	$ \begin{array}{c} -24 \dots + 250 \\ -151 \dots + 332 \\ -150 \dots + 332 \\ \end{array} $ ((B) force-displacement curves from ↓ ↓ measurement of preproce	n SPTs
NiMoCr37 Kh2MFA (A)	31 33	$ \begin{array}{c} -24 \dots + 250 \\ -151 \dots + 332 \\ -150 \dots + 332 \\ \end{array} $ ((B) force-displacement curves from measurement of punch force F_i at	n SPTs essing
NiMoCr37 Kh2MFA (A) [edW] utbue	31 33 *	$ \begin{array}{c} -24 \dots + 250 \\ -151 \dots + 332 \\ -150 \dots + 332 \\ \end{array} $ (((B) force-displacement curves from measurement of punch force F_i at onset of plastic instability	n SPTs
NiMoCr37 Kh2MFA (A) [edW] ubusts		$\begin{array}{c} -24 \dots + 250 \\ -151 \dots + 332 \\ -150 \dots + 332 \end{array}$	(B) force-displacement curves from measurement of punch force F_i at onset of plastic instability PCA	n SPTs essing
NiMoCr37 Kh2MFA (A) ⁰⁰⁰ ⁰⁰⁰ ⁰⁰⁰ ⁰⁰⁰ ⁰⁰⁰ ⁰⁰⁰ ⁰⁰⁰	31 33	$\begin{array}{c} -24 \dots + 250 \\ -151 \dots + 332 \\ -150 \dots + 332 \end{array}$	(B) force-displacement curves from measurement of punch force <i>F_i</i> at onset of plastic instability PCA ↓ ↓	n SPTs
NiMoCr37 Kh2MFA (A) [Wba] [000 000 000 - 000 000 - 000 000 - 000 -		-24 + 250 -151 + 332 -150 + 332 (-150 + 332 (-	(B) force-displacement curves from measurement of punch force F_i at onset of plastic instability PCA extraction of correlation	n SPTs essing Random
NiMoCr37 Kh2MFA (A) [Wba] [Wba	31 33	-24 + 250 -151 + 332 -150 + 332 ((B) force-displacement curves from measurement of punch force F_i at onset of plastic instability extraction of correlation factor β	n SPTs essing Random Forest
NiMoCr37 Kh2MFA (A) 000 00 000 000 - 000 000 - 000 000 - 000 - 00 - 000 - 00 - 000 - 000 - 000 - 000 - 000 - 000 - 000 - 000 - 000 - 000 -	31 33	$\begin{array}{c} -24 \dots + 250 \\ -151 \dots + 332 \\ -150 \dots + 332 \end{array}$	(B) force-displacement curves from measurement of punch force F_i at onset of plastic instability extraction of correlation factor β measurement of preproce PCA Linear regres- sion model	n SPTs essing Random Forest (RF)
NiMoCr37 Kh2MFA (A) (A) (A) (A) (A) (A) (A) (A) (A) (A)	31 33	-24 + 250 -151 + 332 -150 + 332 ((B) force-displacement curves from measurement of punch force F_i at onset of plastic instability extraction of correlation factor β empirical calculation (Linear model (LM)	n SPTs essing Random Forest (RF)
NiMoCr37 Kh2MFA (A) 000 000 000 000 000 000 000 000 000 00	31 33 200 -100 0	$-24 \dots + 250$ $-151 \dots + 332$ $-150 \dots + 332$ ((B) force-displacement curves from measurement of punch force <i>F_i</i> at onset of plastic instability extraction of correlation factor β empirical calculation	n SPTs essing Random Forest (RF)

FIGURE 1 A, UTS σ_{UTS} for the investigated materials and temperatures. B, Scheme



FIGURE 2 A, SPT data at room temperature for materials listed in Table 1. B, SPT data for P91 at different temperatures



FIGURE 3 A, Pairs plot of selected features, temperature and σ_{UTS} . B, Pairs plot of first five principal components, temperature, and σ_{UTS} .

displacement values up to 1.5 mm. Failure occurred for displacements larger than 1.5 mm. An additional feature is the test temperature. Now, each time series is considered to be an observation. Dimensions correspond to the number of displacement sampling points + 1 (temp).

As the number of dimensions (p) increases, the volume of the domain increases exponentially. This, in turn, requires more samples (n) from the domain to provide effective coverage of the domain for a learning algorithm. This problem was introduced earlier as the curse of dimensionality [64]. ML algorithms overcome the curse of dimensionality by making assumptions about the data and structure of the mapping function from inputs to outputs which adds a bias and leads to a loss of the generalization power of the ML model [64]. Ways to approach this problem are feature selection, projection methods, or the application of regularized algorithms.

Another problem when using FDC-data as input data for a ML algorithm is the multicollinearity of the features extracted from these curves. Exemplary correlations between features 1, 5, 45, 95, 150 and the temperature are shown in Figure 3A. Subsequently, PCA is performed on the FDC-data in order to eliminate colinearity in the features. The



FIGURE 4 A Empirical σ_{UTS} vs experimental σ_{UTS} . B, LM based σ_{UTS} vs experimental σ_{UTS} . C, RF based σ_{UTS} vs experimental σ_{UTS}

broad idea behind this scheme is that, in contrast to the original features, the principal components will be uncorrelated. Furthermore, one expects that a small number of principal components will explain most of the variance and therefore provide an accurate representation of the dataset [133]. For the present dataset a total of six principal components explain 99% of the overall variability. Now, the pairs plot of the new features in Figure 3B shows only horizontal or vertical lines between input features (principal components) which is typical for uncorrelated features.

3.3 | ML models

For the estimation of tensile properties such as the UTS σ_{UTS} from SPT data, empirical equations such as (1) or (2) can be used. The force F_i can be associated with the onset of plastic instability and is therefore well suited for a correlation with the UTS σ_{UTS} of the uniaxial tensile test as proposed in [9]. With the number of available data points being rather small a linear regression model (LM) in combination with PCA and a RF based on the original SPT data are proposed for the prediction of σ_{UTS} . The root mean squared error was used to compare the different approaches.

For the evaluation of the performance of the ML approaches 10-fold cross validation was conducted with the results shown in Table 3. Figure 4A-C shows the accuracy of the empirical approach, LM and RF, respectively.

