

## Photoinduced Ultrafast Transition of the Local Correlated Structure in Chalcogenide Phase-Change Materials

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(Received 4 September 2021; revised 19 December 2021; accepted 17 August 2022; published 19 September 2022)

Revealing the bonding and time-evolving atomic dynamics in functional materials with complex lattice structures can update the fundamental knowledge on rich physics therein, and also help to manipulate the material properties as desired. As the most prototypical chalcogenide phase change material,  $\text{Ge}_2\text{Sb}_2\text{Te}_5$  has been widely used in optical data storage and nonvolatile electric memory due to the fast switching speed and the low energy consumption. However, the basic understanding of the structural dynamics on the atomic scale is still not clear. Using femtosecond electron diffraction, structure factor calculation, and time-dependent density-functional theory molecular dynamic simulation, we reveal the photoinduced ultrafast transition of the local correlated structure in the averaged rocksalt phase of  $\text{Ge}_2\text{Sb}_2\text{Te}_5$ . The randomly oriented Peierls distortion among unit cells in the averaged rocksalt phase of  $\text{Ge}_2\text{Sb}_2\text{Te}_5$  is termed as local correlated structures. The ultrafast suppression of the local Peierls distortions in the individual unit cell gives rise to a local structure change from the rhombohedral to the cubic geometry within  $\sim 0.3$  ps. In addition, the impact of the carrier relaxation and the large number of vacancies to the ultrafast structural response is quantified and discussed. Our Letter provides new microscopic insights into contributions of the local correlated structure to the transient structural and optical responses in phase change materials. Moreover, we stress the significance of femtosecond electron diffraction in revealing the local correlated structure in the subunit cell and the link between the local correlated structure and physical properties in functional materials with complex microstructures.

DOI: [10.1103/PhysRevLett.129.135701](https://doi.org/10.1103/PhysRevLett.129.135701)

Because of the growth of the global amount of data, the demand for data storage and processing is increasing exponentially [1]. Chalcogenide phase change materials (PCMs) have been singled out as one of the best classes of prospective materials for all-photonic storage and memory [2] and electronic phase-change memory [1]. A laser or electrical pulse stimulates nonvolatile switching between a crystalline and an amorphous phase with atypically large differences in the optical reflectivity and the electrical resistivity. During such a structural phase switching, the crystallization process is the time-limiting step, while the amorphization process is the energy-intensive step. To optimize these two steps, one hot topic is to modulate the local structural geometry in the atomic level in the most prototypical phase-change material  $\text{Ge}_2\text{Sb}_2\text{Te}_5$  (GST-225) and the similar alloys along the tie line of  $\text{GeTe-Sb}_2\text{Te}_3$  [3–9]. The crystallization speed has been improved from tens of nanoseconds to subnanosecond by introducing prestructural ordering [4] and crystal precursors [5]. On the other hand, the energy consumption for the amorphization can be reduced by introducing the premelting disordering [6] and

controlling the local atomic switching [7]. The recent theoretical studies [8,9] propose that an ultrafast electronic excitation can reduce the energy consumption by introducing a direct solid-solid amorphization bypassing the molten state in GST-225. Overall, engineering the local structures in GST-225 plays a key role in improving the performance of the structural switching.

The crystalline GST-225 is an averaged rocksalt phase, accompanied by local distortions and huge vacancy concentrations, as shown in Fig. 1(a). The anion sublattice is fully occupied by Te atoms, whereas the cation sites are populated by Ge (40%), Sb (40%), and vacancies (20%) randomly [3]. The local structure in an individual unit cell of GST-225 is rhombohedrally distorted with random orientation to eight equivalent distortions along the  $\langle 111 \rangle$  directions [9–11]. Such local distortions with shorter and longer Ge(Sb)—Te bonds is termed as local Peierls distortions as shown in Fig. 1(a) (bottom). From a chemical perspective, the half-filled  $p$  band of Ge (Sb/Te) forms two bonds to the left and right atoms, and this bonding structure is termed resonant bond [12], metavalent bond [13], or