3.4 | Application conclusions

The application presented here aims at the determination of the tensile property σ_{UTS} from SPT data. Altstadt et al [9] proposed an empirical equation which correlates the punch force at onset of plastic instability of the SPT with σ_{UTS} . The correlation factor β for the estimation of σ_{UTS} needs to be determined before the empirical equation can be applied and β is dependent on the sample geometry and the investigated material. The application of ML models on SPT data works well for the prediction of σ_{UTS} , as Table 3 shows. However, the amount of available data is very limited and the generalization power of the LM and RF for other materials, temperatures, geometries has to be critically evaluated. RFs usually do not generalize well. Thus, the application of RF on SPT data might not be sensible for $\sigma_{UTS} < 500$ MPa or $\sigma_{UTS} > 1000$ MPa. More experimental data will have to be collected to verify the RF model further. However, the interpretation of the RF based on SPT data is straightforward and leads to relevant information about the importance of features for the performance of the model.

The LM approach is quite similar to the empirical approach of Altstadt et al. However, not only one specific force value of the SPT is used to correlate to σ_{UTS} but the combination of several principal components generated from the SPT curves



is employed which might justify the slightly higher accuracy of the LM approach. Thus, LM is a great tool to analyze the relationships among the variables but in combination with PCA the interpretability of the resulting model is not intuitive.

Overall, correlations between SPT curves and σ_{UTS} clearly exist and SPT data can be used for the prediction of tensile properties such as σ_{UTS} for five different steels (nine heats). Once the model is established, there is no further need for the conduction of time consuming tensile tests. In the future, the models might have to be re-established on a broader database. For the determination of σ_{UTS} , FEM-simulations might not be necessary anymore. However, for more detailed information about the underlying constitutive laws via inverse analysis, FEM simulations will likely remain important an inevitable.

4 | DISCUSSION AND CONCLUSIONS

This survey addresses applications of machine learning strategies in materials science for material characterization. There exists a wide range of promising applications for ML in materials science, for example, material discovery, molecular dynamics, and global structural prediction. The demand for new approaches dealing with limited data is huge.

It was shown that data-driven approaches play a significant role in materials research in order to find relationships between the structure of a material and its properties. These relationships are often not linear. It is difficult to find generic patterns among multiple length scales and timescales. With experiments only, this cannot be achieved [125]. Therefore, data-mining techniques are indispensable for the recognition of correlations in the (experimental and simulated) data. As the amount of publicly available materials data grows, ML techniques in particular will be able to extract from these data sets scientific principles and design rules that could not be determined through conventional analysis [62].

The majority of early ML applications to materials science employed straightforward and simple-to-use algorithms, like linear kernel models and decision trees. Now, these proofs-of-concept exist for a variety of applications even though there is a lack of benchmarking datasets and standards [133]. To date, ML often cannot realize the expected accuracy when applied to some tasks due to insufficient material data. Therefore, a more accurate model that was trained on a small but accurate data set is only meaningful within the input data space but does not generalize well while a less accurate model on a wide input data space is better at generalization but less exact. Therefore, accelerating the construction of publicly accessible material databases is highly important for the future development of ML in materials science [168]. Another issue that holds back the development of precise ML models is the absence of failure data. In this case, a cultural shift toward the publication of all valid data, may it be positive or negative, is required [133, 168].

The majority of ML approaches in materials science is based on ANNs. However, conventional ANNs still suffer from several weaknesses such as the need for a large number of controlling parameters, the difficulty in obtaining stable solutions, the danger of overfitting and thus the lack of generalization capability [41]. However, ANNs have been enormously successful in understanding complex materials behavior, such as mechanical behavior (flow stress, hardness, tensile strength, fracture strength, and fatigue behavior) of metal alloys subjected to certain heat treatment and/or deformation procedures, as well as in the prediction of micro-structures and phases resulting from heat treatment and/or deformation processes. The most practical way to capture the complex dependence of a desired macroscopic property on the various process parameters is through such learning methods [109]. ANNs have the potential to minimize the need for expensive experimental investigation and/or inspection of structural materials used in various applications, hence resulting in large economic benefits for organizations [148].

In addition to ANN hybrid ML models or ensemble methods work well [41]. For this, multiple independent models are built and the final regression or classification result is usually obtained as an average over the ensemble. In this way, additional noise is introduced into the fitting process and overfitting is avoided [133]. However, there does not exist an overall solution that can be considered the best. The most appropriate model has always to be found specifically for the application and data situation.

Furthermore, this paper focuses on ML based material property prediction from SPT data. Such simple material tests have gained popularity over the last couple of years because even though they are cheap and simple to perform, they make accurate material characterization possible, especially for failure analysis and remaining life assessment of in-service components or structural parts. Nevertheless, a few disadvantages have to be taken into consideration. The small sample size might not represent the bulk material; the sample size effect influences the material properties; and the results of the SPT are sensitive to test parameters. However, also for SPT data, ML based models are popular for material parameter prediction. Most commonly found are ANNs, especially in combination with FEM for data generation. No application of traditional ML models to SPT data was found in the literature. The paper concludes with an application example which

uses FDCs of structural materials as the basis for predicting he UTS. Simple ML approaches presented here, such as linear regression models or RFs provide good results for predicting the UTS based on SPT data, even for a very small database.

As a consequence, it is possible to confirm the benefit of simple ML techniques in predicting mechanical properties such as the UTS based on as simple material tests as the SPT and the authors are sure that ML will positively shape materials science for the years to come.

ACKNOWLEDGEMENTS

This work is part of the Fraunhofer Lighthouse Project ML4P (Machine Learning for Production). The work profited from BiGmax, the Max Planck Society's Research Network on Big-Data-Driven Materials-Science. The data from Section 3 was kindly provided by Dr. Altstadt of the Helmholtz-Zentrum Dresden-Rossendorf, Germany. Open access funding enabled and organized by Projekt DEAL.

CONFLICT OF INTEREST

The authors declare no conflicts of interest.

REFERENCES

- [1] F. Abbassi, T. Belhadj, S. Mistou, and A. Zghal, Parameter identification of a mechanical ductile damage using artificial neural networks in sheet metal forming, Mater. Des. **45** (2013), 605–615.
- [2] J. A. Abdalla and R. Hawileh, Modeling and simulation of low-cycle fatigue life of steel reinforcing bars using artificial neural network, J. Frankl. Inst. 348 (2011), 1393–1403.
- [3] M. Abendroth and M. Kuna, Determination of deformation and failure properties of ductile materials by means of the small punch test and neural networks, Comput. Mater. Sci. 28 (2003), 633–644.
- [4] M. Abendroth and M. Kuna, Identification of ductile damage and fracture parameters from the small punch test using neural networks, Eng. Fract. Mech. **73** (2006), 710–725.
- [5] M. Abendroth and S. Soltysiak, Assessment of material properties by means of the small punch test, in Recent Trends in Fracture and Damage Mechanics, Springer International Publishing, Berlin, Germany, 2015, 127–157.
- [6] V. S. Agarwala, P. L. Reed, and S. Ahmad, Corrosion detection and monitoring-a review, CORROSION 2000, 2000.
- [7] A. Agrawal and A. Choudhary, Perspective: Materials informatics and big data: Realization of the "fourth paradigm" of science in materials science, Appl. Mater. 4 (2016), 053208.
- [8] A. Agrawal, P. D. Deshpande, A. Cecen, G. P. Basavarsu, A. N. Choudhary, and S. R. Kalidindi, Exploration of data science techniques to predict fatigue strength of steel from composition and processing parameters, Integr. Mater. Manuf. Innov. **3** (2014), 90–108.
- [9] E. Altstadt, M. Houska, I. Simonovski, M. Bruchhausen, S. Holmström, and R. Lacalle, On the estimation of ultimate tensile stress from small punch testing, Int. J. Mech. Sci. 136 (2018), 85–93.
- [10] S. Arunkumar, Overview of small punch test, Met. Mater. Int. 26 (2019), 719-738.
- [11] S. Arunkumar and R. V. Prakash, Estimation of tensile properties of pressure vessel steel through automated ball indentation and small punch test, Trans. Indian Inst. Metals **69** (2015), 1245–1256.
- [12] D. J. Atha and M. R. Jahanshahi, Evaluation of deep learning approaches based on convolutional neural networks for corrosion detection, Struct. Health Monit. **17** (2017), 1110–1128.
- [13] B. T. Bastian, J. N, S. K. Ranjith, and C. Jiji, Visual inspection and characterization of external corrosion in pipelines using deep neural network, NDT&E Int. 107 (2019), 102134.
- [14] B. Blaiszik, K. Chard, J. Pruyne, R. Ananthakrishnan, S. Tuecke, and I. Foster, The materials data facility: Data services to advance materials science research, JOM 68 (2016), 2045–2052.
- [15] F. E. Bock, R. C. Aydin, C. J. Cyron, N. Huber, S. R. Kalidindi, and B. Klusemann, A review of the application of machine learning and data mining approaches in continuum materials mechanics, Front Mater **6** (2019), 110.
- [16] G. Bradski and A. Kaehler, Learning OpenCV: Computer vision with the OpenCV library, O'Reilly Media, Inc, Sebastopol, CA, 2008.
- [17] M. Bruchhausen, S. Holmström, I. Simonovski, T. Austin, J. M. Lapetite, S. Ripplinger, and F. de Haan, Recent developments in small punch testing: Tensile properties and DBTT, Theor. Appl. Fract. Mech. **86** (2016), 2–10.
- [18] J. Bulloch, Toughness losses in low alloy steels at high temperatures: An appraisal of certain factors concerning the small punch test, Int. J. Press. Vessel. Pip. 75 (1998), 791–804.
- [19] K. T. Butler, D. W. Davies, H. Cartwright, O. Isayev, and A. Walsh, Machine learning for molecular and materials science, Nature 559 (2018), 547–555.
- [20] T. Byun, E. Lee, J. Hunn, K. Farrell, and L. Mansur, Characterization of plastic deformation in a disk bend test, J. Nucl. Mater. 294 (2001), 256–266.
- [21] E. Campitelli, P. Spätig, R. Bonadé, W. Hoffelner, and M. Victoria, Assessment of the constitutive properties from small ball punch test: Experiment and modeling, J. Nucl. Mater. **335** (2004), 366–378.
- [22] CEN CWA-15627, 15627 Worskshop Agreement: Small punch test method for metallic materials, 2006.
- [23] J. Chakrabarty, A theory of stretch forming over hemispherical punch heads, Int. J. Mech. Sci. 12 (1970), 315–325.