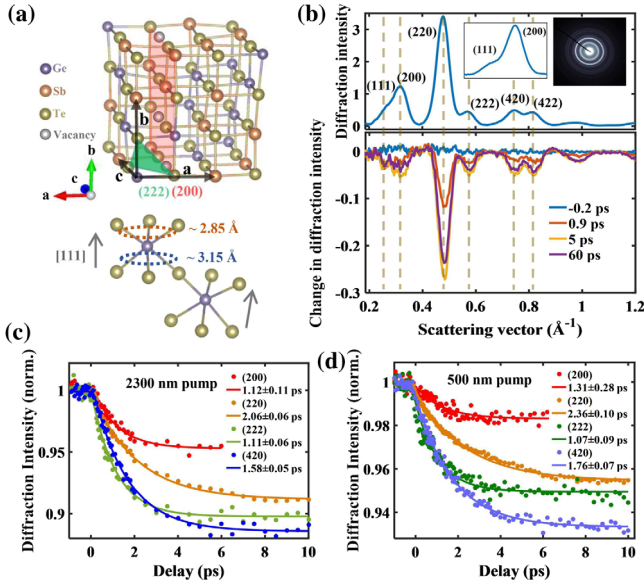


FIG. 1. The crystalline structure and the transient structural dynamics of GST-225. (a) Top: a supercell structure of the rocksalt phase GST-225. The length of three axes of single unit cell is 6.01 Å. Bottom: a schematic illustration of randomly oriented Peierls distortions with longer ( $\sim 3.15$  Å) and shorter ( $\sim 2.85$  Å) Ge(Sb)—Te bonds in the subunit cell. The gray arrows indicate the orientation of the adjacent local rhombohedral structures. (b) Top: the transmission diffraction pattern and the radially averaged intensity distribution (background subtracted) at a negative delay. The inset displays a better separation of (111) and (200) reflections by controlling the current of the magnetic lens positioned after the sample. Bottom: the change in diffraction intensity from  $-0.2$  to 60 ps, by subtracting the intensity distribution at negative delay with 2300 nm and 4.6 mJ/cm<sup>2</sup> laser pump. (c),(d) The temporal evolution of the normalized Bragg reflection intensities with 2300 nm (4.6 mJ/cm<sup>2</sup>) and 500 nm (0.61 mJ/cm<sup>2</sup>) laser pump. The solid lines are the fits with a monoexponential function.

multicenter hyperbonding [14]. The randomly oriented Peierls distortions in GST-225 are a typical characteristic of local correlated structures in crystalline functional materials [15–20]. What is common to material systems with local correlated structures is that there exists a distinction between the local symmetry and the average symmetry imposed by the crystal lattice [15]. Photoexcitation of GST-225 triggers a drastic optical contrast within 100–200 femtoseconds, which has been attributed to the depletion of electrons from the metavalent bonds [21,22]. Besides such an electronic structure modulation, breaking the bond alignment [23] and the medium range order of the lattice structure [24] may also induce a significant change of the optical matrix elements. However, the precise knowledge on the ultrafast structural response is still not clear. The atomic motions after femtosecond laser excitation have been studied extensively by time-resolved molecular dynamics simulation, ultrafast spectroscopy,

and ultrafast electron–x-ray diffraction. Nevertheless, the developed structural response models, including the phonon-driven symmetry change [25,26], the rattling motion [23], the selective bond breaking [8,9], and the simple thermal response [21,22], do not provide a consistent picture. The local Peierls distortions in the rocksalt phase of GST-225, which may associate with the highly damped Raman-inactive phonon mode in Ref. [22], is always neglected in previous studies.

Ultrafast electron–x-ray probes enable a direct access to transient atomic and electronic motions in a broad range of fundamental physical processes after femtosecond laser excitation [27–35]. Here, we report the photoinduced local structural dynamics in phase change material GST-225 revealed with a combination of femtosecond electron diffraction, structure factor calculation and time-dependent density-functional theory molecular dynamic simulations (TDDFT MD). The high temporal resolution of  $\sim 150$  fs in our experimental system [36,37], which is comparable to the highest vibration period of phonons in GST-225 [22], enables a direct access to the ultrafast structural response induced by electronic excitation. A consistent physical scenario toward an ultrafast transition of the local correlated structure is unveiled.

In the experiment, we use a femtosecond laser pulse to excite the GST-225 nanofilm with a thickness of 15 nm (see details in Supplemental Material [38]). A femtosecond electron pulse diffracts off the transient lattice structure and the radially averaged intensity distribution of the rocksalt phase GST-225 are shown in Fig. 1(b) (top). The change in diffraction intensity in Fig. 1(b) (bottom) shows a transient decrease of the Bragg reflection intensities after femtosecond laser excitation. To analyze the detailed temporal evolution, we fit an empirical background function (exponential plus second-order polynomial) and pseudo-Voigt line profiles to the peaks in the radial averages at each pump-probe delay [55].