<u>16 of 21</u>



- [24] M. W. Chang, L. Ratinov, and D. Roth, *Guiding semi-supervision with constraint-driven learning*, Proceedings of the 45th Annual Meeting of the Association of Computational Linguistics, 2007, pp. 280–287.
- [25] S. Chatzidakis, M. Alamaniotis, and L. H. Tsoukalas, Creep rupture forecasting, IJMSTR 2 (2014), 1–25.
- [26] L. Christopher, T. Sasikumar, C. Santulli, and C. Fragassa, Neural network prediction of aluminum-silicon carbide tensile strength from acoustic emission rise angle data, FME Trans. **46** (2018), 253–258.
- [27] J. Correa-Baena, K. Hippalgaonkar, J. van Duren, S. Jaffer, V. R. Chandrasekhar, V. Stevanovic, C. Wadia, S. Guha, and T. Buonassisi, Accelerating materials development via automation, machine learning, and high-performance computing, Joule **2** (2018), 1410–1420.
- [28] S. Datta, F. Pettersson, S. Ganguly, H. Saxén, and N. Chakraborti, Designing high strength multi-phase steel for improved strength-ductility balance using neural networks and multi-objective genetic algorithms, ISIJ Int. **47** (2007), 1195–1203.
- [29] M. de Jong, W. Chen, R. Notestine, K. Persson, G. Ceder, A. Jain, M. Asta, and A. Gamst, A statistical learning framework for materials science: Application to elastic moduli of k-nary inorganic polycrystalline compounds, Sci. Rep. 6 (2016), 34256.
- [30] R. K. Desu, S. C. Guntuku, B. Aditya, and A. K. Gupta, Support vector regression based flow stress prediction in austenitic stainless steel 304, Procedia Mater. Sci. 6 (2014), 368–375.
- [31] H. Dette, N. Neumeyer, and K. Pilz, A simple nonparametric estimator of a strictly monotone regression function, Bernoulli **12** (2006), 469–490.
- [32] P. Dey, J. Bible, S. Datta, S. Broderick, J. Jasinski, M. Sunkara, M. Menon, and K. Rajan, Informatics-aided bandgap engineering for solar materials, Comput. Mater. Sci. 83 (2014), 185–195.
- [33] D. M. Dimiduk, E. A. Holm, and S. R. Niezgoda, Perspectives on the impact of machine learning, deep learning, and artificial intelligence on materials, processes, and structures engineering, Integr. Mater. Manuf. Innov. **7** (2018), 157–172.
- [34] C. Draxl and M. Scheffler, NOMAD: The FAIR concept for big data-driven materials science, MRS Bull. 43 (2018), 676–682.
- [35] P. Dymáček, F. Dobeš, P. Král, and J. Dvořák, *Investigation of fracture in pure aluminium after ECAP by means of small punch test*, Proceedings of the Acta Metallurgica Slovaca-Conference, Vol. 3, 2013, pp. 57–64.
- [36] P. Dymáček and K. Milička, Creep small-punch testing and its numerical simulations, Mater. Sci. Eng. A 510-511 (2009), 444–449.
- [37] D. Erhan, Y. Bengio, A. Courville, and P. Vincent, Visualizing higher-layer features of a deep network, Univ. Montreal 1341 (2009), 1.
- [38] S. Ermon, R. Bras, S. Suram, J. Gregoire, C. Gomes, B. Selman, and R. Van Dover, *Pattern decomposition with complex combinatorial constraints: Application to materials discovery*, 2014. arXiv preprint arXiv:1411.7441.
- [39] G. G. F. H. Allen and R. Sievers, *Crystallographic databases: Information content, software systems, scientific applications*, International Union of Crystallography. Data Commission, Chester, UK, 1987.
- [40] J. Fan, F. Han, and H. Liu, Challenges of big data analysis, Natl. Sci. Rev. 1 (2014), 293–314.
- [41] S. Fang, M. Wang, W. Qi, and F. Zheng, Hybrid genetic algorithms and support vector regression in forecasting atmospheric corrosion of metallic materials, Comput. Mater. Sci. 44 (2008), 647–655.
- [42] J. Field and M. Swain, A simple predictive model for spherical indentation, J. Mater. Res. 8 (1993), 297–306.
- [43] E. Fleury and J. Ha, Small punch tests to estimate the mechanical properties of steels for steam power plant: I. Mechanical strength, Int. J. Press. Vessel. Pip. **75** (1998), 699–706.
- [44] R. Fong and A. Vedaldi, *Interpretable explanations of black boxes by meaningful perturbation*, Proceedings of the IEEE International Conference on Computer Vision, 2017, pp. 3429–3437.
- [45] C. Fragassa, M. Babic, C. P. Bergmann, and G. Minak, Predicting the tensile behaviour of cast alloys by a pattern recognition analysis on experimental data, Metals **9** (2019), 557.
- [46] O. Frolova, E. Roos, K. Maile, and W. Müller, Representation of the heat specific creep rupture behaviour of 9% Cr steels using neural networks, Trans. MLDM **4** (2011), 1–16.
- [47] M. W. Gaultois, A. O. Oliynyk, A. Mar, T. D. Sparks, G. J. Mulholland, and B. Meredig, Perspective: Web-based machine learning models for real-time screening of thermoelectric materials properties, APL Mater. **4** (2016), 053213.
- [48] B. Gülçimen and P. Hähner, Determination of creep properties of a p91 weldment by small punch testing and a new evaluation approach, Mater. Sci. Eng. A **588** (2013), 125–131.
- [49] J. Graser, S. K. Kauwe, and T. D. Sparks, Machine learning and energy minimization approaches for crystal structure predictions: A review and new horizons, Chem. Mater. **30** (2018), 3601–3612.
- [50] C. R. Groom, I. J. Bruno, M. P. Lightfoot, and S. C. Ward, The Cambridge structural database, Acta Crystallogr. Sect. B 72 (2016), 171–179.
- [51] A. K. Gupta, H. N. Krishnamurthy, Y. Singh, K. M. Prasad, and S. K. Singh, Development of constitutive models for dynamic strain aging regime in austenitic stainless steel 304, Mater. Des. 45 (2013), 616–627.
- [52] G. Gutscher, H. C. Wu, G. Ngaile, and T. Altan, Determination of flow stress for sheet metal forming using the viscous pressure bulge (VPB) test, J. Mater. Process. Technol. **146** (2004), 1–7.
- [53] C. Heintze, C. Recknagel, F. Bergner, M. Hernández-Mayoral, and A. Kolitsch, Ion-irradiation-induced damage of steels characterized by means of nanoindentation, Nucl. Instrum. Methods Phy. Res. B 267 (2009), 1505–1508 Proceedings of the 16th International Conference on Ion Beam Modification of Materials.
- [54] N. D. Hoang and V. D. Tran, Image processing-based detection of pipe corrosion using texture analysis and metaheuristic-optimized machine learning approach, Comput. Intell. Neurosci. **2019** (2019), 1–13.
- [55] M. Houska and E. Altstadt, SP tests on p92 (T70175) at RT, Eur Commiss JRC SP tests on P92 (T70175) at RT Version 1.0, 2017.
- [56] N. Huber, A. Konstantinidis, and C. Tsakmakis, Determination of Poisson's ratio by spherical indentation using neural networks—Part I: Theory, J. Appl. Mech. **68** (2000), 218–223.

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- [57] N. Huber, D. Munz, and C. Tsakmakis, Determination of Young's modulus by spherical indentation, J. Mater. Res. 12 (1997), 2459-2469.
- [58] R. Hurst and K. Matocha, The european code of practice for small punch testing Where do we go from here, Metall. J. 63 (2010), 5–11.
- [59] R. Hurst and K. Matocha, Where are we now with the European code of practice for small punch testing, in Determination Mechanical Properties Materials by Small Punch Other Miniature Testing Techniques, K. Matocha, R. Hurst, and W. Sun, Eds., OCELOT SRO, Ostrava, 2012, 4–18.
- [60] A. Husain, R. Sharma, and D. Sehgal, Small punch and indentation tests for structural health monitoring, Process. Eng. **173** (2017), 710–717.
- [61] O. Isayev, D. Fourches, E. N. Muratov, C. Oses, K. Rasch, A. Tropsha, and S. Curtarolo, Materials cartography: Representing and mining materials space using structural and electronic fingerprints, Chem. Mater. **27** (2015), 735–743.
- [62] A. Jain, G. Hautier, S. Ong, and K. Persson, New opportunities for materials informatics: Resources and data mining techniques for uncovering hidden relationships, J. Mater. Res. **31** (2016), 977–994.
- [63] A. Jain, S. P. Ong, G. Hautier, W. Chen, W. D. Richards, S. Dacek, S. Cholia, D. Gunter, D. Skinner, G. Ceder, and K. A. Persson, Commentary: The materials project: A materials genome approach to accelerating materials innovation, APL Mater. **1** (2013), 011002.
- [64] G. James, D. Witten, T. Hastie, and R. Tibshirani, *An introduction to statistical learning*, Vol **112**, Springer, New York, NY, 2013.
- [65] A. Janča, J. Siegl, and P. Haušild, Small punch test evaluation methods for material characterisation, J. Nucl. Mater. 481 (2016), 201–213.
- [66] M. Jiménez-Come, I. Turias, and F. Trujillo, An automatic pitting corrosion detection approach for 316L stainless steel, Mater. Des. 56 (2014), 642–648.
- [67] I. T. Jolliffe and J. Cadima, Principal component analysis: A review and recent developments, Philos. Trans. R. Soc. A **374** (2016), 20150202.
- [68] M. Jordan and T. Mitchell, Machine learning: Trends, perspectives, and prospects, Science 349 (2015), 255–260.
- [69] J. Kameda, A kinetic model for ductile-brittle fracture mode transition behavior, Acta Metall. **34** (1986), 2391–2398.
- [70] J. Kameda and X. Mao, Small-punch and TEM-disc testing techniques and their application to characterization of radiation damage, J. Mater. Sci. 27 (1992), 983–989.
- [71] E. Kim, K. Huang, S. Jegelka, and E. Olivetti, Virtual screening of inorganic materials synthesis parameters with deep learning, NPJ Comput. Mater. **3** (2017), 1–9.
- [72] S. W. Kim and Y. S. Lee, Comparative study on failure prediction in warm forming processes of Mg alloy sheet by the FEM and ductile fracture criteria, Metall. Mater. Trans. B: Process Metall. Mater. Process. Sci. **45** (2013), 445–453.
- [73] S. Kirklin, J. E. Saal, B. Meredig, A. Thompson, J. W. Doak, M. Aykol, S. Rühl, and C. Wolverton, The open quantum materials database (OQMD): Assessing the accuracy of DFT formation energies, NPJ Comput. Mater. **1** (2015), 1–15.
- [74] I. Klevtsov, A. Dedov, and A. Molodtsov, Using of small punch test for determination of tensile properties for power plant steels, Proceedings of the 6th International DAAAM Baltic Conference Industrial Engineering, 2008.
- [75] S. Kohlar, *Gefüge und eigenschaften des warmfesten chromstahls p91*, Wissenschaftlich-Technische Berichte/Helmholtz-Zentrum Dresden-Rossendor HZDR-082 2017, 2017.
- [76] P. Konopík, J. Dzugan, and R. Prochazka, Determination of fracture toughness and tensile properties of structural steels by small punch test and micro-tensile test, Metal Brno Czech Republic **2013** (2013), 15–17.
- [77] K. Kumar, A. Pooleery, K. Madhusoodanan, R. Singh, J. Chakravartty, R. Shriwastaw, B. Dutta, and R. Sinha, Evaluation of ultimate tensile strength using miniature disk bend test, J. Nucl. Mater. **461** (2015), 100–111.
- [78] S. H. Kwon, D. G. Hong, and C. H. Yim, Prediction of hot ductility of steels from elemental composition and thermal history by deep neural networks, Ironmak. Steelmak. **47** (2020), 1176–1187.
- [79] F. Legrain, J. Carrete, A. van Roekeghem, S. Curtarolo, and N. Mingo, How chemical composition alone can predict vibrational free energies and entropies of solids, Chem. Mater. **29** (2017), 6220–6227.
- [80] Z. Li, J. R. Kermode, and A. De Vita, Molecular dynamics with on-the-fly machine learning of quantum-mechanical forces, Phys. Rev. Lett. **114** (2015), 096405.
- [81] Z. Li, S. Wang, W. S. Chin, L. E. Achenie, and H. Xin, High-throughput screening of bimetallic catalysts enabled by machine learning, J. Mater. Chem. A 5 (2017), 24131–24138.
- [82] Y. Lin, J. Zhang, and J. Zhong, Application of neural networks to predict the elevated temperature flow behavior of a low alloy steel, Comput. Mater. Sci. **43** (2008), 752–758.
- [83] T. Linse, M. Kuna, J. Schuhknecht, and H. W. Viehrig, Usage of the small-punch-test for the characterisation of reactor vessel steels in the brittle-ductile transition region, Eng. Fract. Mech. 75 (2008), 3520–3533.
- [84] T. Linse, M. Kuna, and H. W. Viehrig, Quantification of brittle-ductile failure behavior of ferritic reactor pressure vessel steels using the small-punch-test and micromechanical damage models, Mater. Sci. Eng. A **614** (2014), 136–147.
- [85] Q. Liu, X. Zhang, B. Wang, and B. Wang, Control technology of solidification and cooling in the process of continuous casting of steel, Sci. Technol. Cast. Process. 26 (2012), 169–204.
- [86] Y. Liu, T. Zhao, W. Ju, and S. Shi, Materials discovery and design using machine learning, J. Mater. 3 (2017), 159–177.
- [87] L. Lu, M. Dao, P. Kumar, U. Ramamurty, G. Karniadakis, and S. Suresh, Extraction of mechanical properties of materials through deep learning from instrumented indentation, Proc. Natl. Acad. Sci. U. S. A. 117 (2020), 7052–7062.
- [88] A. Lucas, Connecting microscopic physics with the macroscopic properties of materials in introductory physics courses, 2014, arXiv preprint arXiv:1402.2593.