Since the band gap of GST-225 is 0.4–0.5 eV [56], we choose to pump with two different wavelengths, i.e., 2300 nm (0.54 eV) and 500 nm (2.48 eV), in order to investigate the possible impact of the carrier relaxation to the structural response [57]. The relative intensity changes of several Bragg reflections with 2300 and 500 nm optical pump are depicted in Figs. 1(c) and 1(d). We fit the temporal evolution of the intensity with a monoexponential function, convolved with a Gaussian function of 150 fs FWHM to account for the instrument response function. For each reflection, the time constant is nearly the same for 500 and 2300 nm laser excitation. With increasing pump fluence, the temporal evolution of the intensities and the corresponding time constants are summarized in Figs. S1 and S2 and Tables 1 and 2 in Supplemental Material [38]. In these measurements, the density of the excited electrons, i.e.,  $(0.69 - 2.43) \times 10^{15}$  cm<sup>-2</sup>, with 2300 nm laser pump ( $\sim 2-7$  mJ/cm<sup>2</sup>) is

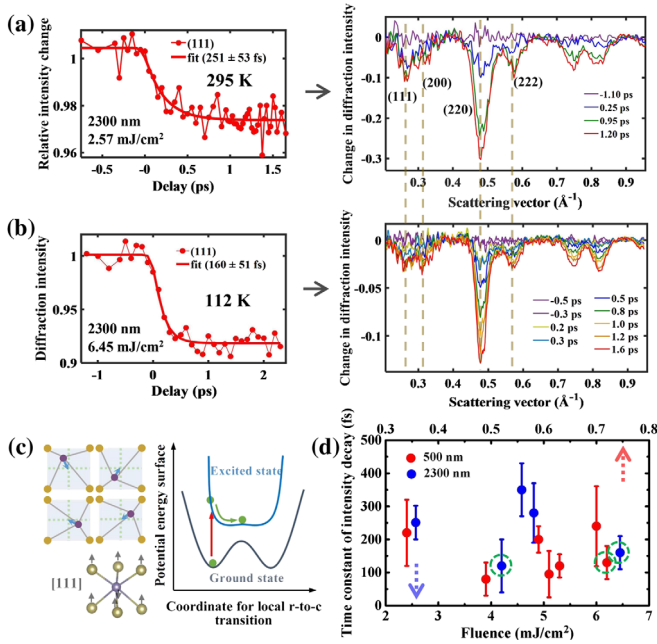


FIG. 2. (a) and (b) with 2300 nm laser pump. Left: normalized intensity change of the (111) reflection as a function of the time delay at room temperature and 112 K, respectively. The solid line is the fit with a monoexponential function. Right: overall change in diffraction intensity from negative to positive delays. (c) The 2D square nets indicate the suppression of the local distortions with random orientation among unit cells. The Ge/Sb atoms (purple balls) move toward the high symmetry position (the corresponding movement of the Te atom is not shown for simplification). In the subunit cell, the sketch of the atomic displacement along the [111] direction and the corresponding potential energy surface change are shown. (d) Time constants of the intensity decay of the (111) reflection with different pump fluences for 500 and 2300 nm laser excitation at room temperature and 112 K. The error bars correspond to 68.3% confidence intervals of the fit. The measurements carried out at 112 K are marked with green balls.

comparable to the density of  $(0.59 - 1.57) \times 10^{15} \text{ cm}^{-2}$  for 500 nm laser excitation ( $\sim 0.3 - 0.8 \text{ mJ}/\text{cm}^2$ ) (see density calculation in Supplemental Material [38]). The similar structural response with 500 and 2300 nm laser excitation is obtained. Whether the carrier relaxation will impact the amorphization process at a higher pump fluence [57] needs to be studied further, but at the low to medium pump fluence, we do not observe distinct structural responses with low to high photon energy excitation.

The local Peierls distortion is randomly oriented along the  $\langle 111 \rangle$  directions in the averaged rocksalt phase of GST, therefore, we focus on the intensity change of the (111) reflection as shown in Fig. 2. To avoid the possible impact of the heat accumulation [58], experiments at room temperature [Fig. 2(a)] and at 112 K [Fig. 2(b)] are performed. The temporal evolution of the intensities in Figs. 2(a) and 2(b) clearly indicate an ultrafast intensity decay of the (111)

reflection. See detail on the data processing in Figs. S3–S6. The time constant of 0.1–0.3 ps for the (111) reflection is significantly smaller than other Bragg reflections in Fig. 1, which may relate to the local Peierls distortion. In conventional crystalline material, the Peierls distortion is generally induced by electronic instability, and a femtosecond laser excitation gives rise to a suppression of such structural distortion by coherent phonons [25,27,29]. Regarding GST-225, the Peierls distortions are local and randomly oriented along the  $\langle 111 \rangle$  directions, therefore, the suppression of adjacent local Peierls distortions across unit cells should be incoherent as illustrated in Fig. 2(c). In particular, the photoexcitation flattens the potential energy surface along the  $\langle 111 \rangle$  directions and enables the opposite movement of the Ge(Sb) and Te atom, giving rise to the local structural transition from the rhombohedral to the cubic geometry. With different pump fluences, the time constant for the intensity decay of the (111) reflection is 0.1–0.3 ps as shown in Fig. 2(d), smaller than the period of the  $A_{1g}$  phonon mode in GeTe [25]. A quasilinear dependence of the amplitude of the (111) intensity change on the pump fluence is shown in Fig. S7 [38].

To quantify the suppression of the local rhombohedral structure, we perform the structure factor calculation. Note that for crystalline materials with local correlated structures, such as the randomly oriented Peierls distortions in adjacent unit cell in GST-225, the diffraction intensity is calculated by a summation over all unit cells included. If the atomic motions associated with such local distortions are uncorrelated across unit cells, the structure factor calculation of a single unit cell can be used to evaluate the diffraction intensity change of the overall lattice. See details about the structure factor calculation in Supplemental Material [38]. Figure 3(a) shows the calculation results on two displacement models, indicating the transition from the rhombohedral to the cubic configuration ( $-0.15$  to  $0 \text{ \AA}$ ). In the first model, only the offset displacement for the Ge atom along the [111] direction is considered because the lighter Ge atom is expected to move more easily than the heavier Te and Sb atoms, while in the second model, the opposite displacement of Ge/Sb and Te atom along the [111] direction is used. The same anisotropic intensity change for the two models is observed, i.e., the intensity of the (111) reflection is decreased significantly while the intensity of the (200) and (220) reflection is unchanged or enhanced. In Fig. 3(b), we extract the experimental intensity change within 0.3 ps, where the intensity decay of the (111) reflection is almost done. As seen, the intensity of the (111) reflection is decreased remarkably while the intensity for the (200) and (220) reflection remains, which is qualitatively in agreement with the structure factor calculation in Fig. 3(a).

For a more precise structure factor calculation, two points need to be considered. First, in the averaged rocksalt phase of GST-225, the length of the shorter Ge–Te bond



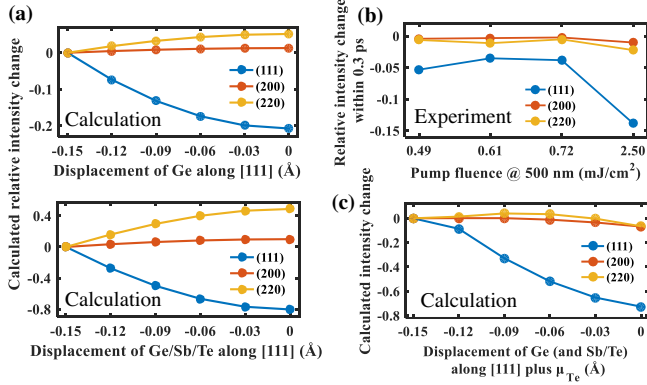


FIG. 3. Calculated and experimental anisotropic intensity change associated with the suppression of the local rhombohedral structure. (a) By structure factor calculation, the relative intensity change of Bragg reflections ( $|F|^2$ ) with (top) only the offset displacement of Ge atom and (bottom) the opposite displacement of Ge/Sb and Te atom along the [111] direction. The zero displacement position indicates the site in an ideal rocksalt phase. The offset displacement of atoms along the [111] direction represents the local rhombohedral structure in individual unit cell. (b) The measured anisotropic intensity change within 0.3 ps, i.e., the larger intensity change of the (111) reflection than that of (200) and (220) reflection. (c) Calculated intensity change with the displacement of Ge from  $-0.15$  to  $0$  Å, Sb from  $-0.08$  to  $0$  Å, Te from  $0.08$  to  $0$  Å, and the vibrations of Te atoms ( $\mu_{Te} \sim 0$  to  $0.2$  Å).

in Ge-centered local rhombohedral structure is  $\sim 2.84$  Å ( $\sim 2.82$  Å), while the shorter Sb—Te bond in Sb-centered local rhombohedral structure is  $\sim 2.91$  Å ( $\sim 2.88$  Å) at 100 K (300 K) [9]. Second, with electronic excitation, the strong local force would drive large vibrations of Te atoms around vacuum sites within 0.5 ps [59]. With the combination of the bond length discrepancy and the local vibrations of Te atoms (see details in Supplemental Material [38]), the calculated intensity change in Fig. 3(c) agrees better with the experimental results in Fig. 3(b).