- [89] X. Ma, Z. Li, L. E. Achenie, and H. Xin, Machine-learning-augmented chemisorption model for CO₂ electroreduction catalyst screening, J. Phys. Chem. Lett. 6 (2015), 3528–3533.
- [90] J. MacInnes, S. Santosa, and W. Wright, Visual classification: Expert knowledge guides machine learning, IEEE Comput. Graph. Appl. 30 (2010), 8–14.
- [91] G. Mahalle, O. Salunke, N. Kotkunde, A. K. Gupta, and S. K. Singh, Neural network modeling for anisotropic mechanical properties and work hardening behavior of inconel 718 alloy at elevated temperatures, J. Mater. Res. Technol. **8** (2019), 2130–2140.
- [92] M. Manahan, A. Argon, and O. Harling, The development of a miniaturized disk bend test for the determination of postirradiation mechanical properties, J. Nucl. Mater. **104** (1981), 1545–1550.
- [93] M. P. Manahan, The development of a miniaturized disk bend test for the determination of post-irradiation mechanical behavior, Ph.D. Thesis, Massachusetts Institute of Technology, 1982.
- [94] A. Mannodi-Kanakkithodi, T. D. Huan, and R. Ramprasad, Mining materials design rules from data: The example of polymer dielectrics, Chem. Mater. **29** (2017), 9001–9010.
- [95] X. Mao, M. Saito, and H. Takahashi, Small punch test to predict ductile fracture toughness JIC and brittle fracture toughness KIC, Scr. Metall. Mater. 25 (1991), 2481–2485.
- [96] X. Mao and H. Takahashi, Development of a further-miniaturized specimen of 3 mm diameter for tem disk (ø 3 mm) small punch tests, J. Nucl. Mater. 150 (1987), 42–52.
- [97] Materials Information Science National Institute of Materials Science, *SuperCon*, https://supercon.nims.go.jp/en/. Accessed July 21, 2020.
- [98] Materials Project, https://materialsproject.org/. Accessed July 21, 2020.
- [99] J. McNaney, G. Lucas, and G. Odette, Application of ball punch tests to evaluating fracture mode transition in ferritic steels, J. Nucl. Mater. **179-181** (1991), 429–433.
- [100] B. Medasani, A. Gamst, H. Ding, W. Chen, K. A. Persson, M. Asta, A. Canning, and M. Haranczyk, Predicting defect behavior in B2 intermetallics by merging ab initio modeling and machine learning, NPJ Comput. Mater. 2 (2016), 1–10.
- [101] L. Meng, P. Breitkopf, B. Raghavan, G. Mauvoisin, O. Bartier, and X. Hernot, Identification of material properties using indentation test and shape manifold learning approach, Comput. Methods Appl. Mech. Eng. **297** (2015), 239–257.
- [102] L. Meng, B. Raghavan, O. Bartier, X. Hernot, G. Mauvoisin, and P. Breitkopf, An objective meta-modeling approach for indentation-based material characterization, Mech. Mater. **107** (2017), 31–44.
- [103] E. Metzbower, J. deLoach, S. Lalam, and H. Bhadeshia, Neural network analysis of strength and ductility of welding alloys for high strength low alloy shipbuilding steels, Sci. Technol. Weld. Join. 6 (2001), 116–124.
- [104] M. A. Meyers and K. K. Chawla, Mechanical behavior of materials, Cambridge University Press, Cambridge, MA, 2008.
- [105] K. Milička and F. Dobeš, Small punch testing of p91 steel, Int. J. Press. Vessel. Pip. 83 (2006), 625–634.
- [106] T. Misawa, T. Adachi, M. Saito, and Y. Hamaguchi, Small punch tests for evaluating ductile-brittle transition behavior of irradiated ferritic steels, J. Nucl. Mater. **150** (1987), 194–202.
- [107] T. Misawa, S. Nagata, N. Aoki, J. Ishizaka, and Y. Hamaguchi, Fracture toughness evaluation of fusion reactor structural steels at low temperatures by small punch tests, J. Nucl. Mater. **169** (1989), 225–232.
- [108] A. Mlinarić, M. Horvat, and V. Šupak Smolčić, Dealing with the positive publication bias: Why you should really publish your negative results, Biochem. Med. 27 (2017), 447–452.
- [109] T. Mueller, A. G. Kusne, and R. Ramprasad, Machine learning in materials science: Recent progress and emerging applications, in Reviews in Computational Chemistry, A. Parrill and K. Lipkowitz, Eds., Wiley Online Library, New Jersey, USA, 2016, 186–273.
- [110] National Academy of Sciences (US). Committee on the Survey of Materials Science, National Academy of Sciences (US). Committee on Science, and Public Policy, Materials and Man's Needs: Materials Science and Engineering; Supplementary Report, Vol. 1, National Academies, 1974.
- [111] National Institute of Materials Science, https://www.nims.go.jp/eng/. Accessed July 21, 2020.
- [112] NIST Materials Data Repository. https://materialsdata.nist.gov/. Accessed July 21, 2020.
- [113] Nomad Repository. https://www.nims.go.jp/eng/. Accessed July 21, 2020.
- [114] W. Oliver and G. Pharr, An improved technique for determining hardness and elastic modulus using load and displacement sensing indentation experiments, J. Mater. Res. 7 (1992), 1564–1583.
- [115] G. Olson, Computational design of hierarchically structured materials, Science 277 (1997), 1237-1242.
- [116] J. O'Mara, B. Meredig, and K. Michel, Materials data infrastructure: A case study of the citrination platform to examine data import, storage, and access, JOM **68** (2016), 2031–2034.
- [117] S. Pattanayak, S. Dey, S. Chatterjee, S. G. Chowdhury, and S. Datta, Computational intelligence based designing of microalloyed pipeline steel, Comput. Mater. Sci. 104 (2015), 60–68.
- [118] I. Peñuelas, C. Betegón, C. Rodríguez, and J. Belzunce, *Inverse methods on small punch tests*, Numerical Simulations-Applications, Examples Theory, INCTECH, Rijeka (Croatia), 2011, pp. 311–330.
- [119] G. Pilania, A. Mannodi-Kanakkithodi, B. Uberuaga, R. Ramprasad, J. Gubernatis, and T. Lookman, Machine learning bandgaps of double perovskites, Sci. Rep. 6 (2016), 19375.
- [120] R. C. Poudel, T. Sakaguchi, and Y. Shimizu, A selective approach on data based quality prediction for quenched and tempered steel reinforcement bars, J. Chem. Eng. Jpn 46 (2013), 294–301.
- [121] B. Puchala, G. Tarcea, E. A. Marquis, M. Hedstrom, H. Jagadish, and J. E. Allison, The materials commons: A collaboration platform and information repository for the global materials community, JOM **68** (2016), 2035–2044.