We further confirm the photoinduced suppression of local Peierls distortions by performing TDDFT MD simulations in the averaged rocksalt phase of GST-225. See details about the simulation in Supplemental Material [38]. The photoexcitation effect is simulated by moving a certain percent of electrons from the valence band maximum states to the conduction band minimum states, similar to the method in Ref. [25]. In order to characterize the bond length discrepancy in the supercell, the bond pairs in each linear triatomic bonding geometry, such as the Te—Ge—Te and the Te—Sb—Te, are tracked during the temporal evolution. The contour map in Fig. 4(a) displays the distribution of the collected bond pairs. At the ground state ( $T = 0$  ps), the elliptical distribution of the contour map in Fig. 4(a) (left) indicates the shorter and the longer bond in most bond pairs, an intrinsic character of the local Peierls distortions. After electronic excitation, as shown in Fig. 4(a) (right), the ellipticity of the contour map is

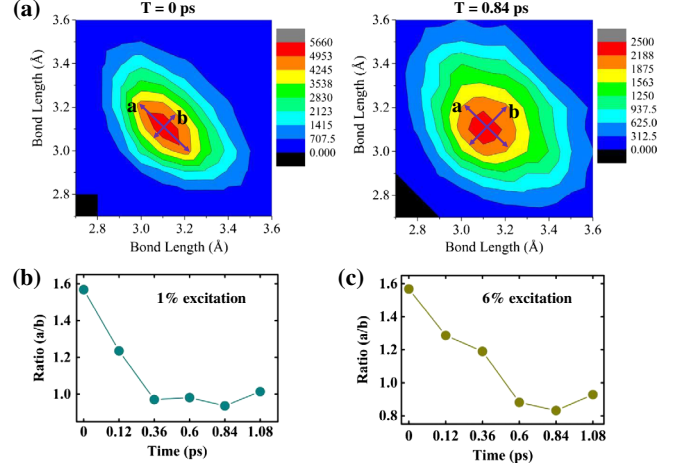


FIG. 4. Simulation results of bond length in linear triatomic bonding geometry in the supercell of GST-225 before and after the electronic excitation. (a) Left: the bond length distribution within the supercell before laser excitation. The tilted distribution indicates a shorter and a longer bond in a linear triatomic bonding geometry. Right: the bond length distribution at 0.84 ps (integration from 0.72 ps to 0.96 ps) after 6% electronic excitation. (b),(c) The temporal evolution of the ratio between the length of line *a* and line *b* [indicated in (a)] with 1% and 6% electronic excitation, respectively. The decrease of the ratio, i.e., the ellipticity, indicates the suppression of the bond length discrepancy in a linear triatomic bonding geometry.

reduced, i.e., the bond length discrepancy in bond pairs is reduced. We quantify the change of the bond length discrepancy by calculating the ellipticity of the contour map, i.e., the ratio between the length of line *a* and line *b* for the brown zone. The maximum electronic excitation in our experiment is  $\sim 1\%$  (see the calculation in Supplemental Material [38]), therefore we perform 1% and 6% electronic excitation in the simulation. The temporal evolutions of the ellipticity are displayed in Figs. 4(b) and 4(c) (each data point is the statistic over  $\pm 120$  fs). The ellipticity has decreased apparently from 1.6 to around 1, indicating the suppression of the local Peierls distortions. Moreover, the change of the ellipticity completes within 0.3–0.6 ps, which agrees with the timescale of the ultrafast intensity decay of the (111) reflection in Fig. 2. Therefore, the simulation result coincides with the physical model of the photoinduced suppression of local Peierls distortions concluded from the experimental results. The time-resolved bond length distribution by counting the Te—Ge—Te and the Te—Sb—Te geometry is shown in Fig. S9 [38].