- [122] P. Raccuglia, K. C. Elbert, P. D. Adler, C. Falk, M. B. Wenny, A. Mollo, M. Zeller, S. A. Friedler, J. Schrier, and A. J. Norquist, Machine-learning-assisted materials discovery using failed experiments, Nature 533 (2016), 73–76.
- [123] R. Rahman and S. Zhafer Firdaus Syed Putra, *Tensile properties of natural and synthetic fiber-reinforced polymer composites*, in *Mechanical and Physical Testing of Biocomposites*, *Fibre-Reinforced Composites and Hybrid Composites*, Elsevier, Sawston, UK, 2019, 81–102.
- [124] M. Raissi, P. Perdikaris, and G. Karniadakis, *Physics informed deep learning (part I): Data-driven solutions of nonlinear partial differential equations*, 2017. arXiv preprint arXiv:1711.10561.
- [125] K. Rajan, Materials informatics, Mater. Today 8 (2005), 38–45.
- [126] S. Rühl, *The inorganic crystal structure database (ICSD): A tool for materials sciences.* https://icsd.products.fiz-karlsruhe.de/. Accessed July 21, 2020.
- [127] J. E. Saal, S. Kirklin, M. Aykol, B. Meredig, and C. Wolverton, Materials design and discovery with high-throughput density functional theory: The open quantum materials database (OQMD), JOM 65 (2013), 1501–1509.
- [128] C. Sainte Catherine, J. Messier, C. Poussard, S. Rosinski, and J. Foulds, Small punch test: Epri-cea finite element simulation benchmark an inverse method for the estimation of elastic plastic behavior, Am. Soc. Test. Mater. **4** (2002), 350–370.
- [129] T. Sangkharat and S. Dechjarern, Using image processing on erichsen cup test machine to calculate anisotropic property of sheet metal, Proc. Manuf. **29** (2019), 390–397.
- [130] I. Sankar, K. Rao, and B. Murhty, Applying ANN to predict mechanical properties in TMT process, IE (I) Journal-MM 90 (2009), 3-6.
- [131] I. Santos, J. Nieves, Y. K. Penya, and P. G. Bringas, Machine-learning-based mechanical properties prediction in foundry production, ICCAS-SICE **2009** (2009), 4536–4541.
- [132] T. Sasikumar, S. Rajendraboopathy, K. Usha, and E. Vasudev, Artificial neural network prediction of ultimate strength of unidirectional T-300/914 tensile specimens using acoustic emission response, J. Nondestruct. Eval. 27 (2008), 127–133.
- [133] J. Schmidt, M. R. Marques, S. Botti, and M. A. Marques, Recent advances and applications of machine learning in solid-state materials science, NPJ Comput. Mater. 5 (2019), 1–36.
- [134] K. Schütt, H. Glawe, F. Brockherde, A. Sanna, K. Müller, and E. Gross, How to represent crystal structures for machine learning: Towards fast prediction of electronic properties, Phys. Rev. B **89** (2014), 205118.
- [135] A. Seko, H. Hayashi, K. Nakayama, A. Takahashi, and I. Tanaka, Representation of compounds for machine-learning prediction of physical properties, Phys. Rev. B 95 (2017), 144110.
- [136] C. Shen, C. Wang, X. Wei, Y. Li, S. van der Zwaag, and W. Xu, Physical metallurgy-guided machine learning and artificial intelligent design of ultrahigh-strength stainless steel, Acta Mater. 179 (2019), 201–214.
- [137] H. Shigemori, M. Kano, and S. Hasebe, Optimum quality design system for steel products through locally weighted regression model, J. Process Control **21** (2011), 293–301.
- [138] H. Shigemori and S. Kawamura, *Optimum quality design support system for steel products using locally-weighted regression model*, Proceedings of the SICE Annual Conference 2007, September, IEEE, 2007, pp. 810–815.
- [139] D. Shin, Y. Yamamoto, M. P. Brady, S. Lee, and J. Haynes, Modern data analytics approach to predict creep of high-temperature alloys, SSRN J. 168 (2018), 321–330.
- [140] T. Shiraiwa, F. Briffod, Y. Miyazawa, and M. Enoki, Fatigue performance prediction of structural materials by multi-scale modeling and machine learning, Proceedings of the 4th World Congress on Integrated Computational Materials Engineering (ICME 2017), 2017, pp. 317–326.
- [141] I. Simonovski, S. Holmström, and M. Bruchhausen, Small punch tensile testing of curved specimens: Finite element analysis and experiment, Int. J. Mech. Sci. 120 (2017), 204–213.
- [142] K. Simonyan, A. Vedaldi, and A. Zisserman, Deep inside convolutional networks: Visualising image classification models and saliency maps, 2013. arXiv preprint arXiv:1312.6034.
- [143] Small punch test methods of metallic materials for in-service pressure equipment, GB/T 29459, https://www.chinesestandard.net/PDF. aspx/GBT29459.2-2012. Accessed July 21, 2020.
- [144] M. Song, K. Guan, W. Qin, and J. A. Szpunar, Comparison of mechanical properties in conventional and small punch tests of fractured anisotropic A350 alloy forging flange, Nucl. Eng. Des. 247 (2012), 58–65.
- [145] Q. Song, A preliminary investigation on materials informatics, Chin. Sci. Bull. 49 (2004), 210.
- [146] T. Sourmail, H. Bhadeshia, and D. MacKay, Neural network model of creep strength of austenitic stainless steels, Mater. Sci. Technol. 18 (2002), 655–663.
- [147] Standard Method for Small Punch Testing of Ultra-High Molecular Weight Polyethylene Used in Surgical Implants, ASTM F2183-02, 2008, Accessed July 21, 2020.
- [148] Z. Sterjovski, D. Nolan, K. Carpenter, D. Dunne, and J. Norrish, Artificial neural networks for modelling the mechanical properties of steels in various applications, J. Mater. Process. Technol. 170 (2005), 536–544.
- [149] J. Struyf and S. Džeroski, *Clustering trees with instance level constraints*, Proceedings of the European Conference on Machine Learning, 2007, pp. 359–370.
- [150] M. Sundararajan, A. Taly, and Q. Yan, Axiomatic attribution for deep networks, 2017. arXiv preprint arXiv:1703.01365.
- [151] S. Swaddiwudhipong, K. K. Tho, Z. S. Liu, J. Hua, and N. S. B. Ooi, Material characterization via least squares support vector machines, Model. Simul. Mater. Sci. Eng. 13 (2005), 993–1004.
- [152] D. Tabor, The hardness and strength of metals, J. Inst. Met. **79** (1951), 1.
- [153] B. Taljat, T. Zacharia, and F. Haggag, Analysis of ball-indentation load-depth data: Part I. determining elastic modulus, J. Mater. Res. 12 (1997), 965–974.



- [154] A. A. Tavassoli, A. Alamo, L. Bedel, L. Forest, J. M. Gentzbittel, J. W. Rensman, E. Diegele, R. Lindau, M. Schirra, R. Schmitt, H. Schneider, C. Petersen, A. M. Lancha, P. Fernandez, G. Filacchioni, M. Maday, K. Mergia, N. Boukos, P. Baluc, E. Alves Spätig, and E. Lucon, Materials design data for reduced activation martensitic steel type EUROFER, J. Nucl. Mater. 329-333 (2004), 257–262 Proceedings of the 11th International Conference on Fusion Reactor Materials (ICFRM-11).
- [155] The Materials Genome Initiative, https://www.mgi.gov/. Accessed July 21, 2020.
- [156] Topological Materials Database, https://www.topologicalquantumchemistry.com/. Accessed September 21, 2020.
- [157] Z. W. Ulissi, A. J. Medford, T. Bligaard, and J. K. Nørskov, To address surface reaction network complexity using scaling relations machine learning and DFT calculations, Nat. Commun. 8 (2017), 1–7.
- [158] A. Vellido, J. D. Martín-Guerrero, and P. J. Lisboa, Making machine learning models interpretable, ESANN 12 (2012), 163–172.
- [159] M. Verleysen and D. François, *The curse of dimensionality in data mining and time series prediction*, Proceedings of the International Work-Conference on Artificial Neural Networks, 2005, pp. 758–770.
- [160] H. W. Viehrig, M. Scibetta, and K. Wallin, Application of advanced master curve approaches on WWER-440 reactor pressure vessel steels, Int. J. Press. Vessel. Pip. 83 (2006), 584–592.
- [161] L. von Rueden, S. Mayer, K. Beckh, B. Georgiev, S. Giesselbach, R. Heese, B. Kirsch, J. Pfrommer, A. Pick, and R. Ramamurthy, *Informed machine learning A taxonomy and survey of integrating knowledge into learning systems*, 2020. arXiv preprint arXiv:1903.12394.
- [162] N. Wagner and J. M. Rondinelli, Theory-guided machine learning in materials science, Front. Mater. 3 (2016), 28.
- [163] C. Wang, C. Shen, Q. Cui, C. Zhang, and W. Xu, Tensile property prediction by feature engineering guided machine learning in reduced activation ferritic/martensitic steels, J. Nucl. Mater. **529** (2020), 151823.
- [164] C. Wang, C. Shen, X. Huo, C. Zhang, and W. Xu, Design of comprehensive mechanical properties by machine learning and high-throughput optimization algorithm in RAFM steels, Nucl. Eng. Technol. **52** (2020), 1008–1012.
- [165] H. Wang, T. Xu, and B. Shou, Determination of material strengths by hydraulic bulge test, Materials 10 (2016), 23.
- [166] Z. X. Wang, H. J. Shi, J. Lu, P. Shi, and X. F. Ma, Small punch testing for assessing the fracture properties of the reactor vessel steel with different thicknesses, Nucl. Eng. Des. 238 (2008), 3186–3193.
- [167] L. Ward, M. Aykol, B. Blaiszik, I. Foster, B. Meredig, J. Saal, and S. Suram, Strategies for accelerating the adoption of materials informatics, MRS Bull. 43 (2018), 683–689.
- [168] J. Wei, X. Chu, X. Y. Sun, K. Xu, H. X. Deng, J. Chen, Z. Wei, and M. Lei, Machine learning in materials science, InfoMat 1 (2019), 338–358.
- [169] Q. Wei, X. Peng, X. Liu, and W. Xie, Materials informatics and study on its further development, Chin. Sci. Bull. 51 (2006), 498–504.
- [170] Y. Wen, C. Cai, X. Liu, J. Pei, X. Zhu, and T. Xiao, Corrosion rate prediction of 3C steel under different seawater environment by using support vector regression, Corros. Sci. 51 (2009), 349–355.
- [171] A. White, The materials genome initiative: One year on, MRS Bull. 37 (2012), 715–716.
- [172] S. Wold, K. Esbensen, and P. Geladi, Principal component analysis, Chemom. Intell. Lab. Syst. 2 (1987), 37-52.
- [173] R. N. Wright, Wire technology, Elsevier, Amsterdam, Netherlands, 2016.
- [174] M. Yetna N'Jock, D. Chicot, X. Decoopman, J. Lesage, J. Ndjaka, and A. Pertuz, Mechanical tensile properties by spherical macroindentation using an indentation strain-hardening exponent, Int. J. Mech. Sci. 75 (2013), 257–264.
- [175] A. Zakutayev, N. Wunder, M. Schwarting, J. D. Perkins, R. White, K. Munch, W. Tumas, and C. Phillips, An open experimental database for exploring inorganic materials, Sci Data 5 (2018), 180053.
- [176] M. Zeiler and R. Fergus, Visualizing and understanding convolutional networks, Proceedings of the European Conference on Computer Vision, 2014, pp. 818–833.
- [177] E. Zhang, M. Yin, and G. Karniadakis, *Physics-informed neural networks for nonhomogeneous material identification in elasticity imaging*, 2020. arXiv preprint arXiv:2009.04525.
- [178] L. Zhang, J. Lei, Q. Zhou, and Y. Wang, Using genetic algorithm to optimize parameters of support vector machine and its application in material fatigue life prediction, Adv. Nat. Sci. 8 (2015), 21–26.
- [179] Y. Zhang and C. Ling, A strategy to apply machine learning to small datasets in materials science, NPJ Comput. Mater. 4 (2018), 1-8.
- [180] Y. Zhu, N. Zabaras, P. Koutsourelakis, and P. Perdikaris, Physics-constrained deep learning for high-dimensional surrogate modeling and uncertainty quantification without labeled data, J. Comput. Phys. **394** (2019), 56–81.
- [181] C. Zurbuchen, Influence of specimen type, crack length and evaluation method on quasi-static and dynamic fracture toughness properties, Proceedings of the ASME Pressure Vessels Piping Conference Volume 5: High Pressure Technology; Nondestructive Evaluation Division; Student Paper Competition, July 2009, pp. 511–517.

How to cite this article: Stoll A, Benner P. Machine learning for material characterization with an application for predicting mechanical properties. *GAMM-Mitteilungen*. 2021;44:e202100003. <u>https://doi.org/10.1002/gamm</u>. 202100003