Using femtosecond electron diffraction, structure factor calculation, and TDDFT MD simulations, we reveal the ultrafast suppression of the local Peierls distortions in the averaged rocksalt phase of GST-225. Distinct from the model of the rattling motion [23], the selective bond breaking [8,9], and the simple thermal response [21,22], a local structural transition from the rhombohedral to the cubic geometry within  $\sim 0.3$  ps is demonstrated. Since the

Peierls distortion is randomly oriented along the  $\langle 111 \rangle$  directions, the atomic displacements associated with the suppression of Peierls distortions are coherent within single unit cell while they are incoherent across unit cells. Therefore, such local structural transition is intrinsically different from the conventional long-range symmetry change, such as the rhombohedral-to-cubic transition in GeTe driven by the coherent  $A_{1g}$  phonon mode [25]. The highly damped Raman-inactive phonon mode observed in GST-225 in Ref. [22] should be attributed to the ultrafast suppression of the local Peierls distortions instead of the vacancy sites. The timescale of the ultrafast local structural transition is comparable to that of the ultrafast dielectric function change [21,22], therefore the local structural transition is expected to contribute to the dielectric function change (see discussion in Supplemental Material [38]).

A direct solid-solid amorphization bypassing the molten state has long been pursued in GST-225 [6–9,57,59,60]. In contrast to the enhancement of the Peierls distortions with rising temperature [9], the identified ultrafast suppression of the local Peierls distortions here is photoinduced. The similar ultrafast structural responses with 2300 and 500 nm laser pump indicate no significant impact of the carrier relaxation to the structural dynamics [57]. We expect upon increasing the pump fluence, the suppression of the Peierls distortions will be the intrinsic process driving the system toward the amorphization phase, which is distinct from the model of the selective breaking of the longer bonds in the local distortion [9,61]. See more discussion on the possible direct amorphization based on the ultrafast local structural transition in Supplemental Material [38].

The local correlated structure (which is also termed as correlated disorder in Ref. [15]), such as the randomly oriented Peierls distortions in GST-225 and BaTiO<sub>3</sub> (see Ref. [15]), is an intrinsic structural character in many functional crystalline materials and is important for the particular function of interest [15,16]. With conventional crystallography technique (such as static x-ray diffraction) determining the space group, only the long-range order is clearly detected and the local correlated structure is hidden. How to characterize the local correlated structure and establish the link between the physical properties and the disorder is challenging. Intrinsically, such local correlated structures by electronic instability are expected to be released in an ultrafast way after femtosecond laser excitation. The corresponding ultrafast atomic motions can be isolated by monitoring the ultrafast intensity changes of diffraction peaks (we present in this Letter and in Ref. [62]), the diffuse scattering [63] and the pair distribution function. Then the local correlated structures can be visualized and identified directly with time-resolved diffraction method, such as ultrafast electron diffraction. We anticipate that the comprehensive analysis of the ultrafast structural responses in our Letter will help to reveal the local correlated structure [15–20] in crystalline functional materials, and deepen the

understanding of the local structural dynamics in halide perovskites [64,65] and order-order–disorder structural transitions in thermoelectric materials [66–69] and ferroelectric-paraelectric transitions [70,71].

All data needed to evaluate the conclusions in the Letter are present in the Letter and the Supplemental Material [38]. Materials related to this Letter may be requested from the corresponding author Yingpeng Qi or the corresponding author Ralph Ernstorfer.

We acknowledge Miquel Rudé and Valerio Pruneri at ICFO-Institut de Ciències Fotòniques for providing the GST-225 samples, Professor Ming Xu and Doctor Meng Xu at Huazhong University of Science and Technology for helpful discussion. Yingpeng Qi acknowledges support by the Sino-German (CSC-DAAD) Postdoc Scholarship Program (Grants No. 201709920054 and No. 57343410). Hélène Seiler acknowledges support by the Swiss National Science Foundation under Grant No. P2SKP2 184100. Work in J. L. U. was supported by the National Natural Science Foundation of China (Grants No. 61922035 and No. 11904118). High-Performance Computing Center (HPCC) at Jilin University for computational resources is also acknowledged. Yingpeng Qi and Ralph Ernstorfer initiated the project. Yingpeng Qi executed the experiments with the help from D. Zahn, T. Vasileiadis, and H. Seiler. Yingpeng Qi did the data analysis and the structure factor calculation. Nianke Chen and Xianbin Li did the TDDFT MD simulation. Yingpeng Qi wrote the manuscript with contributions from all the authors.

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